

NANO EXPRESS

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Photocathodic Protection of 304 Stainless Steel by Bi₂S₃/TiO₂ Nanotube Films Under Visible Light

Hong Li¹, Xiutong Wang^{1*}, Qinyi Wei^{1,2} and Baorong Hou¹

Abstract

We report the preparation of TiO₂ nanotubes coupled with a narrow bandgap semiconductor, i.e., Bi₂S₃, to improve the photocathodic protection property of TiO₂ for metals under visible light. Bi₂S₃/TiO₂ nanotube films were successfully synthesized using the successive ionic layer adsorption and reaction (SILAR) method. The morphology and structure of the composite films were studied by scanning electron microscopy and X-ray diffraction, respectively. UV–visible diffuse reflectance spectra were recorded to analyze the optical absorption property of the composite films. In addition, the influence of Bi₂S₃ deposition cycles on the photoelectrochemical and photocathodic protection properties of the composite films was also studied. Results revealed that the heterostructure comprised crystalline anatase TiO₂ and orthorhombic Bi₂S₃ and exhibited a high visible light response. The photocurrent density of Bi₂S₃/TiO₂ was significantly higher than that of pure TiO₂ under visible light. The sensitization of Bi₂S₃ enhanced the separation efficiency of the photogenerated charges and photocathodic protection properties of TiO₂. The Bi₂S₃/TiO₂ nanotubes prepared by SILAR deposition with 20 cycles exhibited the optimal photogenerated cathodic protection performance on the 304 stainless steel under visible light.

Keywords: TiO₂ nanotube, Bi₂S₃, Stainless steel, Photocathodic protection

Background

304 stainless steel (304SS) is widely used in various industries for its good corrosion resistance and fabricability. However, this material easily deteriorates from pitting corrosion in seawater and chloride-containing solutions [1, 2]. Recently, photocathodic protection for metals has received growing attention from scientists worldwide as a promising and green technology [3–7]. TiO₂ has been extensively investigated as a photoanode for the cathodic protection because of its high chemical stability, low cost, and nontoxicity [8–11]. However, its wide bandgap (~3.2 eV for anatase) restricts its application because of its exclusive activity only under UV irradiation (3–5% of the solar spectrum) [12, 13]. The recombination of photogenerated electrons and holes in the dark results in a low photo-quantum efficiency of

TiO₂. To overcome these defects, TiO₂ nanotube arrays with large specific surface areas were synthesized [14–16] and various strategies were developed to expand its absorption to the visible light range. These strategies include coupling with narrow-bandgap semiconductors (ZnSe, WO₃, SnO₂, CdS, and Ag₂S) [17–21], metals (Ag, Au, Cu, and Bi) [22–24], and nonmetals (N, F, and graphene) [25–27]. Bi₂S₃ is an attractive material because of its narrow bandgap ($E_g = 1.3$ eV) and high photo-to-electron conversion efficiency [28]. The Bi₂S₃/TiO₂ heterostructure can reduce the recombination of photo-generated electrons and holes, and this effect would benefit the photoelectric performance of materials [29–32]. However, no research has been reported on the photogenerated cathodic protection property of Bi₂S₃/TiO₂ nanotubes. Successive ionic layer adsorption and reaction (SILAR) is a promising technique with low cost and simple equipment, which can synthesize continuous and compact film at room temperature [33]. In this study, Bi₂S₃/TiO₂ nanotube films served as photoanode for preventing 304SS corrosion. In the fabrication of the films,

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Bi_2S_3 nanoparticles were prepared by the SILAR method. The morphology, structure, and optical absorption property were studied by scanning electron microscopy (SEM), X-ray diffraction (XRD), and UV–visible (UV–vis) diffuse reflectance spectra, respectively. The influence of Bi_2S_3 deposition cycles on the photoelectrochemical and photocathodic protection properties of the composite films was also investigated in our work.

Methods

TiO_2 nanotubes were first fabricated by anodizing Ti foil in ethylene glycol electrolyte comprised of 0.5 wt% NH_4F and 6 vol% H_2O for 1.5 h and annealing at 450 °C for 1.5 h in air. Then, $\text{Bi}_2\text{S}_3/\text{TiO}_2$ nanocomposites were prepared through the alternate immersion of TiO_2/Ti substrate in the anionic and cationic precursor solutions at room temperature. The cationic precursor solution was composed of 0.01 M $\text{Bi}(\text{NO}_3)_3$ dissolved in 50 ml of acetone. Meanwhile, the anionic precursor solution was composed of 0.01 M Na_2S dissolved in 50 ml of methanol. The TiO_2/Ti substrate was first dipped into the cationic precursor solution for 20 s, and then dipped into the anionic precursor solution for 20 s, rinsed, and dried in air. The Bi_2S_3 synthesized in 10, 20, and 30 deposition cycles were assigned as BST-10, BST-20, and BST-30, respectively.

The morphology of the samples was investigated by SEM (Hitachi S-4800, Japan). The structure of the samples was examined by XRD (Bruker AXSD8 Advance, Germany). The UV–vis diffuse reflectance spectra were obtained on an UV–vis diffuse reflectance spectrophotometer (Hitachi UH4150, Japan). Photoelectrochemical experiments were conducted using a potentiostat/galvanostat (PARSTAT 2273, USA) at room temperature with a Xe lamp (PLS-SXE300C, China) as the visible light source. The open-circuit potential (OCP) of

different coupled photoelectrodes were investigated in a double-cell system (Fig. 1a). A TiO_2 or $\text{Bi}_2\text{S}_3/\text{TiO}_2$ nanotube photoelectrode was placed in a photoanode cell containing a mixed 0.1 M Na_2S and 0.2 M NaOH solution, whereas 304SS was placed in a corrosion cell containing 3.5 wt% NaCl solution. The Pt foil, saturated calomel electrode (SCE), and coupled electrode of TiO_2 and 304SS electrode served as the counter electrode (CE), reference electrode (RE), and working electrode (WE), respectively. Photocurrent curves were measured in 0.2 M Na_2SO_4 solution using an electrochemical workstation (CHI 1010C, China). The TiO_2 or $\text{Bi}_2\text{S}_3/\text{TiO}_2$ composite photoelectrode, SCE, and Pt foil served as the WE, RE, and CE, respectively (Fig. 1b).

Results and Discussion

The morphologies of $\text{Bi}_2\text{S}_3/\text{TiO}_2$ heterostructure were observed by SEM (Fig. 2). TiO_2 nanotube arrays exhibited a well-ordered, high-density, and uniform tubular structure with an average diameter of 60 nm (Fig. 2a). The Bi_2S_3 nanoparticles were successfully deposited on TiO_2 nanotube surfaces through the SILAR method (Figs. 2b–d). For BST-10, the particles distributed irregularly on the mouth of TiO_2 nanotubes (Fig. 2b). When the number of Bi_2S_3 deposition cycle increased to 20, the Bi_2S_3 nanoparticles were deposited regularly on the mouth or wall of TiO_2 nanotubes with about 15 nm in diameter. After undergoing 30 cycles, the amount of nanoparticles significantly increased, and the formation of agglomeration caused the particles to block the nanotubes.

Figure 3a depicts the XRD patterns of TiO_2 and $\text{Bi}_2\text{S}_3/\text{TiO}_2$. Aside from the diffraction peaks of titanium substrate, the peaks at 25.38°, 38.03°, 48.01°, 54.05°, 55.17°, 62.71°, and 70.44° can be indexed to lattice planes (101), (004), (200), (105), (211), (204), and (220) of anatase

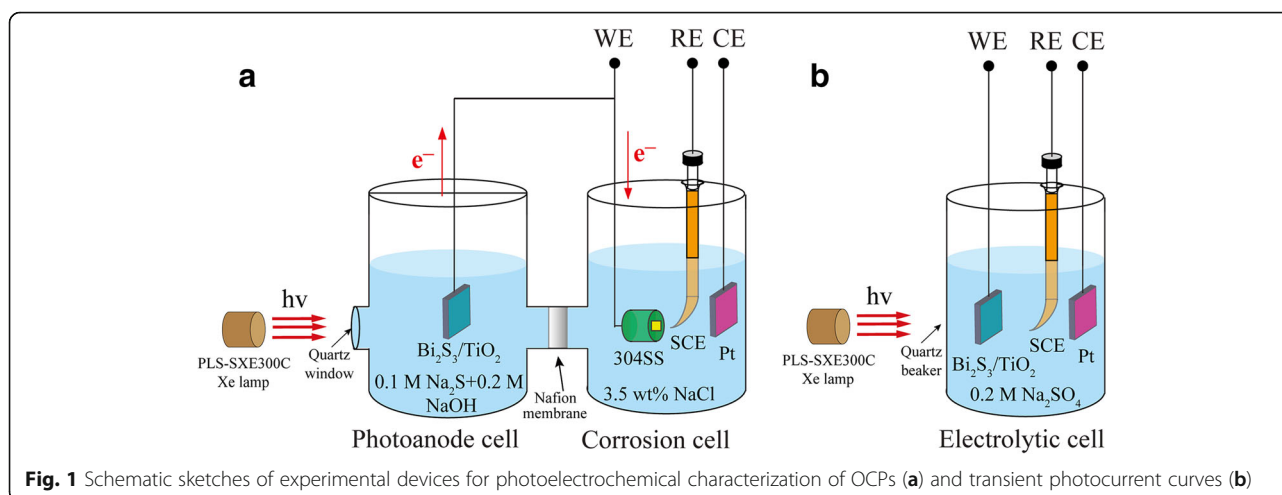


Fig. 1 Schematic sketches of experimental devices for photoelectrochemical characterization of OCPs (a) and transient photocurrent curves (b)

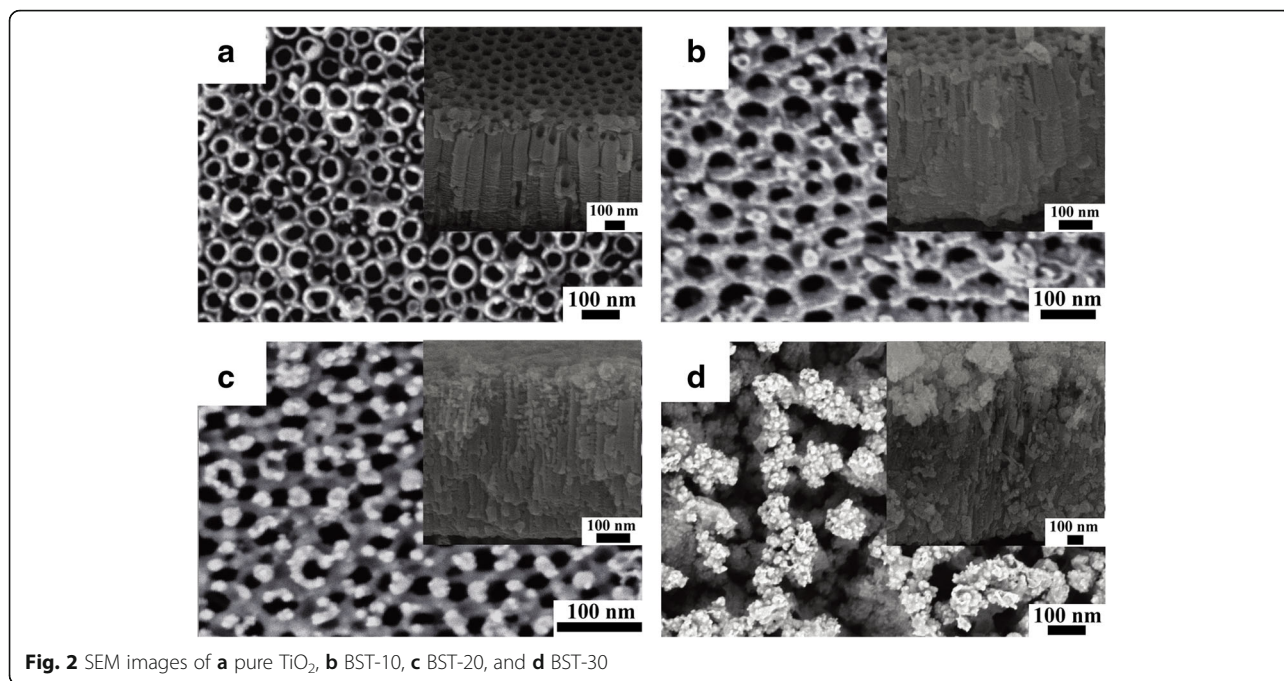


Fig. 2 SEM images of **a** pure TiO₂, **b** BST-10, **c** BST-20, and **d** BST-30

TiO₂, respectively (JCPDS 21-1272). Besides the TiO₂ peaks, the peaks at 27.74° and 32.54° were attributed to lattice planes (211) and (221) of the orthorhombic Bi₂S₃ (JCPDS 17-0320). For Bi₂S₃/TiO₂ nanocomposites, the increase in diffraction peak intensity of Bi₂S₃ with the deposition cycles revealed an increased amount of Bi₂S₃ nanoparticles on the TiO₂ nanotubes. This finding is consistent with the SEM results.

The light absorption abilities of the synthesized Bi₂S₃/TiO₂ nanotube films were assessed by UV-vis spectroscopy (Fig. 3b). Figure 3b shows that TiO₂ nanotubes absorbed mainly in the UV range with a wavelength of

about 380 nm because of the bandgap energy of anatase (3.2 eV). The spectra of Bi₂S₃/TiO₂ exhibit a relatively broad and strong absorption in the visible region, indicating that the Bi₂S₃/TiO₂ nanocomposite is capable of harvesting visible light and acts as a photoanode under visible light [34].

Figure 4a displays the transient photocurrent curves for TiO₂ and Bi₂S₃/TiO₂ photoelectrodes under visible light irradiation. The pure TiO₂ nanotube photoelectrode shows nearly 0 μA/cm² because of weak visible light absorption. However, after Bi₂S₃ nanoparticle sensitization, the transient photocurrent densities of

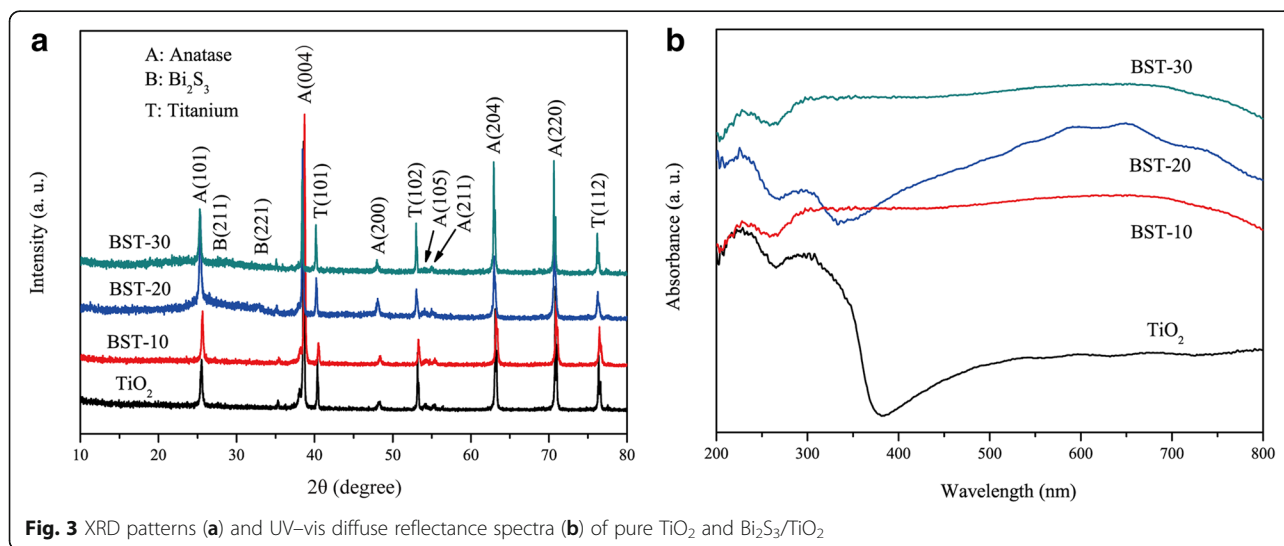


Fig. 3 XRD patterns **(a)** and UV-vis diffuse reflectance spectra **(b)** of pure TiO₂ and Bi₂S₃/TiO₂

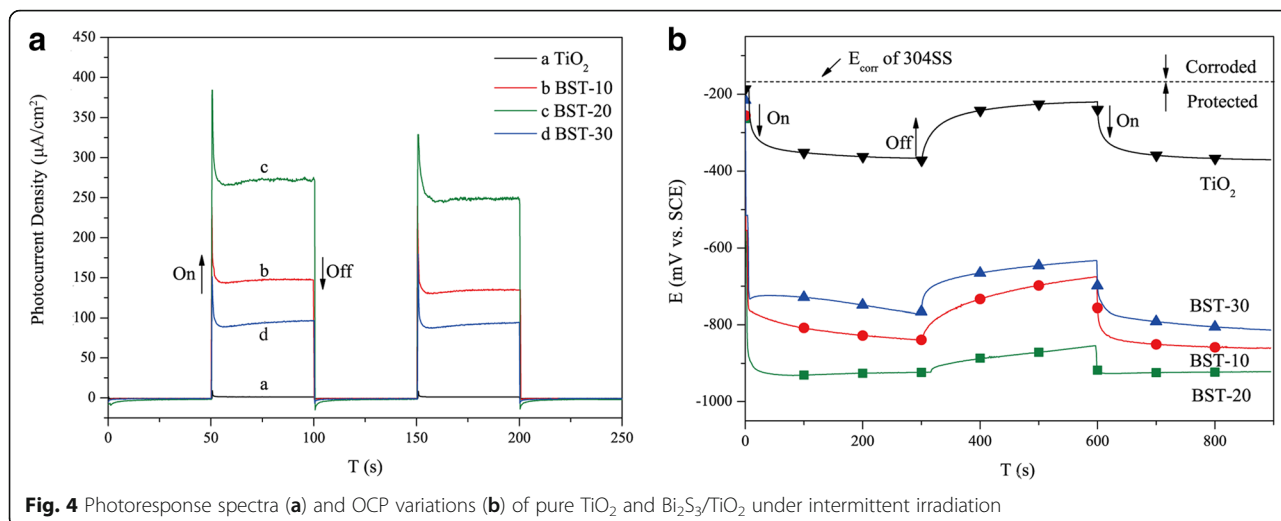


Fig. 4 Photoresponse spectra (a) and OCP variations (b) of pure TiO_2 and $\text{Bi}_2\text{S}_3/\text{TiO}_2$ under intermittent irradiation

$\text{Bi}_2\text{S}_3/\text{TiO}_2$ exhibited an obvious increase, implying that the $\text{Bi}_2\text{S}_3/\text{TiO}_2$ nanocomposite is capable of utilizing visible light and the heterostructure promotes the separation of photogenerated electrons and holes [35]. The transient photocurrent density of BST-20 ($249 \mu\text{A}/\text{cm}^2$) was higher than that of BST-10 ($134 \mu\text{A}/\text{cm}^2$) and BST-30 ($92 \mu\text{A}/\text{cm}^2$), indicating that BST-20 possesses an optimal separation efficiency of the photogenerated electrons and holes.

Figure 4b compares the photogenerated OCPs of 304SS coupled with different TiO_2 nanotubes. When the light was on, the potentials of coupled electrodes all shifted negatively within a few seconds. This effect may be attributed to the cathodic polarization of 304SS which results from the excited photoelectrons transfer from TiO_2 nanotubes to 304SS [36, 37]. After the light was off, the OCP of 304SS coupled to pure TiO_2 returned to a value near the free corrosion potential of bare 304SS, indicating the invalid recombination of the photogenerated electrons and holes in the TiO_2 [38]. By

contrast, the OCPs of 304SS coupled with $\text{Bi}_2\text{S}_3/\text{TiO}_2$ exhibited a slightly positive shift and stayed far below than the free corrosion potential of bare 304SS. The charges stored in the $\text{Bi}_2\text{S}_3/\text{TiO}_2$ composite were released and again transferred to 304SS in the dark. The negative shift of potentials is reportedly an important parameter for evaluating the separation efficiency of photogenerated charges [39, 40]. The increased negative shift of the potentials indicates the increased effectiveness of the cathodic protection of photoanodes. Under visible light, the shift of potentials can be ranked in the following order: TiO_2 (150 mV vs. SCE) < BST-30 (534 mV vs. SCE) < BST-10 (572 mV vs. SCE) < BST-20 (662 mV vs. SCE). Hence, BST-20 possesses the optimal photocathodic protection property for 304SS. This result may be due to the fact that the active sites and light harvesting increased with the rising Bi_2S_3 amount. However, the excessive Bi_2S_3 particles served as the recombination sites of the electrons and holes, which hindered the charge transfer from the $\text{Bi}_2\text{S}_3/\text{TiO}_2$ composite to 304SS.

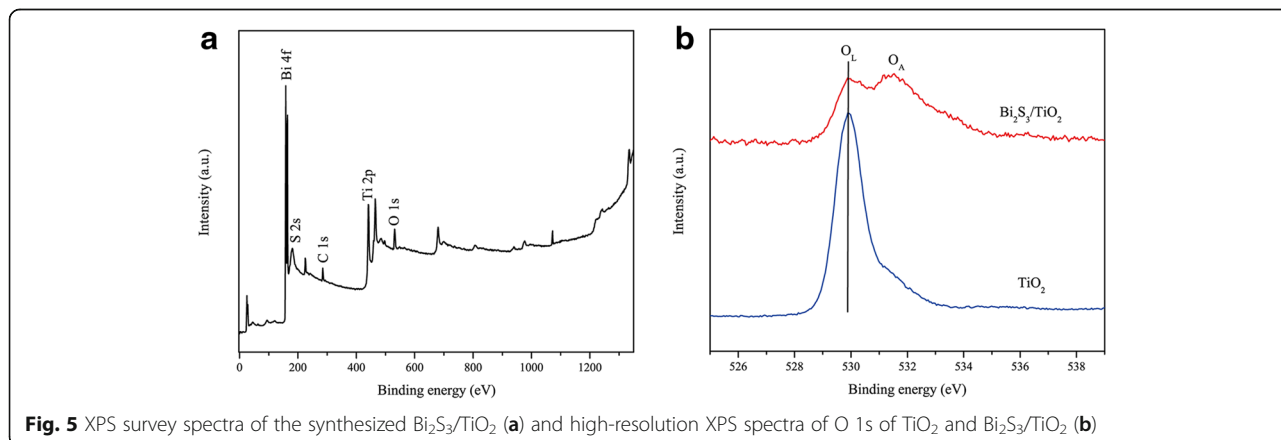


Fig. 5 XPS survey spectra of the synthesized $\text{Bi}_2\text{S}_3/\text{TiO}_2$ (a) and high-resolution XPS spectra of O 1s of TiO_2 and $\text{Bi}_2\text{S}_3/\text{TiO}_2$ (b)

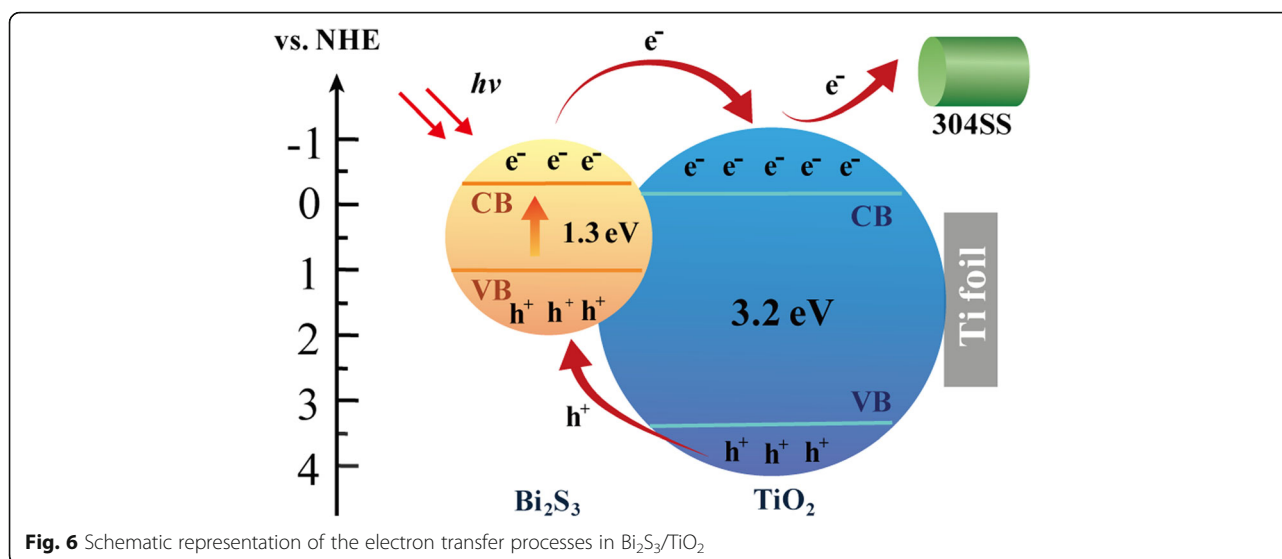


Fig. 6 Schematic representation of the electron transfer processes in $\text{Bi}_2\text{S}_3/\text{TiO}_2$

The X-ray photoelectron spectroscopy (XPS) was measured to investigate the chemical compositions and states of $\text{Bi}_2\text{S}_3/\text{TiO}_2$ (BST-20). The XPS survey spectra revealed the existence of Bi, S, Ti, and O components, in addition to C contaminants (Fig. 5a). As shown in Fig. 5b, the XPS peaks of O 1s at 529.7 eV were analyzed from the lattice oxygen (O_L) in TiO_2 . The peak at 531.6 eV was derived from the adsorbed oxygen (O_A). The O_A was composed of OH species or weak bonding oxygen on the composite surface. The presence of O_A was ascribed to the generation of oxygen vacancy on the sample surface. This suggests that the $\text{Bi}_2\text{S}_3/\text{TiO}_2$ composite exhibits higher photocathodic protection properties than TiO_2 .

Figure 6 shows the schematic diagram of the photoelectric conversion and transportation processes in the $\text{Bi}_2\text{S}_3/\text{TiO}_2$ composite. The Bi_2S_3 nanoparticles can easily absorb the photons in the visible light due to the presence of O_A and the suitable bandgap width of Bi_2S_3 . When the photons were absorbed by the Bi_2S_3 nanoparticles, the photoexcited electrons were generated and transferred from the conduction band (CB) of Bi_2S_3 to the CB of TiO_2 . The photogenerated holes were then shifted from the valence band (VB) of TiO_2 to the VB of Bi_2S_3 . When Na_2S served as a hole-trapping agent, the photogenerated charges were effectively separated. The electrons were finally transferred to the 304SS electrode, and the potential of the 304SS electrode negatively shifted. The 304SS was prevented from corrosion by $\text{Bi}_2\text{S}_3/\text{TiO}_2$ under visible light. Therefore, the more efficient separation of the photogenerated charges in the composite would accelerate the oxidation and reduction reactions and, hence, generate a higher photocathodic protection activity than TiO_2 .

Conclusions

In summary, Bi_2S_3 -nanoparticle-decorated TiO_2 nanotubes were successfully synthesized through the electrochemical anodization and SILAR method. The sensitization of Bi_2S_3 significantly extended the spectral response from UV to the visible region. Consequently, the composite showed higher photocurrents and cathodic protection performance than pure TiO_2 . With increased number of Bi_2S_3 deposition cycles, the increasing grain size and loading of the Bi_2S_3 nanoparticles significantly affected the photocathodic protection activity of the $\text{Bi}_2\text{S}_3/\text{TiO}_2$ nanocomposite. The $\text{Bi}_2\text{S}_3/\text{TiO}_2$ nanotubes prepared by SILAR deposition with 20 cycles exhibited the optimal photocathodic protection property.

Abbreviations

304SS: 304 stainless steel; CB: Conduction band; CE: Counter electrode; O_A : Adsorbed oxygen; OCP: Open-circuit potential; O_L : Lattice oxygen; RE: Reference electrode; SCE: Saturated calomel electrode; SEM: Scanning electron microscopy; SILAR: Successive ionic layer adsorption and reaction; VB: Valence band; WE: Working electrode; XRD: X-ray diffraction

Acknowledgements

This work is financially supported by the Shandong Province Postdoctoral Innovation Project (No. 201601001), China Postdoctoral Science Foundation (No. 2015M582150), and CAS Strategic Priority Project (No. XDA13040404).

Authors' Contributions

HL performed the synthesis and characterization of the $\text{Bi}_2\text{S}_3/\text{TiO}_2$ nanotubes. XW, QW, and BH participated in the characterization. HL supervised the conceptual framework and drafted the manuscript. All authors read and approved the final manuscript.

Competing Interests

The authors declare that they have no competing interests.

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Received: 4 January 2017 Accepted: 24 January 2017

Published online: 31 January 2017

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