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Theoretical studies of non-noble metal single-atom catalyst Ni₁/MoS₂: Electronic structure and electrocatalytic CO₂ reduction

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ABSTRACT Single-atom catalysts (SACs) have aroused significant interest in heterogeneous catalysis in recent years because of their high catalytic selectivity and tunable activity in various chemical reactions. Herein, non-noble metal SACs with 3d-series metal single atoms (M_1) (M = Sc, Ti, V, Cr, Mn,Fe, Co, Ni, Cu, Zn) supported on MoS2 are computationally screened by using first-principles quantum-chemical theory. The Ni₁/MoS₂ catalyst is found to be the most stable among those 3d-series SACs due to the optimal binding energy. In order to provide a fundamental understanding of the intrinsic stability and bonding interaction between the metal single atoms and MoS2 support, the electronic structure, including the spin density populations, charge density difference (CDD), electron localization function (ELF), band structure, density of states (DOS), and crystal orbital Hamiltonian populations (COHP) are systematically examined. The solid-state quantum theory of atoms in molecules (QTAIM) is also applied to further characterize the Ni-S and Mo-S covalent and ionic bonding nature between the metal single atoms and support. It is found that in addition to Ni-S bonding, there exists significant Ni-Mo bonding that is critical for the electronic structure, stability, and catalytic properties of Ni₁/MoS₂ catalyst. As a typical application of this Ni₁/MoS₂ catalyst, the electrocatalytic mechanism and reaction pathway of CO2 reduction reaction (CO2RR) on Ni1/MoS2 catalyst have been investigated. The MoS₂-supported Ni single atoms are found to exhibit high catalytic activity for CO₂RR to methanol. The calculational results provide theoretical insights towards the design of highly efficient SACs on MoS2-based functional materials.

Keywords: quantum theory of atoms in molecules, single-atom catalyst, electronic structure, electrocatalytic, CO₂ reduction

INTRODUCTION

Transition metal dichalcogenides have received extensive attention in recent years due to their distinct physical and chemical properties and robust performance in various applications [1–6]. Particularly, metal disulfides MS_2 (M=Cr,Mn,Fe,Co,Ni,Zn,Mo,W) have been widely studied for electrochemical applications [7–13]. Among these developed MS_2 materials, MoS_2 is identified as an ideal electrocatalytic candidate because of its highly manipulable doping geometries and characteristic electronic structures [14–18], which can be advantageous for CO_2 reduction [19,20]. Nevertheless, MoS_2 has

limited electrocatalytic activity on account of its high-proportioned electrochemically inactive inter-area [21]. The restrictive electron transfer ability also puts restraint on the transport route of electrons, leading to low electrocatalytic reactivity [22]. There are two strategies to enhance the electrocatalytic efficiency of MoS_2 : (1) increasing its intrinsic activity; (2) augmenting the number of active sites [23]. Currently, efforts have been made by introducing promoter cations, which can activate the inert S atoms of the basal plane [24].

Since Zhang, Li, Liu and colleagues [25] proposed the novel concept of single-atom catalyst (SAC) in 2011, the heterogeneous SACs have caused increasingly extensive interest in heterogeneous catalysis and bio-catalysis due to the highly efficient utilization of atoms and well-defined active centers [26-30]. Among them, the transition metal atoms have the advantage of controllable coordination geometries, electronic structures, and spin states of the active center, which can dictate the catalytic behavior and activity of SACs [28,31-33]. Different from nano-catalysts, the transition metal SAC's geometric, electronic, spin and magnetic states can be fine-tuned to generate desirable catalytic selectivity [34]. With these features, SACs can be delicately designed with MoS₂ support to ensure electronic conductivity and maximize the exposed active sites. Therefore, indepth understanding of the enhancing effects of SACs needs to be established for promoting targeted catalytic reactions. With the emergence of new characterization techniques and synthesis strategies (e.g., defect engineering strategies, spatially constrained strategies, and coordinated design strategies [35]), SACs have been proven to be valuable in various catalytic reactions, including oxygen reduction reaction (ORR) [36-40], hydrogen evolution reaction (HER) [41], N₂ reduction reaction (NRR) [42-53], CO₂ reduction reaction (CO₂RR) [54-60], CO oxidation reaction (COOR) [61], and biomass conversion [27].

Electrochemical CO_2RR has been expected as a facile route for converting CO_2 to CO, alcohols, acids and other organics [62,63]. However, owing to the high stability of CO_2 molecules and the complicated reaction network, CO_2RR is kinetically slow and suffers from high overpotential, low Faradaic efficiency, and unsatisfactory selectivity. Moreover, the concurrent HER (2H⁺ + 2e⁻ \rightarrow H₂) is vying on CO_2RR in aqueous electrolytes [64]. Therefore, it is interesting to explore MoS₂-based non-noble metal SACs for CO_2RR electrocatalysts for direct production of alcohols with high selectivity.

Herein, theoretical studies are carried out on the stabilities of SACs with the non-noble 3d-series of metal single atoms (M_1 , M = Sc, Ti, V, Cr, Mn, Fe, Co, Ni, Cu, Zn) supported on MoS_2 . It is

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found that Ni_1/MoS_2 is the most stable SAC among those studied here, which is then selected as a typical example to explore the chemical bonding between the metal atoms and MoS_2 support surface. The electronic structures and electrocatalytic performance for CO_2RR with Ni_1/MoS_2 have been investigated in details.

THEORETICAL AND COMPUTATIONAL DETAILS

Theoretical calculations were carried out based on periodic density functional theory (DFT) using the Vienna ab initio simulation package (VASP 5.3.2) [65,66]. Generalized gradient approach (GGA) was applied using the Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional [67]. The projector-augmented wave (PAW) method with scalar relativistic effects was adopted to account for the interaction of valence electrons and ionic core. The kinetic energy cutoff of 500 eV was used for the planewave basis set. The geometries were fully optimized by using the spin-polarized Kohn-Sham self-consistent field method with force convergence standard of 10⁻⁶ eV and 0.02 eV Å^{-1} , respectively. The Brillouin zone was sampled with a $3 \times 3 \times 1$ Monkhorst-Pack k-point mesh for geometry optimizations and more accurate $5 \times 5 \times 1$ mesh was used for MoS₂ electronic structure analysis. The Bader charges were also calculated to investigate the charge population [68]. Meanwhile, the charge density difference between the fragments (A, B) and the combined A+B was calculated by $\Delta \rho = \rho_{A+B} - \rho_A - \rho_B$, where ρ represents the total electron density.

The 3 × 3 × 1 MoS₂ monolayer supercell (containing 27 atoms) with a 15-Å vacuum region was used to avoid the artificial interaction among adjacent image units. The binding energy ($\Delta E_{\rm M_1}$) of the metal single atom supported on MoS₂ was calculated as

$$\Delta E_{\rm M_1} = E_{\rm M_1/MoS_2} - E_{\rm M_1} - E_{\rm MoS_2}$$

where $E_{\rm M_1/MoS_2}$, $E_{\rm M_1}$, and $E_{\rm MoS_2}$ are the energies for the M₁/MoS₂, bulk metal unit cell-based single atom, and MoS₂ monolayer, respectively. As the energy of the metal single atom is referenced to the unit cell of bulk metal crystal ($E_{\rm M_1} = E_{\rm metal}/n$, here n is the number of metal atoms in the unit cell), all the binding energies of M₁ on MoS₂ are positive (i.e., endothermic), implying that the smaller the binding energy value is, the least endothermic the process is, and therefore the more stable the SAC becomes. The

projected crystal orbital Hamilton population (COHP) was used to analyze the bonding properties between Ni and S atoms in the solid-state material [69].

In the electrochemical calculations, the Gibbs free energy of electrochemical CO₂RR was evaluated based on the computational hydrogen electrode (CHE) model proposed by Nørskov *et al.* [70]. Using the reversible hydrogen electrode (RHE), the free energy of H⁺ + e⁻ in aqueous solution was set to equal to 1/2 H₂, under standard conditions [71]. The free energy (ΔG) of a reaction at ambient temperature (T = 298.15 K) and pressure (P = 1 atm) was calculated as

$$\Delta G = \Delta E + \Delta E_{\text{ZPE}} - T \Delta S + e \Delta U,$$

where ΔE is the total reaction energy at the given temperature and pressure, $\Delta E_{\rm ZPE}$ is the correction of zero-point energy, $T\Delta S$ is the entropy contribution, and $e\Delta U$ denotes the free energy change with the applied electrode potential.

Inasmuch as metal single atoms are bonded with MoS₂, ADF-Band program [72] was further utilized to investigate the single atom charges and metal-support bonding properties to determine the bond critical point (CP), bond path and other properties through a method based on Bader's quantum theory of atoms in molecules (QTAIM) [73]. These solid-state QTAIM calculations were performed with PBE functional and uncontracted TZ2P Slater basis functions which provide fundamental bonding data for SACs.

RESULTS AND DISCUSSION

Single atom binding and stability of M₁/MoS₂

For single atom binding on the MoS_2 monolayer, there are three high-symmetry inequivalent sites to be considered: (1) the top of Mo, (2) the top of S, and (3) the hollow site. For all 10 transition metals (from Sc to Zn) considered, two types of optimized stable structures are obtained: the hollow site (face-centered cubic: Sc, Ti, Mn) as shown in Fig. 1a, and the top of Mo (hexagonal close-packed: Fe, V, Cr, Co, Ni, Cu, Zn) (Fig. 1b), respectively. The bond length of M_1 to S atom ranges from 2.11 to 2.40 Å, implying significant M–S chemical bonding interaction. The exception is for Zn, which has a distance d_{M-S} of 3.52 Å, indicating chemically unbound Zn···S, consistent with the $d^{10}s^2$ closed shell of Zn. The S–M–S angles lie between 85.6° to 106.4° from Sc to Cu.

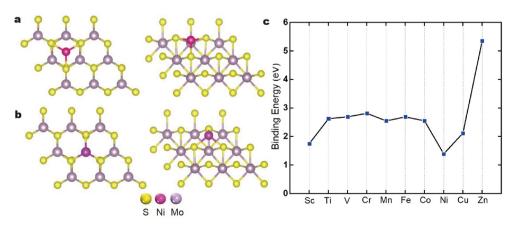


Figure 1 Optimized geometry structures and binding energy (ΔE_{M_1}) for M_1/MoS_2 (M = Sc-Zn) monolayers. Top view and 45° side-view for (a) the hollow site and (b) top site of Mo. (c) Binding energy ΔE_{M_1} for M_1/MoS_2 with metal single-atom energy obtained from the unit cell of bulk metal crystal.

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The binding energy $\Delta E_{\rm M_1}$ of M₁/MoS₂ is plotted in Fig. 1c. As mentioned earlier, all the binding processes of M1 to MoS2 are endothermic when the M₁ atom is taken from the bulk crystal. That is, the smaller positive binding energy represents higher stability of the SACs, which is in contrast to the exothermic (negative) binding energy calculated with reference to isolated M₁ atom. From Fig. 1c, while the least stable is Zn₁/MoS₂ with $\Delta E_{\rm Zn} = 5.37$ eV, the most stable is Ni₁/MoS₂ with $\Delta E_{\rm Ni} = 1.39$ eV, with Sc_1/MoS_2 and Cu_1/MoS_2 being the second and third most stable. The specific binding energy values calculated are listed in Table S1. These binding energies indicate that the M₁/MoS₂ SACs (except Zn) that have been considered in this study are possible targets for experimental synthesis. Given the theoretically predicted highest stability of Ni₁/MoS₂ among these SACs, the following section will focus on the electronic structures and electrocatalysis of this promising SAC hereafter.

To interpret the binding energy trend, the COHP was calculated to analyze the interaction between M_1 (M = Mn, Ni, Cu) and the nearest S atom (Fig. 2). The COHP information of other 3d-series transition metals M_1 (M = Sc, Ti, V, Cr, Fe, Co, Zn) to S is depicted in Fig. S1. In addition, the integrated COHP (ICOHP), which was obtained by integration of COHP over all levels up to the Fermi level, was also calculated. The ICOHP represents a particular bond's contribution to the band-structure energy, thus measuring the bonding strength (the more negative the ICOHP, the stronger the bonding strength). It turns out that there is a nearly linear correlation between ICOHP and $\Delta E_{\rm M}$, which gives a quantitative explanation for the M₁ binding strength to MoS₂ in terms of the bonding orbital populations. Therefore, the bonding orbital populations can be used to account for the observed trends for the binding energies of these metal single atoms. This finding is consistent with the novel covalent metal-support interaction (CMSI) previously found for SACs [74].

Spin and charge density difference of Ni₁/MoS₂

As the 3d-series of transition metals have radially rather contracted 3d orbitals, the spin properties of their SACs are essential for the catalytic properties. Fig. 3a (top and side panels) presents the spin densities of the MoS₂ monolayer. It reveals that, as

expected, there is no obvious spin-polarization for MoS₂. In contrast, when the Ni single atom is supported on MoS₂, it has considerable spin polarization, as shown by the spin density analysis (Fig. 3b), which then causes slight spin polarization at the first nearest neighboring Mo atoms. Therefore, the Ni₁/MoS₂ material can be regarded as spin-polarized SAC through accommodating the magnetic structure of single atom-based sites [75,76].

To further investigate the origin of spin population and charge distribution of Ni₁/MoS₂, the following electronic structure analyses were carried out: (1) charge density differences obtained by comparing the total charge density from Ni₁ + MoS₂ and Ni₁/MoS₂, (2) Bader charges of Ni, Mo, and S atoms from the QTAIM analysis. Fig. 3c shows the charge density difference, from which one can see that the Ni atom has obviously transferred some electron density to the MoS₂ support. The transferred electron density from the Ni atom is primarily situated on the Ni-S bonding region. The two-dimensional (2D) charge density difference is shown in Fig. 3d, which again reveals that there is distinct charge transfer from the Ni atom area. The plane-averaged charge-density difference as calculated by using vaspkit is shown in Fig. 3e [77]. Along Z-coordination, the charge density difference changes from flat to major fluctuation near the Ni atom, and then becomes flat again. By means of the Bader charge analysis (Table S2), compared with the MoS2 support, the Ni atom in Ni₁/MoS₂ has positive charge of +0.39|e⁻|, which causes the electron density of Mo and S to increase by 0.16 and 0.02|e⁻|, respectively. That is, the electron density flows from Ni to Mo and S, which is consistent with the charge density difference results. The charge transfer between Ni and MoS2 is responsible for the enhanced structure stability of SACs because of the ionic interaction between Ni^{δ +} and S^{δ -} [78,79], in addition to the CMSI.

DOS and band structure of Ni₁/MoS₂

To further investigate the electronic structure of the SACs under consideration, the total and projected electron density of states (DOS) and band structure of the MoS_2 support and Ni_1/MoS_2 were studied and the results are shown in Fig. 4a, c. For Ni_1/MoS_2 , the change below -0.41 eV to Fermi level arises from the Ni-S bonding interaction as well as the d-states of Ni atoms.

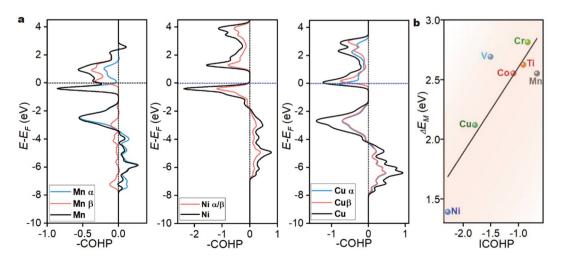


Figure 2 (a) The calculated –COHP between M_1 (M = Mn, Ni, Cu) and S atom of MoS_2 . (b) Illustration of the linear correlation between the ICOHP and the binding energy of M_1/MoS_2 (ΔE_{M_1}).

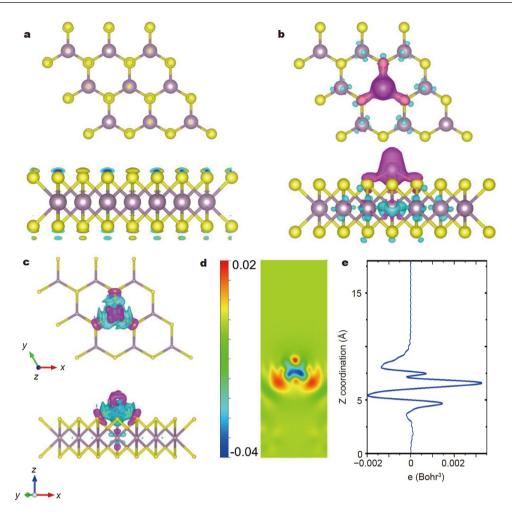


Figure 3 The spin densities of (a) MoS₂ monolayer and (b) Ni₁/MoS₂. The purple area defines the majority spin (α-spin). (c) Charge density differences of Ni₁/MoS₂. The blue and purple areas represent electron accumulation and depletion, respectively. (d) The 2D charge density difference in vertical xz plane, with the color from red to blue representing charges between 0.02 to -0.04 atomic unit (a.u.). (e) The corresponding plane-averaged charge density difference of (d).

Spin-polarized projected DOS (pDOS) reveals that Ni 3d orbitals play a dominant role in bonding with the MoS_2 surface (Fig. 4e), while the Ni 4s/4p states transfer electron density to S and lie much higher in energy. The corresponding partial charge distribution is shown in Fig. 4f, which indeed shows the positively charged Ni atom. The particular pDOS and COHP of Ni d orbitals (z^2 , yz, xz, xy, x^2 – y^2) are shown in Figs S2 and S3. Both of them indicate that the ionic and covalent bonding interactions between Ni and S atoms of the MoS_2 support lay the foundation for the stability of the SACs.

Based on the band structure, the conducting band minimum (CBM) and valence band maximum (VBM) are situated at the Γ points, arising from the Ni atom located at the surface. This electronic feature of Ni single atom is critical for its catalytic performance (*vide infra*). Importantly, there is no state crossing the Fermi level, indicating that the Ni₁/MoS₂ monolayer remains as an indirect gap semiconductor. The symmetric spin states in the DOS imply virtually nonmagnetic behavior of the Ni₁/MoS₂ monolayer as well. From the real space wave functions of the highest occupied molecular orbitals (HOMO) and the lowest unoccupied molecular orbitals (LUMO) at the CBM and VBM (Fig. 4b, d), one can see that the Ni atom has significant orbital distribution at the HOMO of Ni₁/MoS₂, which is viable for

catalytic quantum states. Furthermore, the $\mathrm{Ni_1/MoS_2}$ monolayer undergoes change in electrical conductivity with the Ni atom doped, relevant to the electrocatalytic properties.

ELF and work function analysis of Ni₁/MoS₂

In order to further examine the electronic effects of Ni single atom for MoS₂, the electron localization function (ELF) was analyzed in the *xz* plane [80]. It is clear that there is same-spin electron pair density between Mo–S and Ni–S, while the ionicity in the bonds of the MoS₂ monolayer is verified by the ELF value observed with 0.7 around S and 0.4 around Mo atoms (Fig. 5a). In addition, the work function of the surface was calculated by using vaspkit [77], which is an important parameter for evaluating the charge transfer ability of the interface and semiconductor band alignment. The work functions of MoS₂ and Ni₁/MoS₂ can be calculated with the equation [81]:

$$\Phi = E_{\text{vac}} - E_{\text{F}}$$

where Φ is the work function, $E_{\rm vac}$ represents the electrostatic potential of the vacuum, and $E_{\rm F}$ is the Fermi energy. According to the calculation, the Fermi energy of Ni₁/MoS₂ (Fig. 5c) is higher than that of MoS₂ (Fig. 5b), and the calculated work functions Φ of MoS₂ and Ni₁/MoS₂ are 5.497 and 5.192 eV,

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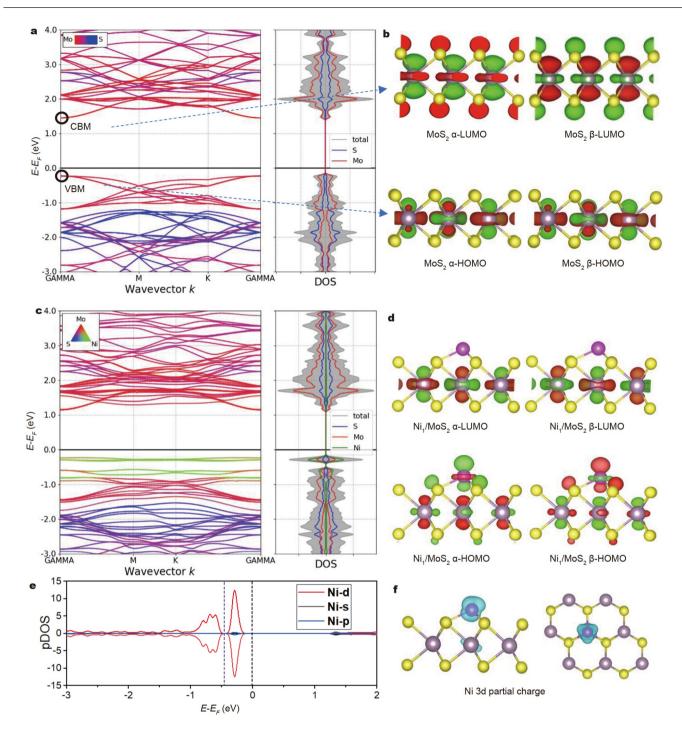


Figure 4 The band structures (with the Fermi energy E_F set to 0) and total DOS of (a) MoS₂ and (c) Ni₁/MoS₂. The α- and β-set real-space wave functions of (b) MoS₂ and (d) Ni₁/MoS₂. (e) The pDOS of Ni (3d, 4s, and 4p) orbitals out of the total DOS. (f) Partial charge distribution of Ni 3d.

respectively. The $\mathrm{Ni_1/MoS_2}$ has lower work function, implying that it is more likely to lose electron compared with $\mathrm{MoS_2}$. Therefore, by supporting Ni atoms on $\mathrm{MoS_2}$, the formed SAC becomes more favorable for the reduction reaction than the support, which can be advantageous for $\mathrm{CO_2}$ reduction.

QTAIM and schematic energy levels of Ni_1/MoS_2

Since SACs have been widely investigated in the past decade, it is interesting to explore the properties of the single atoms and the bonding interaction in Ni₁/MoS₂ by using QTAIM [73]. Based

on the topological analysis of the electron density, QTAIM makes it possible to divide a molecule or solid into constituent atoms on the basis of zero-flux surfaces in the one-electron density gradient field $\nabla \rho$. From the saddle-point properties of the electron density function, there are four types of CPs described as following [68]: nuclear CP (NCP), bond CP (BCP), ring CP (RCP), and cage CP (CCP). Fig. 6a exhibits the Hessian of electron density at the various CPs. There are obvious BCP between Ni–S and Mo–S bonds colored in blue, revealing the direct bonding interaction between Ni single atom and the

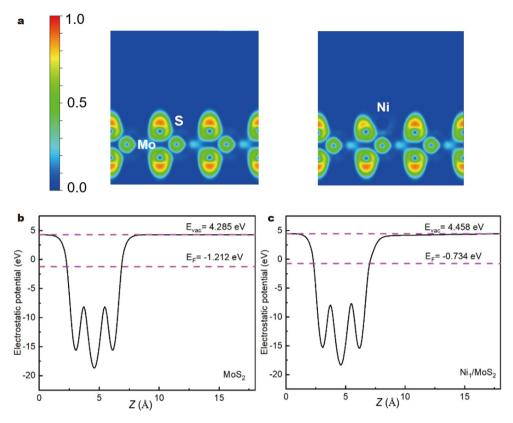


Figure 5 (a) Contour maps of the ELF of MoS_2 and Ni_1/MoS_2 . The probability of same-spin pair density varies from 0 (blue) to 1.0 (red). Electrostatic potentials and work functions of (b) MoS_2 and (c) Ni_1/MoS_2 .

support. In addition, RCP and CCP can also be observed as shown by the gray color. The quantified electron density at the cut-plane containing Ni-S-Mo-S is shown in Fig. 6b. To investigate the bonding interaction between Ni and the nearest neighboring atoms, the BCPs (red dots) and RCPs (green dots) are shown in Fig. 6c. It is interesting to note that in addition to the BCPs of Ni-S and Mo-S, there exists unexpected BCP of Ni-Mo, which implies that the bonding interaction between Ni and Mo atoms also plays a critical role in the electronic structure and catalytic performance of Ni₁/MoS₂. While CMSI mainly occurs between metal single-atom (M₁) and the nonmetal element of the support, this kind of direct metal-metal (here Ni-Mo) bonding has not been fully recognized. This interaction will further enhance the binding and stability of Ni single atom on the MoS₂ support, and regulates the catalytic properties of Ni₁/ MoS₂.

The electron density values at the BCP $\rho(r_{\rm bcp})$, which indicates the strength of the chemical bond, and Laplacian values of the electron density $\nabla^2 \rho(r_{\rm bcp})$ related to the ionic/covalent nature of the bonding as well as the key QTAIM parameters are listed in Table 1. The eigenvalues of the corresponding Hessian matrix are marked as λ_1 , λ_2 , and λ_3 . The ellipticity of a bond is calculated as $\varepsilon = |\lambda_1|/|\lambda_2| - 1$, which characterizes the electron delocalization of the bond under consideration. The $\rho(r_{\rm bcp})$ values decrease in the order of Ni–S > Mo–S > Ni–Mo, indicating that the Ni–Mo bonding is the weakest compared with the bonding of Ni–S and Mo–S. The $\nabla^2 \rho(r_{\rm bcp})$ values for Ni–S, Ni–Mo and Mo–S are all positive, with $|\lambda_1|/|\lambda_3| < 1$, indicating that the bonding interactions are primarily ionic bonds, dominated by the con-

traction of charge away from the BCP toward each of the nuclei [73]. For Ni–S and Ni–Mo, the ellipticity of the bonds are relatively small, ε = 0.057 and 0, respectively, which implies that they behave as σ single bonds because of the cylindrical shape of the bond cross section. The ε reaching 0.294 at Mo–S BCP is interpreted as a measure of the π character because from the single to double bond, the cross section tends to be more elliptic, reaching maximal ellipticity for a typical double bond [82].

The schematic energy levels are presented in Fig. 6d. From the DOS (Fig. 4c) and COHP results, S-2p states are located at the low-energy region of Ni₁/MoS₂, while the bands of Ni-3d and Mo-4d orbitals are broadened by the orbital overlap between Ni or Mo and S atoms. While Ni-3d band is fully occupied, the Mo-4d band is not completely filled, so there exist both occupied and unoccupied states around the Fermi level, consistent with the Mo(IV) oxidation state. Meanwhile, the Ni-4s,4p and Mo-5s,5p orbitals mainly contribute to the unoccupied states in Ni₁/MoS₂ due to the electron transfer and orbital mixing with S orbitals.

Electrocatalytic CO₂RR with Ni₁/MoS₂

The electronic structure features of Ni_1/MoS_2 indicate that this non-noble metal SAC is both stable and reactive for reduction reactions. Therefore, the performance of Ni_1/MoS_2 was evaluated for electrocatalytic CO_2RR . The reaction mechanism of CO_2RR on the Ni_1/MoS_2 catalyst was investigated \emph{via} the CHE model and DFT calculations with PBE functional. The reaction mechanism is summarized in Fig. 7, where the formation of methanol turns out to be the preferred pathway. Here the Ni atoms are found to have highly active catalytic sites for CO_2RR . For electrocatalytic CO_2RR , HER is a potential competing

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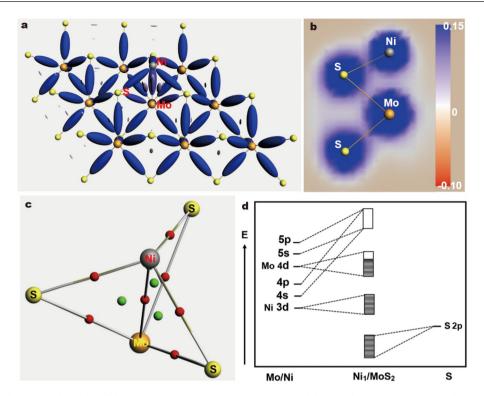


Figure 6 The results of topological analysis of the electron density. (a) Ni_1/MoS_2 Hessian of electron density at various CPs. (b) The electron density at the cut-plane containing Ni-S-Mo-S. (c) The BCPs (red dots) of Ni-S, Ni-Mo, and Mo-S, and the three RCPs (green dots) from QTAIM analysis. (d) Schematic energy levels showing the orbital interaction between Ni-S-Mo-S.

Table 1 The electron density values at BCP ($\rho(r_{bcp})$), Laplacian values of the density ($\nabla^2 \rho(r_{bcp})$), eigenvalues of the Hessian matrix, ratio values between the perpendicular and the parallel curvatures ($|\lambda_1|/|\lambda_3|$) of Ni₁/MoS₂ calculated by using PBE/TZ2P. All quantities are given in a.u.

ьср	$\rho(r_{\rm bcp})$	$\nabla^2 \rho(r_{\rm bcp})$	ε	λ_1	λ_2	λ_3	$ \lambda_1 / \lambda_3 $
Ni-S	0.105	0.212	0.057	-0.111	-0.105	0.428	0.259
Ni-Mo	0.056	0.103	0.000	-0.029	-0.029	0.161	0.180
Mo-S	0.071	0.123	0.294	-0.066	-0.051	0.240	0.275

process. As shown in Fig. S4, the ΔG for the desorption of H* on Ni₁/MoS₂ for HER is higher than that for *CO₂. Therefore, as shown in Fig. 7a, with CO₂ adsorption on the Ni atom of Ni₁/ MoS₂, it gets activated by η^2 - π -bonding with the Ni atom. The chemisorbed *CO₂ can further react with H⁺ + e⁻ to form *COOH, which will form adsorbed *CO and H2O when the OH is attacked by another H+ + e-. Interestingly, because *CO is strongly bound by the Ni atom via 5σ -donation and 2π -backdonation, its desorption from the catalytic sites to generate CO gas is partially suppressed. Instead, hydrogenation of Ni-bonded *CO by H⁺ + e⁻ leads to the formation of *CHO, which is beneficial to further catalytic reaction towards methanol. The different reaction pathways involving intermediates *C(OH), CO(g), *CH₂O, *CHOH, *CH₃O, *CH₂OH, *CH₄O and *CH₃OH were examined (Fig. 7b). For the optimal reaction pathway, the following steps occur:

$$CO_2 + * \rightarrow *CO_2, \tag{1}$$

$$*CO_2 + H^* \rightarrow *COOH,$$
 (2)

$$*COOH + H* \rightarrow *CO + H2O,$$
 (3)

$$*CO + H* \rightarrow *CHO,$$
 (4)

$$*CHO + H* \rightarrow *CH_2O,$$
 (5)

$$*CH_3O + H^* \rightarrow *CH_3O,$$
 (6)

$$*CH_3O+ H* \rightarrow *CH_3OH,$$
 (7)

$$*CH_3OH \rightarrow CH_3OH + *.$$
 (8)

Compared with the electronic structure of Ni_1/MoS_2 as discussed before, the Ni_1 doping on MoS_2 changes the charge and spin densities, which favors strong adsorption of CO, resulting in superior electrocatalytic CO_2RR performance toward methanol [83,84].

CONCLUSIONS

Heterogeneous SACs have recently aroused significant interest and become the frontier in catalysis science. This novel kind of catalysts provides an opportunity for tuning the catalytic properties using the surface atoms of the support. Therefore, developing non-noble metal catalysts via the idea of SACs looks promising. In this article, the first principles quantum theory has been used to computationally screen M_1/MoS_2 ($M_1 = Sc$, Ti, V, Cr, Mn, Fe, Co, Ni, Cu, Zn) with the first series of transition metals. The electronic structure of Ni_1/MoS_2 has been studied in

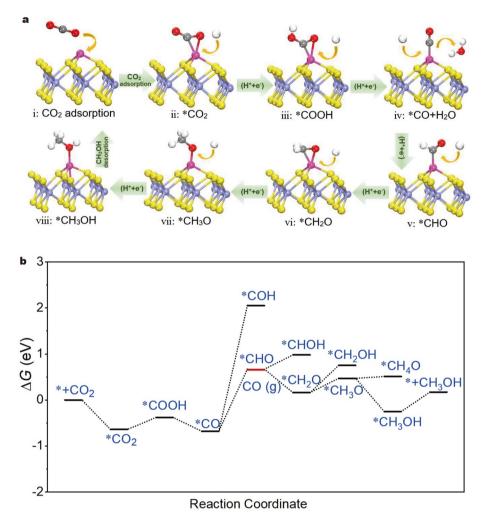


Figure 7 The reaction mechanism for CO_2RR into CH_3OH on Ni_1/MoS_2 . (a) The structures of the reaction intermediates; (b) the energy profile for different pathways calculated using the CHE model and DFT method with PBE functional.

details using solid state analysis approaches, including ELF, DOS, and band structure. The bonding interactions have been further examined by using COHP and QTAIM. The spin densities and charge density difference have been changed with Ni₁ doped on MoS₂, which influences the electrocatalytic performance for CO₂RR. The optimal reaction pathway of CO₂RR to methanol has also been investigated. The Ni single atoms have exhibited highly active catalytic sites for this reaction. Further exploration of this kind of non-noble metal SACs for other reduction reactions would be interesting.

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Supplementary information Supporting data are available in the online version of the paper.



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非贵金属单原子催化剂Ni₁/MoS₂的电子结构及CO₂还原电催化性能

于琦*

摘要 单原子催化剂(SACs)因其在化学反应中具有高选择性及催化活性,近年来在多相催化领域已引起广泛关注.本文采用第一性原理理论计算,对MoS2负载的非贵金属SACs进行计算化学筛选,考察了3d金属单原子(M=Sc,Ti,V,Cr,Mn,Fe,Co,Ni,Cu,Zn)负载的SACs的稳定性.我们发现,Ni₁/MoS₂催化剂具有最佳的结合能,在此3d系列催化剂结构中最稳定.为分析SACs的稳定性与成键作用,本文系统地研究了Ni₁/MoS₂的电子结构,包括使用自旋密度、电荷密度差分(CDD)、电子局域化函数(ELF)、能带结构、态密度(DOS)以及局部晶体轨道哈密顿量(COHP).此外,还应用分子中原子的固态量子理论(QTAIM)进一步表征了Ni-S、Ni-Mo及Mo-S键的共价性与离子性.此外,为研究Ni₁/MoS₂的电催化应用,对CO₂还原反应(CO₂RR)制甲醇的反应机理与路径进行了分析.计算表明,Ni₁/MoS₂对于CO₂RR具有较高的催化活性.本文为MoS₂基功能材料高效SACs的设计提供了理论依据.