



Synthesis of PdH_{0.43} nanocrystals with different surface structures and their catalytic activities towards formic acid electro-oxidation

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ABSTRACT The synthesis of nanocrystals (NCs) with defined morphology and surface structure provides an effective way to investigate the structure-activity relationship of nanocatalysts, and it will facilitate the design of nanocatalysts with excellent catalytic performance. In this paper, we developed a facile method to synthesize PdH_{0.43} NCs with the shape of cube, octahedron and rhombic dodecahedron (RD), whose surface facets are {100}, {111} and {110}, respectively. The as-prepared PdH_{0.43} NCs are highly stable and exhibit enhanced catalytic activity and extremely low overpotential towards electro-oxidation of formic acid compared with the commercial Pd black and three types of Pd NCs. The specific activity of the cubic PdH_{0.43} NCs is more than five times that of the commercial Pd black and two times that of the cubic Pd NCs. Among the three types of PdH_{0.43} NCs with different surface structure, the activity order is followed by PdH_{0.43} {100} > PdH_{0.43} {111} > PdH_{0.43} {110}.

Keywords: alloy, nanoparticles, palladium hydrides, surface structure, electrocatalytic activity

INTRODUCTION

The physical-chemical properties of noble metal based nanocrystals (NCs) are highly dependent on their size, composition and surface structure [1,2]. Controllable synthesis of noble metal based NCs with different surface structures is an important approach for achieving unique properties as well as enhanced performance in many important application fields [3,4]. Therefore, in the past decade, intensive studies have been carried out on controlling the noble NCs with various crystal facets exposed and exploring the relationship between exposed crystal

facets and properties [5–7].

Pt-based electrocatalysts are commercially predominant for fuel cells, but they are high cost and in poor reserve [8,9]. As an alternative solution, Pd-based electrocatalysts have attracted much attention [10–12]. Palladium is also well-known for its high affinity with hydrogen, and it can adsorb large volumetric quantities of hydrogen at ambient condition to form PdH_x. Due to this unique property, Pd plays an important role in various fields of applications including hydrogen detection [13], hydrogen storage [14], hydrogen purification [15], catalysis for alkyne hydrogenation [16], and so on. On the other hand, when forming PdH_x by the adsorption of hydrogen, the electronic structure of Pd is modified, and thus its catalytic activity is also tuned compared with pure Pd catalysts. So far, the palladium hydrides are typically formed by direct exposure of the metal Pd to hydrogen gas at certain condition [17,18]. Besides, PdH_x can be also formed *via* a mild chemistry method by reacting Pd particle with aqueous borohydride [19] or by applying negative potential to drive the evolved hydrogen atoms into the lattice of Pd NCs [20]. However, these as-prepared PdH_x nanoparticles are usually unstable due to spontaneous releasing of hydrogen in ambient condition, and thus they are difficult to be good candidate as promising catalysts. To date, PdH_x nanoparticles were sparsely explored as the electrocatalyst for fuel cells, let alone PdH_x NCs with well refined surface structures.

Very recently, unexpected stable PdH_{0.43} nanoparticles *via* a facile treatment of commercial Pd black with amines were successfully synthesized [21,22], exhibiting superior catalytic activity toward formic acid electro-oxidation

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[21]. To optimize the electrocatalytic performance of the stable PdH_{0.43} nanoparticles, it is thus necessary to study the relevance between surface-structure and catalytic activity. In this paper, we report three types of PdH_{0.43} NCs with different morphologies enclosed with {111}, {100} and {110} facets respectively by a two-step synthesis method. Compared with the pure Pd NCs and the commercial Pd black, the three as-prepared PdH_{0.43} nanocatalysts served as highly efficient catalysts in the formic acid electro-oxidation reaction. They not only improve the maximum oxidation current density of the formic acid, but also greatly reduce the oxidation overpotential with the activity order as PdH_{0.43} {100} > PdH_{0.43} {111} > PdH_{0.43} {110}.

EXPERIMENTAL SECTION

Chemicals and materials

Palladium(II) chloride (PdCl₂), sodium chloropalladite (Na₂PdCl₄), citric acid (C₆H₈O₇), polyvinylpyrrolidone (PVP, Mw=5500), L-ascorbic acid (AA), potassium iodide (KI), potassium bromide (KBr), sulfuric acid (H₂SO₄), formic acid (HCOOH) and ethanol (C₂H₅OH) were purchased from Sinopharm Chemical Reagent Co. Ltd. Commercial Pd black was purchased from Johnson Matthey. Hexadecyl trimethyl ammonium chloride (CTAC), hexadecyl trimethyl ammonium bromide (CTAB) and oleylamine (C₁₈H₃₇N) were purchased from J&K. *n*-Butylamine (C₄H₁₁N), *n*-hexylamine (C₆H₁₅N), ethanediamine (C₂H₈N₂), phenylamine (C₆H₇N), dodecylamine (C₁₂H₂₇N) and octadecylamine (C₁₈H₃₉N) were purchased from Alfa Aesar. Nitrogen (N₂, 99.99%) and mixed gas (4% CO + 96% N₂) were purchased from Linde Industrial Gases. All chemicals were used as received without further purification.

Preparation of PdH_{0.43} NCs

PdH_{0.43} NCs were prepared by a two-step method. Firstly, the cubic, octahedral and rhombic dodecahedral (RD) Pd NCs were synthesized *via* the modified method based on the references [23–25], respectively. Then, 2 mg of cubic/octahedral/RD Pd NCs was mixed with 10 mL of *n*-butylamine under magnetic stirring. The suspension was transferred to a Teflon-lined stainless-steel autoclave with a capacity of 25 mL. The sealed autoclave was heated from room temperature to 200°C in 60 min and kept at that temperature for 6 h, before it was naturally cooled to room temperature. The products were collected by centrifugation (9500 rpm for 5 min) and washed several times with ethanol.

Characterizations

The morphology and crystal structure of the as-prepared products were observed by scanning electron microscopy (SEM, Hitachi S4800) operating at acceleration voltage of 15 kV and high-resolution transmission electron microscopy (HRTEM, JEOL JEM 2100) with an acceleration voltage of 200 kV. The samples were prepared by dropping ethanol dispersion of samples onto carbon-coated copper TEM grids. The crystal phase of the as-prepared products was determined by the powder X-ray diffraction (XRD) pattern using a Rigaku Ultima IV X-ray diffractometer with Cu K α radiation. The surface states and chemical compositions of products were analyzed by X-ray photoelectron spectroscopy (XPS, K-Alpha+, Thermo fisher Scientific). All binding energies were calibrated using the contaminant carbon (C 1s = 284.8 eV).

Electrocatalytic measurement

Cyclic voltammograms (CV) were recorded using an electrochemical workstation (CHI 760e, Shanghai Chenhua Co., China). A glassy carbon electrode (diameter 5 mm) was carefully polished and washed before each experiment. The Pd or PdH_{0.43} nanocatalysts were dispersed in ethanol and 5% nafion (volume ratio: 0.02%). The concentrations of the three catalysts were 1.0 mg mL⁻¹. In a typical experiment, 5 μ L of the suspensions were deposited on the glassy carbon electrode as the working electrode. A carbon electrode and a saturated calomel electrode (SCE) were served as the counter electrode and the reference electrode, respectively. It should be noted that in order to remove possible organic adsorbates before other electrochemical measurement, the prepared Pd and PdH_{0.43} catalysts were firstly electrochemically cleaned by continuous potential cycling between -0.25 and 0.90 V in 0.5 mol L⁻¹ H₂SO₄ solution until a stable CV curve was obtained. For specific activity, the peak currents of nanocatalysts were all normalized to the electrochemically active surface area (ECSA). The ECSA was determined according to the equation ECSA = Q / q^0 , where Q is the electric charge calculated from the area of the coulombic charge of the reduction of PdO at 25°C in N₂-saturated 0.5 mol L⁻¹ H₂SO₄ solution, and q^0 is 430 μ C cm⁻² [26]. The catalytic activity was measured by CV in a solution containing 0.5 mol L⁻¹ H₂SO₄ + 0.25 mol L⁻¹ HCOOH at a scan rate of 50 mV s⁻¹ from -0.25 to 0.80 V. For the CO stripping experiments in 0.5 mol L⁻¹ H₂SO₄ solution, it was first purged with N₂ for 10 min, and then, the electrode was put into it, followed by CO purging for 15 min to allow the complete adsorption of CO onto the catalysts. After that, the electrode

was transferred to the electrochemical cell with $0.5 \text{ mol L}^{-1} \text{ H}_2\text{SO}_4$ solution bubbled with N_2 for 10 min for CO stripping measurement. The CO stripping voltammogram was recorded from -0.25 to 0.80 V with a scan rate of 20 mV s^{-1} .

RESULTS AND DISCUSSION

The cubic, octahedral and RD $\text{PdH}_{0.43}$ NCs were prepared *via* a facial *n*-butylamine treatment of prefabricated Pd NCs. The TEM (Fig. 1a–c) and SEM images (Fig. S1) show that the shapes of the as-prepared $\text{PdH}_{0.43}$ NCs are well maintained from their precursors of Pd NCs (Figs S2, S3). The cubic, octahedral and RD $\text{PdH}_{0.43}$ NCs are enclosed by $\{100\}$, $\{111\}$ and $\{110\}$ facets, respectively. The Fourier transition images (Fig. 1g–i) based on HRTEM images of the single $\text{PdH}_{0.43}$ cubic, octahedral and RD NCs (Fig. 1d–f) confirm the nature of single crystallinity and the predominant surface facets of these nano-crystalline samples being $\{100\}$, $\{111\}$ and $\{110\}$ facets, re-

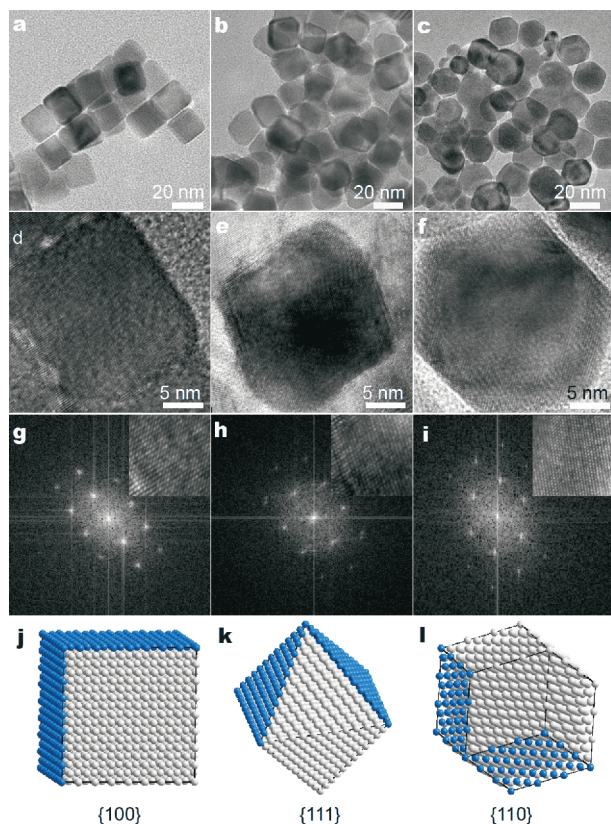


Figure 1 TEM and HRTEM images of $\text{PdH}_{0.43}$ NCs with (a, d) cubic, (b, e) octahedral and (c, f) RD shapes. (g–i) The corresponding Fourier transition images, the inset is the enlarged lattice fringes in (d–f). (j–l) The typical schematic models of cube, octahedron and RD of metallic fcc structures.

spectively. The sizes of the three types of $\text{PdH}_{0.43}$ NCs are 21.1, 26.5 and 28.4 nm, respectively (Fig. S4). XRD patterns of the three synthesized $\text{PdH}_{0.43}$ NCs are shown in Fig. 2, the three patterns are similar, and all the diffraction peaks shift to low 2θ compared with Pd face centered cubic (fcc) structure with lattice parameter of 3.890 \AA (JCPDS No. 46-1043). The negative shifts indicate the expansion of the Pd lattice, and the cell constant of the as-prepared three $\text{PdH}_{0.43}$ NCs is determined to be ca. $3.996(1) \text{ \AA}$. It is about 2.7% larger than that of Pd NCs, suggesting incorporation of hydrogen in Pd lattice structure, that is, the successful formation of palladium hydrides in the synthetic condition. Based on the relationship between lattice parameter and composition within palladium hydride, the H/Pd ratio in the three types of as-prepared products is estimated to be 0.43, and thus the palladium hydrogen compounds are denoted as $\text{PdH}_{0.43}$. The H-content (0.43) in the three as-prepared palladium hydrides NCs with different surface structures is smaller than the minimum β -phase content (0.58) for the bulk samples, due to the nano-scaled effect [27].

XPS was conducted to study the chemical state and valence band structure of Pd element in $\text{PdH}_{0.43}$ NCs and the precursor Pd NCs. The high resolution XPS Pd 3d spectra show the presence of Pd ($3d_{3/2}$) and Pd ($3d_{5/2}$) peaks (Fig. 3a–c). The binding energy of these Pd NCs (335.25 eV) is very close to the value generally reported for metallic Pd in the literature [28], whereas the binding energy of Pd ($3d_{5/2}$) in $\text{PdH}_{0.43}$ NCs is 0.42 eV higher than that of Pd NCs, suggesting the adsorbate-induced surface core-level shift and the contribution from the formation of PdH_x . The valence band structures of the samples are

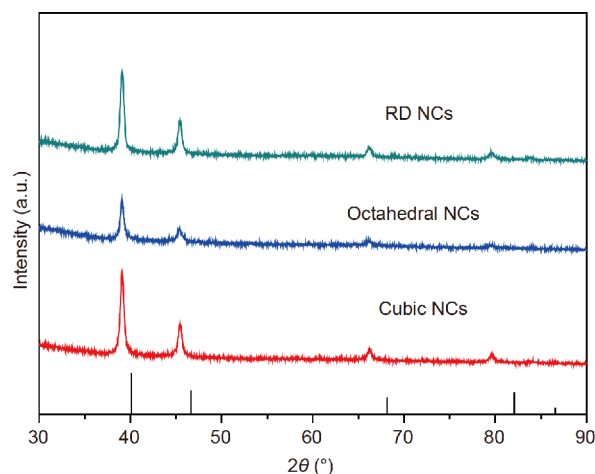


Figure 2 XRD patterns of $\text{PdH}_{0.43}$ with different morphologies. The black rod lines represent the standard Pd fcc structure (JCPDS No. 46-1043).

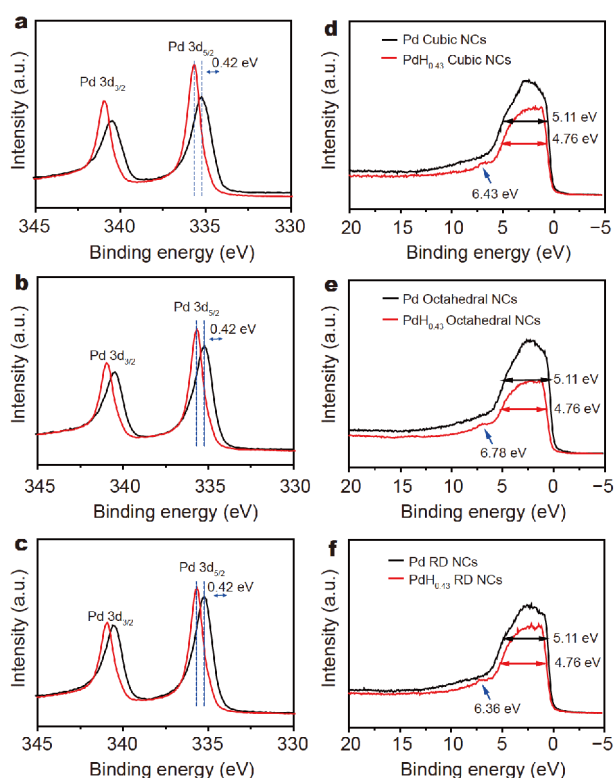


Figure 3 (a–c) XPS Pd 3d spectra and (d–f) XPS valence band structures of PdH_{0.43} NCs (red curve) and Pd NCs (black curve) with different morphologies.

shown in Fig. 3d–f. Compared with Pd NCs, the full width at half maximum are reduced from ca. 5.11 eV in Pd NCs to 4.76 eV in PdH_{0.43} NCs. The narrowed band width of 0.35 eV is caused by the larger Pd–Pd distance in the hydride [21]. At the same time, the small emission peaks were also observed at 6.43, 6.78 and 6.36 eV in the valence bands for cubic, octahedral and RD PdH_{0.43} NCs, respectively, which may arise from the strong interaction between the Pd 4d and H 1s due to the formation of PdH_x [29]. All of these XPS evidences indicate that the precursor Pd NCs are completely converted to PdH_{0.43} NCs, in good agreement with XRD analyses.

Our previous studies demonstrated the primary amine can release hydrogen along with the formation of corresponding secondary amine and imine in the presence of Pd catalyst [21]. We tried a series of amine to produce PdH_x NCs from Pd cubic NCs. Besides *n*-butylamine, *n*-hexylamine and ethanediamine can easily release hydrogen and lead to the formation of PdH_{0.43} NCs, no PdH_x product was detected in the presence of octadecylamine and dodecylamine with relative longer alkyl chains (Fig. S5). Meanwhile, we got the mixture of Pd and

PdH_{0.43} NCs in the presence of oleylamine and phenylamine in this synthesis environment (Fig. S5). Therefore the selection of primary amine is crucial for the formation of high purity PdH_{0.43} NCs. It is worth noting that when we tried *n*-butylamine treatment on three precursors of Pd NCs at ambient condition for three days, a subtle variation was observed. The shoulder peaks of Pd 3d in XPS spectra (Fig. S6a–c) are obviously different from the single peaks of Pd NCs or PdH_{0.43} NCs in Fig. 3a–c. The shoulder peaks include adsorbate-induced surface core-level shift components, indicating distinct hydrogen incorporation. However, the XRD patterns (Fig. S6d) match well with that of standard Pd fcc structure, indicating H atoms are merely adsorbed on the surface of Pd NCs. We thus propose the formation of PdH_x NCs could divide into two steps, that is, the released H₂ is adsorbed on the surface of palladium at first, and then the dissociated hydrogen atoms intercalate into the crystal lattice of palladium.

We further conducted the time-dependent experiments in the formation of three types of PdH_{0.43} NCs. From the time tracking XRD patterns (Fig. 4a–c), we found the cubic, octahedral and RD PdH_{0.43} NCs were thoroughly converted after 0.5, 4 and 6 h of reaction, respectively, indicating different forming rates for PdH_{0.43} NCs with different morphologies. The forming rate of PdH_{0.43} is consistent with the hydrogen absorption rate of Pd NCs, in which the order is Pd cube > Pd octahedron > Pd RD. Strikingly, those PdH_{0.43} NCs are quite stable, their β -phase structure and H content of 0.43 are well preserved even after being kept at ambient condition for one year (their XRD patterns are shown in Fig. S7). The stability of three types of as-prepared PdH_{0.43} NCs was further tested by annealing the samples in an argon atmosphere at 300°C for 2 h. As shown in Fig. 4d, after heat treatment, the cubic PdH_{0.43} NCs maintain unchanged, RD PdH_{0.43} NCs become Pd NCs thoroughly, while octahedral PdH_{0.43} NCs turn out to be the mixture of Pd and PdH_x NCs. Among the three types of PdH_{0.43} NCs, the heat stability obeys the order of PdH_{0.43} {100} > PdH_{0.43} {111} > PdH_{0.43} {110}. The time-dependent and the heat-stability experiments reveal the difference regarding the hydrogen adsorption of Pd NCs and desorption of PdH_{0.43} NCs with three low-index facets. A recent exploration also found two low-index facets ({100} and {111}) yielded significantly different desorption rates of hydrogen, where the desorption was thought to be governed by the surface-limited recombination of hydrogen atoms to form H₂ [30].

The successful syntheses of cubic, octahedral and RD

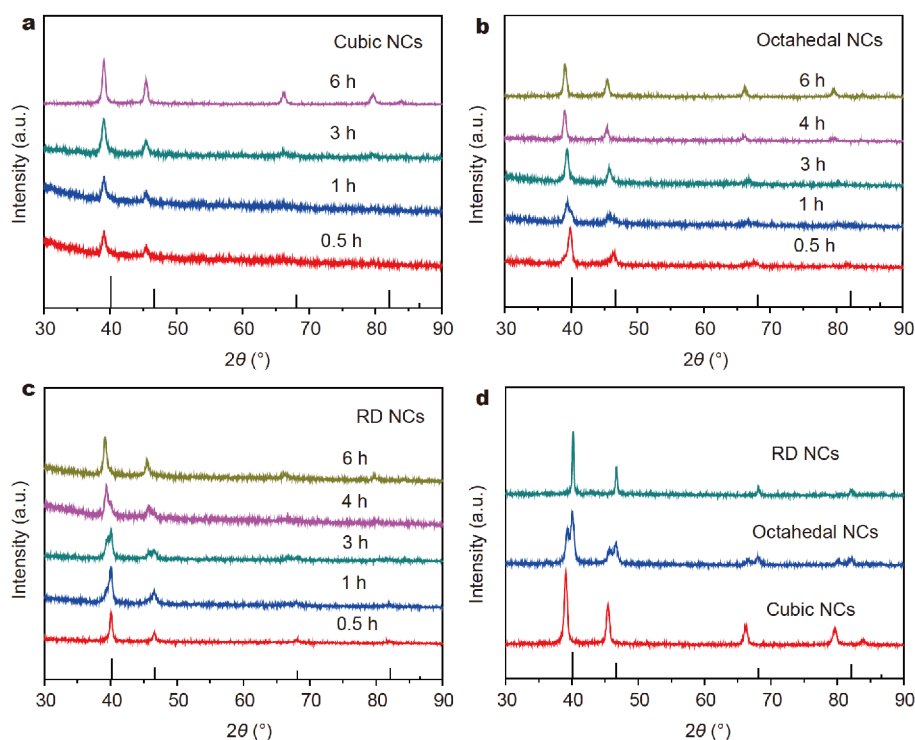


Figure 4 (a–c) Time tracking of XRD patterns during the synthesis of PdH_{0.43} with reaction times of 0.5, 1, 3, 4 and 6 h. (d) XRD patterns of PdH_{0.43} NCs after annealing under Ar atmosphere at 300°C for 2 h. The black rod lines represent the standard Pd fcc structure (JCPDS No. 46-1043).

PdH_{0.43} NCs with similar size but different surface structures ($\{100\}$, $\{111\}$ and $\{110\}$) provide us a good opportunity to study the structure-activity relationship among them. Here, we chose the formic acid electro-oxidation reaction as a probe reaction, three untreated Pd NCs and commercial Pd black as the references. CV measurements were employed to examine the electro-catalytic activity of different catalysts. CV curves recorded in 0.5 mol L⁻¹ H₂SO₄ solution exhibit distinct differences for PdH_{0.43} and Pd NCs (Figs S8, S9). The peak currents of H desorption of the PdH_{0.43} NCs are dozens of times higher than that of the Pd NCs, which can be attributed to the formation of β -phase PdH_x [20]. Fig. 5a–c show CV curves for PdH_{0.43} and Pd NCs in 0.5 mol L⁻¹ H₂SO₄ + 0.25 mol L⁻¹ HCOOH solution (the specific catalytic activities and peak potentials of the catalysts are also summarized in Table S1). It can be found the peak current at positive scan on the cubic, octahedral and RD PdH_{0.43} NCs is 11.1, 9.6, 7.8 mA cm⁻², while the value of Pd NCs with the same morphology is 5.2, 4.2 and 3.8 mA cm⁻², respectively. The peak current for commercial Pd black is 2.1 mA cm⁻². Thus, the specific activity of the cubic PdH_{0.43} NCs is more than 5 times that of the commercial Pd black and more than 2 times that of the cubic Pd NCs.

In fact, all three PdH_{0.43} samples exhibit enhanced specific activity compared with Pd NCs and the Pd black, suggesting the enhanced catalytic performance towards formic acid oxidation of palladium hydride. It should be noted that the specific activity order of both PdH_{0.43} and Pd NCs with three basal facets is followed by $\{100\} > \{111\} > \{100\}$. For a better comparison, we also compared the current density of different nanocatalysts at a fixed potential of 0.16 V (Table S1). Three PdH_{0.43} nanocatalysts still exhibit the enhanced catalytic activity than those of the Pd nanocatalysts. Strikingly, besides the enhanced peak current density, the oxidation peak potentials of formic acid on the three types of PdH_{0.43} NCs shift to much low potential. For instance, the peak potential of the cubic PdH_{0.43} NCs is as low as 54 mV, which is much lower than that of the cubic Pd NCs (245 mV) and the commercial Pd black (316 mV), respectively. To largely decrease overpotential of formic acid oxidation reaction is a remarkable improvement for Pd-based NCs. Meanwhile, the peak potential for octahedral and RD PdH_{0.43} are 61 and 63 mV, respectively. The similarity in decreasing peak potential on PdH_{0.43} NCs with different basal facets implies it is the incorporation of hydrogen plays a vital role in affecting the oxidation potential sig-

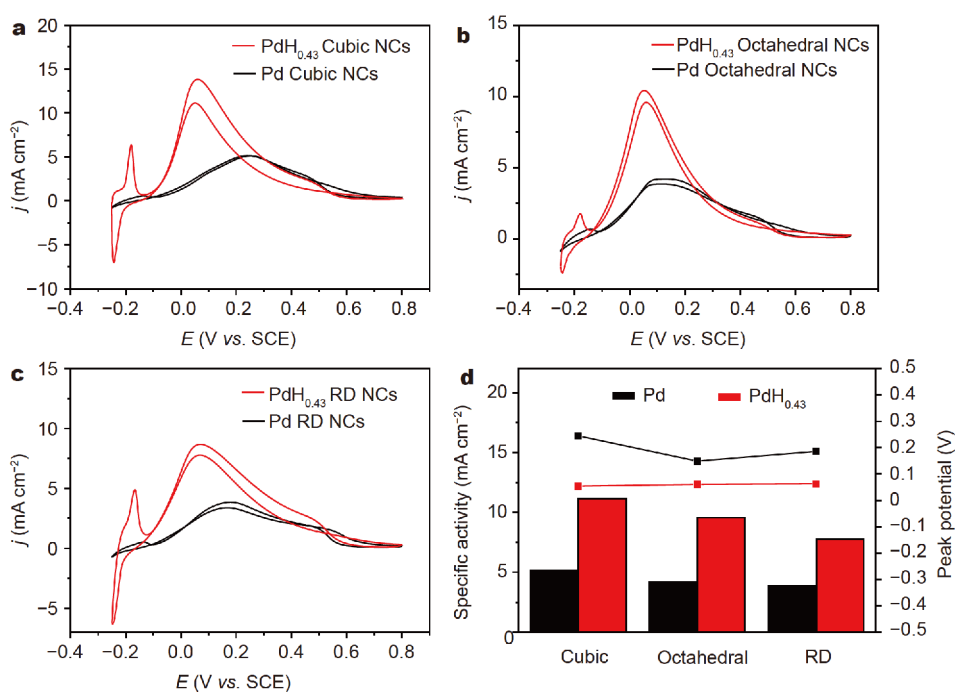


Figure 5 (a–c) CVs measured on PdH_{0.43} and Pd NCs with different morphologies in 0.5 mol L⁻¹ H₂SO₄ + 0.25 mol L⁻¹ HCOOH (scan rate: 50 mV s⁻¹). (d) Comparison of specific activity (column) and peak potential (line) towards formic acid oxidation on PdH_{0.43} and Pd NCs.

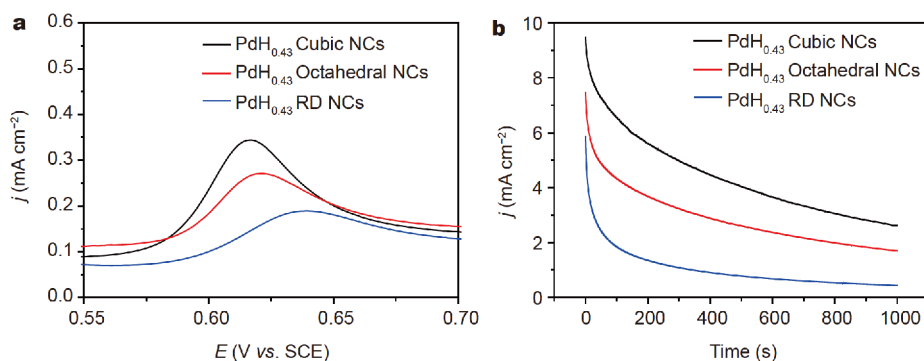


Figure 6 (a) CO-stripping curves in 0.5 mol L⁻¹ H₂SO₄ (scan rate: 20 mV s⁻¹). (b) $i-t$ curves for formic acid oxidation at 0 V (vs. SCE, at 298 K) on three PdH_{0.43} NCs with different surface structures.

nificantly, but not the surface structure. Combining the experimental data and our previous studies, the enhanced catalytic performance could be attributed to the distinct electronic structure of PdH_x. The interaction of H with Pd would lead to the downshift of the Pd d-band center [21,31–35], which weakens the bonding interaction between noble metal atoms and reaction intermediates and thereby lows the onset oxidation potential. We further conducted CO electro-oxidation experiment to check the CO-tolerance property of the three as-prepared PdH_{0.43} NCs (Fig. S10). As shown in Fig. 6a, The CO-stripping curves reveal the onset potentials of CO electro-oxidation

on different PdH_{0.43} NCs have a gradual upward trend with the order as cubic NCs < octahedral NCs < RD NCs. It means the PdH_{0.43} NCs with {100} facets have the best CO-tolerance property among the three PdH_x NCs. In addition, the CO onset potential of PdH_{0.43} NCs is lower than that of Pd NCs (Table S1), which proves their CO-tolerance property is improved. The catalytic durability of the three PdH_{0.43} NCs was tested by chronoamperometric technique. The current-time ($i-t$) curves in Fig. 6b show cubic PdH_{0.43} NCs keep the highest catalytic performance among the three NCs over the test. At the same time, the three PdH_{0.43} NCs were collected and tested by powder

XRD after the electrochemical test (Fig. S11). By comparing the diffraction peaks of samples before and after electrochemical test, one can find that the phase structure and the composition of PdH_{0.43} is unchanged, confirming good electrocatalytic stability of the three PdH_{0.43} NCs toward the formic acid oxidation at working potential from -0.25 to 0.9 V.

CONCLUSIONS

In summary, we have successfully prepared three types of PdH_{0.43} NCs with different morphologies, which are similar in size and enclosed with the facets of {100}, {111} and {110}, respectively. The electrochemical characterizations exhibit the enhanced catalytic performance when applying those PdH_{0.43} NCs to the anode reaction of formic acid fuel cell compared with Pd NCs and commercial Pd black. The enhanced peak current density and negative shift of peak potential suggest that the as-prepared PdH_{0.43} NCs have great potential in increasing both the current density and working voltage of fuel cells. Furthermore, the catalytic activity sequence for palladium hydride NCs with different surface structures follows the order of PdH_{0.43} {100} > PdH_{0.43} {111} > PdH_{0.43} {110}. Most importantly, the present approach for the preparation of PdH_x with well-defined surface structure is facile and cost-effective, and therefore it is suitable for rational design of Pd-based nanocatalysts with promising performance.

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Conflict of interest The authors declare no conflict of interest.

Supplementary information Experimental details and supporting data are available in the online version of the paper.



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不同表面结构PdH_{0.43}纳米晶的制备及其甲酸电氧化性能

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摘要 制备具有确定形貌和表面结构的纳米晶为研究其构效关系提供了有效途径, 且有利于设计开发具有优异催化性能的纳米催化剂. 本研究发展了一种制备不同形貌PdH_{0.43}纳米晶的简易方法, 其形貌分别为立方体、八面体和菱形十二面体, 对应的裸露晶面分别为{100}, {111}和{110}晶面. 不同形貌PdH_{0.43}纳米晶都非常稳定. 其与商业Pd黑和三种纯Pd纳米晶相比, 在甲酸电催化氧化反应中显示出更高的催化活性和极低的氧化过电位. 立方体PdH_{0.43}纳米晶的催化活性分别高于商业Pd黑的5倍和立方体Pd纳米晶的2倍. 不同表面结构PdH_{0.43}纳米晶的催化活性依次为PdH_{0.43} {100} > PdH_{0.43} {111} > PdH_{0.43} {110}.