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Investigation on the organic–inorganic hybrid crystal $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$: structure, phase transition, thermal property, structural geometry, and dynamics

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Understanding the physical properties of the organic–inorganic hybrid $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ is essential to expand its applications. The single $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystals were grown and their comprehensive properties were investigated. The crystals had a monoclinic structure with the space group $P2_1/n$ and lattice constants of $a = 8.8651$ (5) Å, $b = 11.9938$ (6) Å, $c = 13.3559$ (7) Å, and $\beta = 91.322^\circ$. The transition temperature from phase I to phase II was determined to be 388 K. Variations in the ^1H nuclear magnetic resonance chemical shifts of NH_2 and ^{14}N NMR chemical shifts according to the temperature changes in the cation were attributed to vibrations of NH_2 groups at their localization sites. The ^1H and ^{13}C spin–lattice relaxation times ($T_{1\rho}$) in phase II changed significantly with temperature, indicating that these values are governed by molecular motion. The $T_{1\rho}$ values were much longer in phase I than in phase II, which means energy transfer was difficult. Finally, the activation energies for phases I and II were considered. According to the basic mechanism of $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystals, organic–inorganic materials may have potential applications in various fields.

Significant attention has been paid to organic–inorganic hybrid compounds owing to their diverse applications as catalysts, sensors, functional smart coatings, fuel and solar cells, light-emitting diodes, light-emitting transistors, and perovskite photovoltaic cells^{1–12}. Perovskite type organic–inorganic metal halides are lead-based $\text{CH}_3\text{NH}_3\text{PbX}_3$ ($X = \text{Cl}, \text{Br}, \text{I}$) compounds with a 3D structure. However, $\text{CH}_3\text{NH}_3\text{PbX}_3$ -based photovoltaics demonstrate high instability under typical environmental conditions, notably moisture, as well as high toxicity owing to the bioaccumulation of Pb ^{13–18}. Thus, researchers have suggested the substitution of Pb with other low-toxicity or eco-friendly metals such as Cu and Zn to develop Pb -free photovoltaic devices¹⁹. The urgent need to develop eco-friendly hybrid perovskite solar cells has recently been highlighted. For example, two-dimensional $[\text{NH}_3(\text{CH}_2)_n\text{NH}_3]\text{BX}_4$ ($n = 2, 3, \dots, B = \text{Mn}, \text{Co}, \text{Cu}, \text{Zn}, \text{Cd}; X = \text{Cl}, \text{Br}, \text{I}$) and $[(\text{C}_n\text{H}_{2n+1}\text{NH}_3)]_2\text{BX}_4$ organic–inorganic hybrids have recently been proposed as an alternative to these materials^{20–27}. In addition, it is needed to study for $[\text{NH}_2(\text{CH}_3)_n]_2\text{BX}_4$ ^{28–33}, which has a different hydrogen-bond structure from $[\text{NH}_3(\text{CH}_2)_n\text{NH}_3]\text{BX}_4$, which has three H atoms bonded to one N. The overall structure of these materials consists of a 1D inorganic network. The isolated building blocks in these 1D perovskites provide a large degree of freedom for the dynamic motion of organic ammonium cations, which can trigger a disorder-to-order transition. Such materials are expected to act as proton conductors owing to the availability of hydrogen bonds.

Dimethylammonium tetrabromocuprate (II) $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ is a member of the $[\text{NH}_2(\text{CH}_3)_n]_2\text{BX}_4$ family. In this group of compounds, the individual BX_4 tetrahedral anions are isolated and surrounded by $[\text{NH}_2(\text{CH}_3)_2]^+$ cations. However, the properties and characteristics of $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ have not yet been reported.

In this study, the single crystals of $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ were grown using an aqueous solution method, and their structure, the phase-transition temperature (T_c), and the thermal property was considered. The coordination geometries around ^1H , ^{13}C , and ^{14}N atoms were investigated by obtaining the chemical shifts of the ^1H

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magic-angle spinning (MAS) nuclear magnetic resonance (NMR), ^{13}C MAS NMR, and static ^{14}N NMR as functions of temperature. In addition, the ^1H and ^{13}C NMR spin–lattice relaxation times ($T_{1\rho}$), which represent the energy transfer surrounding the ^1H and ^{13}C atoms of the cation, respectively, were discussed, and their activation energies (E_a) were determined. The single-crystal structure and physical properties observed in this study are expected to provide important information on the basic mechanism for various applications.

Methods

Crystal growth

Single crystals of $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ were prepared by dissolving dimethylammonium bromide (Aldrich, 98%) and CuBr_2 (Aldrich, 98%) at a ratio of 2:1 in tertiary distilled water. To make a saturated solution, the mixed material was heated, stirred, and filtered through filter paper. The prepared saturated solution was placed in a beaker, covered with filter paper, to let natural evaporation in an apparatus with a constant temperature of 300 K. Several dark brown colored single crystals with sizes of $6 \times 3 \times 1$ mm were obtained after a few days.

Characterization

The structure and lattice parameters of the $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystals were determined at 300 K using the single-crystal X-ray diffraction (SCXRD) system of the Seoul Western Center at the Korea Basic Science Institute (KBSI). SCXRD measurements were performed using a diffractometer with a graphite-monochromated Mo-K α target with a wavelength of 0.71073 Å under a cold nitrogen flow (-50 °C) (Bruker D8 Venture PHOTON III M14). The data was collected using SMART APEX3 (Bruker 2016) and SAINT (Bruker, 2016). The crystal structure was solved using direct methods and refined using the full-matrix least-squares method³⁴. All hydrogen atoms are presented in their geometric positions. Additionally, powder X-ray diffraction (PXRD) patterns were measured using an XRD system at 300 K with the same target used for SCXRD.

Differential scanning calorimetry (DSC) results were performed in the temperature range of 200–423 K using a DSC instrument (TA Instruments, DSC 25) under a nitrogen atmosphere. The thermogram was measured using the sample of 6.1 mg at a heating rate of 10 °C/min.

Thermogravimetric analysis (TGA) results were performed at a heating rate of 10 °C/min under dry nitrogen gas. The thermogram was collected using a 15.92 mg sample while heating from room temperature to 900 K.

The MAS NMR chemical shifts and spin–lattice relaxation time $T_{1\rho}$ of the $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystals were measured using a solid-state NMR spectrometer (AVANCE II+, Bruker) at the Seoul Western Center of the KBSI. The Larmor frequency for the ^1H NMR experiment was 400.13 MHz, while that for the ^{13}C NMR experiment was 100.61 MHz. The samples were placed in a cylindrical zirconia rotor and then subjected to MAS NMR measurements at a spinning rate of 10 kHz to reduce the spinning sidebands. Chemical shifts were measured using adamantane and tetramethylsilane as the standards for ^1H and ^{13}C , respectively. The 1D NMR spectra of ^1H and ^{13}C were obtained with a delay time of 0.2–2 s. To obtain $T_{1\rho}$ values, the one-pulse method was used, and the delay times were within 0.4–4 s, and the 90° pulse for ^1H and ^{13}C was used to 3.45–7.5 μs and 5–5.5 μs , respectively. The ^{13}C $T_{1\rho}$ values were obtained by varying the duration of the ^{13}C spin-locking pulse applied after cross-polarization (CP) preparation. Static ^{14}N NMR chemical shifts were recorded using the one-pulse method at a Larmor frequency of 28.90 MHz with NH_4NO_3 as the standard sample. NMR experiments above 430 K were not possible because of the limitations of the NMR instrument. The temperature was maintained nearly constant within the error range of ± 0.5 °C, even when the N_2 gas flow rate and heater current were adjusted.

Results and discussions

Crystal structure

The SCXRD results of the $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystals were obtained at 300 K. The crystals had a monoclinic structure with the $P2_1/n$ space group and lattice constants of $a = 8.8651$ (5) Å, $b = 11.9938$ (6) Å, $c = 13.3559$ (7) Å, $\beta = 91.322$ (2)°, and $Z = 4$. A perspective view of the atomic arrangement in the unit cell of a $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystal is shown in Fig. 1a,b. Here, the $[\text{CuBr}_4]^{2-}$ tetrahedra and $[\text{NH}_2(\text{CH}_3)_2]$ cations are linked by hydrogen bonds. Specifically, the crystal structure consists of discrete, slightly deformed CuBr_4 tetrahedra linked to the organic cations through N–H...Br hydrogen bonds^{28–30,33}. The SCXRD data for the $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystals are shown in Table 1, and the corresponding bond lengths and angles are presented in Table 2. The CIF file result of SCXRD for crystal structure at 300 K is shown in the Supplementary information S1.

In addition, PXRD experiments were performed at 300 K. The PXRD patterns obtained in the 2 θ measurement range of 8°–50° are shown in red color in Fig. 2. The PXRD pattern simulated by SCXRD structural parameters was consistent with that determined from the PXRD experiment at 300 K. The peaks observed in this diffractogram were indexed using the Mercury program as shown in Fig. 2.

Phase transition temperature

The DSC thermogram of the crystals was measured in the temperature range of 200–423 K at a heating rate of 10 °C/min. Figure 3 shows a strong endothermic peak at 388 K with an enthalpy of 22.26 kJ/mol. The two phases were denoted as phase I, which refers to the region above 388 K, and phase II, which refers to the region below 388 K.

Thermal property

The TGA curves shown in Fig. 4 were measured as the temperature increased. In the TGA curve, the partial decomposition temperature (T_d) represents a weight loss of 2% at 432 K, which means the material is thermally stable up to 432 K. The molecular weight of the $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystal decreased abruptly as the temperature increased owing to partial decomposition. The amounts of the sample remaining from the partial decomposition

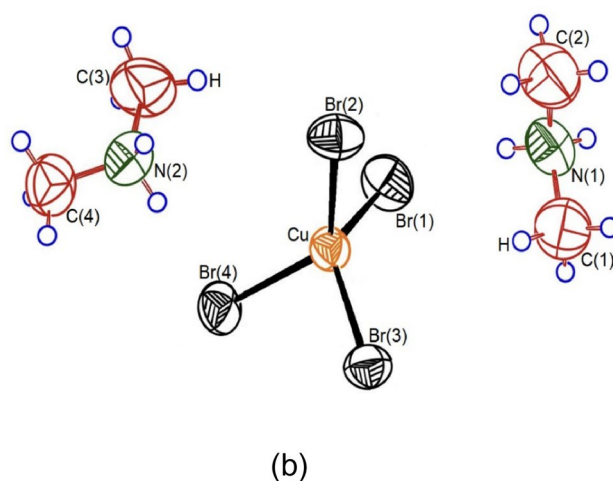
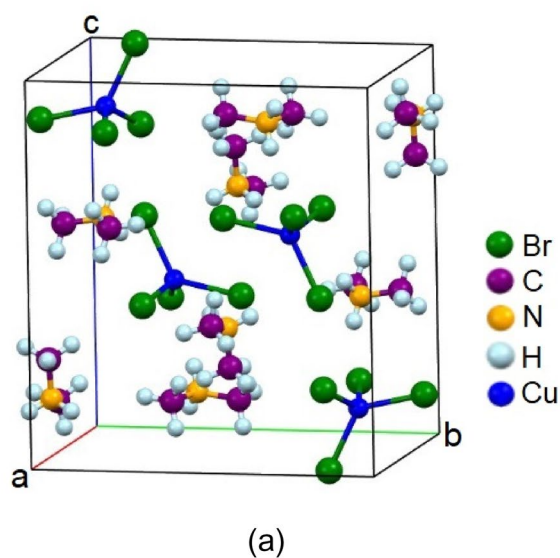
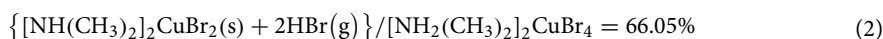


Figure 1. (a) Structure of the $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystal at 300 K (CCDC No. 2290529). (b) Thermal ellipsoid plot for $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystal structure at 300 K.

of HBr and 2HBr were obtained from a total molecular weight of 475.37 mg using the TGA data and following chemical reactions^{35,36}.



Molecular weight losses of 27% and 34%, which were attributed to the decomposition of HBr and 2HBr, respectively, were observed. The initial weight loss of 27% occurred at 545 K, while the second weight loss of 34% occurred in the range of 577 K.

The endothermic peak at 390 K observed in the DTA curve, which is presented as the differential form of the TGA curve, was in good agreement with the T_C shown in the DSC results. Complete weight loss occurred at temperatures above 850 K. Morphology in the crystals with increasing temperature were confirmed by optical polarizing microscopy to understand their thermal properties based on the TGA results. As shown in the inset in Fig. 4, as the temperature increases from 300 to 373 K, the crystal morphology does not change. However, when the temperature reaches 430 K, the single crystal begins to melt, and a considerable amount of it melts at 490 K. From the DSC, TGA, and polarizing microscopy experiments, the T_C , T_d , and melting temperature of the crystals were determined to be $T_C = 388$ K, $T_d = 432$ K, and $T_m = 490$ K, respectively.

Chemical formula	C ₄ H ₁₆ N ₂ CuBr ₄
Weight	475.37
Crystal structure	Monoclinic
Space group	<i>P2₁/n</i>
T (K)	300
<i>a</i> (Å)	8.8651 (5)
<i>b</i> (Å)	11.9938 (6)
<i>c</i> (Å)	13.3559 (7)
α (°)	90
β (°)	91.322 (2)
γ (°)	90
Z	4
V (Å ³)	1419.70
Wavelength (Å)	0.71073
θ range for data collection (°)	2.283–28.331
Reflections collected	29,993
Independent reflections	3535 (<i>R</i> _{int} = 0.0951)
Goodness-of-fit on <i>F</i> ²	0.990
Final <i>R</i> indices [<i>I</i> > 2σ(<i>I</i>)]	<i>R</i> ₁ = 0.0408, <i>wR</i> ₂ = 0.0690
<i>R</i> indices (all data)	<i>R</i> ₁ = 0.1156, <i>wR</i> ₂ = 0.0861

Table 1. Structure of single crystal for [NH₂(CH₃)₂]₂CuBr₄ at 300 K (CCDC No. 2290529).

Cu–Br(1)	2.3511 (8)
Cu–Br(2)	2.3772 (8)
Cu–Br(3)	2.3917 (9)
Cu–Br(4)	2.3918 (8)
N(1)–C(1)	1.447 (7)
N(1)–C(2)	1.456 (7)
N(2)–C(3)	1.437 (7)
N(2)–C(4)	1.461 (7)
N–H	0.89
C–H	0.96
Br(1)–Cu–Br(2)	100.02 (3)
Br(1)–Cu–Br(3)	131.86 (4)
Br(2)–Cu–Br(3)	101.03 (3)
Br(1)–Cu–Br(4)	100.43 (3)
Br(2)–Cu–Br(4)	128.13 (4)
Br(3)–Cu–Br(4)	99.62 (3)

Table 2. Bond-lengths (Å) and bond-angles (°) for [NH₂(CH₃)₂]₂CuBr₄ at 300 K.

¹H and ¹³C MAS NMR chemical shifts

The ¹H NMR chemical shifts of the [NH₂(CH₃)₂]₂CuBr₄ crystals were recorded at phases I and II as shown in Fig. 5. As expected, the two ¹H signals of NH₂ and CH₃ in the cation were detected. At low temperatures, the ¹H NMR spectra of NH₂ and CH₃ completely overlapped and only one signal was obtained. The two ¹H signals corresponding to NH₂ and CH₃ began to separate at temperatures above 260 K. The ¹H chemical shift for NH₂ at 300 K was recorded at 6.81 ppm, while that for CH₃ was obtained at 4.67 ppm. In phase II below *T*_C, changes in the chemical shifts with increasing temperature are indicated by dotted lines. The ¹H chemical shifts for CH₃ were independent of temperature, whereas those for NH₂ shifted positively with increasing temperature. The two ¹H chemical shifts for NH₂ and CH₃ changed discontinuously near *T*_C.

Furthermore, the linewidths of the ¹H spectrum in phase II were very broad because the two signals overlapped, whereas those in phase I were relatively thin owing to their complete separation. The sidebands for ¹H in NH₂ and CH₃ are represented by open circles and asterisks, respectively. A disadvantage of spinning is that it may lead to the presence of spinning sidebands. These are spurious signals (i.e. peaks) that result from the modulation of the magnetic field at the spinning frequency. The peaks always appear on either side of any large genuine peak at a separation equal to the spinning rate. Near *T*_C, the linewidths rapidly changed from a Gaussian shape to a Lorentzian one.

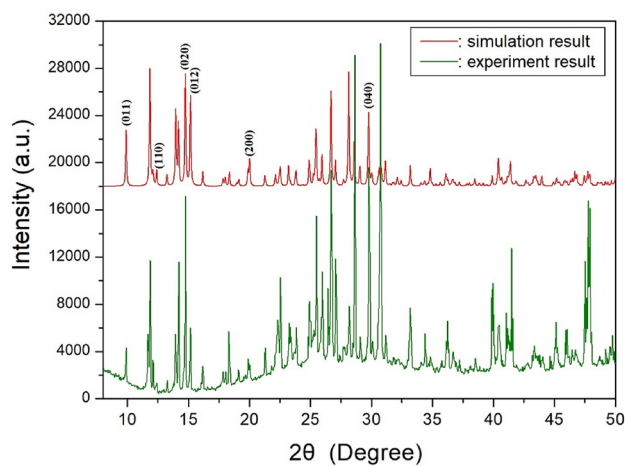


Figure 2. Powder XRD and simulated XRD patterns of $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ at 300 K.

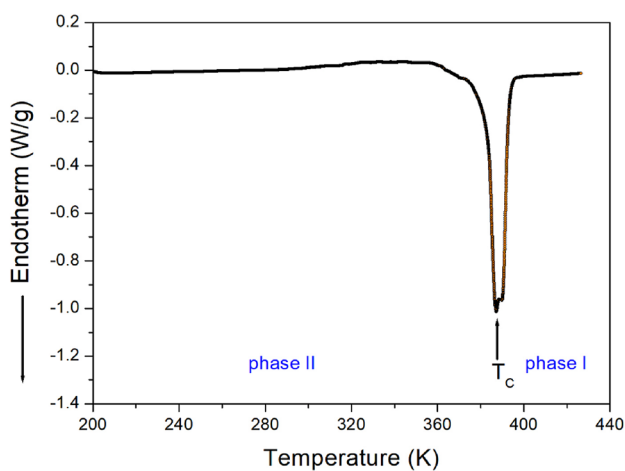


Figure 3. Differential scanning calorimetry (DSC) curve of $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ measured at a heating rate of $10^\circ\text{C}/\text{min}$.

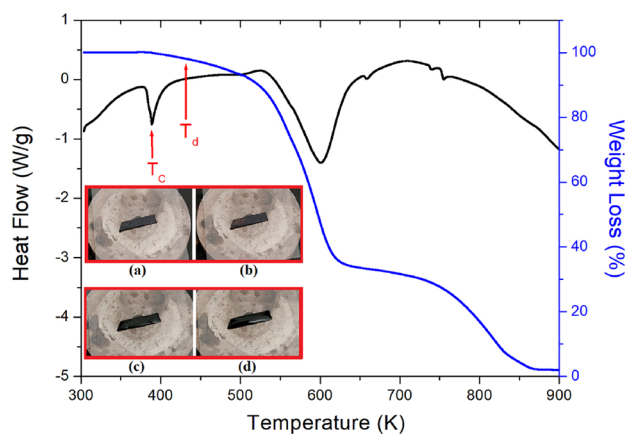


Figure 4. Thermogravimetric analysis (TGA) and differential thermal analysis (DTA) results of $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ (Inset: Morphology in crystal by polarizing microscopy at (a) 300 K, (b) 373 K, (c) 430 K, and (d) 490 K for $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$).

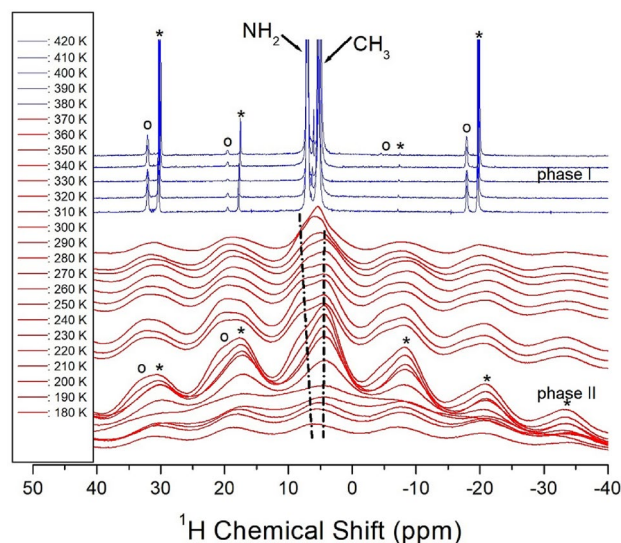


Figure 5. ^1H MAS NMR chemical shifts of NH_2 and CH_3 in $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ at phases II and I. The open circles and asterisks are denoted sidebands for ^1H in NH_2 and CH_3 , respectively.

The ^{13}C NMR chemical shifts of $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ were also measured in phases I and II as a function of temperature, as shown in Fig. 6. Because the signals above 340 K could not be well detected, chemical shifts above this temperature were measured using the one-pulse method. In phases I and II, only one ^{13}C NMR signal was observed, and one signal of ^{13}C at 48.10 ppm was observed at 180 K. At T_c values above 380 K, this signal suddenly shifted to 82.32 ppm. This abrupt shift is related to the phase transition caused by structural changes between phases I and II. The changes in the ^{13}C NMR chemical shifts reflect changes in the coordination geometry around ^{13}C .

Static ^{14}N NMR chemical shifts

The static NMR spectra of ^{14}N in NH_2 at the center of the cation in the $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ single crystals were recorded as a function of temperature in the range of 200–430 K, and the longest direction of the single crystal and the applied magnetic field of 9.4 T were measured in the directions perpendicular to each other. ^{14}N has a spin number of 1, and its signals are expected to show two resonances owing to its quadrupole interactions³⁷. The ^{14}N NMR spectrum was challenging to obtain because of the low Larmor frequency (28.90 MHz). The chemical shifts in the ^{14}N NMR spectra at various temperatures are shown in Fig. 7. The structural geometries of N(1) and N(2) in the two $[\text{NH}_2(\text{CH}_3)_2]^+$ groups were determined based on the ^{14}N NMR chemical shifts. The resonance pairs for ^{14}N are indicated by the same symbol. The N(1) chemical shifts represented by red squares decrease slightly with increasing temperature, whereas the N(2) chemical shifts represented by blue circles decrease

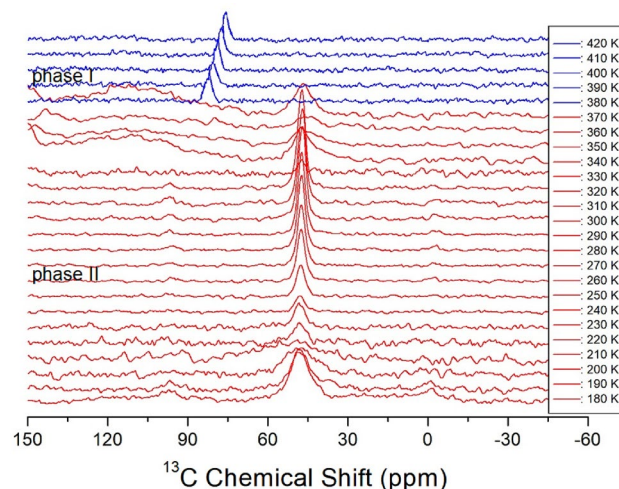


Figure 6. ^{13}C MAS NMR chemical shifts in $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ at phases II and I.

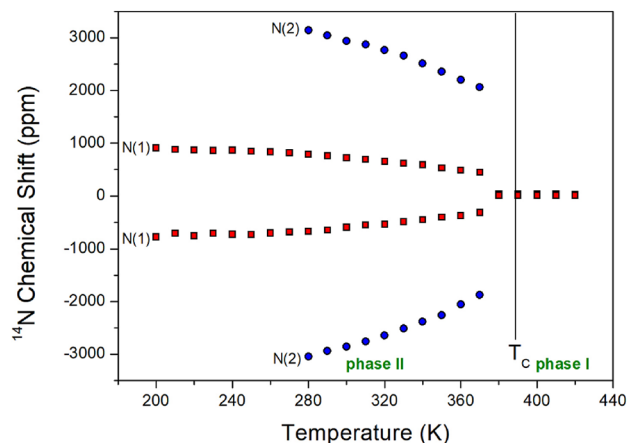


Figure 7. Static ^{14}N NMR chemical shifts in $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ at phases II and I.

abruptly. The $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ structure consists of complex CuBr_4 anions and two $[\text{NH}_2(\text{CH}_3)_2]$ cations, as shown in Fig. 1. The ^{14}N NMR spectrum of N(2) was difficult to detect at low temperatures because of the wide area outside the observed chemical shift range. In phase I, the four ^{14}N spectra of the two sets of signals were reduced to two ^{14}N spectra of only one set. The abrupt changes in the N(1) and N(2) chemical shifts between phases I and II indicate changes in the coordination geometry of the surrounding environments around ^{14}N ; N(1) and N(2) with different surrounding environments exist in phase II, whereas only one N site with the same surroundings exists in phase I; temperature changes in ^{14}N NMR static chemical shifts may be associated with vibrations of NH_2 groups at their localization sites.

^1H and ^{13}C NMR spin–lattice relaxation times

The intensities of the ^1H and ^{13}C NMR spectral peaks were measured according to the increase in delay time to obtain $T_{1\rho}$. The decay curves of the changes in intensities and delay times are represented by the following equation^{37–39}:

$$I(t) = I(0)\exp(-t/T_{1\rho}), \quad (3)$$

where $I(t)$ is the intensities of the peaks at time t and $I(0)$ is the intensities of the peaks at time $t=0$. The $T_{1\rho}$ values for ^1H and ^{13}C in $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ were obtained using Eq. (3), and the results are shown in Figs. 8 and 9, respectively, as functions of the inverse temperature.

In the case of ^1H , the chemical shifts for ^1H in NH_2 and CH_3 in phase II were not completely separated, thus, their $T_{1\rho}$ was obtained as a single value. The $T_{1\rho}$ values were strongly dependent on temperature, and $T_{1\rho}$ rapidly decreased as the temperature increased, reaching a minimum value of 0.47 ms at 230 K, as shown in Fig. 8. $T_{1\rho}$ is lengthened again with further increases in temperature, indicating molecular motion according to Bloembergen–Purcell–Pound theory³⁸. The minimum value of $T_{1\rho}$ is clearly due to the reorientational motion of ^1H in NH_2 and CH_3 . In the case of ^{13}C , the $T_{1\rho}$ values shown in Fig. 9 decrease as temperature increases and

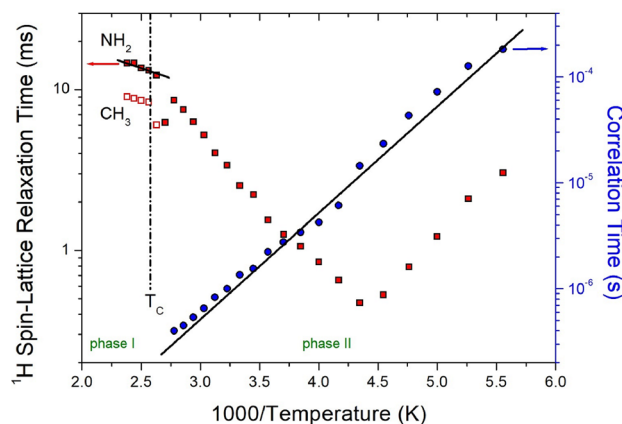


Figure 8. ^1H NMR spin–lattice relaxation times in $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ at phases II and I. The slope of solid line at phases II and I is represented the activation energy by the correlation times and the relaxation times as a function of inverse temperature, respectively.

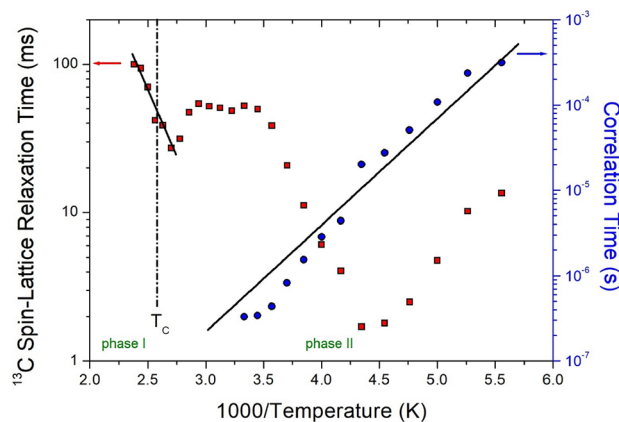


Figure 9. ^{13}C NMR spin–lattice relaxation times in $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ at phases II and I. The slope of solid line at phases II and I is represented the activation energy by the correlation times and the relaxation times as a function of inverse temperature, respectively.

increase abruptly at temperatures above 230 K. Compared with the $T_{1\rho}$ of ^1H , the $T_{1\rho}$ of ^{13}C showed a minimum value of 1.70 ms at 230 K, and the ^{13}C $T_{1\rho}$ pattern indicated active molecular motion. In addition, the $T_{1\rho}$ values of ^1H and ^{13}C rapidly increased in phase I. However, the $T_{1\rho}$ of ^{13}C was approximately 10 times longer than that of ^1H . In phase II, the experimental values of $T_{1\rho}$ can be expressed by the correlation time τ_C for reorientational motion and are given by^{39,40}:

$$\begin{aligned} (1/T_{1\rho}) = & R\{4\tau_C/[1 + \omega_C^2\tau_C^2] + \tau_C/[1 + (\omega_C - \omega_H)^2\tau_C^2] + 3\tau_C/[1 + \omega_C^2\tau_C^2] \\ & + 6\tau_C/[1 + (\omega_C + \omega_H)^2\tau_C^2] + 6\tau_C/[1 + \omega_H^2\tau_C^2]\}, \end{aligned} \quad (4)$$

where R is a constant; ω_1 is the spin-lock field; and ω_C and ω_H are the Larmor frequencies for carbon and protons, respectively. The τ_C value can be obtained from the condition that $T_{1\rho}$ is at a minimum when ω_1 is 1. As the $T_{1\rho}$ curves exhibited minima, the coefficient R in Eq. (4) can be obtained. Using this R , based on the $T_{1\rho}$ values observed over the temperature range investigated and the frequency power of ω_1 given in the experiment, the τ_C value at various temperatures could be obtained. The ω_1 values for ^1H and ^{13}C were 69.44 and 50 kHz, respectively. Local field fluctuations are caused by thermal motion, which is activated by thermal energy. τ_C is generally expressed as an Arrhenius-type equation based on the E_a for molecular motion and temperature as follows^{36,37}:

$$\tau_C = \tau_C^0 \exp(-E_a/k_B T) \quad (5)$$

where E_a is the activation energy and k_B is the Boltzmann constant. The magnitude of E_a depends on molecular dynamics. The logarithmic scale of τ_C represented by blue circles versus $1000/T$ is shown in Figs. 8 and 9 to determine the molecular dynamics of the crystal. In phase II, the activation barriers are 19.38 ± 1.70 kJ/mol (^1H) and 23.88 ± 4.48 kJ/mol (^{13}C), and their values are the same within the experimental error. The phenomenon showing the minimum relaxation time for both $T_{1\rho}$ (^1H) and $T_{1\rho}$ (^{13}C) is related to the same relaxation process, namely the reorientation of CH_3 around its own $\text{C}3$ axis. On the other hand, the changes in the relaxation times for Arrhenius-type random motions as functions of $T_{1\rho}$ for ^1H and ^{13}C in phase I are described in terms of slow motions; for $\tau_C \ll \omega_C$ (or ω_H), $T_{1\rho} \sim \tau_C = \tau_0 \exp(-E_a/k_B T)$, where ω_C (or ω_H) denotes the Larmor frequency. The E_a values for ^1H and ^{13}C obtained from the logarithmic scale of $T_{1\rho}$ in phase I were 5.11 ± 1.24 and 33.36 ± 9.90 kJ/mol, respectively. The difference in E_a values between phases I and II was greater for ^1H than for ^{13}C .

Conclusion

The physicochemical properties of organic–inorganic hybrid $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystals were investigated in this study. The monoclinic structure of this crystal was determined using SCXRD, and its phase transition temperature T_C was determined as 388 K. Its thermal stability at approximately 432 K was not good. ^1H , ^{13}C , and ^{14}N NMR spectroscopy provided valuable information on the hydrogen, carbon, and nitrogen environments, as well as their connectivity within the crystal molecule; the ^{13}C and ^{14}N NMR chemical shifts abruptly changed near the T_C , thus suggesting that the surrounding environment changes with temperature. By contrast, the ^1H chemical shifts for NH_2 rather than CH_3 changed rapidly over the temperature range investigated. The variations in the ^1H NMR chemical shifts for NH_2 and ^{14}N NMR chemical shifts according to the changes in temperature in the cation were associated to vibrations of NH_2 groups at their localization sites. Finally, the ^1H and ^{13}C $T_{1\rho}$ values, which represent the extent of energy transfer surrounding the ^1H and ^{13}C atoms of the cation, changed significantly with the temperature in phase II, indicating that these values are governed by the large degree of freedom for the molecular motions of organic cations. The $T_{1\rho}$ values for ^1H and ^{13}C indicated similar molecular motions, but the $T_{1\rho}$ values in phase I were much longer than those in phase II, indicating difficulties in energy transfer. The ^{13}C E_a determined from the $T_{1\rho}$ results for molecular motion was larger than the ^1H E_a , and the E_a

for ^1H and ^{13}C near the T_C changes greater. Based on their basic mechanism of $[\text{NH}_2(\text{CH}_3)_2]_2\text{CuBr}_4$ crystals, it is expected that the potential applications in various fields will be possible.

Data availability

The datasets used and/or analysed during the current study are available from the corresponding author on reasonable request.

Received: 30 August 2023; Accepted: 21 November 2023

Published online: 29 November 2023

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Acknowledgements

This work was supported by the National Research Foundation of Korea (NRF) grant, funded by the Korean government (MSIT) (2023R1A2C2006333). The work was supported by the Basic Science Research Program of the National Research Foundation of Korea (NRF), funded by the Ministry of Education, Science, and Technology (2016R1A6A1A03012069).

Author contributions

A.R.L. designed the project, NMR experiment, and wrote the manuscript. C.N. performed X-ray experiment.

Competing interests

The authors declare no competing interests.

Additional information

Supplementary Information The online version contains supplementary material available at <https://doi.org/10.1038/s41598-023-48015-6>.

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