

# SCIENTIFIC REPORTS

OPEN

## Self-assembly preparation of SiO<sub>2</sub>@Ni-Al layered double hydroxide composites and their enhanced electrorheological characteristics

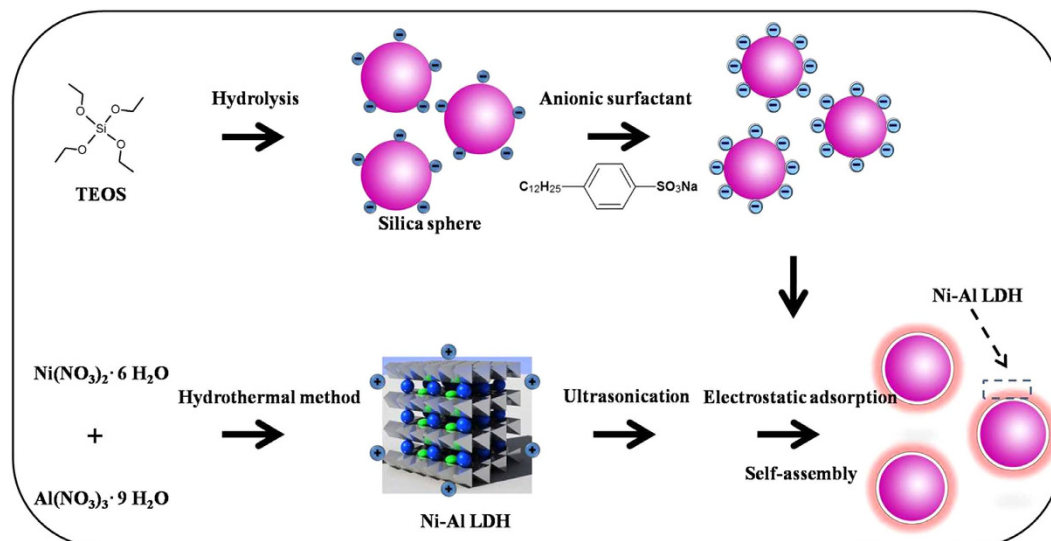
Received: 06 October 2015  
Accepted: 17 November 2015  
Published: 16 December 2015

Xuqiang Ji<sup>1,\*</sup>, Wenling Zhang<sup>1,\*</sup>, Lei Shan<sup>2</sup>, Yu Tian<sup>2</sup> & Jingquan Liu<sup>1</sup>

The core-shell structured SiO<sub>2</sub>@Ni-Al layered double hydroxide (LDH) composites were prepared via self-assembly of Ni-Al LDH on the surface of SiO<sub>2</sub> spheres. Only coating a layer of ultrathin Ni-Al LDH sheet, the resulting SiO<sub>2</sub>@Ni-Al LDH composites exhibit significantly enhanced electrorheological (ER) characteristics compared to conventional bare SiO<sub>2</sub> spheres. The monodispersed SiO<sub>2</sub> spheres with average diameters of 260 nm were synthesized by the hydrolysis of tetraethyl orthosilicate (TEOS), while the shell part, Ni-Al LDH sheet was prepared by the hydrothermal procedure. The morphology of the samples was investigated via scanning transmission electron microscopy (STEM), scanning electron microscopy (SEM) and transmission electron microscopy (TEM). The structure of the samples was characterized by X-ray diffraction (XRD). The species and distribution of elements in samples were confirmed by X-ray photoelectron spectroscopy (XPS), Energy dispersive analysis of X-ray (EDX) and elemental mapping in STEM. Subsequently, the ER characteristics of the composites dispersed in insulating oil were characterized by a rotational rheometer. The electric field-stimulated rheological performances (yield stress, viscosity, modulus, etc.) were observed under an external electric field, which is different from the Newtonian state in the free electric field.

There are many examples of biological and inorganic materials with complicated and efficient hierarchical morphologies in natural world, such as marine coccolith, radiolarian shell and clay, exhibiting superior performance<sup>1–5</sup>. To synthesize inorganic materials with natural materials' structure for the development of catalysis<sup>6</sup>, electronic<sup>7</sup>, lithium secondary battery<sup>8</sup>, and many other applications to meet the challenges of green energy and sustainability, has been one of the hotspots in material chemistry fields. Here, the layered double hydroxides (LDH) are hydroxide-like clays with the empirical formula,  $[M^{II}_{1-x}M^{III}_x(OH)_2]^{x+}[A^{n-}]_{x/n} \cdot nH_2O$ , where M<sup>II</sup> and M<sup>III</sup> are divalent and trivalent metals such as Mg, Ni, Co, Al, and Fe, A<sup>n-</sup> can be almost any anion with the charge number of n<sup>9–11</sup>. Due to their special and complicated structure and anion exchange property, LDH materials have been applied to many areas including catalysts<sup>12–16</sup>, anion exchangers<sup>17–20</sup>, electrodes for alkaline secondary batteries and supercapacitor<sup>21,22</sup>, especially for electrorheological (ER) fluids<sup>23–25</sup>. Compared to other clay minerals including smectites, kaolins, Pyrophyllite, talo etc., LDH sheets possess distinctive advantages, such as positive charges on its surface due to the existence of metal ions (Ni<sup>+</sup>, Al<sup>3+</sup>, Mg<sup>2+</sup> etc.) and negative charges among the interlayers<sup>26</sup>, which are more sensitive to electric stimulus for the application of ER systems. ER fluids are smart materials that experience continuous and reverse changes in rheological properties in a few milliseconds under an applied electric field<sup>27–30</sup>. In general, it involves a dispersed phase such as polyaniline (PANI), polypyrrole (PPy), polystyrene (PS), silica, titania, zeolite, clay or other polarizable or semiconducting particles and a continuous phase, including silicone oil or corn oil, etc.<sup>24,31–33</sup>. The solid-like property emerges when electric field is applied across the ER fluids,

<sup>1</sup>College of Chemical Science and Engineering, Laboratory of Fiber Materials and Modern Textile, The Growing Base for State Key Laboratory, Collaborative Innovation Centre for Marine Biomass Fibers, Qingdao University, Qingdao 266071, China. <sup>2</sup>State Key Laboratory of Tribology, Tsinghua University, Beijing 100084, China. \*These authors contributed equally to this work. Correspondence and requests for materials should be addressed to W.Z. (email: wzhangqd@qdu.edu.cn) or J.L. (email: jliu@qdu.edu.cn)



**Figure 1.** Schematic diagram of the stepwise preparation of electro-responsive  $\text{SiO}_2$ @Ni-Al LDH composites.

with the formation of chain-like or columnar structures aligned to the electric field. Due to the fast response to electric field, controllable mechanical properties and no waste discharge, the ER fluid has been paid more and more attention in an automotive industry such as shock absorbers, engine mounts and clutches<sup>34,35</sup>.

The inorganic materials (aluminosilicate, zeolite, alumina,  $\text{TiO}_2$ ,  $\text{SiO}_2$  etc.) and the semiconducting polymer (PANI, copolypyrrole, copolyaniline, polyacene quinone radicals etc.) have been widely utilized to prepare ER suspensions due to their low cost and anisotropic morphology<sup>36–48</sup>. Recently, much attention has been focused on the layered inorganic materials, such as LDH sheet, graphene or graphene oxide (GO) based hybrids. However, the aggregation of LDH sheet usually greatly weakens the material's ER properties, thus significantly limits the applications in an automotive industry field. The core-shell structured composites with controllable morphology, orientation, and dimensionality have evoked considerable interest because of their remarkable ER properties, low cost and well controlled particle size<sup>49,50</sup>.

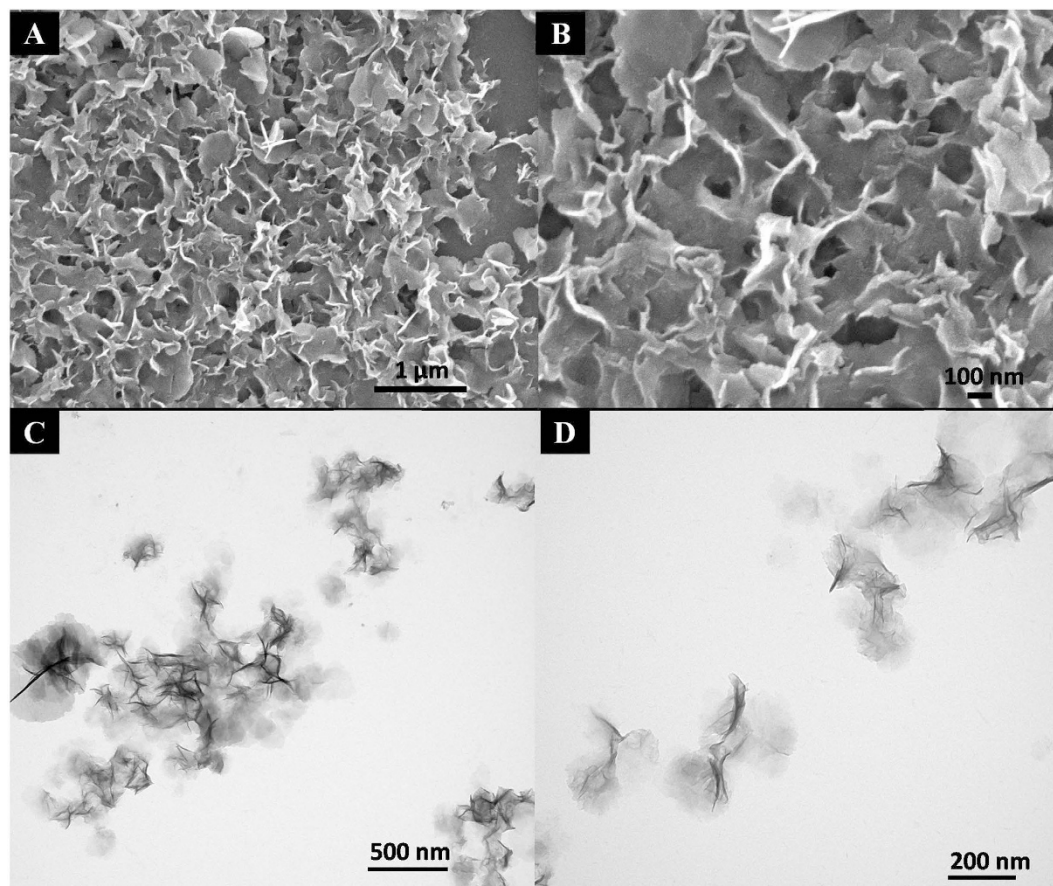
Diverse composite nanoparticles with core-shell structure have been found to show excellent ER properties. PS microspheres coated with GO sheets were prepared using  $\pi$ - $\pi$  stacking interactions<sup>51</sup>. PANI-coated anisotropic snowman-like poly(methyl methacrylate) (PMMA) particles were also prepared with an anionic surfactant as ER fluid suspension particles<sup>52</sup>. Core-shell structured titania/PANI and PS/PANI microspherical composites were synthesized via *in situ* polymerization in the presence of titanium oxide or PS nanospheres<sup>42,53</sup>.  $\text{SiO}_2/\text{TiO}_2$  particles were prepared by means of a co-precipitation process by Wu and his coworkers<sup>54</sup>. Monodispersed PMMA microbeads coated with MWNT-NH<sub>2</sub> were also prepared via a grafting reaction<sup>55</sup>. To prepare core-shell structured biomimetic materials, Mirkin and co-workers have explored diatom silica walls as the core template to integrate with inorganic nanoparticles. The nanoparticles formed a near-monolayer along the surface morphology and shape of the diatom template<sup>56</sup>. And a facile approach related to direct self-assembly of inorganic nanostructures under mild conditions without the utilization of harsh chemical treatments is under explored urgently. Seeking for greener synthetic routes to avoid the use of toxic reagents and realize time-saving has been one of the major concerns of the global vision of world economy.

Inspired by the previous work, here,  $\text{SiO}_2$ @Ni-Al LDH core-shell structure was prepared via self-assembly of Ni-Al LDH on  $\text{SiO}_2$  sphere as shown in Fig. 1. As the core material, monodispersed  $\text{SiO}_2$  with diameters of 260 nm were synthesized by the hydrolysis of tetraethyl orthosilicate (TEOS) induced using the catalysis of ammonia in ethanol medium, while the shell part, Ni-Al LDH, was prepared by the hydrothermal method. Subsequently, the ER characteristic of the  $\text{SiO}_2$ @Ni-Al LDH composite dispersed in insulating oil was characterized by a rotational rheometer.

## Results and Discussion

Generally, the surface of the bare  $\text{SiO}_2$  sphere in ethanol is negatively charged with a zeta potential value of  $-15.8$  mV, the zeta potential value for Ni-Al LDH sheets is 16.18 mV. Therefore, the electrostatic interactions between the negatively charged  $\text{SiO}_2$  sphere and positively charged Ni-Al LDH sheet could take place. Due to the low zeta potential, the anionic surfactant was also introduced to enhance the adsorption between  $\text{SiO}_2$  sphere and Ni-Al LDH sheet. The morphology of Ni-Al LDH sheet is shown in Fig. 2. It can be observed from the SEM and TEM images, Ni-Al LDH sheet is irregular and thin hexagonal platelet with a mean lateral size of 300 nm, while most of the LDH sheets stack with each other.

The morphology of  $\text{SiO}_2$ @Ni-Al LDH composites was analyzed through SEM (Fig. 3A,B) and TEM (Fig. 3C,D) images. The  $\text{SiO}_2$  spheres are found to be monodispersed with a uniform size of  $\sim 260$  nm. The final  $\text{SiO}_2$ @Ni-Al LDH composites exhibited a roughly spherical morphology with curly Ni-Al LDH sheets coated on the surface. The



**Figure 2.** (A,C) SEM and TEM images of Ni-Al LDH sheets; (B,D) High magnification SEM and TEM images of Ni-Al LDH sheets.

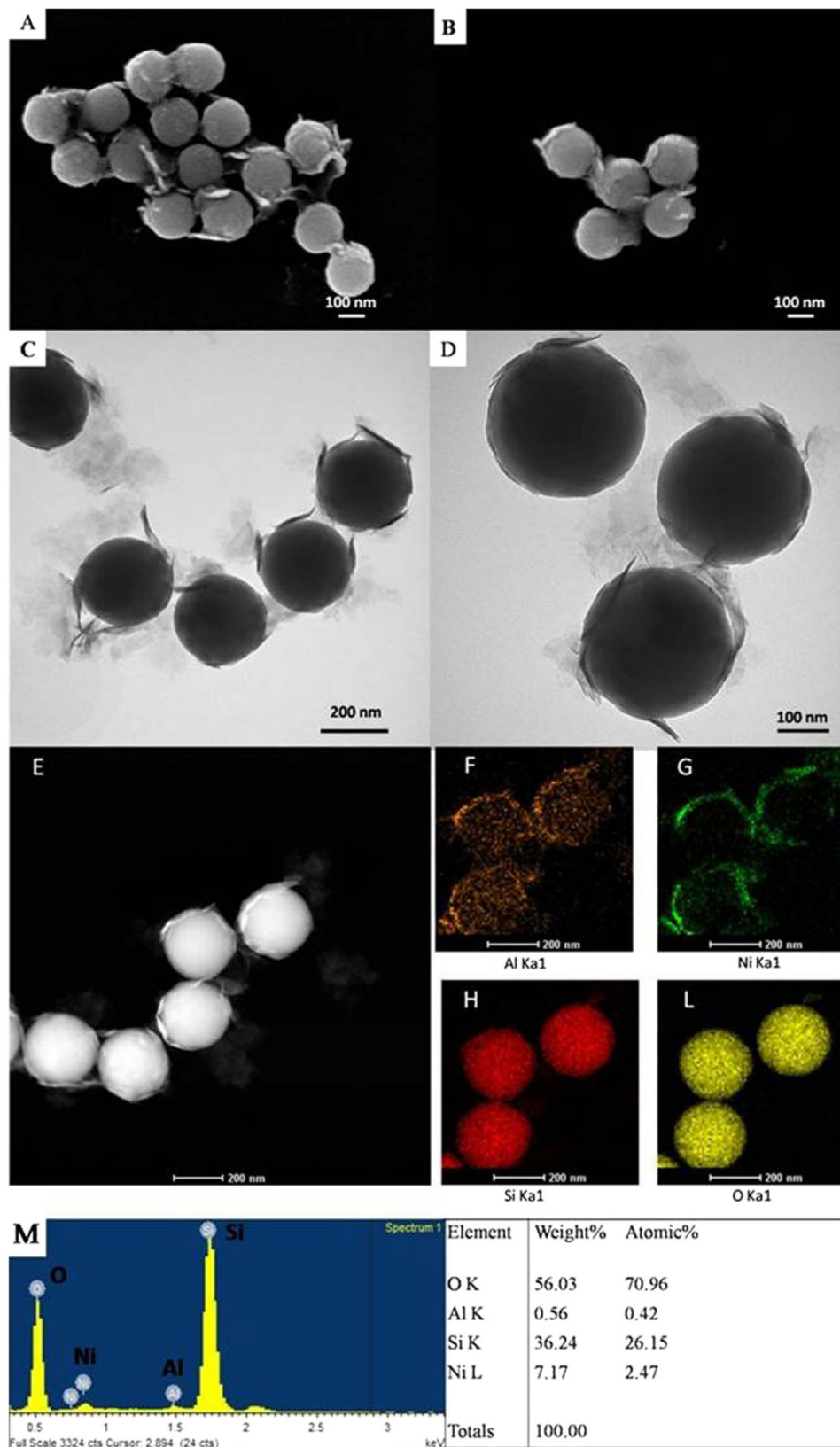
presence of Ni-Al LDH sheet in the composites was also proved through EDX (Fig. 3M), the weight percentages of Ni and Al elements were calculated to be 7.17% and 0.56% respectively.

A dark-field STEM analysis of  $\text{SiO}_2$ @Ni-Al LDH composites was also used to characterize the distribution of the elements and the result is shown in Fig. 3E. It can be seen that the  $\text{SiO}_2$ @Ni-Al LDH composites have a representative core-shell structure. In addition, element mapping of the identical spheres showed the space distribution of Si, O, Ni, Al. As shown in Fig. 3H,L, the core part is filled with intense Si and O signals. Meanwhile, the Ni, Al elements are mainly located on the shell part, indicating the successful coating of the Ni-Al LDH sheet.

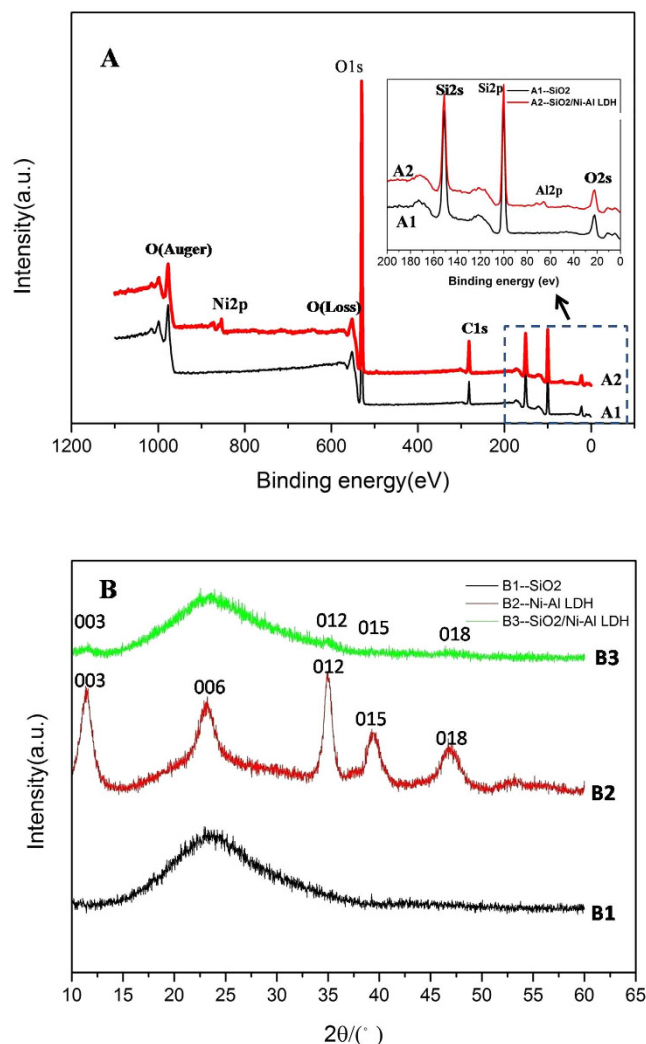
The XRD patterns of  $\text{SiO}_2$  sphere, Ni-Al LDH sheet, and  $\text{SiO}_2$ @Ni-Al LDH composites are shown in Fig. 4B. The diffraction pattern of  $\text{SiO}_2$  sphere depicts a reflection characteristic of amorphous silica<sup>57</sup>. After the coating with the Ni-Al LDH sheet, peaks related to 003, 006, 012 and 015 originating from Ni-Al LDH sheet can be observed in addition to those resulted from silica. The composition of  $\text{SiO}_2$  sphere and  $\text{SiO}_2$ @Ni-Al LDH composites was investigated using the XPS technique. As shown in Fig. 4A, the XPS survey spectrum of  $\text{SiO}_2$ @Ni-Al LDH composites revealed the presence of Si, O, Ni, Al elements, implying the formation of  $\text{SiO}_2$ @Ni-Al LDH composites. Taken together, the results of SEM, TEM, EDX, STEM, XRD and XPS evidenced the successful synthesis of  $\text{SiO}_2$ @Ni-Al LDH composites.

In general, the shear stress of the bare silica particles based suspensions are not very sensitive to the external electric field stimulus due to their weak ER activity<sup>58</sup>. Therefore, coating a layer of ultrathin Ni-Al LDH sheet, is an effective way to enhance the ER activity of the bare  $\text{SiO}_2$  spheres. Here, the prepared  $\text{SiO}_2$ @Ni-Al LDH composites exhibited improved ER performance compared to the conventional bare  $\text{SiO}_2$  particles, which have been described below:

The shear stress-shear rate behaviour of the  $\text{SiO}_2$ @Ni-Al LDH composites (10 wt% particle concentration) was tested using a rotational rheometer equipped with a high voltage generator subjected to different electric field strengths. As shown in Fig. 5A, the shear stress increased linearly with the increasing shear rate without an external electric field, which is consistent with a Newtonian fluid behaviour<sup>59</sup>. However, when a high electric field was applied, the ER fluid exhibited a large increase in shear stress because of the formation of chain-like structure. This performance was similar to a traditional Bingham fluid. The shear stress of bare  $\text{SiO}_2$  particle based ER fluid exhibited slightly changes with the increasing of electric field strength shown in Fig. 6A, however, pure Ni-Al LDH based ER fluid resulted in electric short at even 1.05 kV/mm due to its relatively higher electric conductivity (Fig. 6B). Figure 5B shows the shear viscosity of  $\text{SiO}_2$ @Ni-Al LDH composites as a function of the shear rate. The



**Figure 3.** (A,B) SEM images (C,D) TEM images of  $\text{SiO}_2$ @Ni-Al LDH composites; (E) Dark-field STEM image of  $\text{SiO}_2$ @Ni-Al LDH composites. The elemental mapping for Al (F), Ni (G), Si (H) and O (L) elements in the core-shell structure of  $\text{SiO}_2$ @Ni-Al LDH composites; and (M) EDX spectrum of  $\text{SiO}_2$ @Ni-Al LDH composites.



**Figure 4.** (A) XPS survey spectra of SiO<sub>2</sub> sphere and SiO<sub>2</sub>@Ni-Al LDH composites. (B) XRD patterns of SiO<sub>2</sub> sphere, Ni-Al LDH sheet, and SiO<sub>2</sub>@Ni-Al LDH composites, respectively.

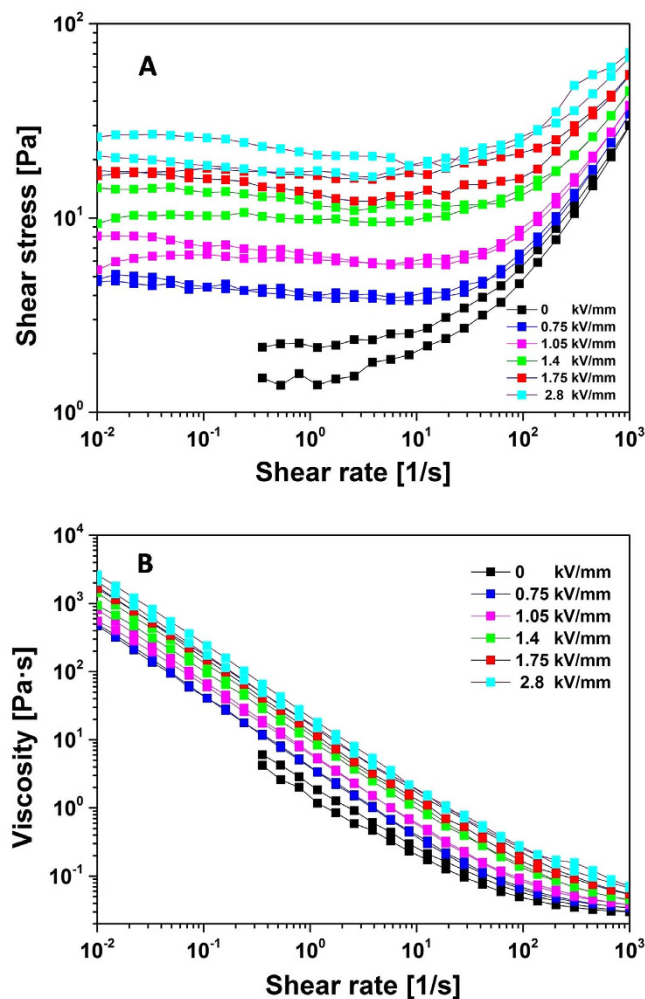
ER fluid of SiO<sub>2</sub>@Ni-Al LDH composites showed Newtonian fluid characteristics without an electric field and typical shear thinning behaviour<sup>60</sup> under different electric fields.

Furthermore, the dynamic yield stress was plotted as a function of electric fields in log-log scale shown in Fig. 7. The correlation between the yield stress ( $\tau_y$ ) and electric fields ( $E$ ) is fitted by the equation  $\tau_y \propto E^m$ , where  $m$  equals to 1.5, which is consistent with the conduction model. The conductivity mismatch between the particles and the liquid medium is considered to be the critical factor to result in the aligned array along the field direction in ER system.

The viscoelastic properties of the ER fluids was examined by the oscillation measurements<sup>61</sup>. As shown in Fig. 8A, the linear viscoelastic range was clearly shown in the amplitude sweep. In addition, the storage modulus ( $G'$ ) was larger than the loss modulus ( $G''$ ) under an electric field, indicating a higher degree of solid-like behaviour displayed by the ER fluid. When the strain amplitude was beyond a critical value, both  $G'$  and  $G''$  decreased sharply because of the breakdown of the structure of the ER fluid.

Figure 8B shows the  $G'$  and  $G''$  in the frequency sweep for the SiO<sub>2</sub>@Ni-Al LDH ER fluid in the linear viscoelastic region under different electric fields. The suspension exhibited solid-like behaviour, in which  $G'$  was substantially larger than  $G''$ , and  $G'$  remained almost constant over a broad frequency range. The larger  $G'$  with increasing electric field strength reflects the higher ER effect<sup>41</sup>.

In order to verify the sensitivity and the stability of the ER fluid to an electric field, the steady shear flow was operated at a fixed shear rate of  $1 \text{ s}^{-1}$  under applied square voltage pulse ( $t = 20 \text{ s}$ ). The shear stress under various electric field strengths is shown in Fig. 9. When exposed to electric field, the shear stress of the ER fluids jumped to higher level from the zero-field shear stress and dropped to zero-field level again right away when the electric field is removed. This phenomenon indicates the fast and reversible transition in the as-prepared ER fluid corresponding to the electric field.



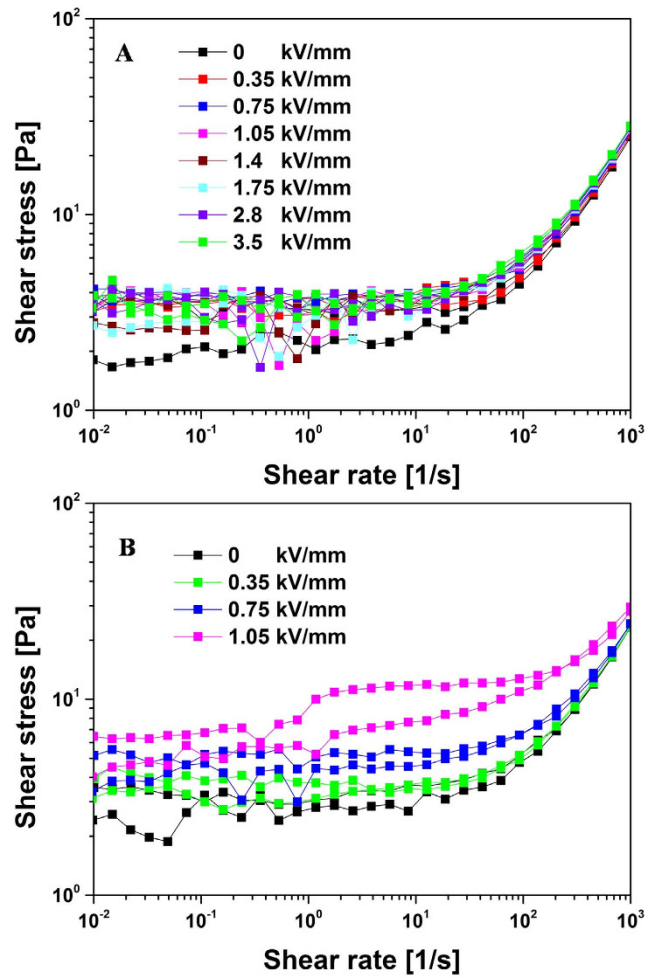
**Figure 5.** Flow curves of shear stress (A) and viscosity (B) vs. shear rate for the SiO<sub>2</sub>@Ni-Al LDH composites (particle concentration, 10 wt%)-based ER fluid.

## Methods

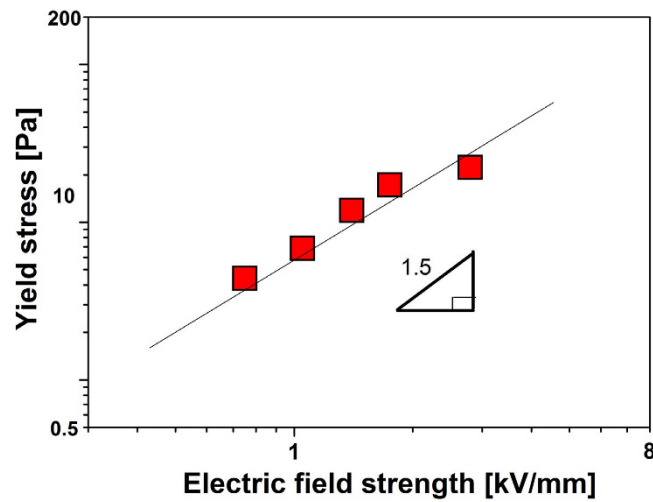
All chemical reagents were used as received from commercial sources without further purification. Ultrapure water was made by the Flom ultrapure water system and used in all experimental and washing processes.

**Synthesis of SiO<sub>2</sub> sphere and Ni-Al LDH sheets.** The TEOS ( $\geq 28\%$ , 4.37 g) was firstly added into ethanol ( $\geq 99.5\%$ , 50 mL), then the mixture of 2.23 g H<sub>2</sub>O, 7.7 mL ammonium (28%) and 40 ml ethanol was dropwise added into the above solution. After 20 hours' hydrolysis reaction, the mixture was separated in a high speed centrifugal machine to collect the precipitation. Then, the white SiO<sub>2</sub> sphere was dispersed in ultrapure water and centrifuged to collect the white precipitate, this process was repeated until the supernatant became clear. The sample was further dried under high vacuum to obtain the pure SiO<sub>2</sub> spheres. For the preparation of Ni-Al LDH sheet, certain amounts of Ni(NO<sub>3</sub>)<sub>2</sub>·6 H<sub>2</sub>O (99%, 1.308 g), Al(NO<sub>3</sub>)<sub>3</sub>·9 H<sub>2</sub>O (99%, 0.8622 g) and urea (99%, 0.9468 g) were added to 150 mL ultrapure water. Then the mixture was kept in an autoclave pressure vessel at 95 °C for 24 h. Finally, the solution were centrifuged and washed several times with ultrapure water. The clean Ni-Al LDH sheets were dried at 75 °C for 12 h in a vacuum oven. Before using, the Ni-Al LDH sheet dispersed in ethanol was broken using ultrasonic waves for 6 h.

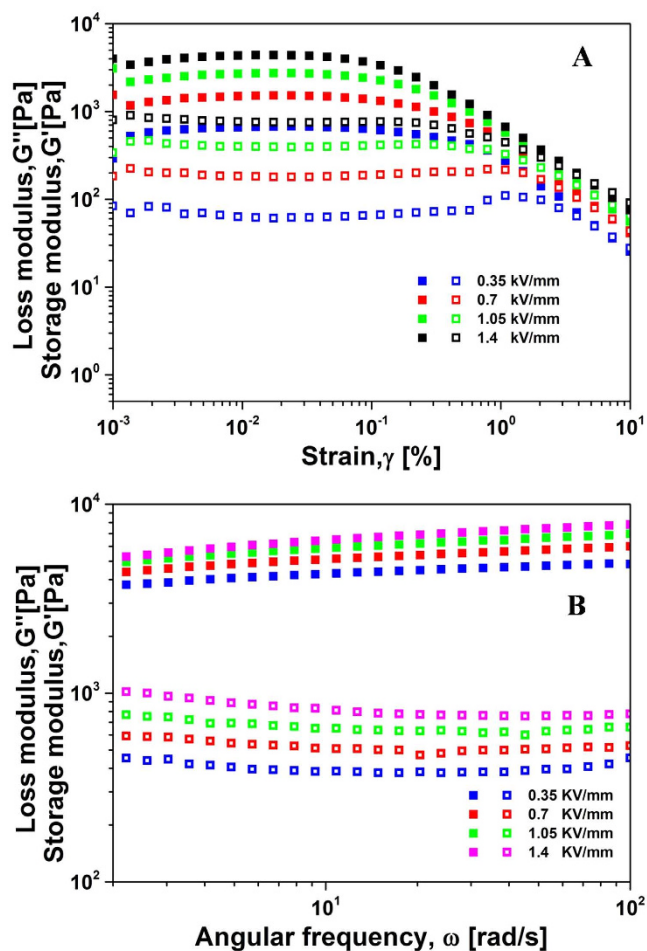
**Preparation of SiO<sub>2</sub>@Ni-Al LDH composites by self-assembled process.** SiO<sub>2</sub>@Ni-Al LDH composites were prepared by electrostatic adsorption between SiO<sub>2</sub> sphere and Ni-Al LDH sheet. SiO<sub>2</sub> sphere (0.1 g) was dispersed in 20 ml ethanol by ultrasonication for 15 min. The sodium dodecyl benzene sulfonate (SDBS) (98%, 0.01 g) was then added into the above solution, followed by magnetic stirring for 1 h. After centrifugation, the supernatant was discarded. The precipitation was dispersed in 20 ml ethanol under sonication. Dispersion of Ni-Al LDH sheet in ethanol (0.0044 g Ni-Al LDH sheet, 5 ml ethanol) was added into the above solution. The resulting mixture was stirred for 5 h and then centrifuged. The precipitate was collected and then dried at 60 °C for 12 h to afford SiO<sub>2</sub>@Ni-Al LDH composites.



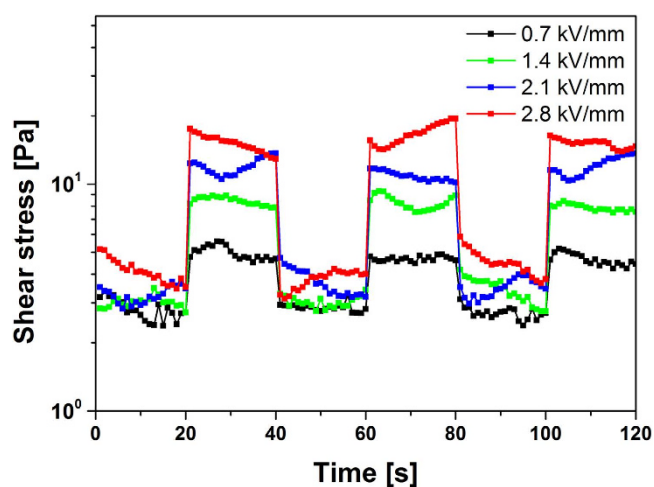
**Figure 6.** Flow curves of shear stress vs. shear rate for the single SiO<sub>2</sub> particle (A) and Ni–Al LDH (B) (concentration, 10 wt%)-based ER fluid.



**Figure 7.** Yield stress as a function of electric field strength for 10 wt % SiO<sub>2</sub>@Ni–Al LDH composite based ER fluid.



**Figure 8.** (A) Amplitude sweep for  $\text{SiO}_2$ @Ni-Al LDH composites (particle concentration, 10 wt%)-based ER fluid at  $6.28 \text{ rad s}^{-1}$  of angular frequency. (B) Frequency sweep for the  $\text{SiO}_2$ @Ni-Al LDH composites (particle concentration, 10 wt%)-based ER fluid with a strain amplitude of 0.01% ( $G'$ : closed symbols;  $G''$ : open symbols).



**Figure 9.** (A) Shear stress of  $\text{SiO}_2$ /Ni-Al LDH composites based ER fluids at a fixed shear rate of  $1 \text{ s}^{-1}$  in the electric field with a square voltage pulse ( $t = 20 \text{ s}$ ).



**Characterization.** Field-emission scanning electron microscopy (FE-SEM) (Model JSM-2010, JEOL) was used to observe the morphology of Ni-Al LDH sheet and SiO<sub>2</sub>@Ni-Al LDH composites. Transmission electron microscopy (TEM) (Model H-800, Hitachi, Tokyo, Japan) was performed to analyze the morphology of Ni-Al LDH sheets and SiO<sub>2</sub>@Ni-Al LDH composites. Scanning transmission electron microscopy (STEM) (Model JEM 2100F, JEOL) was carried out to characterize the distribution of the elements. X-ray photoelectron spectroscopy (XPS) tests were measured using an ESCALAB MK-II electron spectrometer with an Al K $\alpha$  X-ray source. X-ray diffraction (XRD) (Model DMax/rA, Rigaku, Japan) was performed using a rotating anode X-ray diffractometer with Cu K $\alpha$  radiation ( $\lambda = 0.154$  nm) at 40 kV. The ER characteristics were examined using a rotational rheometer (Physica MCR301, Anton Paar) equipped with a coaxial cylinder.

## Conclusions

The SiO<sub>2</sub>@Ni-Al LDH composites were successfully fabricated via self-assembly of Ni-Al LDH sheet on SiO<sub>2</sub> sphere based on electrostatic interactions. SEM, TEM, EDX, STEM, XRD and XPS analyses were carried out to confirm the successful preparation of the core-shell SiO<sub>2</sub>@Ni-Al LDH composites. The SiO<sub>2</sub>@Ni-Al LDH composites displayed an enhanced ER characteristic compared to bare SiO<sub>2</sub> suspension when dispersed in silicone oil, highlighting the potential applications of the SiO<sub>2</sub>@Ni-Al LDH composites in an automotive industry such as shock absorbers, engine mounts and clutches.

## References

- Kawasumi, M., Hasegawa, N., Kato, M., Usuki, A. & Okada, A. Preparation and mechanical properties of polypropylene-clay hybrids. *Macromolecules* **30**, 6333–6338 (1997).
- Ogata, N., Kawakage, S. & Ogihara, T. Structure and thermal/mechanical properties of poly (ethylene oxide)-clay mineral blends. *Polymer* **38**, 5115–5118 (1997).
- Sylwester, E., Hudson, E. & Allen, P. The structure of uranium (VI) sorption complexes on silica, alumina, and montmorillonite. *Geochim. Cosmochim. Acta* **64**, 2431–2438 (2000).
- Kim, J. *et al.* Synthesis and electrorheological characterization of polyaniline and Na<sup>+</sup>-montmorillonite clay nanocomposite. *Int. J. Mod. Phys. B* **15**, 657–664 (2001).
- Mann, S. The chemistry of form. *Angew. Chem. Int. Ed.* **39**, 3392–3406 (2000).
- Zampieri, A. *et al.* Biotemplating of Luffa cylindrica sponges to self-supporting hierarchical zeolite macrostructures for bio-inspired structured catalytic reactors. *Mater. Sci. and Eng.: C* **26**, 130–135 (2006).
- Shin, H.-C., Corno, J. A., Gole, J. L. & Liu, M. Porous silicon negative electrodes for rechargeable lithium batteries. *J. Power Sources* **139**, 314–320 (2005).
- Chia, S., Urano, J., Tamanoi, F., Dunn, B. & Zink, J. I. Patterned hexagonal arrays of living cells in sol-gel silica films. *J. Am. Chem. Soc.* **122**, 6488–6489 (2000).
- Evans, D. G. & Duan, X. Preparation of layered double hydroxides and their applications as additives in polymers, as precursors to magnetic materials and in biology and medicine. *Chem. Commun.*, 485–496 (2006).
- Zhao, Y., Li, F., Zhang, R., Evans, D. G. & Duan, X. Preparation of layered double-hydroxide nanomaterials with a uniform crystallite size using a new method involving separate nucleation and aging steps. *Chem. Mater.* **14**, 4286–4291 (2002).
- Khan, A. I. & O'Hare, D. Intercalation chemistry of layered double hydroxides: recent developments and applications. *J. Mater. Chem.* **12**, 3191–3198 (2002).
- Climent, M. J., Corma, A. & Iborra, S. Heterogeneous catalysts for the one-pot synthesis of chemicals and fine chemicals. *Chem. Rev.* **111**, 1072–1133 (2010).
- Greenwell, H. C., Holliman, P. J., Jones, W. & Velasco, B. V. Studies of the effects of synthetic procedure on base catalysis using hydroxide-intercalated layer double hydroxides. *Catal. Today* **114**, 397–402 (2006).
- Patzkó, Á. *et al.* ZnAl-layer double hydroxides as photocatalysts for oxidation of phenol in aqueous solution. *Colloid Surf. A-Physicochem. Eng. Asp.* **265**, 64–72 (2005).
- Sels, B. *et al.* Layered double hydroxides exchanged with tungstate as biomimetic catalysts for mild oxidative bromination. *Nature* **400**, 855–857 (1999).
- Dula, R. *et al.* Layered double hydroxide-derived vanadium catalysts for oxidative dehydrogenation of propane: Influence of interlayer-doping versus layer-doping. *Appl. Catal. A-Gen.* **230**, 281–291 (2002).
- Prasanna, S. V., Kamath, P. V. & Shivakumara, C. Synthesis and characterization of layered double hydroxides (LDHs) with intercalated chromate ions. *Mater. Res. Bull.* **42**, 1028–1039 (2007).
- Meyn, M., Beneke, K. & Lagaly, G. Anion-exchange reactions of layered double hydroxides. *Inorg. Chem.* **29**, 5201–5207 (1990).
- Liu, Z. *et al.* Synthesis, anion exchange, and delamination of Co-Al layered double hydroxide: assembly of the exfoliated nanosheet/polyanion composite films and magneto-optical studies. *J. Am. Chem. Soc.* **128**, 4872–4880 (2006).
- Fogg, A. M., Dunn, J. S. & O'Hare, D. Formation of Second-Stage Intermediates in Anion-Exchange Intercalation Reactions of the Layered Double Hydroxide [LiAl<sub>2</sub>(OH)<sub>6</sub>] Cl·H<sub>2</sub>O As Observed by Time-Resolved, *In Situ* X-ray Diffraction. *Chem. Mater.* **10**, 356–360 (1998).
- Kamath, P. V., Therese, G. H. A. & Gopalakrishnan, J. On the existence of hydrotalcite-like phases in the absence of trivalent cations. *J. Solid State Chem.* **128**, 38–41 (1997).
- Gao, Z. *et al.* Graphene nanosheet/Ni<sup>2+</sup>/Al<sup>3+</sup> layered double-hydroxide composite as a novel electrode for a supercapacitor. *Chem. Mater.* **23**, 3509–3516 (2011).
- Dong, Y., Liu, Y., Yin, J. & Zhao, X. Preparation and enhanced electro-responsive characteristic of graphene/layered double-hydroxide composite dielectric nanoplates. *J. Mater. Chem. C* **2**, 10386–10394 (2014).
- Leroux, F. *et al.* Percolation network of organo-modified layered double hydroxide platelets into polystyrene showing enhanced rheological and dielectric behavior. *J. Mater. Chem.* **20**, 9484–9494 (2010).
- Li, M. *et al.* Preparation, solubility, and electrorheological properties of carbon nanotubes/poly (methyl methacrylate) nanocomposites by *in situ* functionalization. *Compos Part A: Appl. Sci. Manufac* **40**, 413–417 (2009).
- Klopprogge, J. T., Komarneni, S. & Amonette, J. E. Synthesis of smectite clay minerals: a critical review. *Clays Clay Miner.* **47**, 529–554 (1999).
- Tao, R. & Jiang, Q. Simulation of structure formation in an electrorheological fluid. *Phys. Rev. Lett.* **73**, 205 (1994).
- Hu, H. *et al.* Microwave-assisted covalent modification of graphene nanosheets with chitosan and its electrorheological characteristics. *Appl. Surf. Sci.* **257**, 2637–2642 (2011).
- Kim, S. T., Lim, J. Y., Park, B. J. & Choi, H. J. Dispersion-Polymerized Carbon Nanotube/Poly (methyl methacrylate) Composite Particles and their Electrorheological Characteristics. *Macromol. Chem. Phys.* **208**, 514–519 (2007).
- Zhang, M. *et al.* Microdroplet-based universal logic gates by electrorheological fluid. *Soft Matter* **7**, 7493–7497 (2011).
- Yin, J., Chang, R., Shui, Y. & Zhao, X. Preparation and enhanced electro-responsive characteristic of reduced graphene oxide/polypropylene composite sheet suspensions. *Soft Matter* **9**, 7468–7478 (2013).

32. Yin, J., Chang, R., Kai, Y. & Zhao, X. Highly stable and AC electric field-activated electrorheological fluid based on mesoporous silica-coated graphene nanosheets. *Soft Matter* **9**, 3910–3914 (2013).
33. Yin, J., Wang, X., Chang, R. & Zhao, X. Polyaniline decorated graphene sheet suspension with enhanced electrorheology. *Soft Matter* **8**, 294–297 (2012).
34. Choi, S., Han, Y., Sohn, J. & Choi, H. Bingham characteristics of polymer-based electrorheological fluids with different electrode gaps and materials. *J. Appl. Polym. Sci.* **114**, 3636–3644 (2009).
35. Tan, K. *et al.* A simple one dimensional robot joint based on the ER linear reversing mechanism. *J. Intel. Mater. Syst. Str* **13**, 533–537 (2002).
36. Gamota, D. & Filisko, F. High frequency dynamic mechanical study of an aluminosilicate electrorheological material. *J. Rheol.* **35**, 1411–1425 (1991).
37. Filisko, F. E. & Radzilowski, L. H. An intrinsic mechanism for the activity of alumino-silicate based electrorheological materials. *J. Rheol.* **34**, 539–552 (1990).
38. Cho, M., Choi, H., Chin, I.-J. & Ahn, W.-S. Electrorheological characterization of zeolite suspensions. *Microporous Mesoporous Mater.* **32**, 233–239 (1999).
39. Zhang, W., Choi, H. & Leong, Y. Facile fabrication of graphene oxide-wrapped alumina particles and their electrorheological characteristics. *Mater. Chem. Phys.* **145**, 151–155 (2014).
40. Hong, J.-Y. & Jang, J. A comparative study on electrorheological properties of various silica-conducting polymer core-shell nanospheres. *Soft Matter* **6**, 4669–4671 (2010).
41. Yin, J., Zhao, X., Xiang, L., Xia, X. & Zhang, Z. Enhanced electrorheology of suspensions containing sea-urchin-like hierarchical Cr-doped titania particles. *Soft Matter* **5**, 4687–4697 (2009).
42. Liu, Y. D., Park, B. J., Kim, Y. H. & Choi, H. J. Smart monodisperse polystyrene/polyaniline core-shell structured hybrid microspheres fabricated by a controlled releasing technique and their electro-responsive characteristics. *J. Mater. Chem.* **21**, 17396–17402 (2011).
43. Goodwin, J. W., Markham, G. M. & Vincent, B. Studies on model electrorheological fluids. *J. Phys. Chem. B* **101**, 1961–1967 (1997).
44. Cho, M. S., Choi, H. J. & To, K. Effect of ionic pendent groups on a polyaniline-based electrorheological fluid. *Macromol. Rapid Commun.* **19**, 271–273 (1998).
45. Choi, H. J., Cho, M. S. & Jhon, M. S. Electrorheological properties of poly (acene quinone) radical suspensions. *Polym. Adv. Technol.* **8**, 697–700 (1997).
46. Yin, J., Shui, Y., Dong, Y. & Zhao, X. Enhanced dielectric polarization and electro-responsive characteristic of graphene oxide-wrapped titania microspheres. *Nanotechnology* **25**, 045702 (2014).
47. Zhang, W. L. & Choi, H. J. Fast and facile fabrication of a graphene oxide/titania nanocomposite and its electro-responsive characteristics. *Chem. Commun.* **47**, 12286–12288 (2011).
48. Zhang, W. L., Park, B. J. & Choi, H. J. Colloidal graphene oxide/polyaniline nanocomposite and its electrorheology. *Chem. Commun.* **46**, 5596–5598 (2010).
49. Oh, S. Y. & Kang, T. J. Electrorheological response of inorganic-coated multi-wall carbon nanotubes with core-shell nanostructure. *Soft Matter* **10**, 3726–3737 (2014).
50. Liu, Y. D. & Choi, H. J. Electrorheological fluids: smart soft matter and characteristics. *Soft Matter* **8**, 11961–11978 (2012).
51. Zhang, W. L., Liu, Y. D. & Choi, H. J. Graphene oxide coated core-shell structured polystyrene microspheres and their electrorheological characteristics under applied electric field. *J. Mater. Chem.* **21**, 6916–6921 (2011).
52. Liu, Y. D., Fang, F. F. & Choi, H. J. Core-shell structured semiconducting PMMA/polyaniline snowman-like anisotropic microparticles and their electrorheology. *Langmuir* **26**, 12849–12854 (2010).
53. Wang, B. *et al.* Double template assisting synthesized core-shell structured titania/polyaniline nanocomposite and its smart electrorheological response. *Compos. Sci. Technol.* **86**, 89–100 (2013).
54. Wu, J. *et al.* Preparation and electrorheological characteristics of uniform core/shell structural particles with different polar molecules shells. *Colloid Surf. A-Physicochem. Eng. Asp.* **410**, 136–143 (2012).
55. Zhang, K., Lim, J., Choi, H. & Seo, Y. Core-shell structured carbon nanotube/poly (methyl methacrylate) composite and its electrorheological characteristics. *Diamond Relat. Mater.* **17**, 1604–1607 (2008).
56. Rosi, N. L., Thaxton, C. S. & Mirkin, C. A. Control of Nanoparticle Assembly by Using DNA-Modified Diatom Templates. *Angew. Chem.* **116**, 5616–5619 (2004).
57. Yan, N. *et al.* Hollow porous SiO<sub>2</sub> nanocubes towards high-performance anodes for lithium-ion batteries. *Sci. Repor.* **3**, 1568 (2013).
58. Kim, S. D., Zhang, W. L., Choi, H. J., Seo, Y. P. & Seo, Y. Electrorheological activity generation by graphene oxide coating on low-dielectric silica particles. *RSC. Adv.* **4**, 62644–62650 (2014).
59. Cho, M. S., Cho, Y. H., Choi, H. J. & Jhon, M. S. Synthesis and electrorheological characteristics of polyaniline-coated poly (methyl methacrylate) microsphere: size effect. *Langmuir* **19**, 5875–5881 (2003).
60. Zhu, J. *et al.* *In situ* stabilized carbon nanofiber (CNF) reinforced epoxy nanocomposites. *J. Mater. Chem.* **20**, 4937–4948 (2010).
61. Cho, M. S., Choi, H. J. & Ahn, W.-S. Enhanced electrorheology of conducting polyaniline confined in MCM-41 channels. *Langmuir* **20**, 202–207 (2004).

## Acknowledgements

This work was supported by the National Natural Science Foundation of China (51173087), and Qingdao (12-1-4-2-2-jch) and Taishan Scholars Program and for financial support.

## Author Contributions

X.J., W.Z. and J.L. conceived and designed the experiments. X.J., W.Z., L.S. and Y.T. executed the experiments. X.J., W.Z., L.S., Y.T. and J.L. analyzed the results. Y.T. provided the equipment to measure the rheological characteristics. X.J., W.Z., Y.T. and J.L. composed the paper. All authors reviewed the manuscript.

## Additional Information

**Competing financial interests:** The authors declare no competing financial interests.

**How to cite this article:** Ji, X. *et al.* Self-assembly preparation of SiO<sub>2</sub>@Ni-Al layered double hydroxide composites and their enhanced electrorheological characteristics. *Sci. Rep.* **5**, 18367; doi: 10.1038/srep18367 (2015).



This work is licensed under a Creative Commons Attribution 4.0 International License. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in the credit line; if the material is not included under the Creative Commons license, users will need to obtain permission from the license holder to reproduce the material. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>