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Coexistence of multiple metastable polytypes in rhombohedral bismuth

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Derivative structural polytypes coexisting with the rhombohedral A7 structure of elemental bismuth (Bi) have been discovered at ambient condition, based on microstructure analyses of pure Bi samples treated under high pressure and high temperature conditions. Three structures with atomic positions close to those of the A7 structure have been identified through first-principles calculations, showing these polytypes energetically comparable to the A7 structure under ambient condition. Simulated diffraction data are in excellent agreement with the experimental observations. We argue that previously reported some variations of physical properties (e.g., density, electrical conductivity, and magnetism) in bismuth could be due to the formation of these polytypes. The coexistence of metastable derivative structural polytypes may be a widely occurring phenomenon in other elemental materials.

Bismuth as a typical semimetal has attracted extensive research interests. Numerous studies have been conducted on Bi to investigate the electrical properties^{1–5} and carrier transport behaviors^{6–15}, thermoelectric properties^{16,17} and phase transitions under high pressure and high temperature (HPHT)^{18–22}. Under ambient condition, Bi is commonly designated as a rhombohedral lattice (space group R-3m, Strukturbericht A7), which is characterized by a pair of atoms spaced non-equidistantly along the trigonal axis in a Peierls distortion of the simple cubic structure⁸. Alternatively, the structure can be described either as a hexagonal lattice with six atoms per unit cell or as a pseudo-cubic lattice with one atom per unit cell⁵. However, it is well-known that the observed X-ray diffraction (XRD) patterns of Bi cannot be fully accounted for with the rhombohedral A7 structure, which is designated as (marked by) “Doubtful (?)” in the PDF card (JCPDS PDF 44-1246) of the International Center for Diffraction Data and labeled with “§” in the periodic table of elements²³.

Some long-standing unusual behaviors in physical properties have also been noted in Bi, possibly resulting from the structure uncertainty. In 1928, Kapitza reported inconsistent changes in specific resistance when Bi crystals were subjected to strong magnetic field²⁴, and attributed the observations to small lattice distortions in Bi structure. Later, slight density variations of Bi single crystals were reported²⁵. In 1931, Goetz and Focke found that Bi single crystals grown under a strong oriented magnetic field exhibit a decrease in density and an increase in specific resistance²⁶. The applied magnetic field appeared to affect the Bi lattice during solidification, leading to changes in the density and resistance. Some earlier observations were attributed to a so-called “mosaic block” structure^{27,28} with slightly different lattice constants. A variety of formation mechanisms were proposed for this mosaic structure based on different assumptions, such as the influence of impurities²⁷, allotropic changes²⁹, and deformation twinning²⁵. Nonetheless, the atomic configuration of the mosaic structure in Bi crystals remains elusive.

In this paper, we report structural polytypes of Bi evidenced by both experimental observations and first-principles calculations. We performed structural characterizations on pure Bi samples treated at the conditions of different pressure and temperature with XRD and high-resolution transmission electron microscopy (HRTEM) and discovered a remarkable structural diversity in Bi after the treatments. Using crystal structure prediction techniques, we identified three new structures of bismuth that are closely related to the well-known A7 structure and differ from each other only by minute lattice distortions. First principles calculations indicate these structures have very similar energies, explaining the observed coexistence of multiple metastable polytypes in Bi. Such a structural diversity is intrinsic, not due to extrinsic action from the impurities or deformation, and maybe leads to some property variations (such as density, melting point, electrical conductivity, and magnetism) in pure

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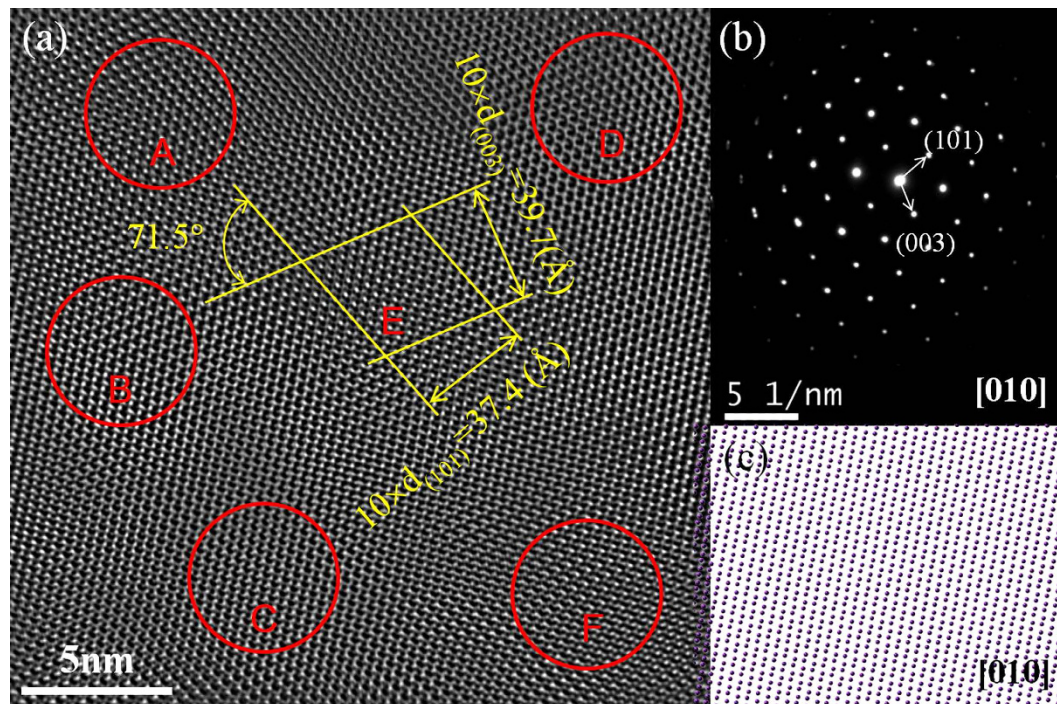


Figure 1. TEM micrograph of Bi treated under the pressure of 2 GPa and the temperature 2273 K.

(a) HRTEM image viewed along the [101] zone axis of the A7 structure. Different lattice distortion regions are labeled by A–F. Measurements of lattice d-spacings and interplanar angles for the A7 structure are shown in region E; (b) Overall SAED pattern corresponding to (a); (c) simulated atomic distribution observed along [010] direction in A7-Bi structure.

Bi under normal conditions. These polytypes provide a structural interpretation for the so-called “mosaic block” structure previously speculated.

Results and Discussion

An HRTEM image of the Bi sample treated under the condition of high pressure of 2 GPa and temperature of 2273 K viewed along [010] zone axis is presented in Fig. 1a with the corresponding selected area electron diffraction (SAED) patterns shown in Fig. 1b. Simulated atomic distribution of the ideal A7-Bi structure viewed along the same zone axis is shown in Fig. 1c. In Fig. 1a, several regions with distinct atomic configurations are highlighted (red circles, labeled A–F), which are clearly different from the ideal A7 atomic distribution (Fig. 1c). The boundaries between these lattice distortion regions (Fig. 1a A–F) are fuzzy and poorly defined. For clarity, these regions are displayed side-by-side in Fig. 2, and the d-spacings and interplanar angles measured from these regions are summarized in Table 1. All these regions show very similar d-spacings and interplanar angles, with deviations smaller than 0.1 \AA and 0.3° , respectively. We note that the d-spacings ($3.74 \pm 0.03 \text{ \AA}$ and $3.97 \pm 0.03 \text{ \AA}$) and the interplanar angles ($71.52 \pm 0.05^\circ$) determined from region E perfectly match those of the (003) and (101) planes of the A7 structure. Therefore, the SAED pattern (Fig. 1b) taken from the entire area of Fig. 1a does not show clear evidence of structural variation, even though changes in lattice parameters and clearly differences in lattice fringes are observed. It should be noticed that these lattice distortion regions may be formed in only nano-scale (about 2–10 nm) basing on observations of the HRTEM images.

The shifts of Bi atoms from the A7 structure are so small (less than 0.1 \AA) that HRTEM images and the corresponding SAED patterns alone cannot offer solid proof of structural polytypes. To account for the subtle lattice distortions observed in HRTEM measurement, we theoretically investigated a number of possible structural variations, in which atoms were shifted slightly from the equilibrium sites of the A7 structure without significant energy increase. Three new structures based on CALYPSO searches along with the known A7 structure and phase II (structure of Bi at pressure higher than 2.7 GPa) are shown in Fig. 3, where each structure is depicted with biatomic layers to highlight the structural similarities among them. These newly identified structures (Fig. 3b–d) are designated as M (monoclinic with space group C2/m), T1 (triclinic with space group P-1), and T2 (another triclinic structure with space group P1) phases, respectively. The symmetry groups, enthalpies, densities, and lattice parameters of these structures are listed in Table 2. Unit cells and atomic positions are shown in Fig. S1 (see the Supporting Information) and Table 3, respectively.

The bismuth biatomic layer can be constructed by chair-formed hex-atomic rings as building blocks. In Fig. 3f Bi atoms labelled as Bi1, Bi3, and Bi5 in the hex-atomic ring constitute the upper sub-layer, while those labelled as Bi2, Bi4, and Bi6 constitute the lower sub-layer. The Bi structures are constructed by stacking a series of biatomic layers. The biatomic layers in A7, M, T1, and Phase II structures are all parallel to the *ab* plane of the lattice. The A7 structure is comprised of ...ABCABC... stacked biatomic layers A, B, and C along the *c* axis, with each layer

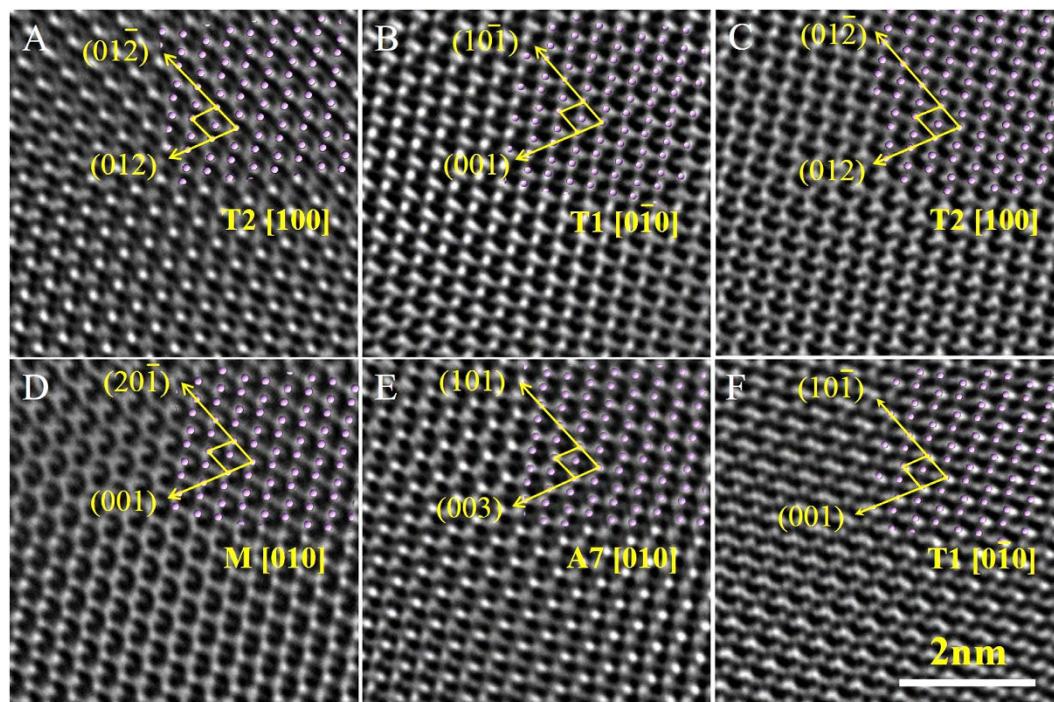


Figure 2. Enlarged atomic-resolution electron micrographs for (A–F) areas in HRTEM image shown in Fig. 1. Corresponding atomic arrangements for given Bi structures are projected on the micrographs.

Measuring data in HR-TEM image			Corresponding d-spacings and angles in predicted Bi structures		
Area in HR-TEM image	d-spacing (Å) Deviation: ±0.03	Angle (°) Deviation: ±0.05	Structure and zone axis	d-spacing (Å)	Angle (°)
A	$d_1 = 3.99$	71.38	T2-Bi [100]	$d_{(012)} = 4.0$	71.34
	$d_2 = 3.76$			$d_{(01-2)} = 3.7453$	
B	$d_1 = 3.97$	71.54	T1-Bi [0-10]	$d_{(001)} = 4.0323$	71.50
	$d_2 = 3.76$			$d_{(10-1)} = 3.7593$	
C	$d_1 = 3.97$	71.36	T2-Bi [100]	$d_{(012)} = 4.0$	71.34
	$d_2 = 3.76$			$d_{(01-2)} = 3.7453$	
D	$d_1 = 3.95$	71.67	C-Bi [010]	$d_{(001)} = 3.9683$	71.64
	$d_2 = 3.76$			$d_{(20-1)} = 3.7453$	
E	$d_1 = 3.97$	71.52	A7-Bi [010]	$d_{(003)} = 3.9370$	71.56
	$d_2 = 3.74$			$d_{(101)} = 3.7453$	
F	$d_1 = 3.97$	71.43	T1-Bi [0-10]	$d_{(001)} = 4.0323$	71.50
	$d_2 = 3.76$			$d_{(10-1)} = 3.7593$	

Table 1. Correspondence between the measured data for different areas in HR-TEM and corresponding d-spacings and angles in predicted Bi polytype structures.

spanning one third of the lattice period. For M, T1, and phase II structures, each biatomic layer spans one lattice period along the *c* axis. T2 structure (Fig. 3d), however, possesses the most complex crystal structure: The repeating unit consists of four chair-formed hex-atomic rings with distinct bond angles and bond lengths, as detailed in Fig. 3g. Each biatomic layer can only be constituted by moving the units along the [100] and [2-2 1] directions. These layers are parallel to the (0 2 4) plane of T2 structure, with an...ABAB... stacking along the *b* axis.

The chair-formed hex-atomic rings in these structures differ slightly in the bond angle and bond length. The corresponding angles (α_1 – α_6) and lengths (L1–L6) are thus introduced (see Fig. 3f), and listed in Table 2 to characterize each structure. The A7 structure has a unique chair conformation with identical bond angle of 94.54° and bond length of 3.1 Å. For the newly identified Bi polytypes, the bond angle and length show gradually increased diversities, leading to a gradually reduced symmetry from M to T1, and to T2. However, we note that deviations in the bond angle, bond length, and enthalpy are only within 0.5%, 0.8%, and 0.04%, respectively, for the Bi polytypes compared with the A7 structure.

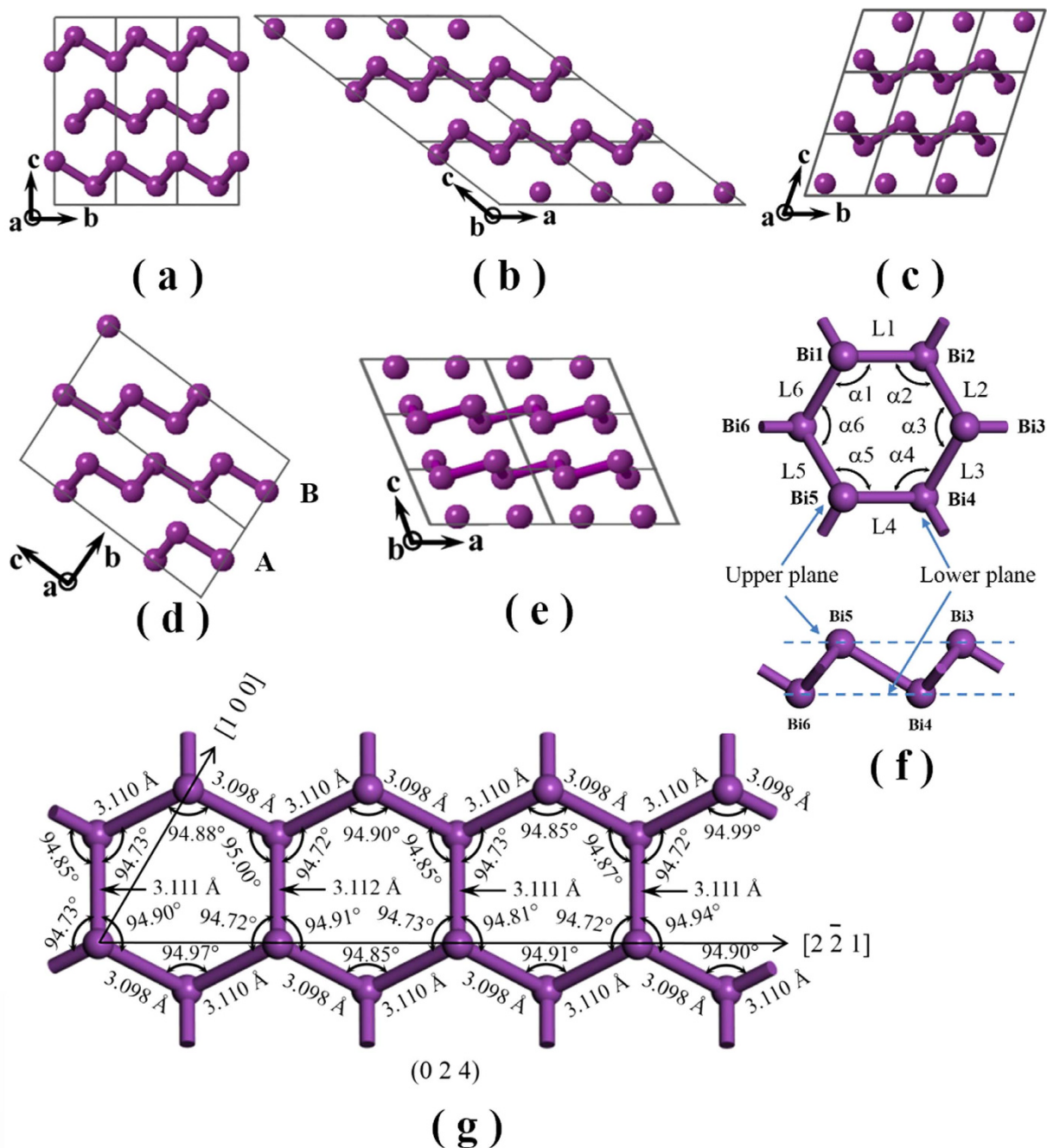


Figure 3. Top views of possible Bi polytype crystal structures. (a–e) are views along the *a* or *b* axis for A7, M, T1, T2, and Bi-phase II, respectively. The double atomic layers are described by the atoms connected by illustrative bonding sticks; (f) the typical chair-form hexatomic unit, and (g) the distorted chair-form hexatomic unit for the T2 structure.

Dynamic stability of the new structures was verified by phonon spectrum calculations in Fig. S2 (see the Supporting Information), which show no imaginary frequencies in the Brillouin zone. Thermodynamic stability of these derivative structures was also examined, and the results are shown in Fig. 4. Enthalpy differences of the new structures relative to the A7 structure are less than 3 meV at pressures between 0 and 2.7 GPa. Such small differences suggest all these new structures are stable within this pressure range. At pressures higher than 2.7 GPa, our calculations show that the most stable structure is Bi-phase II. This is consistent with the known phase diagram of Bi¹⁸.

The very small enthalpy differences among the A7, M, T1, and T2 structures suggest that very small energy perturbations, due to either phase transition from phase II on pressure release or magnetic treatment, may alter the structure of Bi and that these derivative polytypes can co-exist with the A7 structure. This may be the structural origin of the crystalline imperfections¹² and the anisotropic defects¹⁵ or the speculated “mosaic block”

Structure	Symmetry group	Lattice parameter	Enthalpy (eV/atom)	Density (g/cm ³)	layer structural characteristic
A7	R-3m (D_{3d}^5)	a = b = 4.5540 Å c = 11.8281 Å $\alpha = \beta = 90^\circ \gamma = 120^\circ$	-3.8783	9.80	$\alpha_1 = \alpha_2 = \alpha_3 = \alpha_4 = \alpha_5 = \alpha_6 = 94.54^\circ$ L1 = L2 = L3 = L4 = L5 = L6 = 3.100 Å
M	C2/m (C_{2h}^3)	a = 7.8873 Å b = 4.5572 Å c = 6.5836 Å $\alpha = \gamma = 90^\circ$ $\beta = 143^\circ$	-3.8777	9.75	$\alpha_1 = \alpha_2 = \alpha_4 = \alpha_5 = 94.55^\circ \alpha_3 = \alpha_6 = 94.70^\circ$ L1 = L4 = 3.102 Å L2 = L3 = L5 = L6 = 3.098 Å
T1	P-1 (C_i^1)	a = 4.5738 Å b = 4.5786 Å c = 4.8088 Å $\alpha = 90.31^\circ$ $\beta = 118.25^\circ \gamma = 60.04^\circ$	-3.8769	9.47	$\alpha_1 = \alpha_4 = 94.98^\circ \alpha_2 = \alpha_5 = 94.78^\circ \alpha_3 = \alpha_6 = 94.61^\circ$ L1 = L4 = 3.098 Å L2 = L5 = 3.123 Å L3 = L6 = 3.106 Å
T2	P1 (C_1^1)	a = 4.5767 Å b = 4.7786 Å c = 13.3039 Å $\alpha = 86.07^\circ$ $\beta = 90.02^\circ \gamma = 90.02^\circ$	-3.8781	9.56	$\alpha_1 = 94.92 \pm 0.07^\circ \alpha_2 = 94.92 \pm 0.08^\circ \alpha_4 = 94.91 \pm 0.06^\circ$ $\alpha_5 = 94.88 \pm 0.07^\circ \alpha_3 = \alpha_6 = 94.725 \pm 0.005^\circ$ L1 = L4 = 3.098 Å L2 = L5 = 3.1115 ± 0.005 Å L3 = L6 = 3.110 Å
Phase II	C2/m (C_{2h}^3)	a = 7.0015 Å b = 6.2556 Å c = 3.3749 Å $\alpha = \gamma = 90^\circ$ $\beta = 112.71^\circ$	-3.8406	10.07	$\alpha_1 = \alpha_3 = \alpha_4 = \alpha_6 = 92.91^\circ \alpha_2 = \alpha_5 = 145.28^\circ$ L1 = L2 = L4 = L5 = 3.277 Å L3 = L6 = 3.199 Å

Table 2. Calculated symmetry groups, lattice parameters, enthalpies, densities, and double-atomic layer characteristics of the structural polytypes.

Bi structures	Atoms	Wyckoff	Atomic Positions		
			x/a	y/b	z/c
M	Bi1	4i	1.02900	0	0.79310
T1	Bi1	2i	0.96552	0.26962	0.20288
T2	Bi1	1a	0.12504	0.97126	0.99179
	Bi2	1a	0.62463	0.52827	0.50851
	Bi3	1a	0.12485	0.02910	0.75843
	Bi4	1a	0.62519	0.47073	0.74186
	Bi5	1a	0.12432	0.02877	0.25836
	Bi6	1a	0.62468	0.47042	0.24194
	Bi7	1a	0.12495	0.96996	0.49170
	Bi8	1a	0.62466	0.52955	0.00859

Table 3. Atomic positions in new polytype structures of Bi.

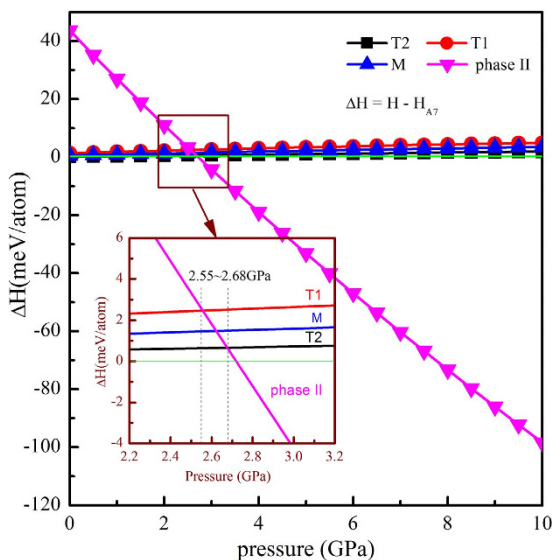


Figure 4. Enthalpy differences (ΔH) of the predicted Bi polytype structures relative to A7. Inset shows the details of ΔH at the pressure range from 2.2 to 3.2 GPa.

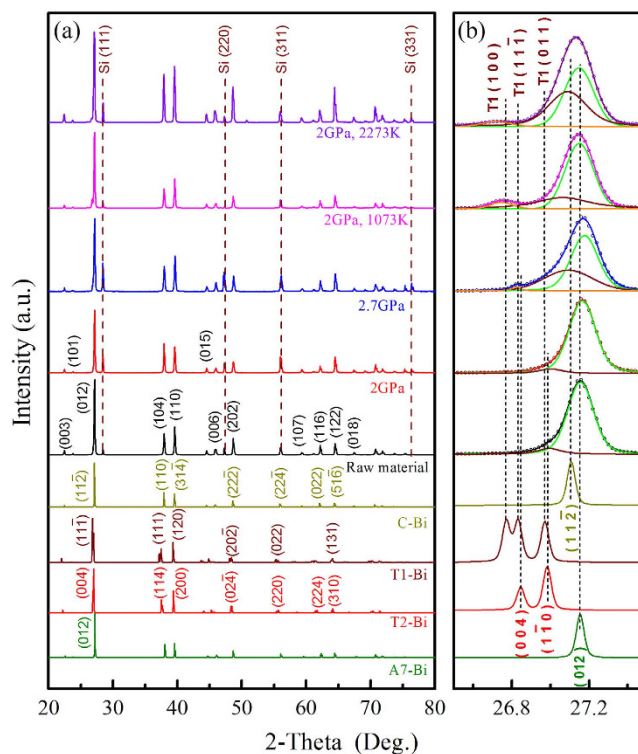


Figure 5. XRD patterns of various Bi samples as compared with those calculated from the new structural polytypes. **(a)** Experimental XRD spectra corresponding to the Bi samples recovered from raw material, cold-pressing at 2 GPa and 2.7 GPa, HT-HP treating at 2 GPa-1073 K and 2 GPa-2273 K, and the calculated XRD spectra of M, T1, T2 and A7 structures. **(b)** The enlarged spectra at the range of 26.5~27.5°.

structure^{25–28} in pure Bi. The density of calculated A7 structure is 9.80 g/cm³, in agreement with the experimental value reported by Kapitza²⁴. The densities of M, T1, and T2 structures are all slightly lower than that of A7, with the lowest being 9.47 g/cm³ in T1 structure. Thus, the coexistence of various structural polytypes explains well the density variations measured by Goetz in Bi single crystals solidified under different magnetic field strength²⁶.

Table 1 summarizes d-spacings and interplanar angles measured from HRTEM for regions A–F of Fig. 2 in comparison with the calculated ones. Simulated atomic positions viewed along the corresponding directions of the structural polytypes are superimposed on the HRTEM images in the upper right corners (Fig. 2). The excellent agreement between simulated and observed images suggest that region E corresponds to a projection along the [010] direction of the A7 structure, regions A and C to a projection along T2 [100], regions B and F to a projection along T1 [0 – 1 0], and region D to a projection along M [010].

The structural polytypes can also be detected in XRD patterns (Fig. 5), which further confirm the coexistence of the derivative polytypes in the bulk samples including the raw Bi and the Bi specimens treated under the different conditions of high pressure and temperature. XRD peaks generally match the rhombohedral A7 structure (PDF#44-1246), but with additional weak peaks near the main peaks that cannot be explained by the A7 structure alone. The calculated XRD patterns of the M, T1, T2, and A7 structures are also shown in Fig. 5 for comparison. For illustration purposes, we concentrate on the main peak in the XRD patterns, which are magnified in Fig. 5b for the two-theta range from 26.5° to 27.5°. Clearly, there are one or more additional peaks other than the A7 (0 1 2) peak at 27.16° in all the samples. A small peak at 26.99°, corresponding well to T2 (1 – 1 1), is present in the raw material (untreated) and the sample cold-pressed to 2 GPa, suggesting that small amounts of the derivative polytype T2 phase coexisted in the samples. Two additional peaks at 26.82° and 27.09° present in the sample cold-pressed to 2.7 GPa (i.e., above the A7–phase II boundary of 2.52 GPa) can be attributed to the T1 (1 1 – 1) and M (1 1 – 2), respectively. In the samples treated at 2 GPa/1073 K and 2 GPa/2273 K, the peaks at 26.75° and 27.09° may correspond to the diffraction lines of T1 (1 0 0) and M (1 1 – 2), respectively (Fig. 5b). Note that the last two samples underwent melting and recrystallization, a more intense new peak is present at a smaller angle.

Conclusion

Based on the experimental and theoretical investigations, we establish the structural diversity as an essential property for element Bi, in particular for samples treated at high pressures and high temperatures. Such diversity leads to the formation of secondary structures, which may be considered as derivative structures of the A7 phase. In our samples, these phase have only been observed on the nanoscale basing on the HRTEM images. Coexisting derivative structural polytypes may be a general phenomenon. One example is the monoclinic diamond (M-diamond) recently reported in bulk nanotwinned diamond material³⁰. In that case, phase transition from carbon onion nanoparticles to diamond and localized stress heterogeneity across diamond twin domains may be the cause. The

exceptional thermal and mechanical properties observed in such multi-phase material suggest that coexisting structural polytypes may have interesting potentials in tailoring physical properties of materials.

Methods

Experimental details. Bi granules (Alfa Aesar, purity 99.999%) were compressed into cylinders (6 mm in diameter and 6 mm in height) and each was placed into a BN capsule with 8 mm outer diameter. High-pressure and high-temperature (HP-HT) experiments were performed using a China-type large volume cubic press with a maximum of 1400 tons on every WC anvil³¹. Pyrophyllite cubes with edge length of 49 mm served as both pressure medium and gasket. The samples were first compressed to required pressure, and then heated to the desired temperature for 1 hour by resistive heating with graphite tube heaters (outer diameter 10 mm and inner diameter 8 mm). The HP-HT conditions are presented in Table S1 (see the Supporting Information). For all the experiments, temperature was measured *in situ* with a type C thermocouple (W5/Re26). Pressure was estimated with previously obtained calibration curves based on the electrical resistance change³² during the phase transitions of Bi at room temperature. After HP-HT treatment, the samples were quenched to room temperature at a cooling rate of ~100 K/s before the pressure was unloaded. Bulk samples recovered at ambient condition had diameters of approximately 5.5 mm and heights of 5.3–5.7 mm. Both the raw granules and the bulk samples after HP-HT treatment were ground in an agate mortar with a pestle, for XRD (Rigaku D/MAX-PC/2500, operated at 40 kV and 200 mA with a Cu-K α target) and high resolution transmission electron microscopy (HRTEM) (JEM-2010 with an accelerating voltage of 200 kV). The HRTEM was carefully calibrated with silicon at the selected magnification. The change in microstructure due to grinding is expected to be the same for all samples; thus the difference observed in HRTEM and XRD analyses are considered due to HP-HT treatment.

First-Principles Calculations. The software package of the crystal structure analysis by particle swarm optimization (CALYPSO)³³ was employed to search possible low-energy structures for Bi. We examined thousands of candidate structures with simulation cell sizes ranging from 2 to 12 atoms, at pressures of 0–6 GPa. Structural relaxations were performed using the density functional theory (DFT) within the generalized gradient approximation (GGA)³⁴ implemented in the Vienna Ab Initio Simulation Package (VASP)³⁵. Projector augmented-wave pseudopotential with 6s²6p³ electrons (as is the case of the valence for Bi) was adopted. The energy cutoff was 300 eV, and appropriate Monkhorst-Pack k meshes were chosen to ensure that enthalpy calculations converged to better than 0.01 meV per atom for each structure. Phonon frequencies were generated using the Cambridge Serial Total Energy Package (CASTEP) with finite displacement theory³⁶. Conventional DFT schemes that apply to the GGA usually overestimate lattice constant parameters³⁷. In order to make direct comparison between calculated lattice parameters and the experimentally observed values, we adjusted hydrostatic pressure in computing the lattice parameters of the A7 structure, until they match the parameters for the rhombohedral structure (PDF#44-1246). This hydrostatic pressure was identified for 1.2 GPa and then applied to all other possible structures in first-principles calculations. Finally, XRD patterns of these structures were simulated by Reflex tools package of the Materials Studio (MS) software, to help interpreting the experimentally observed XRD patterns.

References

- Smith, G. E., Baraff, G. A. & Rowell, J. M. Effective g factor of electrons and holes in bismuth. *Phys. Rev.* **135**, A1118–1124(1964).
- Mangez, J. H., Issi, J. P. & Heremans, J. Transport properties of bismuth in quantizing magnetic fields. *Phys. Rev. B* **14**, 4381–4385(1976).
- Édel'man, V. S. Electrons in bismuth. *Adv. Phys.* **25**, 555–613(1976)
- Kopelevich, Y., Medina Pantoja, J. C., da Silva, R. R. & Moehlecke, S. Universal magnetic-field-driven metal-insulator-metal transformations in graphite and bismuth. *Phys. Rev. B* **73**, 165128:1–7(2006).
- Hofmann, Ph. The surfaces of bismuth: Structural and electronic properties. *Prog. Surf. Sci.* **81**, 191–245(2006).
- Issi, J.-P. Low temperature transport properties of the group V semimetals. *Aust. J. Phys.* **32**, 585–628(1979).
- Behnia, K., Balicas, L. & Kopelevich, Y. Signatures of electron fractionalization in ultraquantum bismuth. *Science* **317**, 1729–1731(2007).
- Fritz, D. M. *et al.* Ultrafast bond softening in bismuth: mapping a solid's interatomic potential with X-rays. *Science* **315**, 633–636(2007).
- Behnia, K. Elemental complexity. *Science* **321**, 497–498(2008).
- Li, L. *et al.* Phase transitions of dirac electrons in bismuth. *Science* **321**, 547–550(2008).
- Yang, H. *et al.* Phase diagram of bismuth in the extreme quantum limit. *Nat. Commun.* **1**, 47(2010).
- Zhu, Z., Collaudin, A., Fauqué, B., Kang, W. & Behnia, K. Field-induced polarization of dirac valleys in bismuth. *Nat. Phys.* **8**, 89–94(2012).
- Fuseya, Y., Ogata, M. & Fukuyama, H. Transport properties and diamagnetism of dirac electrons in bismuth. *J. Phys. Soc. Jpn.* **84**, 012001:1–22 (2015)
- Zhu, Z. *et al.* Landau spectrum and twin boundaries of bismuth in the extreme quantum limit. *PNAS* **109**, 14813–14818(2012).
- Küchler, R. *et al.* Thermodynamic evidence for valley-dependent density of states in bulk bismuth. *Nat. Mater.* **13**, 461–465(2014).
- Heremans, J. & Thrush, C. M. Thermoelectric power of bismuth nanowires. *Phys. Rev. B* **59**, 12579–12583(1999).
- Hostler, S. R. *et al.* Thermoelectric properties of pressed bismuth nanoparticles. *Superlattice. Microst.* **43**, 195–207(2008).
- Tonkov, Y. E. & Ponyatovsky, E. G. Phase Transformations of Elements Under High Pressure. *Series: Advances in Metallic Alloys* (eds Eskin, D. G., Fridlyander, J. N.), 148–157 (CRC Press, 2004).
- Vaišnys, J. R. & Kirk, R. S. Effect of pressure on the electrical properties of bismuth. *J. Appl. Phys.* **38**, 4335–4337(1967).
- Compy, E. M. Phase Diagram of Bismuth at Low Temperatures. *J. Appl. Phys.* **41**, 2014–2018(1970)
- Lu, M. *et al.* Low-temperature electrical-transport properties of single-crystal bismuth films under pressure. *Phys. Rev. B* **53**, 1609–1615(1996).
- Iwasaki, H., Chen, J. H. & Kikegawa, T. Structural study of the high-pressure phases of bismuth using high-energy synchrotron radiation. *Rev. Sci. Instrum.* **66**, 1388–1390(1995).
- Wittwer, J. Periodic table of the elements. © 2003–2015 Vertex42 LLC. All rights reserved. Available at: <http://www.vertex42.com/Files/pdfs/2/periodic-table.pdf> (Accessed: 28th June 2015)
- Kapitza, P. The Study of the specific resistance of bismuth crystals and its change in strong magnetic fields and some allied problems. *P. R. Soc. Lond. A-Conta.* **119**, 358–443(1928).
- Goetz, A. On the experimental evidence of the mosaic structure of Bi single-crystals. *Proc. N. A. S.* **16**, 99–105(1930).

26. Goetz, A. & Focke, A. B. Density and conductivity of bismuth single crystals grown in magnetic fields with relation to their mosaic structure. *Phys. Rev.* **37**, 1044–1056(1931).
27. Zwicky, F. On the imperfections of crystals. *Proc. N. A. S.* **15**, 253–259(1929).
28. Zwicky, F. On mosaic crystals. *Proc. N. A. S.* **15**, 816–822(1929).
29. Goetz, A. & Hergenrother, R. C. X-ray studies of the thermal expansion of bismuth single crystals. *Phys. Rev.* **40**, 643–661(1932).
30. Huang, Q. *et al.* Nanotwinned diamond with unprecedented hardness and stability. *Nature* **510**, 250–253(2014).
31. Liu, X. *et al.* A large volume cubic press with a pressure-generating capability up to about 10Gpa. *High Pressure Res.* **32**, 239–254(2012).
32. Decker, D. L., Bassett, W. A., Merrill, L., Hall, H. T. & Barnett, J. D. High-pressure calibration: A critical review. *J. Phys. Chem. Ref. Data* **1**, 773–835(1972).
33. Wang, Y. C., Lv, J., Zhu, L. & Ma, Y. M. Crystal structure prediction via particle-swarm optimization. *Phys. Rev. B* **82**, 094116:1–8(2010).
34. Perdew, J. P., Burke, K. & Ernzerhof, M. Generalized gradient approximation made simple. *Phys. Rev. Lett.* **77**, 3865–3868(1996).
35. Kresse, G. & Furthmüller, J. Efficient iterative schemes for *ab initio* total-energy calculations using a plane-wave basis set. *Phys. Rev. B* **54**, 11169–11186(1996).
36. Clark, S. J. *et al.* First principles methods using CASTEP. *Z. Kristallogr.* **220**, 567–570(2005).
37. Haas, P., Tran, F. & Blaha, P. Calculation of the lattice constant of solids with semilocal functionals. *Phys. Rev. B* **79**, 085104:1–10(2009).

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Author Contributions

Y.J.T. conceived the project. D.L.Y., Y.S., G.Y.S., Y.B.W., Z.Y.L. and Y.J.T. designed the experiments. Y.S., Z.S.Z. and D.L.Y. performed the HPHT experiments, W.T.H. and D.L.Y. performed HRTEM observations and analyses. Y.S. and D.L.Y. performed XRD measurements and the first principles calculations. Y.S., D.L.Y., W.T.H. and H.J.L. analyzed the data. D.L.Y., Y.S., B.X., Y.B.W. and G.Y.S. co-wrote the paper. All authors discussed the results and commented on the manuscript.

Additional Information

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