# Fabrication of Efficient Organic-Inorganic Perovskite Solar Cells in Ambient Air 

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#### Abstract

Although many groups have been trying to prepare perovskite solar cells (PSCs) in ambient air, the power conversion efficiency (PCE) is still low. Besides, the effect of moisture on the formation of perovskite films is still controversial. In this paper, we studied the effect of moisture on the formation of perovskite films in detail, and found that moisture can speed up the crystallizing process of $\mathrm{Pbl}_{2}$ films to form poor-quality films with large grain size and surface roughness, while, for the conversion of $\mathrm{Pbl}_{2}$ to perovskite films, a small amount of moisture is not adverse, and even beneficial. On this basis, we report the successful fabrication of efficient mesoporous PSCs with PCE of $16.00 \%$ under ambient air conditions at $25 \%$ relative humidity by adding a small amount of n-butyl amine into the solution of $\mathrm{Pbl}_{2}$ to enhance the quality of $\mathrm{Pbl}_{2}$ films and thus to achieve high-quality perovskite films with smooth surface, large crystal grains, and high crystal quality.


Keywords: Ambient air, Moisture, N-butyl amine, Perovskite solar cells
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## Background

Organic-inorganic perovskite solar cells (PSCs) have become a pacemaker in the photovoltaic community with a rapid increase in the power conversion efficiency (PCE) from initial $3.8 \%$ in 2009 to a recently reported $22.7 \%$ [1-3], due to the high absorption coefficient, low exciton-binding energy, long charge carrier diffusion length, and high mobility of the organic-inorganic perovskite materials [4-12]. Unfortunately, the organic-inorganic perovskite materials are very sensitive to moisture due to the hygroscopic nature of the organic components [13], so the fabrication and long-term stability of PSCs in ambient air has been considered as one of the major challenges for future large-scale application. Interface engineering and encapsulation technology are widely used to improve the stability of PSCs in ambient air, which gains obvious effect [14, 15]. To avoid moisture in ambient air during the fabrication process of PSCs, most groups prepare PSCs inside a $\mathrm{N}_{2}$-filled glove box. There are also several research groups, who found that when perovskite materials were prepared in glovebox under inert atmosphere, the resulting perovskite films remained poorly crystalline, but, once exposed to a

[^0]special controlled humidity atmosphere, a rapid crystallization into highly oriented crystallites was observed [16-19]. However, for future mass production, it is the best choice to fabricate highly efficient PSCs with a facile and simple way in ambient air, neither glove box nor a special controlled humidity atmosphere.

Recently, strategies have been achieved to develop the air-processed PSCs, and they can be generally divided into two methods: (i) finding unique and simple manufacturing processes to achieve high-quality perovskite films in ambient air; (ii) exploring new perovskite materials with fundamentally good air stability. For the first method, preheating substrates has been employed as a simple and effective manufacturing process to fabricate PSCs in ambient air. One group reported a highest PCE of $7.9 \%$ by preheating substrates at $200{ }^{\circ} \mathrm{C}$ before the one-step spin-coating of perovskite films in ambient air [20]. For the second method, $\mathrm{CsPbBr}_{3}$-based inorganic PSCs have been fabricated in ambient air which exhibits the highest PCE of $7.78 \%$ [21]. Besides, Tai et al. reported a type of efficient and stable perovskite solar cells prepared in ambient air by using lead (II) thiocyanate precursor instead of $\mathrm{PbI}_{2}$. Upon optimization, the devices demonstrated average PCEs over $13 \%$, together
with the maximum value of $15 \%$ [22]. However, the authors did not study the role moisture plays on the crystallizing process of $\mathrm{PbI}_{2}$ films and conversion of $\mathrm{PbI}_{2}$ to perovskite films in detail. Although many groups have been trying to prepare PSCs in ambient air, the PCE is still low, seldom reaches $16 \%$ as reported [13]. Besides, the effect of moisture on the formation of perovskite films is still controversial.
In this work, we report the successful fabrication of efficient mesoporous PSCs with PCE of $16.00 \%$ under ambient air conditions at $25 \%$ relative humidity (RH) by adding a small amount of n-butyl amine (BTA) into the solution of $\mathrm{PbI}_{2}$ to enhance the quality of $\mathrm{PbI}_{2}$ films and thus to achieve high-quality perovskite films with smooth surface, large crystal grains, and high crystal quality. Besides, to study the effect of moisture on the formation of perovskite films, the performance of mesoporous PSCs prepared under different RHs without BTA additive has been investigated in detail. Combining SEM and XRD characterizations of $\mathrm{PbI}_{2}$ and perovskite films with and without BTA additive, it is clear that moisture can speed up the crystallizing process of $\mathrm{PbI}_{2}$ films to form poor-quality films with large grain size and surface roughness, while, for the conversion of $\mathrm{PbI}_{2}$ to perovskite films, a small amount of moisture is not adverse, and even beneficial.

## Methods

## Fabrication of Perovskite Solar Cells

A mesoporous device structure was adopted for the device fabrication, as shown in Fig. 1a. Fluorine-doped transparent conducting $\mathrm{SnO}_{2}$-coated glass substrates (FTO) with a sheet resistance of $7 \Omega \mathrm{sq}^{-1}$ were cleaned with acetone, ethanol, isopropanol, deionized water, and isopropanol respectively. A compact $\mathrm{TiO}_{2}\left(\mathrm{c}-\mathrm{TiO}_{2}\right)$ layer was deposited on the FTO substrates by spin coating at 3000 rpm for 30 s (repeat twice, followed by annealing at $150{ }^{\circ} \mathrm{C}$ for 15 min for each time), and then the $\mathrm{c}-\mathrm{TiO}_{2}$ layer was annealed at $500^{\circ} \mathrm{C}$ for 30 min in air. After cooling to the
room temperature, a mesoporous $\mathrm{TiO}_{2}\left(\mathrm{mp}-\mathrm{TiO}_{2}\right)$ layer was deposited by spin-coating at 5000 rpm for 45 s using a $\mathrm{TiO}_{2}$ paste (18NRD) diluted in EtOH (1:7, weight ratio). After drying at $80^{\circ} \mathrm{C}$ for 40 min , the $\mathrm{mp}-\mathrm{TiO}_{2}$ layer was sintered at $500{ }^{\circ} \mathrm{C}$ for 30 min . Once cooling to the room temperature, the film was dipped in the aqueous solution of $\mathrm{TiCl}_{4}$ for 30 min at $70^{\circ} \mathrm{C}$, rinsed with deionized water, and finally annealed at $500{ }^{\circ} \mathrm{C}$ for 30 min . After that, perovskite films were prepared by the two-step spin-coating method as follows. Firstly, 1 M of $\mathrm{PbI}_{2}$ in N,N-dimethylformamide (DMF) (adding a small amount of BTA into the solution) was spin coated onto the $\mathrm{mp}-\mathrm{TiO}_{2}$ layer at 3000 rpm for 30 s , and then annealed at $70^{\circ} \mathrm{C}$ for 15 min . After the $\mathrm{PbI}_{2}$ films cooled down to room temperature, the solution of methylammonium (MA) iodide was spin coated onto $\mathrm{PbI}_{2}$ films at 4000 rpm for 45 s . Finally, the samples were annealed at $100^{\circ} \mathrm{C}$ for 30 min to grow into $\mathrm{MAPbI}_{3}$ films. Cooling down to room temperature, the 2,2',7,7'-Tetrakis[N,N-di(4-methoxyphenyl)amino]-9,9'-s-piro-bifluorene (Spiro-OMeTAD) layer was spin-coated at 2000 rpm for 45 s , where 80 mg Spiro-OMeTAD in 1 mL chlorobenzene solution was employed with the addition of $28.8 \mu \mathrm{~L}$ 4-tert-butylpyridine (TBP) and $17.7 \mu \mathrm{~L}$ lithium bis(trifluoromethanesulfonyl)imide (Li-TFSI) solution ( 520 mg Li-TFSI in 1 mL acetonitrile). Finally, Ag back electrode was deposited by thermal evaporation. The active area of the device was $0.1 \mathrm{~cm}^{2}$.
During the fabrication process, four solar cells were fabricated in each FTO substrate. Among which, if the maximum PCE deviation is less than $3 \%$ in at least three solar cells with higher PCE values, then their performance parameters will be recorded.

## Characterization

Current density-voltage ( $\mathrm{J}-\mathrm{V}$ ) characteristic curves of photovoltaic cells were measured by a source meter (Keithley, 2400) with a solar simulator (Zolixss150) under $100 \mathrm{~mW} \mathrm{~cm}{ }^{-2}$ AM 1.5 G illumination; the light intensity was calibrated by means of a silicon reference
(a)

(b)


Fig. 1 a A schematic diagram of the mesoporous PSCs. b The SEM cross-section image of the device, with the structure $\mathrm{FTO} / \mathrm{c}-\mathrm{TiO} / \mathrm{O}_{2} / \mathrm{mp}-\mathrm{TiO}{ }_{2} /$ $\mathrm{MAPbl}_{3} /$ Spiro-OMeTAD/Ag


Fig. 2 Box Charts of $\mathbf{a} V_{O C}, \mathbf{b} J_{S C}, \mathbf{c}$ FF, and $\mathbf{d}$ PCE for mesoporous PSCs prepared under different RHs at $30^{\circ} \mathrm{C}$ without the adding of BTA during the two-step spin-coating of perovskite films


Fig. 3 SEM images of $\mathrm{Pbl}_{2}$ films on $\mathrm{FTO} / \mathrm{c}-\mathrm{TiO}_{2} / \mathrm{mp}-\mathrm{TiO}_{2}$ substrates without (a) and with (b) BTA additive; and corresponding $\mathrm{MAPbl} \mathrm{A}_{3}$ films without (c) and with (d) BTA additive prepared under 25\% RH
solar cell. The active area of the devices was $0.1 \mathrm{~cm}^{2}$. Typical J-V curves were obtained from scanning in reverse bias direction at the step width of 200 mV . The voltage was scanned from 1.2 to -0.2 V at a rate of $100 \mathrm{mV} \mathrm{s}^{-1}$. The J-V measurement was carried out in ambient air. An emission Hitachi S-4800 was used to obtain scanning electron microscopy (SEM) images with the electron beam acceleration in the range of 15 to 60 KV . The characteristic X-ray diffraction (XRD) patterns were recorded between $10^{\circ}$ and $70^{\circ}$, using the $\mathrm{Cu}-\mathrm{K} \alpha$ radiation at $1.5405 \AA$. The light absorbance spectra were required by using Cary 5000 UV-Vis spectrophotometer in the wavelength range of 200 to 1200 nm with 1 nm increment. All the measurements of these films were carried out in ambient air without humidity control.

## Results and Discussion

To study the effect of moisture on the formation of perovskite films, two-step spin-coating experiments under different RHs at $30^{\circ} \mathrm{C}$ without the adding of BTA were designed, and the corresponding statistic results of detailed photovoltaic parameters are shown in Fig. 2. With the increase of RH from 0 to $15 \%$, all of the photovoltaic parameters, including open-circuit voltage ( $\mathrm{V}_{\mathrm{OC}}$ ), short-circuit current density ( $\mathrm{J}_{\mathrm{SC}}$ ), fill factor (FF), and PCE, are improved obviously. As reported, a small amount of moisture could promote ion diffusion in the precursor film, facilitate the perovskite crystal growth, and thus induce a rapid crystallization into highly oriented crystallites [13, 23]. Therefore, better performance of PSCs under $15 \%$ RH was observed, compared with the PSCs fabricated in the glovebox ( $0 \% \mathrm{RH}$ ). Going on to increase the RH, the photovoltaic parameters, $\mathrm{V}_{\mathrm{OC}}, \mathrm{J}_{\mathrm{SC}}, \mathrm{FF}$, and PCE, begin to drop sharply, as shown in Fig. 2. When the RH rises to $45 \%$, the average value of $\mathrm{V}_{\mathrm{OC}}, \mathrm{J}_{\mathrm{SC}}, \mathrm{FF}$, and PCE drops to $1.00 \mathrm{~V}, 9.84 \mathrm{~mA} / \mathrm{cm}^{2}, 51.02 \%$, and $5.02 \%$, respectively. The drastic decline of PCE under $45 \%$ RH is mainly caused by the sharp decrease in $\mathrm{J}_{\mathrm{SC}}$. It was reported that too much moisture could cause poor surface morphology and even decomposition of perovskite films, so the JSC of PSCs dropped sharply under $45 \%$ RH [18]. According to the results above, the optimal humidity conditions for two-step spin-coating of perovskite films in ambient air without the adding of BTA is $15 \% \mathrm{RH}$, and the corresponding highest PCE is $13.21 \%$ (average PCE is $12.48 \%$ ), which is too low to meet future mass production. Besides, the results above are still insufficient to explain the role that moisture plays on the formation of perovskite films during two-step spin-coating.
To improve the performance of the PSCs prepared in ambient air and further investigate the role that moisture plays on the formation of perovskite films, a small amount of BTA was added into the solution of $\mathrm{PbI}_{2}$. BTA, who has strong volatility, well infiltrative property,
and strong Lewis base nature [13], will isolate the $\mathrm{PbI}_{2}$ films from part of the moisture in ambient air, help the $\mathrm{PbI}_{2}$ solution to spread out on the substrates easily and homogeneously, and greatly slow down the crystallization rate to form high-quality $\mathrm{PbI}_{2}$ films.

As known, the morphology, such as grain sizes, surface roughness, and pinholes, of perovskite films plays an important role in the performance final PSCs. For the typical two-step spin-coating process of perovskite films, controlling the morphology of $\mathrm{PbI}_{2}$ films is a key strategy for controlling the morphology of perovskite films [13, 19, 24]. However, it is disappointing to prepare high-quality $\mathrm{PbI}_{2}$ films in ambient air with $25 \% \mathrm{RH}$, as illustrated by the SEM image shown in Fig. 3a, which exhibits inhomogeneous and porous structure with large grain size and surface roughness. The poor quality of the $\mathrm{PbI}_{2}$ films under $25 \% \mathrm{RH}$ can be mainly due to the moisture-induced rapid crystallization of $\mathrm{PbI}_{2}$ films. After the addition of a small amount of BTA into the $\mathrm{PbI}_{2}$ solution, a full coverage,


Fig. $4 \times R D$ spectra of $\mathrm{Pbl}_{2}$ films (a) and $\mathrm{MAPbl}_{3}$ films (b) on quartz substrates prepared with and without BTA additive, in ambient air under $25 \% \mathrm{RH}$. The inset of (a) is the rocking curve of the main diffraction peak of $\mathrm{Pbl}_{2}$ films at $12.69^{\circ}$
continuous, and homogeneous $\mathrm{PbI}_{2}$ film with small grain size and low surface roughness is obtained, as presented in Fig. 3b. The high-quality $\mathrm{PbI}_{2}$ films can be attributed to the strong volatility, well infiltrative property, and strong Lewis base nature of BTA, who will isolate the $\mathrm{PbI}_{2}$ films from part of the moisture in the ambient air, help the $\mathrm{PbI}_{2}$ solution to spread out on the substrates homogeneously, and greatly slow down the crystallization rate to form high-quality $\mathrm{PbI}_{2}$ films in ambient air with $25 \% \mathrm{RH}$. As stated earlier, for the typical two-step spin-coating process of perovskite films, controlling the morphology of $\mathrm{PbI}_{2}$ films is a key strategy for controlling the morphology of perovskite films [13, 19, 24]. By virtue of the high-quality $\mathrm{PbI}_{2}$ films shown in Fig. 3b, high-quality $\mathrm{MAPbI}_{3}$ films composed of densely packed big crystal grains without any pinholes are prepared as shown in Fig. 3d, while inhomogeneous $\mathrm{MAPbI}_{3}$ films with small grain size and amounts of pinholes are obtained using the poor-quality $\mathrm{PbI}_{2}$ films as shown in Fig. 3c. In addition, the high-quality $\mathrm{MAPbI}_{3}$ films shown in Fig. 3d is converted from $\mathrm{PbI}_{2}$ in ambient air with $25 \% \mathrm{RH}$, indicating a small amount of moisture ( $25 \% \mathrm{RH}$ ) is not adverse, and even beneficial for the conversion of $\mathrm{PbI}_{2}$ to perovskite films.
The crystal quality of as-grown $\mathrm{PbI}_{2}$ and $\mathrm{MAPbI}_{3}$ films, prepared in ambient air with and without BTA additive, was characterized by the XRD measurement. Figure 4a, b shows the XRD patterns of $\mathrm{PbI}_{2}$ and $\mathrm{MAPbI}_{3}$ films respectively, and it can be seen that the $\mathrm{PbI}_{2}$ and $\mathrm{MAPbI}_{3}$ films with and without BTA additive exhibit almost the same crystalline phase. As shown in Fig. 4a, the XRD patterns of $\mathrm{PbI}_{2}$ films with and without BTA additive show the intense diffraction peak at $12.69^{\circ}$, corresponding to
the characteristic peak of $\mathrm{PbI}_{2}$. However, the peak at $12.69^{\circ}$ reduces significantly in the $\mathrm{PbI}_{2}$ film with BTA additive, which can be explained as follows. On the one hand, as mentioned above, BTA has well infiltrative property and can help the $\mathrm{PbI}_{2}$ solution to spread out on the substrates easily and homogeneously. On the other hand, grain size of the $\mathrm{PbI}_{2}$ film with BTA additive is much smaller than the $\mathrm{PbI}_{2}$ film without BTA additive, as evidenced by the SEM images in Fig. 3 and the increase in full width at half maximum (FWHM) with BTA additive shown inset of Fig. 4a. Figure 3b shows the XRD patterns of $\mathrm{MAPbI}_{3}$ films prepared with and without BTA additive. As can be seen, the diffraction peaks were present at $2 \theta$ values of $14.06^{\circ}, 20.00^{\circ}, 23.45^{\circ}, 28.42^{\circ}, 31.86^{\circ}, 40.59^{\circ}$, and $43.21^{\circ}$ corresponding to the reflection planes of (110), (112), (202), (220), (310), (224), and (404) of the tetragonal perovskite structure [25], respectively. Additionally, the characteristic peak of $\mathrm{PbI}_{2}$, at $12.69^{\circ}$, is also observed in both the two $\mathrm{MAPbI}_{3}$ films prepared with and without BTA additive. The films prepared under ambient conditions lead to the incomplete conversion of $\mathrm{PbI}_{2}$ to $\mathrm{MAPbI}_{3}$, due to the termination of nucleation and growth of the perovskite caused by the formation of relatively continuous capping layer on the surface [26]. It is reported that a little bit of $\mathrm{PbI}_{2}$ can improve the performance of PSCs by passivating the defects in perovskite films [19, 26]. Furthermore, the $\mathrm{MAPbI}_{3}$ film prepared without BTA additive shows much higher intensity for the peak at $12.69^{\circ}$, compared with the film prepared with BTA additive. This suggests that there are too much $\mathrm{PbI}_{2}$ residual in the $\mathrm{MAPbI}_{3}$ film prepared without BTA additive attributed to the poor-quality $\mathrm{PbI}_{2}$ film without


Fig. 5 UV-Vis absorption spectrum of $\mathrm{MAPbl}_{3}$ films on quartz substrates with and without BTA additive. Insets are cross-section SEM images of $\mathrm{MAPb}_{3}$ films prepared with (a) and without (b) BTA additive


Fig. $6 \mathrm{~J}-\mathrm{V}$ characteristics of the PSCs fabricated with and without BTA additive in ambient air with $25 \% \mathrm{RH}$ under AM 1.5 G one sun $\left(100 \mathrm{~mW}^{-2}\right)$ illumination, the inset is the detailed photovoltaic performance

BTA additive leading to the insufficient reaction between $\mathrm{PbI}_{2}$ and MAI.
Combining SEM and XRD results above, it is clear what role moisture plays in the two-step spin-coating of $\mathrm{MAPbI}_{3}$ films in ambient air. For the $\mathrm{PbI}_{2}$ films, moisture can speed up the crystallizing process to form poor-quality $\mathrm{PbI}_{2}$ films with large grain size and surface roughness. However, for the conversion of $\mathrm{PbI}_{2}$ to the $\mathrm{MAPbI}_{3}$ films, a small amount of moisture ( $25 \% \mathrm{RH}$ ) is not adverse, and even beneficial.
UV-Vis absorption spectrum of $\mathrm{MAPbI}_{3}$ films prepared with and without BTA additive is presented in Fig. 5. All the two samples show an absorbance at a threshold of about 780 nm in the overall visible region, indicating the formation of $\mathrm{MAPbI}_{3}$ crystallites [27]. As can be seen, the $\mathrm{MAPbI}_{3}$ films with BTA additive shows lower absorbance, which is attributed to its relatively smaller thickness compared with the films without BTA additive, as confirmed in the cross-section SEM images of $\mathrm{MAPbI}_{3}$ films (insets of Fig. 5). Besides, the weak absorption shoulder at about 510 nm , which appears in all the two spectra, is a characteristic feature of $\mathrm{PbI}_{2}$, implying the residual of $\mathrm{PbI}_{2}$ as confirmed by the XRD measurement.
The $\mathrm{MAPbI}_{3}$ films prepared with and without BTA additive was then used to construct PSCs with the structure of $\mathrm{FTO} / \mathrm{c}-\mathrm{TiO}_{2} / \mathrm{mp}-\mathrm{TiO}_{2} / \mathrm{MAPbI}_{3} /$ Spiro-OMeTAD/ Ag , and the corresponding $\mathrm{J}-\mathrm{V}$ characteristics of the devices under AM 1.5G one sun ( $100 \mathrm{~mW} \mathrm{~cm}{ }^{-2}$ ) illumination are shown in Fig. 6, the inset of which is the photovoltaic parameters. The highest PCE values in the records were adopted here for comparison. The device using perovskite films prepared without BTA additive at
$25 \%$ RH showed the highest PCE of $11.38 \%$, with $\mathrm{J}_{\mathrm{SC}}$ of $19.97 \mathrm{~mA} / \mathrm{cm}^{2}, \mathrm{~V}_{\mathrm{OC}}$ of 0.98 V , and FF of $58.15 \%$. When introducing BTA additive, the devices showed significant improvements in all of the four photovoltaic parameters. There into, the device using perovskite films prepared with BTA additive showed the highest PCE of $16.00 \%$, which is improved by $\sim 40 \%$ compared with the PSCs using perovskite films prepared without BTA additive, with $\mathrm{J}_{\mathrm{SC}}$ of $22.29 \mathrm{~mA} / \mathrm{cm}^{2}, \mathrm{~V}_{\mathrm{OC}}$ of 1.10 V , and FF of $65.25 \%$, which was attributed to the high-quality perovskite films with smooth surface, large crystal grains, and high crystal quality.

## Conclusions

In conclusion, we studied the effect of moisture on the formation of perovskite films in detail, and found that moisture can speed up the crystallizing process of $\mathrm{PbI}_{2}$ films to form poor-quality films with large grain size and surface roughness, while, for the conversion of $\mathrm{PbI}_{2}$ to $\mathrm{MAPbI}_{3}$ films, a small amount of moisture is not adverse, and even beneficial. On this basis, by adding a small amount of BTA into the solution of $\mathrm{PbI}_{2}$ to enhance the quality of $\mathrm{PbI}_{2}$ films and thus to achieve high-quality perovskite films with smooth surface, large crystal grains, and high crystal quality, we fabricated mesoporous PSCs with PCE of $16.00 \%$ under ambient air conditions at $25 \% \mathrm{RH}$. The results may pave a new way for fabricating efficient and reproducible PSCs under ambient air condition.

[^1]substrates; Jsc: Short-circuit current; J-V: Current density-voltage; Li-TFSI: Lithium bis(trifluoromethanesulfonyl)imide; $\mathrm{MA}:$ Methylammonium; $\mathrm{mp}^{-\mathrm{TiO}_{2}}$ : MesoporousTiO $\mathrm{M}_{2}$ PCE: Power conversion efficiency; PSCs: Perovskite solar cells; RH: Relative humidity; SEM: Scanning electron microscopy; Spiro-OMeTAD: 2,2',7,7'-
Tetrakis[N,N-di(4-methoxyphenyl)amino]-9,9'-spiro-bifluorene; TBP: 4-Tertbutylpyridine; $\mathrm{V}_{\text {Oc: }}$ : Open-circuit voltage; XRD: X-ray diffraction

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## Availability of Data and Materials

The datasets supporting the conclusions of this article are included within the article.

## Authors' Contributions

JW carried out the experiment and characterization, analyzed results, and participated in the draft of the manuscript. JJD designed the experiment, supervised the research, and revised the manuscript. SXC, HYH, JX, and HL offered the technique supports. All authors read and approved the final manuscript.

## Competing Interests

The authors declare that they have no competing interests

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[^1]:    Abbreviations
    BTA: N-butyl amine; $\mathrm{c}-\mathrm{TiO}_{2}$ : Compact $\mathrm{TiO}_{2}$; DMF: $\mathrm{N}, \mathrm{N}$-dimethylformamide; FF: Fill factor; FTO: Fluorine-doped transparent conducting $\mathrm{SnO}_{2}$-coated glass

