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Magnetic control of transverse electric polarization in BiFeO_3

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Numerous attempts have been made to realize crossed coupling between ferroelectricity and magnetism in multiferroic materials at room temperature. BiFeO_3 is the most extensively studied multiferroic material that shows multiferroicity at temperatures significantly above room temperature. Here we present high-field experiments on high-quality mono-domain BiFeO_3 crystals reveal substantial electric polarization orthogonal to the widely recognized one along the trigonal c axis. This novel polarization appears to couple with the domains of the cycloidal spin order and, hence, can be controlled using magnetic fields. The transverse polarization shows the non-volatile memory effect at least up to 300 K.

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Magnetism and ferroelectricity are two representative subjects in condensed matter physics. The quest to understand their fundamentals has clarified how various materials can exhibit such properties, resulting in the development of various indispensable devices. However, these properties of solids are usually mutually exclusive in materials¹. Although some exceptional materials, exhibiting both properties, have been recognized as ferroelectric magnets for over 40 years, the rich physics attributed to their cross-correlation had long remained unknown. The recent discovery of magnetic control of the electric polarization in manganite² has attracted significant attention in pure and applied science. The following large number of studies revealed three major microscopic scenarios to induce the electric polarization by the spin system³: spin current^{4–6}, exchange striction^{7,8} and spin-dependent p - d hybridization mechanisms^{9–11}.

The first scenario, called spin-current or inverse Dzyaloshinskii–Moriya mechanism, predicted the emergence of the electric polarization (P) expressed by the vector spin chirality of the adjacent spins ($\mathbf{S}_{i,j}$) and the unit vector connecting these sites (\mathbf{e}_{ij}) as $\mathbf{P} \propto \mathbf{e}_{ij} \times (\mathbf{S}_i \times \mathbf{S}_j)$. This simple expression supported the excavation of multiferroic materials in which electric polarization can be controlled by external magnetic fields. The vector spin chirality in localized spin system is usually caused by the frustration between spin interactions. Inevitably, the magnetic ordering temperature is low in the frustrated spin systems. A large number of efforts have been devoted to realize high-temperature multiferroic and/or magnetoelectric (ME) materials^{12–15}, aiming to apply multiferroics to practical devices such as low-power-consumption memory devices. Although some of them exhibit significant ME effects at room temperature^{13–15}, the electric polarization is absent or negligibly small at zero magnetic field and, hence, external magnetic fields are required to control the dielectric state.

BiFeO₃ is unique, because it exhibits substantial multiferroicity well above room temperature¹⁶. It belongs to the trigonal $R3c$ space group (Fig. 1) below $\sim 1,100$ K, owing to the displacement of ions along and normal to the $\langle 111 \rangle$ direction of the perovskite cubic unit (c axis of the trigonal cell) and the rotation of the ions around the same axis. Spontaneous polarization emerges along this axis with the magnitude exceeding $400,000 \mu\text{C m}^{-2}$ (ref. 17), which is superior to those of typical ferroelectric compounds such

as BaTiO₃. As BiFeO₃ shows eight-fold degeneracy in the directions of the trigonal c axis, its actual crystals often contain multiple crystallographic domains.

The spins of Fe³⁺ ions form a cycloidal order with the propagation vector pointing $\langle 110 \rangle$ direction of the trigonal lattice below ~ 640 K^{18–20}. Such a non-collinear state can be explained by the Ginzburg–Landau theory, including the ME coupling term²¹, or equivalently by the spin Hamiltonian, including Dzyaloshinskii–Moriya vectors normal to the c axis^{22,23}. According to the spin-current mechanism, this cycloidal spin order involves P along the c axis. This magnetically induced component of P can be removed at the transition from the cycloidal to the canted-antiferromagnetic state induced by a high magnetic field^{24,25}, and indeed observed by Popov *et al.*²⁴. Quantitative evaluation of the ME effect can be done only through the experiments on mono-domain crystals, otherwise the crystallographic domain boundaries, which are known to be conductive²⁶, deteriorate the intrinsic signal. Recent high-field experiments on the mono-domain crystals showed the change in P at the transition to be $\sim 210 \pm 30 \mu\text{C m}^{-2}$ below 400 K as a projected value on the cubic principal axis²⁷. Details of the ME tensor remain open albeit indispensable to understand the multiferroicity of this material. In particular, we have to pay attention to the threefold degeneracy in the choice of the magnetic propagation vector that should affect the ME effects observed at low fields²⁸.

In the present study, we performed experiments along the trigonal coordinates to obtain a fundamental understanding of the ME effects in BiFeO₃. Recently developed high-quality BiFeO₃ single crystals grown using the laser-diode heating floating-zone method²⁹ enabled these experiments. The results unexpectedly reveal the transverse (normal to the cycloidal spin rotation plane) component of the electric polarization $> 800 \mu\text{C m}^{-2}$, which is controllable using magnetic fields. This novel component cannot be understood by the spin-current mechanism. However, on a bond with lower symmetry, the anti-symmetric pair of the adjacent spins can couple to \mathbf{P} in a generic formula $P_\alpha \propto \sum_\beta d^{\alpha\beta} (\mathbf{S}_i \times \mathbf{S}_j)_\beta$, where d is a tensor and non-zero components of it are determined by a local symmetry of the bonds^{30,31}. In fact, all the tensor components of d are allowed on nearest-neighbour bonds in BiFeO₃, which induce the coupling between the transverse component of the electric polarization and cycloidal spin structure. As a result, we can reasonably reproduce the novel transverse component, which is confined to the magnetic domains and, hence, is easily controlled using a moderately high magnetic field even at 300 K. Further, BiFeO₃ shows the irreversible ME effect up to 300 K, indicating its potential application in non-volatile memory devices.

Results

Magnetization and the longitudinal electric polarization.

Figure 2a shows the magnetization (M) curves for BiFeO₃. Most of the data were measured at 4.2 K, unless otherwise stated. Magnetic fields (H) were applied along the a axis, normal to the ca plane, and along the c axis, which are defined as the X , Y and Z directions, respectively (Figs 1 and 2d). For the H applied along the X direction, the M first slowly increased and then steeply increased at ~ 18 T. It then varied almost linearly with increasing H (black line). Linearly extrapolating the high-field M - H curve (dashed black line) gave a finite intercept of $0.048 \pm 0.002 \mu_B$ per Fe at the zero-field, which is consistent with the emergence of the high-field canted antiferromagnetic phase²⁵. The H applied along the Y direction (blue line) exhibited a trace almost identical to that of the H applied along the X direction. The difference between the blue and black traces is barely distinguishable in

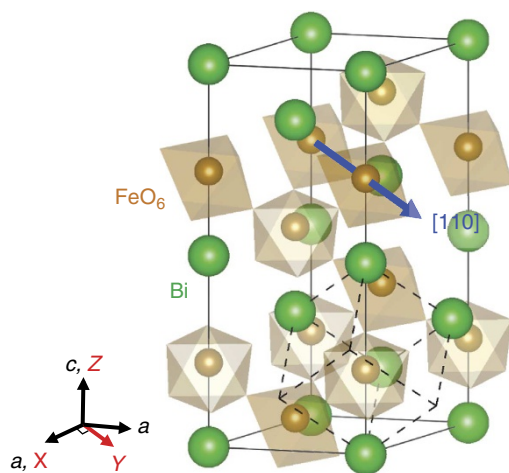


Figure 1 | Schematic crystal structure of BiFeO₃. Thick blue arrow represents a direction of a spin modulation vector. Dashed lines represent the pseudo-cubic unit in the perovskite structure. X , Y and Z denote Cartesian coordinates.

Fig. 2a. The M for $H||Z$, on the other hand, gradually increased in the low field, steeply increased with increasing H between 22 and 26 T, and then varied linearly with increasing H up to 56 T. Extrapolating the M - H curve from the high-field phase to the zero-field indicated ferromagnetic components of $0.009 \pm 0.002 \mu_B$ per Fe, significantly lower than the trigonal plane ones, suggesting that the ferromagnetic moment was mostly confined to the trigonal plane.

Careful observation of the magnetization curve revealed hysteresis in the low-field region. Figure 2b shows the steady-field magnetization with increasing H to 7 T. The M - H curve for the H applied along the Y direction was hysteretic, as shown by the thin blue line. The M - H curve for the second field sweep coincided with the upper trace without showing hysteresis (solid blue symbols). The M - H curve for the H applied along the X direction (not shown) showed similar hysteretic behaviour. The M - H curve for the H applied along the Z direction, on the other hand, showed a reversible profile even for the first sweep (thin red line). We interpreted these hysteretic behaviours as the re-orientation of the magnetic domains in BiFeO_3 .

Figure 2c shows the magnetic-field-induced change in the P along the Z direction (ΔP_Z). The effects of the H applied along the X and Y directions are almost identical. The difference between the blue and black traces is barely identifiable. The P_Z first gradually increased and then steeply increased at the magnetic transition with increasing H . The P_Z continuously changed above the transition field. The H applied along the Z direction also caused P_Z to first gradually increase then steeply increased at the transition. The low-order ME effect above the transition field showed the opposite sign to that of the in-plane fields. Although we could not determine the sign of the ΔP_Z with respect to the non-centrosymmetric crystal, the sign of the polarity relative to those of the different field directions is meaningful, because we changed the field direction while holding the wiring.

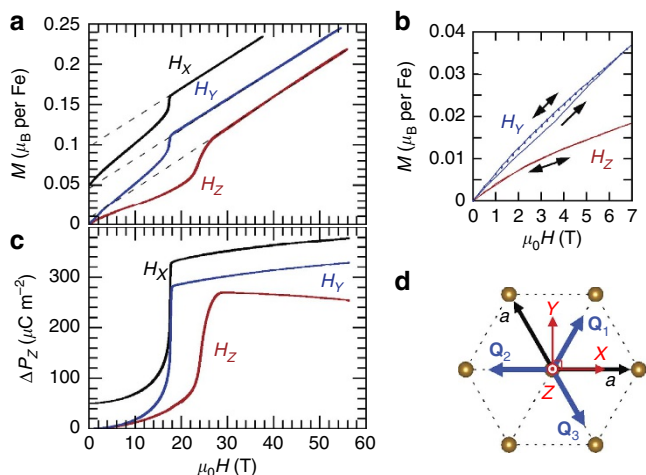


Figure 2 | Magnetic-field dependence of magnetization and electric polarization along the Z axis. (a) Magnetization curves for BiFeO_3 measured at 4.2 K for H applied along X (black), Y (blue), and Z (red) directions. The trace for H_x was vertically offset by $0.05 \mu_B$ per Fe for clarity. (b) Magnetization curves measured at 4.2 K up to 7 T. Thin lines and solid symbols represent data measured during first and second field scans, respectively. (c) Magnetic-field-induced changes in polarization measured along Z direction (ΔP_Z) for fields applied along X , Y and Z directions at 4.2 K. The data for H_x was vertically offset by $50 \mu\text{C m}^{-2}$ for clarity. (d) Schematic drawing of Cartesian coordinates X , Y and Z . Thick blue arrows represent direction of modulation vectors for Q_1 , Q_2 and Q_3 magnetic domains.

Magnetic control of the transverse electric polarization.

Figure 3a shows the change in the transverse component, P_Y , for the H applied along the X , Y and Z directions. The thin black line shows that P_Y steeply changed in the first trace for the $H||X$ sweep at $\mu_0 H \leq 7$ T. Here, μ_0 is the space permeability. P_Y then further increased $\sim 100 \mu\text{C m}^{-2}$ at the transition field and moderately changed with increasing H up to 56 T. P_Y followed almost the same trace down to 10 T when the magnetic field was decreased, but did not significantly change in $\mu_0 H \leq 7$ T. The second sweep (thick black line) traced the profile of the previous down-sweep without showing any prominent hysteresis. Applying H_Y also produced an irreversible P - H curve but in the opposite direction (thin blue line), while $\mu_0 H_Y \leq 7$ T during the virgin scan. Further, ΔP_Y at the transition field also showed the opposite sign from that for $H||X$ and showed an almost linear ME effect with increasing H to the highest field. The P - H curve for H_Z , on the other hand, showed marginal hysteresis for $\mu_0 H < 20$ T, steeply increased when $\mu_0 H \sim 26$ T and then showed the linear ME effect. The observed ΔP_Y amounted to 500 – $800 \mu\text{C m}^{-2}$ and dominated the widely recognized P_Z component caused by the cycloidal spin order (Fig. 1c).

Figure 3b shows the H dependence of P_X . The field-induced change in P_X for H_X and H_Y is much smaller than those in P_Y for the same, which remarkably contrasts with the almost isotropic magnetic properties observed in the magnetization measurements. Contrary to the small change in P_X when the in-plane fields were applied, P_X considerably changed only when H_Z was applied. The solid red line in Fig. 3b shows that the ΔP_X amounted to $800 \mu\text{C m}^{-2}$ at the transition field and then showed the linear ME effect when the field was higher. This considerable change, however, was less reproducible. The dashed red line represents the P_X - H_Z profile measured for another piece of BiFeO_3 crystal. The change measured at the transition became much smaller and showed the opposite sign of the previous one, whereas the high-field phase showed a similar linear ME effect in the same sign.

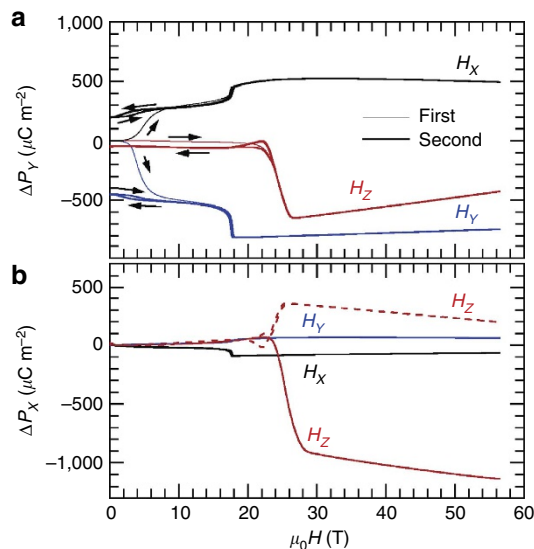


Figure 3 | Magnetic-field dependence of electric polarization along the X and Y axis. (a) Magnetic field dependence of P_Y for H applied along X (black), Y (blue) and Z (red) directions. Thin and thick lines represent traces during first and second field scans, respectively. (b) Field-induced changes in P_X for H applied along all three directions. Dashed red line shows P_X - H_Z curve measured for a different piece of BiFeO_3 crystal.

Discussion

The origin of the prominent transverse polarization could be anti-symmetric spin pair-dependent electric polarization $P_x \propto \sum_{\beta} d^{x\beta} (\mathbf{S}_i \times \mathbf{S}_j)_{\beta}$, where $d^{x\beta}$ is a coupling between spin and electric polarization^{30,31}. For example, the cycloidal screw state with \mathbf{Q}_2 have finite $(\mathbf{S}_i \times \mathbf{S}_{i+x})_Y [(\mathbf{S}_i \times \mathbf{S}_{i+y})_Y]$ on bonds along $\langle 100 \rangle$ ($\langle 010 \rangle$) on a pseudo-cubic unit cell (Supplementary Fig. 1), which produces $P_Y \propto d^{YY} [(\mathbf{S}_i \times \mathbf{S}_{i+x})_Y - (\mathbf{S}_i \times \mathbf{S}_{i+y})_Y]$ in addition to the conventional term $P_Z \propto d^{ZY} [(\mathbf{S}_i \times \mathbf{S}_{i+x})_Y - (\mathbf{S}_i \times \mathbf{S}_{i+y})_Y]$ due to the spin current mechanism. From the observed values of ΔP and the reported cycloidal spin structure²⁰, the magnitudes of the coupling are estimated to be $|d^{YY}| = 6.4 \times 10^{-32}$ C m and $|d^{ZY}| = 4.5 \times 10^{-32}$ C m as described in Supplementary Note 1. The results indicate that the coupling of P_Y is larger than that of P_Z . Owing to the symmetry of the crystal, the cycloidal screw states with \mathbf{Q}_1 , \mathbf{Q}_2 and \mathbf{Q}_3 modulations can produce in-plane components of electric polarization (P_T) normal to the spin rotation plane, which is spanned by the c axis and the spin modulation vector for each cycloidal domain (Fig. 4a,b). Applying magnetic fields is expected to favour the domains showing the spin rotation plane normal to the field. For example, applying H_Y stabilizes the \mathbf{Q}_2 domain as illustrated in Fig. 4a. Here, $P_X = 0$ and $P_Y = P_T$ (Fig. 4a). Applying H_X , on the other hand, favours the \mathbf{Q}_1 and \mathbf{Q}_3 domains; hence, $P_X \sim 0$ if the BiFeO₃ contains the same number of each domain, while $P_Y = -P_T/2$ regardless of the number of each domain (Fig. 4b). In this scenario, P_X becomes small when the field is applied along the X or Y directions, as was experimentally observed, whereas it can become large when the field is applied at intermediate angles within the X - Y plane. The response to applying H_Z would be highly sensitive to possible field misalignment, as slight tilting towards a certain in-plane direction would determine the magnetic domains, that is, magnitude and direction of the P_T , which are both consistent with the results shown in Fig. 3b.

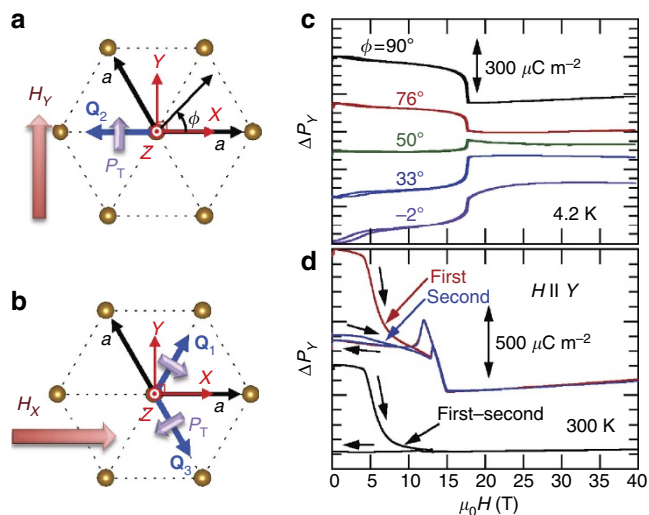


Figure 4 | Magnetic-field dependence of electric polarization along the Y axis at various field direction and at temperature of 300 K. Schematic drawings of magnetic field dependence of the magnetic domains for (a) $H \parallel Y$ and (b) $H \parallel X$. Short violet arrows represent direction of transverse electric polarization vectors (P_T) for \mathbf{Q}_1 , \mathbf{Q}_2 and \mathbf{Q}_3 magnetic domains. (c) Magnetic field dependence of P_Y measured at various angles in range $-2^\circ \leq \phi \leq 90^\circ$. (d) P_Y - H_Y curve measured at 300 K. Red and blue lines represent traces for first and second field scans, respectively. Black line shows difference between them. Data in Fig. 4c,d were vertically offset for clarity.

To investigate these hypotheses, we studied the field dependence of P_Y by changing the field direction within the X - Y plane. Figure 4c shows P_Y plotted as functions of the H measured at various azimuthal angles, ϕ , from the X direction. The change in P_Y at the transition field changed its sign between 50° and 76° with increasing the ϕ , which is consistent with the expectation from our simple domain model in which the most stable domain changed from \mathbf{Q}_3 ($P_Y = -P_T/2$) to \mathbf{Q}_2 ($P_Y = P_T$) at $\phi = 60^\circ$.

The results so far suggest that a large transverse polarization was coupled with magnetic domains of BiFeO₃. The irreversible magnetic and dielectric behaviour for $\mu_0 H < 10$ T suggests that the re-orientation of the magnetic domains was preserved even after the H was removed. To check whether the memory effect could be observed even at room temperature, we measured P_Y - H_Y at 300 K. As shown in Fig. 4d, P irreversibly changed under the weak field only during the first sweep even at 300 K. The difference between the first (red) and second (blue) profiles shows that the irreversible component of P_Y amounted to $600 \mu\text{C m}^{-2}$ at 300 K, as shown by the black line, which is almost comparable with or even larger than those measured at 4.2 K. Further, P steeply changed when the $\mu_0 H$ was between 12 and 14 T. Our P - H measurements at various temperatures revealed that this anomaly is smoothly connected with the one measured at low temperature when $\mu_0 H \sim 18$ T (Supplementary Fig. 2 and Supplementary Note 2) and, hence, can be interpreted as the magnetic transition to the canted antiferromagnetic state.

Although our simple interpretation partly explains the observed change in the in-plane polarization, this domain model alone cannot explain the irreversible component measured at $\mu_0 H < 10$ T. According to this simple model, applying the H along the Y direction stabilized the \mathbf{Q}_2 domain, which would have the largest component of $P \parallel Y$. As shown in Fig. 4d, however, the initial point of the first sweep showed higher polarization than that of the second sweep, where we believe only the \mathbf{Q}_2 domain existed. The origin for this additional P remains open at present. Judging from the magnetization and linear ME effect above 27 T (the red traces in Figs 2a and 3a), the spin canting with the weak ferromagnetic moment of $0.1 \mu_B$ per Fe along the Z direction can induce $P_Y \sim 200 \mu\text{C m}^{-2}$. Possible emergence of weak ferromagnetic moment at the domain wall region³² might contribute somewhat to this extra component. Further systematic study will be necessary to solve this mystery.

Owing to the large electric polarization along the c axis, that is, the corresponding lattice distortion, spins in BiFeO₃ exhibit the cycloidal order. According to the generalized microscopic theory for anti-symmetric spin-dependent polarization^{30,31}, this spin order involves parasitic spin-driven electric polarizations along with and also normal to the c axis. In BiFeO₃, the latter coupling d^{YY} is larger than the former one d^{ZY} caused by the spin current mechanism. This transverse component is confined to the magnetic domains and, hence, is easily controlled by H . The energy required for domain rotation can be estimated from the magnetization curve shown in Fig. 2b as $\oint B dM = 1.18 \times 10^3 \text{ J m}^{-3}$ at 4.2 K. From this energy density, we can expect that transverse polarization of $600 \mu\text{C m}^{-2}$ can be flopped by an electric field of 2 MV m^{-1} . This crude estimation reasonably agrees with the reported change in volume fraction of magnetic domains by electric field of 1.3 MV m^{-1} along the cubic axis³³. Despite this small energy, the domain arrangement is preserved after removal of the H even at 300 K. These results indicate potential use of BiFeO₃ as a three-state non-volatile memory corresponding to the three \mathbf{Q} vectors \mathbf{Q}_1 , \mathbf{Q}_2 and \mathbf{Q}_3 .

In conclusion, we studied the magnetic field dependence of the electric polarization in BiFeO₃ single crystals grown using the laser-diode heating floating-zone method. The electric polarization normal to the c axis showed prominent field-induced change.

The irreversible low-field polarization was memorized by the field sweep. Such non-volatile memory effect was also observed even at 300 K, suggesting that the BiFeO₃ can be useful for novel memory devices.

Methods

Experiments in high magnetic fields. Pulsed magnetic fields up to 56 T were generated by using pulse magnets installed at the Institute for Solid State Physics at the University of Tokyo. The magnetization was measured using the induction method in pulsed fields. We used a commercial magnetometer (MPMS; Quantum Design) for the magnetization measurements up to 7 T, as shown in Fig. 2b. The field-induced changes in the electric polarization were calculated by numerically integrating the polarization currents³⁴. To subtract the contribution of the voltage induced in the wiring, we averaged the signals measured during the positive and negative field sweeps (Supplementary Fig. 3 and Supplementary Note 3).

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Author contributions

M.T. planned and performed the experiments in collaboration with M.A. and A.M. Single crystals of BiFeO₃ were grown by T.I. and prepared for experiments by M.A. and H.K. Theoretical calculations were done by S.M. and N.F. All authors discussed the results.

Additional information

Supplementary Information accompanies this paper at <http://www.nature.com/naturecommunications>

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