



Research Article

The synergistic effects study between metal oxides and graphene on far-infrared emission performance



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Abstract

Materials with high far-infrared (FIR) emissivity are quiet promising in modern medical care field, owing to its accelerating effects to blood circulation and metabolism. In this work, we synthesized two serials composites: MnO₂/graphene nanoplate (GNPs) and Fe₂O₃/GNPs via in situ hydrothermal process. Positive synergistic effects on FIR emission were shown between either MnO₂ or Fe₂O₃ and GNPs. In particular, MnO₂/GNPs (2%) showed a super high FIR emissivity of 0.981 compared with pristine MnO₂ (0.940) and GNPs (0.878) in the wavelength of 8–14 μm. The FIR emissivity of Fe₂O₃/GNPs (1%) was up to 0.953 that was much higher than those of pristine Fe₂O₃ (0.877) and GNPs as well. Benefiting from the characteristics of graphene such as large specific surface, lightweight and thin-layer, MnO₂ and Fe₂O₃ nanocrystals grew on graphene surface and formed different morphologies. A coralloid-like morphology was formed to MnO₂/GNPs (2%), suggesting a highly porous surface that meant more unstable atoms or molecules were exposed. The morphologies of metal oxide/GNPs were thought to relate with the good synergistic effects between GNPs and metal oxides. Therefore, this work provides that metal oxides/graphene composites have promising application for high FIR emission materials.

Keywords Graphene · MnO₂ · Fe₂O₃ · Far-infrared emission

1 Introduction

Since the heightened awareness of human healthy, materials with FIR emission are quite popular in medical care products including warm-up cloth, bedding, footare, knee-cap and waistband, owing to that human body is easy to absorb FIR ray with wavelength 8–14 μm that can accelerate the blood circulation and metabolism [1–3]. Hence, materials with high FIR emissivity are in great demand nowadays. Graphene and its derivatives have attracted numerous attention for advanced materials due to the unique thermal and electrical conductivity, high mechanical strength and optical properties [4–10]. However, the research on the FIR absorbing and emitting capability of graphene is rarely studied. Hu et al. coated graphene

nanoplate on cotton fabric by using polyurethane as adhesive [11]. The FIR emissivity of cotton increased from 0.867 to 0.911 after finishing, showing a novel potential application to FIR emission material for graphene. Traditionally, metal oxides from the III~V periodic elements show good FIR emissivity, such as Al₂O₃ [12], MgO [12, 13], TiO₂ [14, 15], SiO₂, Cr₂O₃ [16], ZrO₂ [17], MnO₂ [18], Fe₂O₃ [19] and ZrC [20]. The current FIR emission materials are mainly prepared through blending or coating these metal oxides directly into matrix. Shim et al. simply blended ceramic compounds (MU-4 N and RT-3, complex of SiO₂, Fe₂O₃, Al₂O₃, CaO and Na₂O) to hydrophilic polyurethane films to form film laminate textiles [21]. With a 20 wt.% loading, the FIR emissivity of prepared textile was 0.89. Although FIR emissivity increased, the high loading of inorganic

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particle would influence the mechanical properties. Above all, graphene is a novel promising candidate for high FIR emission material compared with traditional FIR emission additives. To develop graphene composites with high FIR emissivity is meaningful for modern medical care technologies. Herein, we prepared composites of graphene and metal oxides (MnO_2 , Fe_2O_3) with high FIR emissivity via hydrothermal process in this work. The impact of GNP content on FIR emissivity of metal oxide was studied. With a 2 wt.% loading of GNPs, the FIR emissivity of MnO_2 raised from 0.94 to 0.981 and that of Fe_2O_3 with 1 wt.% GNPs raised from 0.877 to 0.953. The results revealed apparent synergistic effect between GNPs/ MnO_2 (or Fe_2O_3). In presence of GNPs, MnO_2 and Fe_2O_3 crystals grew on graphene surface with a different way compared with themselves. A coraloid-like morphology was formed to the MnO_2 with a 2 wt.% loading of graphene, suggesting a highly porous MnO_2 /graphene composites. By measuring the BET nitrogen absorption and desorption curve, it was clear that GNPs addition increased the specific surface area and the pore volume. Hence, the morphologies of metal oxide/GNPs were thought to relate with the good synergistic effects between GNPs and metal oxides.

2 Experimental

2.1 Material

GNPs (KNG-G5) was supplied by Xiamen Knano Graphene Technology Co., Ltd. KMnO_4 was purchased from Xilong Scientific Co., Ltd. H_2SO_4 , H_2O_2 , and $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ were purchased from Sinopharm Chemical Reagent Co., Ltd.

2.2 The preparation of GNPs/ MnO_2 composite

GNPs/ MnO_2 composites were synthesized via the redox reaction between GNPs and KMnO_4 under hydrothermal process. Briefly, GNPs (0.2 g) was dispersed in 100 mL deionized water via ultrasonic vibration for 1 h. A certain amount of KMnO_4 was solved in the GNPs dispersion. Subsequently, the final mixture suspension was reacted in a 150 mL hydrothermal reactor under 150 °C for 6 h. Then the reactor was opened after cooling down naturally. The raw solid product was filtered and washed by deionized water for three times. The obtained GNPs/ MnO_2 composites was dried under 80 °C for 1 day. For comparison, pure birnessite-type MnO_2 was also synthesized.

2.3 The preparation of GNPs/ Fe_2O_3 composites

GNPs/ Fe_2O_3 composites were synthesized via a similar hydrothermal process between GNPs and FeCl_3 with

that of GNPs/ MnO_2 composites. The difference is that the temperature and duration of hydrothermal process were 200 °C and 10 h, respectively.

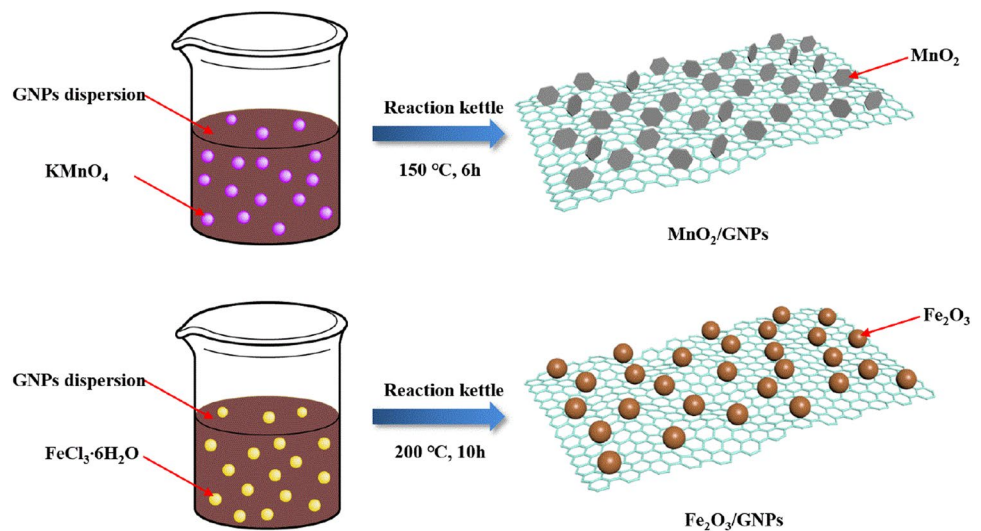
2.4 Characterization

The micro morphology of the as-prepared materials was observed by a field-emission scanning electron microscopy (SEM, NJSM-6700F, JEOL, Japan) and transmission electron microscope (TEM, JEOL-2100F, Japan). The crystallographic structures were determined by a powder XRD system (D8-Advance, Bruker, Germany) equipped with Cu K α radiation ($\lambda = 0.15406$ nm) with a scan rate (2θ) of 5°/min and an accelerating voltage of 40 kV. The Raman spectra were recorded from 500 to 3500 cm^{-1} on a Raman spectrometer (Raman, LABRAM, HJY, France) equipped with a 532 nm wavelength laser. FIR emissivity was measured by dual-band emissivity meter (IR-2, Shanghai Chengbo Optoelectronic Technology, China) with a range of 8–14 μm . The specific surface area, pore volume and pore size was measured by the Brunauer–Emmett–Teller (BET) method via nitrogen adsorption–desorption measurements at a liquid nitrogen temperature using a NOVA 2000e analyzer (Quantachrome, USA).

3 Results and discussion

Scheme 1 illustrated the hydrothermal procedure to fabricate the MnO_2 /GNPs and Fe_2O_3 /GNPs composites. Firstly, GNPs (0.2 g) was dispersed in 100 mL deionized water under ultrasonic vibration for 1 h. A certain amount of KMnO_4 was then solved in the GNPs dispersion. MnO_2 crystals formed and grew on GNPs surface via the redox reaction between GNPs and KMnO_4 , when the final mixture suspension was reacted in a 150 mL hydrothermal reactor at 150 °C for 6 h. Fe_2O_3 /GNPs composites were synthesized via the redox reaction between GNPs and FeCl_3 under a similar hydrothermal process with that of MnO_2 /GNPs composites. The difference is that the temperature and duration of hydrothermal process were 200 °C and 10 h, respectively. Table 1 listed the self-prepared MnO_2 /GNPs and Fe_2O_3 /GNPs composites at various loading of GNPs.

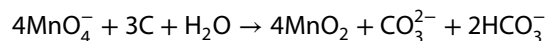
The morphology of MnO_2 , MnO_2 /GNPs, Fe_2O_3 and Fe_2O_3 /GNPs composites were observed by SEM and TEM. Seen in Fig. 1a, MnO_2 prepared via the hydrothermal process presented a spherical flower structure formed by many nanoflakes. Its diameter ranged from 2 to 4 μm . The large specific surface area of GNPs enabled metal oxide easily grown on GNPs via hydrothermal process. In Fig. 1b, it is notable that MnO_2 /GNPs showed a highly porous surface with uniformly diameter, indicating that a layer of MnO_2

Scheme 1 The hydrothermal process of MnO₂/GNPs and Fe₂O₃/GNPs composites**Table 1** All the samples and their FIR emissivity

| Sample | GNPs content (wt.%) | FIR emissivity |
|---|---------------------|----------------|
| GNPs | 100 | 0.878 |
| MnO ₂ /GNPs-20 | 20 | 0.940 |
| MnO ₂ /GNPs-10 | 10 | 0.964 |
| MnO ₂ /GNPs-5 | 5 | 0.975 |
| MnO ₂ /GNPs-2 | 2 | 0.981 |
| MnO ₂ | 0 | 0.940 |
| Fe ₂ O ₃ /GNPs-10 | 10 | 0.869 |
| Fe ₂ O ₃ /GNPs-5 | 5 | 0.909 |
| Fe ₂ O ₃ /GNPs-3 | 3 | 0.910 |
| Fe ₂ O ₃ /GNPs-1 | 1 | 0.953 |
| Fe ₂ O ₃ | 0 | 0.877 |

particles covered GNPs uniformly during the hydrothermal process. As shown in Fig. 1b, a coralloid-like morphology was shown due to the resembling of highly porous MnO₂ on the surface of GNPs. Figure 1c, d showed the TEM images of MnO₂/GNPs nanocomposite at different magnifications, respectively. The numerous tiny nanoflakes of MnO₂ connected with each other and uniformly dispersed on the surface of GNPs (Fig. 1c). Compared with the microsphere in Fig. 1a, the size of MnO₂ grown on GNPs was smaller than that of pristine MnO₂. In Fig. 1d, the interface between GNPs and MnO₂ layer was clearly shown. MnO₂ nanoflakes grew directly on the surface of GNPs, forming ordered aligned MnO₂ arrays, rather than deposited. The interplanar spacing of the nanoflakes was measured to be 0.704 nm, which was much close to the as-reported value for birnessite-type of MnO₂ [22, 23].

The porous MnO₂ structure was reported to easily form via a hydrothermal process of KMnO₄ solution [24]. The formation of MnO₂ was from the decomposition of KMnO₄, which was expressed as:



In present of GNPs, the decomposition rate of KMnO₄ might be accelerated owing to a possible catalytic effect of GNPs. Under hydrothermal process, the self-assembling rate of MnO₂ nanocrystallines could not catch up the production rate. The MnO₂ nanocrystallines initially grown on GNPs then acted as the new nucleation sites, resulting that the nanoflakes was formed along with the surface of GNPs. Therefore, a highly porous MnO₂ layer covered on the surface of GNPs and the size of MnO₂ particles was smaller than that of pristine MnO₂.

In Fig. 2a, pristine Fe₂O₃ synthesized via the hydrothermal process showed an elliptic spherical structure along

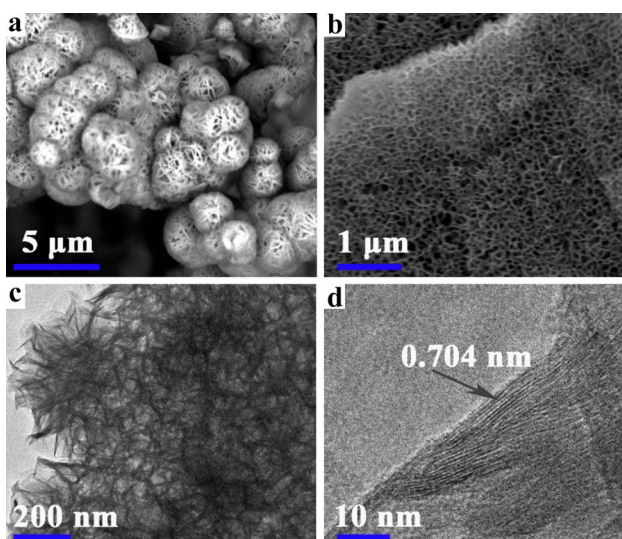
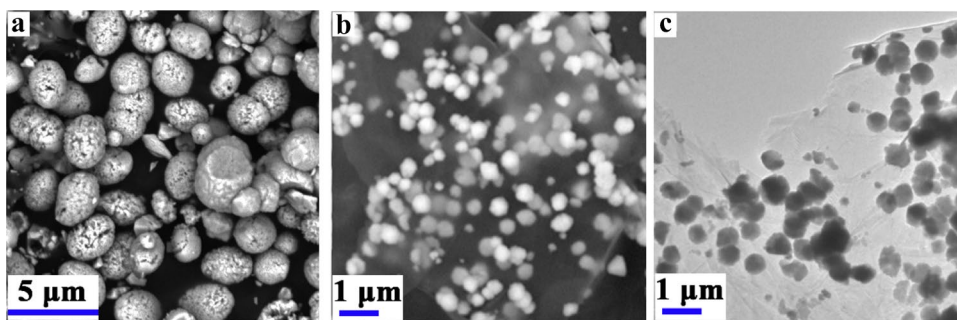
**Fig. 1** SEM images of MnO₂ (a), MnO₂/GNPs (b) and TEM images of MnO₂/GNPs (c and d)

Fig. 2 SEM images of Fe₂O₃ (a), GNPs/Fe₂O₃ (b) and TEM images of GNPs/Fe₂O₃ (c)



with a path length ranged from 0.8 to 2.5 μm. In present of GNPs, Fe₂O₃ particles grew on the surface of GNPs and their sizes were smaller than 0.5 μm, as shown in Fig. 2b. The TEM image of GNPs/Fe₂O₃ (Fig. 2c) showed that Fe₂O₃ particles were solid spheres, rather than hollow ones. Notably, Fe₂O₃ particles were loosely rather than connected with each other.

Figure 3a described the XRD patterns of GNPs and MnO₂/GNPs-2 at different loading of GNPs. The diffraction peaks at 26.4° on XRD pattern of GNPs was assigned to the (002) reflection of GNPs. This peak appeared due to the stacking of GNPs sheets. On XRD patterns of MnO₂/GNPs, the diffraction peak of GNPs was rather weak. On one side, GNPs loadings was only 2 wt.%. On another side, the coating of the MnO₂ layer was thick and intensely covered all over the GNPs sheet, which might decrease the intensity of (002) reflection peak of GNPs. The diffraction 2θ peaks at 21.2°, 26.9°, 35.6° and 65.6° were assigned to the birnessite-type MnO₂ (JCPDS number 80-1098) composed of 2D edge-shared MnO₆ octahedral layers with K⁺ cations and H₂O molecules in the interlayer space. The two peaks at 21.2° and 26.9° were related with the (001) and (002) basal reflections, respectively, while the two peaks at 35.6° and 65.6°

were related with (20 //11 l) and (02 //31 l) diffraction, respectively. The XRD patterns of Fe₂O₃ synthesized by the hydrothermal process was assigned to the hematite α-Fe₂O₃ (JCPDS number 33-0664). The 2θ peaks at 24.2°, 33.2°, 35.6°, 40.9°, 49.6°, 54.1°, 57.6°, 62.5° and 64.0° were related with the (012), (104), (110), (113), (024), (116), (122), (214) and (300) basal reflections, respectively. On XRD pattern of Fe₂O₃/GNPs-1 (Fig. 3b), the reflection of GNPs was still visible at 26.4° since spherical Fe₂O₃ was dispersedly grown on GNPs rather than with a manner of intensely porous coating as like the MnO₂.

The Raman spectra further confirmed the structure feature of GNPs, MnO₂/GNPs and Fe₂O₃/GNPs. In Fig. 4, pristine GNPs showed three characteristic peaks at 1350 (D band) and 1575 cm⁻¹ (G band), respectively. Along with the growth of MnO₂, the three peaks showed at the same positions on Raman spectrum of MnO₂/GNPs. The birnessite-type MnO₂ showed three characteristic peaks located at 500, 575, and 651 cm⁻¹, respectively, which was in agreement with the reported data [25]. α-Fe₂O₃ showed feature peaks at 222, 290, 405, 606 and 1300 cm⁻¹ [26]. On Raman spectrum of Fe₂O₃/GNPs, the peak at 1300 cm⁻¹ was overlapped with the D-band of GNPs as one broad peak. Above all, the results of Raman test were accordance

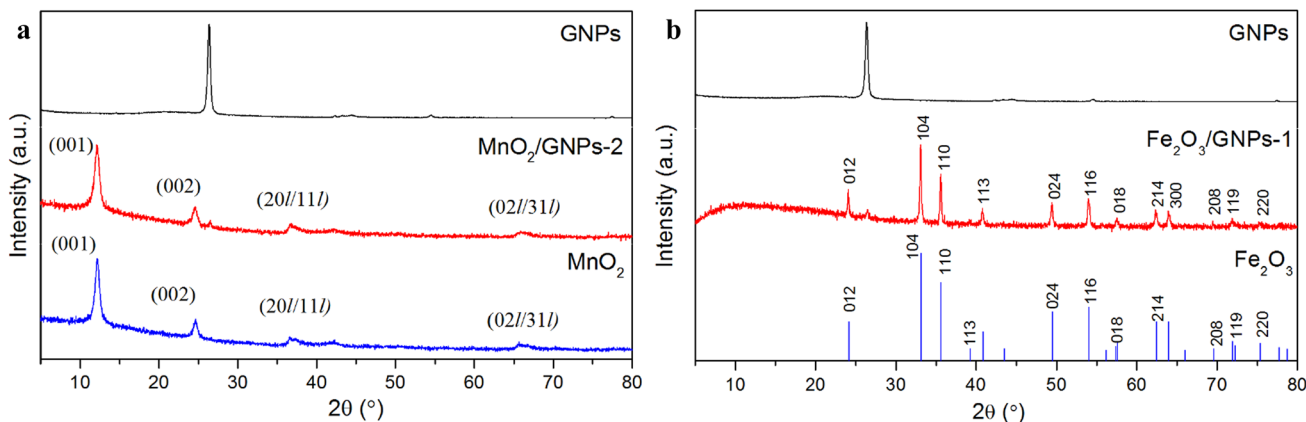


Fig. 3 The XRD curves of GNPs, MnO₂, MnO₂/GNPs, Fe₂O₃ and Fe₂O₃/GNPs

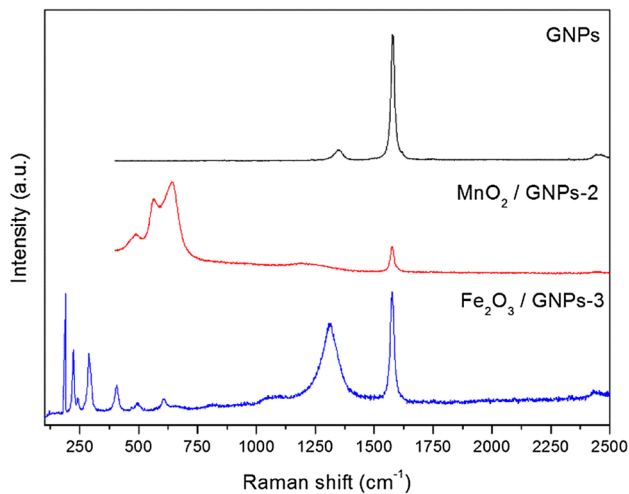


Fig. 4 Raman spectra of GNP, MnO₂/GNP and Fe₂O₃/GNP

with those of XRD. The synthesis of MnO₂/GNP and Fe₂O₃/GNP were successful.

The FIR emission performances were measured with a wavelength range from 8 to 14 μm. The emissivity (ϵ) is defined as the ratio of radiant emittance of an object to that of a blackbody. Generally, ϵ of one material drops in the range of $0 < \epsilon < 1$.

A good FIR emission material has ϵ close to 1. Figure 5a, b described the variation of ϵ of MnO₂/GNP and Fe₂O₃/GNP with different loadings of GNP. To show the synergistic effect of GNP and metal oxide (MO), the calculated (*Cal.*) ϵ of GNP/MO was calculated as following equation:

$$Cal.\epsilon = \epsilon_{GNP} \times W_{GNP} + \epsilon_{MO} \times W_{MO}$$

ϵ of pristine MnO₂ and GNP were 0.940 and 0.878, respectively. In Fig. 5a, *Cal.* ϵ of MnO₂/GNP was linearly reduced

and the value ranged in the above two limit ϵ values as the GNP loadings raised. Although the experimental (*Exp.*) ϵ followed the similar trend, the value was much higher than the *Cal.* ϵ of MnO₂/GNP. The *Exp.* ϵ of MnO₂/GNP (2%) reached up to 0.981 which was higher than 0.940 of pristine MnO₂. The results presented a positive synergistic effect existed between GNP and MnO₂. This phenomenon existed in Fe₂O₃/GNP system as well. Pristine Fe₂O₃ had almost equal ϵ (0.877) with GNP. *Cal.* ϵ of each Fe₂O₃/GNP (1, 3, 5 and 10%) were all close to 0.878. However, *Exp.* ϵ of Fe₂O₃/GNP (1, 3, 5 and 10%) were 0.953, 0.910, 0.909 and 0.869, respectively. When GNP loadings were 1, 3 and 5%, the synergistic effect between GNP and Fe₂O₃ greatly improved the FIR emission performance of pristine GNP and Fe₂O₃ (Fig. 5b). Therefore, the FIR ray emission of MnO₂ and Fe₂O₃ were improved owing to the addition of GNP.

Table 2 collected the data of specific surface area (SA), pore volume and pore diameter of pristine MnO₂ and MnO₂/GNP (2%). The average pore diameter of MnO₂/GNP (2%) was 3.8 nm as well as the pore size of pristine MnO₂. It was reasonable that GNP afforded the surface for MnO₂ microcrystalline to align its structure, while it did not change the pore forming style. The large specific surface of GNP increased the SA of MnO₂ from 40.2 to 50.7 m²/g. The pore volume increased from 0.04 to 0.32 mL/g as well, resulting from the more porous structure of MnO₂/GNP. It was considered that the enlarged specific surface

Table 2 The specific surface area, pore volume and pore diameter of pure MnO₂ and MnO₂/GNP

| Sample | SA(m ² /g) | Pore volume (mL/g) | Average pore diameter (nm) |
|-----------------------|-----------------------|--------------------|----------------------------|
| MnO ₂ | 40.2 | 0.04 | 3.8 |
| MnO ₂ /GNP | 50.7 | 0.32 | 3.8 |

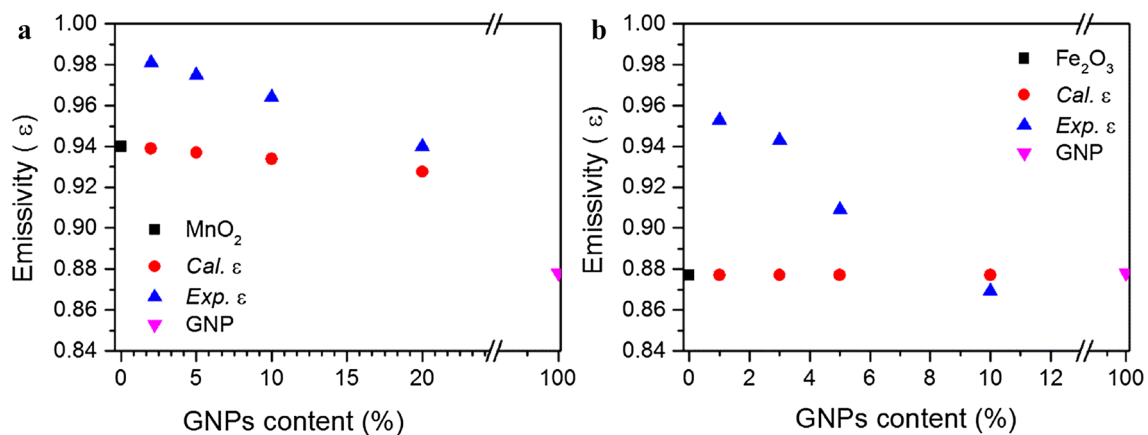


Fig. 5 The FIR emissivity of MnO₂/GNP (a) and Fe₂O₃/GNP (b)

brought about the increase of surface energy, owing to the increase of exposed molecule number on the surface. Herein, MnO_2/GNPs had larger SA than that of pristine MnO_2 , indicating that more unstable atoms or molecules were exposed, resulting to a high FIR emissivity of MnO_2/GNPs . This is a preliminary judgment for the positive synergistic effect between MnO_2 and GNP, while it was poor to explain the mechanism for the $\text{Fe}_2\text{O}_3/\text{GNPs}$ system.

4 Conclusion

In summary, metal oxides (MnO_2 and Fe_2O_3)/GNPs composites were synthesized by in situ hydrothermal process. Both MnO_2/GNPs and $\text{Fe}_2\text{O}_3/\text{GNPs}$ showed positive synergistic effect on FIR emission. $\text{MnO}_2/\text{GNPs}(2\%)$ showed a higher FIR emissivity of 0.981 compared with pristine MnO_2 and GNP. The coralloid-like morphology of MnO_2/GNPs played important role on increasing FIR emissivity due to the enlarged specific surface area and pore volume. The FIR emissivity of $\text{Fe}_2\text{O}_3/\text{GNPs}$ (1%) was up to 0.953 that was much higher than those of pristine Fe_2O_3 and GNP, while the synergistic effect between Fe_2O_3 and GNP needed to be studied in future. According to the FIR emissivity of current market products (> 0.8), this work shows that metal oxides/graphene composites is one promising choice for the high FIR emission materials in medical care application.

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Compliance with ethical standards

Conflicts of interest The authors declare that they have no conflict of interest.

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