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# 3D Interconnected $V_6O_{13}$ Nanosheets Grown on Carbonized Textile via a Seed-Assisted Hydrothermal Process as High-Performance Flexible Cathodes for Lithium-Ion Batteries

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## Abstract

Three-dimensional (3D) free-standing nanostructured materials have been proven to be one of the most promising electrodes for energy storage due to their enhanced electrochemical performance. And they are also widely studied for the wearable energy storage systems. In this work, interconnected  $V_6O_{13}$  nanosheets were grown on the flexible carbonized textile (c-textile) via a seed-assisted hydrothermal method to form a 3D free-standing electrode for lithium-ion batteries (LIBs). The electrode exhibited a specific capacity of  $170 \text{ mA h g}^{-1}$  at a specific current of  $300 \text{ mA g}^{-1}$ . With carbon nanotube (CNT) coating, its specific capacities further increased 12–40% at the various current rates. It could retain a reversible capacity of  $130 \text{ mA h g}^{-1}$ , 74% of the initial capacity after 300 cycles at the specific current of  $300 \text{ mA g}^{-1}$ . It outperformed most of the mixed-valence vanadium oxides. The improved electrochemical performance was ascribed to the synergistic effect of the 3D nanostructure of  $V_6O_{13}$  for feasible  $\text{Li}^+$  diffusion and transport and highly conductive hierarchical conductive network formed by CNT and carbon fiber in c-textile.

**Keywords:** Interconnected  $V_6O_{13}$  nanosheets, Seed-assisted hydrothermal, Lithium-ion batteries

## Background

Vanadium oxides (e.g.,  $V_6O_{13}$ ,  $V_3O_7$ ,  $V_2O_5$ ) are cathode materials applicable for high-energy lithium-ion battery (LIB), due to their low cost, high specific capacities, and the abundance of vanadium element [1–6]. Among the oxides,  $V_6O_{13}$  has been considered as an excellent candidate of the cathode material [7–14]. Its theoretical capacity and energy density can reach  $417 \text{ mA h g}^{-1}$  and  $890 \text{ Wh kg}^{-1}$  when lithiated to the final product,  $\text{Li}_8V_6O_{13}$  [2, 8]. However,  $V_6O_{13}$  electrodes have suffered short cycle life and low rate capability for a long time because  $V_6O_{13}$ 's electronic conductivity decreases when lithiated while  $\text{Li}^+$

diffusion coefficients ( $10^{-8}$  to  $10^{-9} \text{ cm}^2 \text{ S}^{-1}$ ) are low [7, 9]. Constructing free-standing 3D nanostructures is an effective method to solve the above problems. 3D nanostructure can enhance ion/electron transport/diffusion while effectively avoids self-aggregation [15–20]. For example, Yu et al. synthesized 3D  $V_6O_{13}$  nanotextiles assembled from interconnected nanogrooves via a facile solution-redox-based self-assembly route with  $\text{MnO}_2$  template at room temperature. In a voltage range of 1–4 V,  $V_6O_{13}$  nanotextiles exhibited reversible capacities of 326 and  $134 \text{ mA h g}^{-1}$  at 20 and  $500 \text{ mA g}^{-1}$ , respectively, and a capacity retention of above 80% after 100 cycles at  $500 \text{ mA g}^{-1}$  [2]. Tong et al. fabricated  $V_6O_{13}$  cathode supported by a steel mesh with wrinkles by the similar route. The free-standing electrode with a loading amount of  $V_6O_{13}$  up to  $2.0 \text{ mg cm}^{-2}$  was obtained. At a current density of  $500 \text{ mA g}^{-1}$ , the  $V_6O_{13}$  electrode demonstrated an initial capacity of  $225 \text{ mA h g}^{-1}$  that deteriorated to around

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150 mA h g<sup>-1</sup> after 500 cycles [21]. However, above research involved the two-step electrodeposition and removal of MnO<sub>2</sub>. Direct growth of mixed-valence vanadium oxide nanostructure with good electrochemical property remains a great challenge [22]. Meanwhile, previous studies have not demonstrated V<sub>6</sub>O<sub>13</sub>-based flexible cathode, which has a potential use in the wearable devices.

Herein, we proposed a simple hydrothermal process to successfully grow interconnected V<sub>6</sub>O<sub>13</sub> nanosheets on the carbonized textile to fabricate a 3D free-standing electrode. It exhibited specific capacities of 161 and 105 mA h g<sup>-1</sup> at the specific currents of 300 and 1200 mA h g<sup>-1</sup>, respectively. With carbon nanotube (CNT) coating to further improve its conductivity, its specific capacities increased to 170 and 140 mA h g<sup>-1</sup>. Meanwhile, its cycling performance was also improved. It could retain 74% of initial capacity with CNT coating, compared with 50% retention without CNT coating after 400 cycles at 300 mA h g<sup>-1</sup>. The improvement on the electrochemical performance was mainly ascribed to the synergistic effect of the 3D nanostructure of V<sub>6</sub>O<sub>13</sub> and hierarchical conductive network.

## Methods

### Synthesis of c-textile

The commercially available bamboo cloth was soaked in a solution with 2.5 g NaF and 60 ml H<sub>2</sub>O for 1 h and dried for 5 h in 120 °C oven. The dried textile was carbonized at 800 °C in N<sub>2</sub> for 30 min to obtain c-textile.

### Growth of 3D V<sub>6</sub>O<sub>13</sub> Nanostructure on c-textile

3D V<sub>6</sub>O<sub>13</sub> nanostructure was grown on c-textile by a seed-assisted hydrothermal method. c-textile was slightly oxidized in the condensed nitric acid (80 wt%) for 30 min. V<sub>2</sub>O<sub>5</sub> powder (1 mg) was added to 5 ml deionized water and then ultrasonicated for 15 min to obtain a suspension. The oxidized c-textile was then immersed into the suspension for 2 h, dried, and heated at 300 °C for 10 min to grow the seed of vanadium oxide on c-textile. V<sub>2</sub>O<sub>5</sub> powder (16 mg) was added to 224 μl of 30 wt% H<sub>2</sub>O<sub>2</sub> and stirred for 10 min to obtain a brown solution. It was then diluted with additional 40 ml distilled water and stirred for 30 min. After the solution was transferred into a 25-ml stainless-steel autoclave, the oxidized c-textile was immersed into the solution. The autoclave was kept at 180 °C for 48 h, then the sample was washed with distilled water and alcohol and dried at 60 °C for 8 h to finally get the flexible 3D free-standing V<sub>6</sub>O<sub>13</sub> nanostructure supported with flexible c-textile. CNT was further coated on V<sub>6</sub>O<sub>13</sub> nanostructure by repeatedly dipping it into NMP suspension (0.5 mg/mL) of multi-walled CNT and drying to produce a V<sub>6</sub>O<sub>13</sub>/CNT composite electrode.

## Characterization of Materials

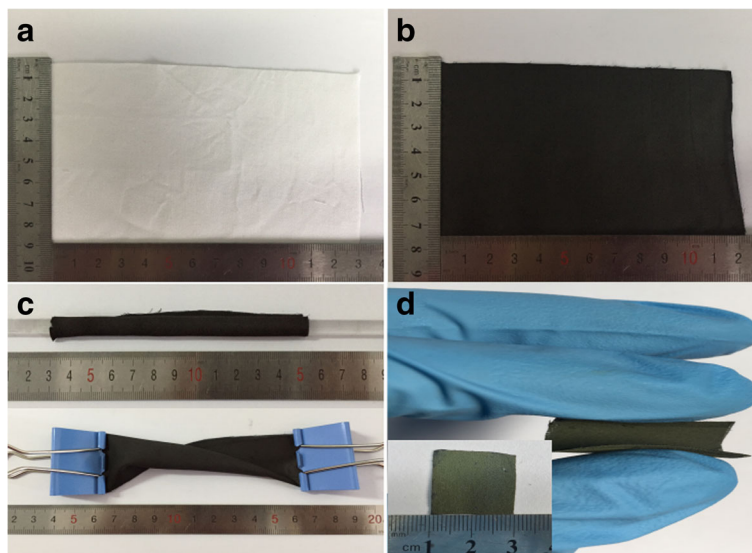
Morphology of the product was observed by a scanning electron microscopy (SEM, Philips XL30 FEG) and a transmission electron microscopy (TEM, JEOL JEM-2010). X-ray photoelectron spectroscopy (XPS) analyses (K-Alpha) were performed using a monochromatic Al K $\alpha$  source.

## Battery Fabrication and Electrochemical Measurements

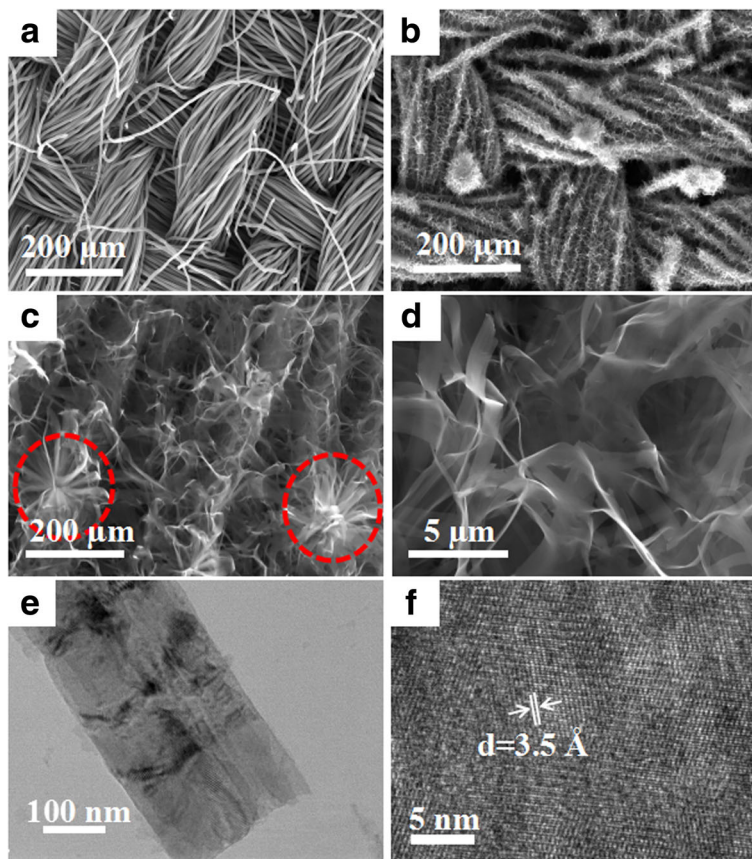
Standard CR2016-type coin cells were assembled in an argon-filled glove box (Vigor Inc. Suzhou, China) with V<sub>6</sub>O<sub>13</sub> electrode as the working electrode with a mass loading of ~1 mg cm<sup>-2</sup>. A lithium foil was used as the counter electrode; 1 mol LiPF<sub>6</sub> in a mixture of ethylene carbonate (EC), diethyl carbonate (DEC), and dimethyl carbonate (DMC) with a volume ratio of 1:1:1 was used as the electrolyte, and a polypropylene film was used as the separator. The assembled cells were electrochemically cycled between 1.5 and 4.0 V vs. Li/Li<sup>+</sup> for galvanostatic charge/discharge on a LAND battery test system (Wuhan Kingnuo Electronics Co., Ltd., China) at 25 °C. Electrochemical impedance spectroscopy (EIS) studies were conducted with Autolab PGSTAT302N workstation in the frequency range of 10 mHz to 10 kHz.

## Results and Discussion

The schematic of the growth of 3D V<sub>6</sub>O<sub>13</sub> interconnected nanosheets on c-textile was shown in Additional file 1: Figure S1. The textile (Fig. 1a) was firstly carbonized at 800 °C to obtain c-textile (Fig. 1b). SEM images (Fig. 2a) showed that c-textile was composed of weaved bundles of carbon fibers with a diameter of ~5 μm. c-textile exhibited excellent flexibility and mechanical strength. It was able to be rolled and twisted as shown in Fig. 1c. The square resistivity of the c-textile was measured to be 5 Ω/sq. with the four-probe method. Thus, it was used as a promising flexible support/collector for the electrode materials. It was then immersed in VO<sub>x</sub> suspension, dried, and kept at 300 °C for 10 min to grow the seed crystals. Its weight change was undetectable (<0.1 mg). After being immersed in vanadium oxide (VO<sub>x</sub>) sol solution for the hydrothermal growth, the black c-textile was covered with a layer of a yellow-green thin film; however, its flexibility was kept, as shown in Fig. 1d. Its resistivity increased to 50 Ω/sq. SEM images (Fig. 2b, c) further showed that it was composed of several micron-long and several hundred nanometer-wide interconnected nanosheets, as building blocks to construct 3D nanostructure on c-textile. High-resolution TEM image (Fig. 2f) showed well-defined lattice fringes of the grown nanosheets. The spacing of 3.5 Å in the lattice fringe was consistent with (110) interplanar distance of the orthogonal V<sub>6</sub>O<sub>13</sub> phase (PDF card No.71-2235) which was in agreement with the XRD



**Fig. 1** Optical images of **a** commercially available textile, **b** carbonized textile, **c** rolled and twisted c-textile, and **d** c-textile with grown  $V_6O_{13}$  at the rolled state, inset: at the flat state

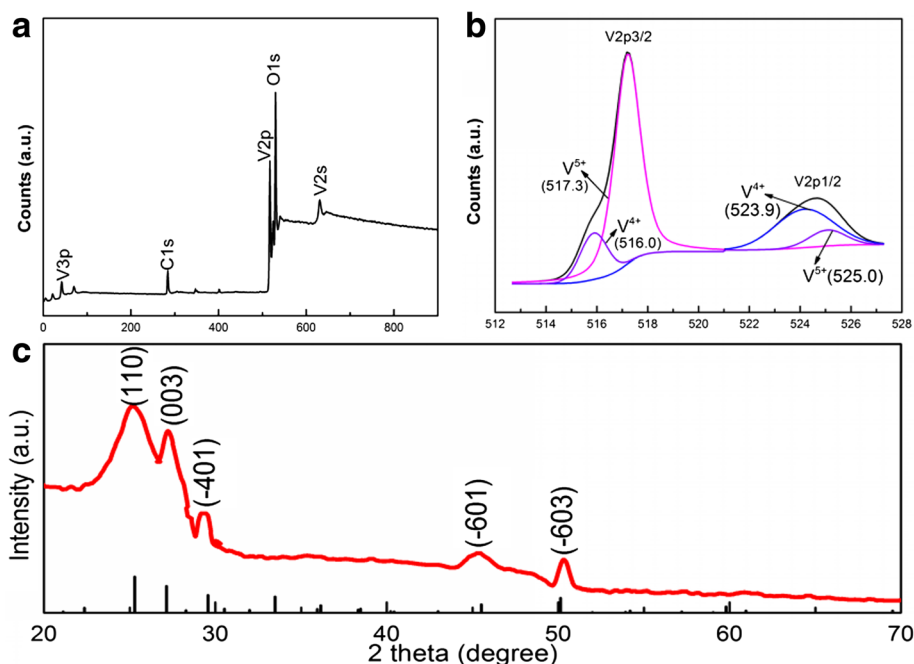


**Fig. 2** Microstructure of 3D free-standing interconnected  $V_6O_{13}$  nanosheets on c-textile: **a, b** low resolution SEM images of c-textile without and with nanosheets, respectively; **c, d** high-resolution SEM images of interconnected nanosheets grown on c-textile; **e, f** low- and high-resolution TEM images of the nanosheet, respectively

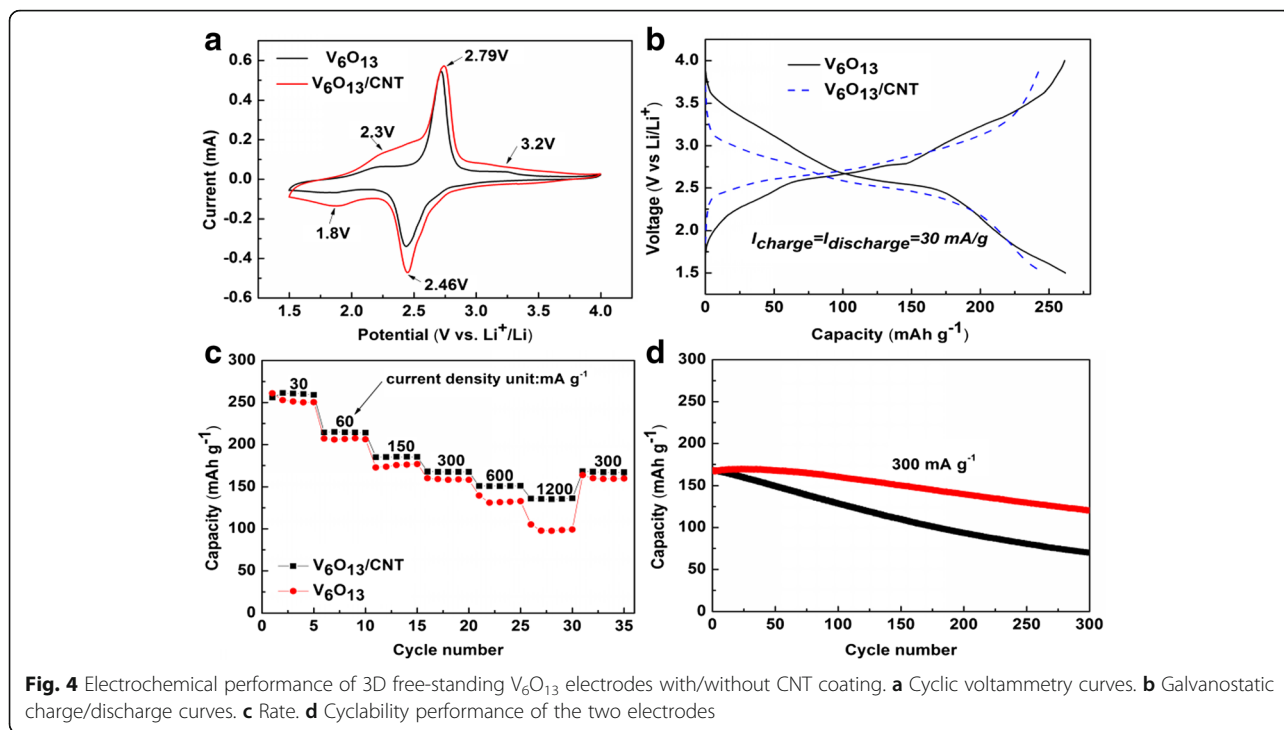
pattern (Fig. 3c). The growth mechanism was that seed crystal firstly nucleated on the sites with an oxygen-bearing functional group [23, 24]. Then during the hydrothermal process in the  $\text{VO}_x$  aqueous solution, interconnected  $\text{V}_6\text{O}_{13}$  nanosheets were continuously grown on the seed crystals. As for the formation of 3D structured microflowers, it might be due to the several seed crystals aggregated at the same location for growth of the nanosheets. To further determine the valence state of vanadium element in  $\text{V}_6\text{O}_{13}$ , XPS analyses were conducted on the synthesized interconnected  $\text{V}_6\text{O}_{13}$  nanosheets. The survey XPS scan (Fig. 3a) revealed that the sample was composed of V, O, C, and N elements. The binding energies for vanadium 2p3/2 and 2p1/2 were identified in Fig. 3b at 516.0 and 523.9 eV for  $\text{V}^{4+}$  and 517.3 and 525.0 eV for  $\text{V}^{5+}$ , respectively. It was well consistent with the chemical state of vanadium in  $\text{V}_6\text{O}_{13}$  reported [25–27]. The above results confirmed that 3D  $\text{V}_6\text{O}_{13}$  nanostructures were successfully grown on c-textile via a simple seed-assisted hydrothermal process.

To investigate the electrochemical performance of 3D  $\text{V}_6\text{O}_{13}$  nanostructures grown on c-textile, half-cell coin batteries were assembled with a  $\text{V}_6\text{O}_{13}$  electrode as the cathode and lithium foil as the anode. Figure 4a displayed typical cyclic voltammetry (CV) curves of  $\text{V}_6\text{O}_{13}$  electrode in a scan rate of  $0.2 \text{ mV s}^{-1}$  between 1.5 and 4.0 V (vs.  $\text{Li}/\text{Li}^+$ ). The main redox peaks at 2.8/2.5 V could be easily identified. Broad anodic peak at  $\sim 3.2$  and 2.3 V and cathodic peak at  $\sim 1.8$  V could also be observed [11, 28]. The appearance of the above peaks indicated multi-step phase

transitions, and the location of peak voltages was consistent with the previously reported ones [2]. Figure 4b showed the galvanostatic discharge/charge curve for the first cycle at the specific current of  $30 \text{ mA g}^{-1}$ . Multiple poorly defined voltage plateaus could be identified. In the discharge curve, two sloped plateaus were identified at 2.3 and 2.8 V, corresponding to the anodic peaks. When the specific currents increased from 30 to 150, 300, 600, and  $1200 \text{ mA g}^{-1}$ , the specific capacities were 253, 176, 161, 133, and  $105 \text{ mA h g}^{-1}$ . The good electrochemical properties were due to the 3D nanostructure composed of  $\text{V}_6\text{O}_{13}$  nanosheets. Such open structure could contact with electrolytes very well and shorten the  $\text{Li}^+$  transport and diffusion path. The morphology of the original  $\text{V}_6\text{O}_{13}$  electrode and the cycled electrode with SEI was examined under SEM (Additional file 1: Figure S2). The morphology of 3D interconnected nanosheets was retained during the cycling. This further suggests the importance of the 3D nanostructure on the structural integrity of  $\text{V}_6\text{O}_{13}$  electrode during the electrochemical cycling. However, the conductivity of  $\text{V}_6\text{O}_{13}$  decreased as lithiation proceeded [7, 29]. Moreover, the length of nanosheet in the  $\text{V}_6\text{O}_{13}$  electrode reached several tens of microns. Only a small portion of individual nanosheet is directly connected with the conductive carbon fibers of c-textile, which served as the collector. It could hinder the electron transfer during the charge/discharge process. To further enhance the conductivity, and thus the electrochemical properties of the 3D free-standing  $\text{V}_6\text{O}_{13}$  electrode, it was immersed in

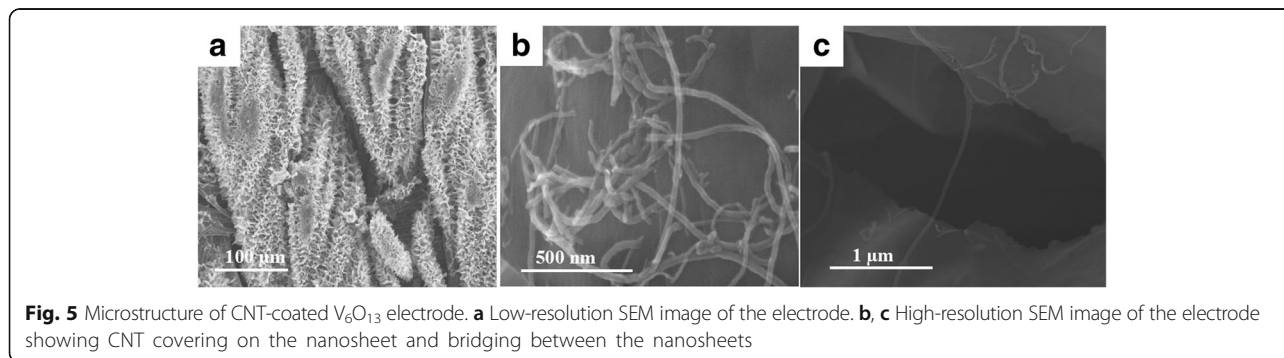


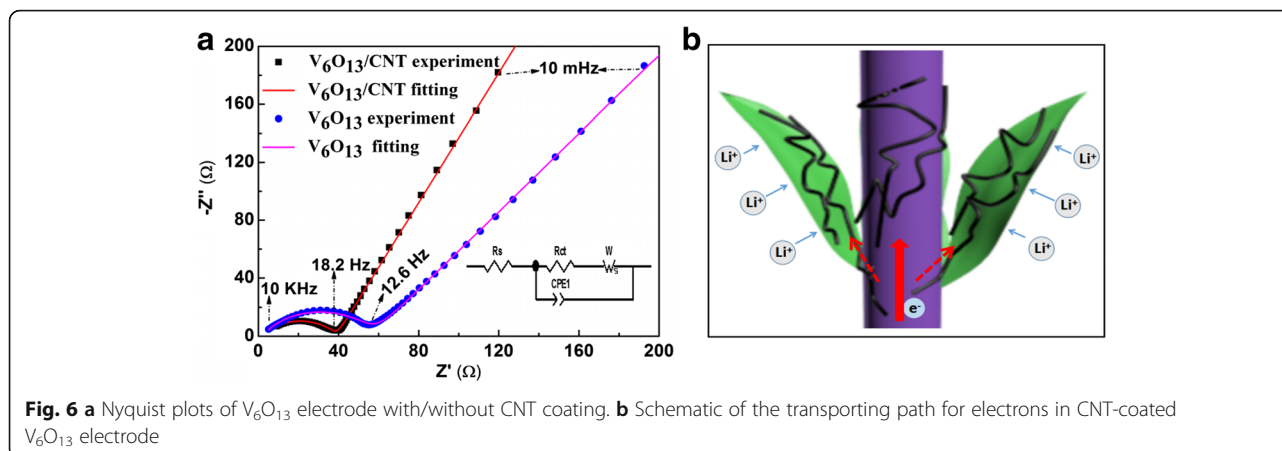
**Fig. 3** **a** A survey XPS spectrum of 3D free-standing interconnected  $\text{V}_6\text{O}_{13}$  nanosheets grown on c-textile. **b** Spectrum of  $\text{V}_{2p}$  and  $\text{O}_{1s}$  with a fitted curve. **c** The XRD patterns of  $\text{V}_6\text{O}_{13}$



CNT dispersion to dip-coat CNT on its surface. Figure 5a, b showed SEM images of the  $V_6O_{13}$  electrode with CNT. CNT was successfully deposited in the plane of  $V_6O_{13}$  nanosheets and intimately contacted with them. Even the bridging was built through CNTs between neighboring nanosheets, as shown in Fig. 5c. As expected, the resistivity of the  $V_6O_{13}$  electrode with CNT decreased from 50 to 20  $\Omega$ /sq. After the coating of CNTs, the redox peak appeared at the same position on CV profile (Fig. 4a) while peak currents increased. It indicated fast kinetic of electrochemical reaction in  $V_6O_{13}$  electrode with CNT. The  $V_6O_{13}$  electrode with CNT exhibited better rate performance compared with the electrode without CNT coating, as shown in Fig. 4c. The specific discharge capacities were 261, 185, 170, 153, and 140  $\text{mA h g}^{-1}$  at the specific currents of 30, 150, 300, 600, and 1200  $\text{mA g}^{-1}$ , respectively, corresponding to 12~40% increase compared with the

composite cathode without CNTs. To further verify the role of CNT, we calculated the lithium ion diffusion coefficient with cyclic voltammetry. The  $V_6O_{13}/\text{CNT}$  anodic and cathodic diffusion coefficients were  $4.79 \times 10^{-8}$  and  $2.01 \times 10^{-8} \text{ cm}^2 \text{ s}^{-1}$ , higher than  $V_6O_{13}$  electrode's  $2.42 \times 10^{-8}$  and  $1.7 \times 10^{-8} \text{ cm}^2 \text{ s}^{-1}$ , respectively (and the associated discussion is in Additional file 1: Figure S3). Nyquist plots (Fig. 6a) of  $V_6O_{13}$  electrode and  $V_6O_{13}$  electrode with CNT displayed similar shapes, a semicircle shape in the high-to-medium frequency domain and an inclined line in the low-frequency regions, corresponding to electrochemical reaction impedance (charge transfer process) and diffusion process of lithium ions. The inset is the equivalent circuit used to fit Nyquist plots. In the circuit, CPE is the constant phase angle element and W is the Warburg impedance.  $R_s$  and  $R_{ct}$  represent the ohmic resistance (total resistance of the electrolyte, separator, and





electrical contacts) and the charge transfer resistance, respectively [22, 30]. Additional file 1: Table S1 listed the parameters used to fit the plots.  $R_{ct}$  for the  $V_6O_{13}/CNT$  electrode was calculated to be  $37.24 \Omega$ , lower than that of  $V_6O_{13}$  ( $55.58 \Omega$ ). This decrease in charge transfer resistance was ascribed to the addition of CNT. The mechanism was illustrated in Fig. 6b. CNT intimately connected with  $V_6O_{13}$  nanosheets for faster electron transferring. Furthermore, CNTs and carbon fiber in the c-textile composed hierarchical conductive network for better electron conducting. The cyclability of  $V_6O_{13}$  electrodes was shown in Fig. 4d. At the specific current of  $300 \text{ mA g}^{-1}$ , the electrode with CNT coating could maintain 74% of the initial capacity of  $170 \text{ mA h g}^{-1}$  after 300 charge/discharge cycles, while the  $V_6O_{13}$  electrode only retained 42% of its initial capacity. It outperformed most of the low dimensional mixed-valence vanadium oxides or their 3D nanostructure listed in Additional file 1: Table S2. The better cyclability of  $V_6O_{13}$  electrode with CNT might be ascribed to the following reasons: (1) Reinforced with CNT,  $V_6O_{13}$ 's mechanical properties were improved. (2) Even if  $V_6O_{13}$  nanostructure was broken during the discharge/charge process, it was still attached to CNT and could be electrochemically activated. (3) Self-segregation of  $V_6O_{13}$  nanosheets was limited by the appearance of CNT. (4) CNT coating might be a valid barrier to alleviate side reaction of vanadium oxide with electrolyte, if any. Thus, CNT coating can be a facile alternative way to improve the conductivity of the 3D nanostructure, other than carbon coating and polymeric coating which usually require tremendous chemical synthesis work [14]. The overall electrochemical performance of  $V_6O_{13}$  cathode was limited by the conductivity of carbon cloth, the Li diffusivity in  $V_6O_{13}$  materials, and electron transfer between  $V_6O_{13}$  nanostructures and the carbon cloth. In the future work, further improvement can be made in the following ways: (1) reducing the resistance of the carbon cloth substrate, (2) doping  $V_6O_{13}$  with sulfur to improve its

diffusivity of lithium ion, and (3) coating the  $V_6O_{13}$  with conductive polymer coating.

### Conclusions

In summary, we successfully fabricated 3D free-standing  $V_6O_{13}$  nanostructure composed of interconnected nanosheets via a facile seed-assisted hydrothermal process as a cathode for LIB. The electrode exhibited good electrochemical performance. It could be further improved by coating 3D  $V_6O_{13}$  nanostructure with CNT, outperforming most of the mixed-valence vanadium oxides. Its excellent performance was due to its open 3D nanostructure and hierarchical conductive network composed of CNT in nanoscale and carbon fiber in microscale. The design of 3D nanostructure with the building block (e.g., nanowire, nanosheet) combined with constructing of the hierarchical conductive path by CNT coating can be extended to other electrode materials for better electrochemical performance.

### Additional file

**Additional file 1:** 3D interconnected  $V_6O_{13}$  nanosheets grown on carbonized textile via a seed-assisted hydrothermal process as high-performance flexible cathodes for lithium-ion batteries. (DOCX 1386 kb)

### Abbreviations

3D: Three dimensional; CE: Coulombic efficiency; CNT: Carbon nanotube; c-textile: Carbonized textile; CV: Cyclic voltammetry; DEC: Diethyl carbonate; DMC: Dimethyl carbonate; EC: Ethylene carbonate; EIS: Electrochemical impedance spectroscopy; LIB: Lithium-ion battery; SEM: Scanning electron microscopy; TEM: Transmission electron microscopy

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### Authors' contributions

HT and ZB conceived the project and supervised the study. SX conducted the material preparation, electrochemical evaluation, and structural analysis and drafted the manuscript. DC and PG helped with the electrochemical studies. All authors read and approved the final manuscript.

**Competing interests**

The authors declare that they have no competing interest.

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