



Research Article

Synthesis of β -AgVO₃ nanowires by hydrothermal and precipitation routes: a comparative study



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Abstract

Silver vanadate, especially β -AgVO₃, have a wide range of technological applications due to remarkable biological, optical, and electrical features. The structure, morphology, and properties of β -AgVO₃ are directly affected by synthesis conditions. A comparative study between two different synthesis routes (precipitation and hydrothermal) of β -AgVO₃ was systematically investigated in this work. X-ray diffraction, Raman spectroscopy, scanning electron microscopy, zeta potential, and UV–visible spectrometry results showed that the hydrothermal method produced more homogeneous samples of β -AgVO₃, with elevated purity and crystallinity. In addition, the sample of β -AgVO₃ obtained by hydrothermal method had a wire-like morphology while that obtained by precipitation had irregular shape. Finally, the hydrothermal procedure showed more elevated reproducibility.

Keywords Silver vanadate · β -AgVO₃ · Precipitation synthesis · Hydrothermal method

1 Introduction

Transition metal oxides, especially vanadates, are functional materials with increasing attention in nanotechnological fields. Their interesting optical, electrical, biological, and thermal properties are expanding their use in many industrial areas, such as photocatalyst [1], batteries [2], sensors [3], solar cells [4], anti-bacterial additive in paints [5, 6], and biomedical devices [7]. Among the vanadate compounds, silver vanadates are considered one of the most interesting, due to their remarkable properties, including higher crystallinity, narrow band gap, and significant anti-bacterial activity [8].

Silver and vanadium atoms show several oxidation states, which allow the formation of different silver vanadium oxides, which may contain Ag, V and O in stoichiometric and non-stoichiometric ratios. Some common

forms are Ag₂V₄O₁₁, Ag₃VO₄, Ag₄V₂O₇ and AgVO₃ [9]. Their properties are directly dependent on their composition, size, crystalline structure and surface characteristics. The most common form of silver vanadate is AgVO₃, a stable one-dimensional material with several applications, such as: sensors, batteries, photocatalytic for dye photodegradation, antimicrobial agents, and additive for dental prostheses [10–12].

AgVO₃ can be crystallized in four different structures. α -AgVO₃, γ -AgVO₃ and δ -AgVO₃ are metastable and β -AgVO₃ is the only one thermodynamically stable. α -AgVO₃ and β -AgVO₃ show one-dimensional morphologies, and they can be found in the form of nanowires, nanorods or nanotubes. The α -AgVO₃ phase can become irreversibly β -AgVO₃ at temperatures close to 200 °C [13]. It has monoclinic crystalline system, space group C_m,

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and crystallographic parameters $a = 17,87$, $b = 3,58$ and $c = 8,036$ Å (JCPDS 29-1154) [10].

Silver vanadates present the most complex structures among the vanadate compounds due to their various number of phases [14]. Different reaction conditions, such as precursor materials, reagent stoichiometry, synthesis method, reaction temperature, among others, can generate silver vanadates with different structures and properties [15, 16]. Zhang et al. [17] prepared $\text{Ag}_2\text{V}_4\text{O}_{11}$ nanowires, $\alpha\text{-AgVO}_3$ microrods, and $\beta\text{-AgVO}_3$ nanowires using the hydrothermal method with different parameters. The authors showed that different phases can be obtained by controlling the hydrothermal reaction parameters, as pH, time, and precursors. De Oliveira et al. [18] observed similar behavior. They produced $\beta\text{-AgVO}_3$ crystals by precipitation method at 30, 60, and 90 °C to determine the effect of temperature on the materials properties. According to the results, the sample obtained at 90 °C presented higher crystallinity degree than the samples obtained at 30 and 60 °C. Consequently, the band gap energy was lower for the sample produced at 90 °C (1.88 eV), when compared with the samples produced at 30 °C (2.04 eV) and 60 °C (2.03 eV).

A complete understanding of the synthesis process can allow the production of materials with improved structure, morphology, and properties. In this way, this work specifically intends to explain detailly the effects of two synthesis routes, hydrothermal and precipitation, on size, morphology, structure, and properties of $\beta\text{-AgVO}_3$ powders. The influence of Ag/V molar ratio on $\beta\text{-AgVO}_3$ crystallographic phases was also analyzed, since it has a great effect on the electrochemical and antimicrobial properties [17].

2 Materials and methods

2.1 Synthesis of $\beta\text{-AgVO}_3$ by precipitation route

$\beta\text{-AgVO}_3$ nanowires were synthesized by a precipitation reaction of silver nitrate (AgNO_3 , Neon, 99.8%) and ammonium metavanate (NH_4VO_3 , Neon, 99%) using two different molar ratios of Ag/V: 1/1 and 2/1. 1 mmol of AgNO_3 and 1 mmol of NH_4VO_3 were separately dissolved in 35 mL of distilled water at 60 °C, under magnetic stirring for 20 min. NH_4VO_3 solution was mixed with AgNO_3 solution, forming $\beta\text{-AgVO}_3$ precipitate. The precipitate was filtered and washed with distilled water several times. Then, the material was dried in conventional furnace at 60 °C for 12 h. The samples were labeled as P1 (Ag/V: 1/1) and P2 (Ag/V: 2/1).

2.2 Synthesis of $\beta\text{-AgVO}_3$ by hydrothermal route

The mixtures obtained in step 2.1 (P1 and P2 with respective supernatant, before filtered step) were placed in a Teflon-lined stainless-steel autoclave, heated to 180 °C for 16 h. The time and temperature of reaction were defined based in the $\alpha \rightarrow \beta$ AgVO_3 phase transformation. Under hydrothermal conditions (relatively high pressures), the metastable $\alpha\text{-AgVO}_3$ phase can be irreversible transformed into $\beta\text{-AgVO}_3$ at temperatures above 180 °C and reaction time higher than 15 h [16, 17]. The system was cooled to room temperature. Then, the collected product was filtered and washed with distilled water. Finally, the material was dried in conventional furnace at 60 °C for 12 h. For this synthesis, the sample produced with molar ratio of Ag/V: 1/1 was labeled as H1, while the Ag/V: 2:1 was labeled H2.

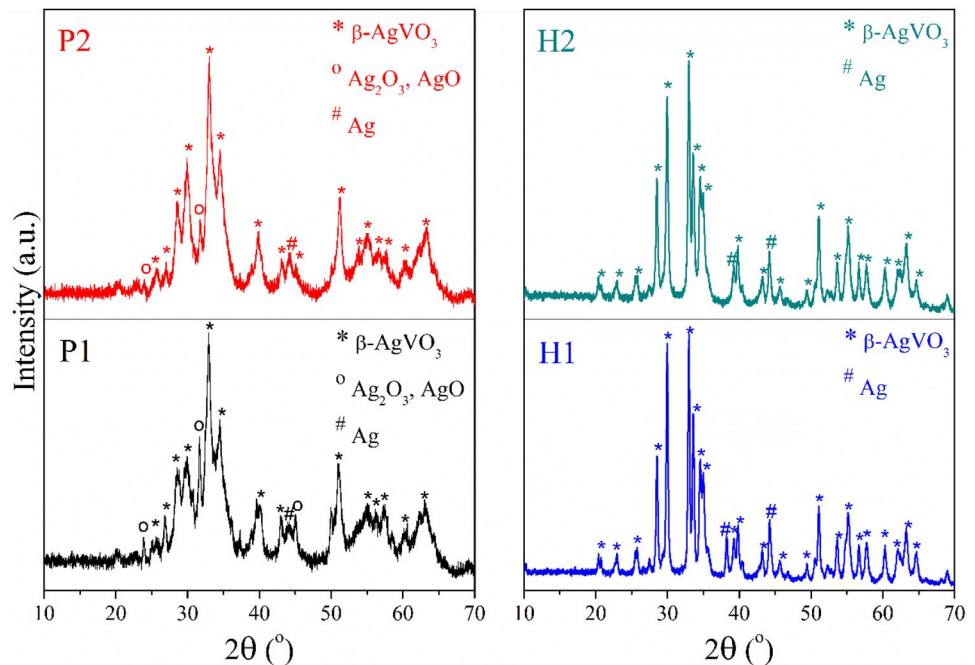
3 Characterizations

X-ray diffraction (XRD) analysis was performed using a PANalytical Philips X'Pert diffractometer operating with $\text{CuK}\alpha$ radiation ($\lambda = 1.54178$ Å) at 45 kV and 40 mA. The data were collected in the range $10^\circ < 2\theta < 70^\circ$ with $\Delta\theta = 0.02^\circ$ and 30 s per step. Raman spectroscopy was performed in a Horiba LabRam HR Evolution equipment with 532 nm laser excitation. Field emission scanning electron microscopy (FE-SEM) were carried out in a Jeol JSM-5310 at accelerating voltage of 30 kV. Zeta potential values were obtained in dynamic light scattering equipment, Beckman Coulter Delsa Nano. Samples were previously dispersed in deionized water (pH = 5.0) and the measurements were performed at 25 °C at an angle of 15 °. The equation of Smoluchowski [19] was used to obtain zeta potential. UV-Vis absorption spectrum was obtained in a Thermo Scientific Evolution 220 UV-Visible spectrophotometer by measure of absorbance from 800 nm to 260 nm using silica cuvette.

4 Results and discussion

Figure 1 shows the X-ray diffraction (XRD) pattern of the resulting silver vanadate nanostructures prepared by precipitation and hydrothermal method. Samples P1 and P2 exhibited diffractions peaks indexed to monoclinic $\beta\text{-AgVO}_3$ (JCPDS 29-1154), cubic silver oxide phases with chemical formula of Ag_2O_3 (JCPDS 01-072-0607) and AgO (JCPDS 01-076-1489) and metallic silver (JCPDS 81-1740). Samples H1 and H2 showed diffractions peaks attributed to monoclinic $\beta\text{-AgVO}_3$ (JCPDS 29-1154) and

Fig. 1 XRD patterns samples P1, P2, H1 and H2



metallic silver (JCPDS 81-1740) [6, 20]. Therefore, the two peaks attributed to Ag was verified in all samples. Ag is an interesting metal for biological and for batteries applications, since it may improve the antimicrobial activity and enhance the electrochemical performance due to the improved conductivity [12]. Figure 1 also shows that samples H1 and H2 are constituted only of β -AgVO₃ besides metallic silver, while samples P1 and P2 showed diffractions peaks indexed to silver oxides (Ag₂O₃ and AgO) along with β -AgVO₃.

The crystallinity degree of each samples was estimated using the background of the XRD patterns using the Krimm and Tobolsky [21, 22]. The crystallinity degree of multiphase samples (CD) was calculated from the ratio between the sum of all crystalline diffracted areas (A_C) and the total area of diffraction (A_T = crystalline + amorphous), as shown in Eq. 1:

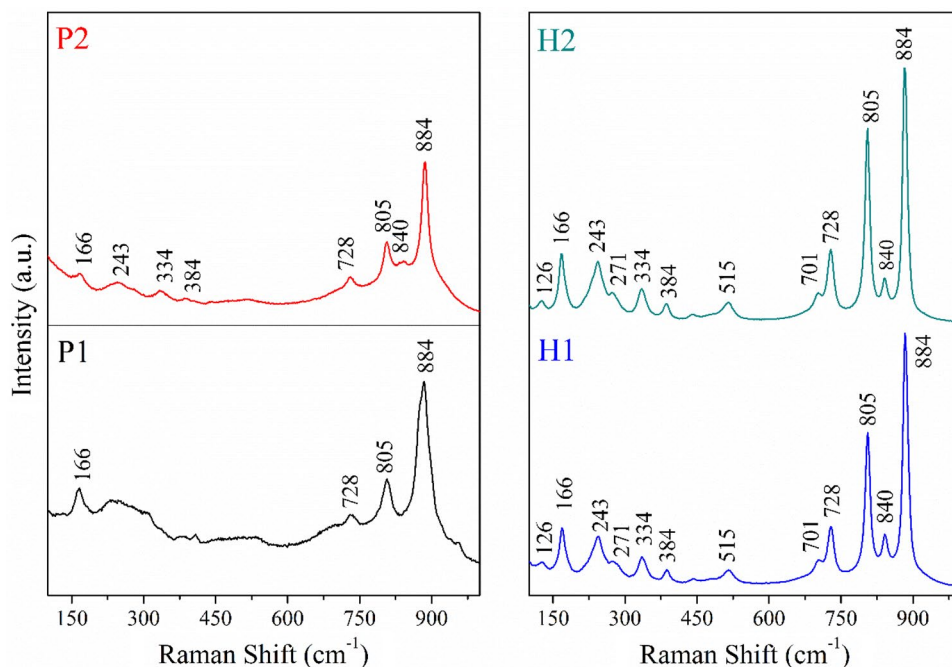
$$CD = \left(\frac{A_C}{A_T} \right) \times 100 \quad (1)$$

Sample P1 and P2 showed average crystalline degrees of 68% and 63%, while, H1 and H2 were 81% and 79%, respectively. Therefore, the hydrothermal synthesized samples showed increased crystalline phase when compared to precipitated samples. The higher values of crystalline degree of samples H1 and H2 is due to high temperature and pressure of the hydrothermal process, which were 180 °C and about 1.7 MPa. In the hydrothermal process, the precursors dissolve in the solvent and recrystallize in nanostructured form. Initially,

tiny crystalline nuclei of α -AgVO₃ emerge in the supersaturated solution system, which grow in the form of microrods, following the “Ostwald-ripening process”. In the first 4 h of reaction, longer α -AgVO₃ microrods are formed. Then, with longer reaction time, the α -AgVO₃ microrods split into β -AgVO₃ nanowires. The phase/morphology transformation of α -AgVO₃ microrods into β -AgVO₃ nanowires, under hydrothermal conditions, occurs at around 180 °C and relatively high pressure (1.7 MPa) [17]. The hydrothermal conditions guarantee the formation of nanomaterials with excellent organization and crystallinity, reducing the amorphous structure and controlling the grain size [23].

Figure 2 displays the Raman spectra of samples P1, P2, H1 and H2. All samples presented peaks related to formation of β -AgVO₃. The most intense peak at 884 cm⁻¹ can be attributed to stretching vibrations of V–O–Ag, Ag–O–Ag and O–V–O [24–28]. That band at 840 cm⁻¹ is related to the stretching vibrations of V–O groups in (V₂O₇)⁻⁴ ion or to Ag–O–V stretching vibrations [24, 27, 28]. The band at 805 cm⁻¹ is associated with stretching vibrations of Ag–O–Ag bridges [25, 27, 28], while that band at 728 cm⁻¹ is related to VO₄ deformation modes [24–27]. The stretching vibrations of Ag–O–V and asymmetric stretching of V–O–V in the polymeric chains of metavanadate are assigned to bands at 701 and 515 cm⁻¹ [24–28]. Raman bands at 384 and 334 cm⁻¹ are attributed to the asymmetric deformation of VO₄³⁻ tetrahedron [29, 30]. The bands around 271 and 243 cm⁻¹ are related to the asymmetrical and symmetrical folding of VO₄³⁻ tetrahedron, respectively. Those bands below 200 cm⁻¹ are related to

Fig. 2 Raman spectra of samples P1, P2, H1 and H2



the collective vibration modes of the phonon vibrations of the crystal lattice [31, 32].

Figure 2 shows that the Raman bands of samples H1 and H2 are related to β -AgVO₃. Raman bands of samples P1 and P2 shows the disappearance of some peaks related to β -AgVO₃ at frequencies (under 500 cm⁻¹). The XRD patterns, in general, reveal the long-range ordering of the materials in the length of periodicity of 40 to 100 Å, showing structural information of several unit cells [28, 33, 34]. In this way, a periodicity over a length of at least 50 Å is necessary to obtain an XRD pattern from a specific material. This requirement makes XRD technique insensitive to structural changes that occur under such range order [34]. On the other hand, Raman spectroscopy is very sensitive to the structure symmetry, being suitable for molecular-level investigations [35]. The presence of any disorder in the system, especially those caused by oxygen displacement, will result in drastic changes in the Raman spectrum of a perfect crystal lattice [36, 37].

In general, the intensity of Raman bands become less intense and wider when samples have structural disorders, which results in a breakdown of translational symmetry [36]. Broad modes could also be related to disorder from metal–oxygen vacancies [38]. Raman bands of a perfect crystal shows narrow lines [28, 33]. Samples P1 and P2 showed weak and overlapped bands suggesting a great amount of structural disorders, as metal–oxygen vacancies. However, samples H1 and H2 showed narrow bands with high intensity, indicating more ordered structures. The results of XRD and Raman spectroscopy suggested that samples H1 and H2 have more ordered structures with

higher crystallinity, when compared to samples P1 and P2. One-dimensional structures with high crystallinity are suitable for several applications. In the electrical field, wire-like crystalline β -AgVO₃ can be used as cathode for lithium batteries due to the higher transport of electrons [12, 17].

Figure 3 shows the SEM micrographs of samples P1, P2, H1 and H2. A significant difference in the crystal shapes can be observed. Samples P1 and P2 are constituted of

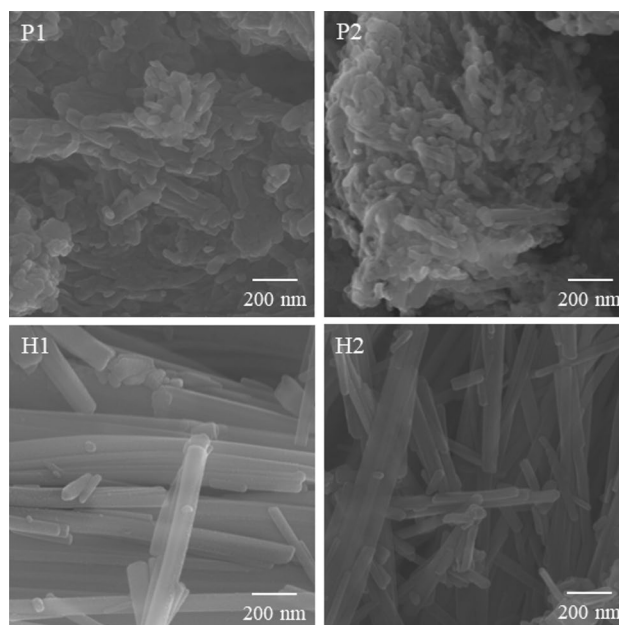


Fig. 3 SEM micrographs of samples P1, P2, H1 and H2

particles of irregular shape with some rodlike structures. However, these structures are irregular and heterogeneous, with different diameters and lengths. In the precipitation route, the nucleation plays a major role when compared to the crystal growth. In this case, most of the solute crystallize in the nucleation stage, as a mass precipitation process. The particles are formed by the ordered aggregation of several nucleus rather than by crystal growth [39]. As a result, irregular and heterogeneous particles are produced, since there is not enough time to grow and rearrange in preferential directions to generate a one-dimensional material. Sample P2 was more agglomerated, but it has more rodlike structures when compared to sample P1. This result indicates that higher Ag/V ratio was more favorable for one-dimensional silver vanadate production in precipitation reaction.

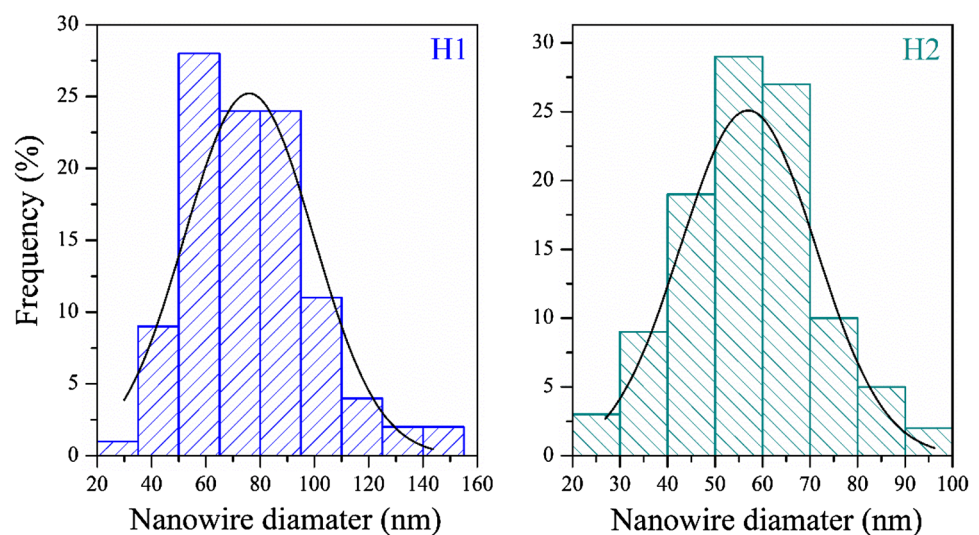
Figure 3c, d reveal that the hydrothermal method produced samples with one-dimensional morphology with length of several tens of micrometers. Both samples H1 and H2 showed uniform and homogeneous wirelike structures, without the presence of particles with other shapes. This is an indication that hydrothermal methodology can produce one-dimensional (wirelike) β -AgVO₃ particles with good uniformity. As mentioned before, in the hydrothermal route, the precursors gradually dissolve and recrystallize in the form of microrods. Since the reaction take place in higher temperatures and longer times, there is enough time to the particle grow in the energy favorable direction. First, α -AgVO₃ microrods are produced, which are splitted into β -AgVO₃ nanowires [17, 40]. One-dimensional nanostructures have great potential to be used in electronics, optoelectronics, and memory devices. β -AgVO₃ wirelike morphology allows an efficient transport of electrons, expanding their use as cathode materials for lithium batteries [11, 16].

The diameters of the nanowires of samples H1 and H2 were determined from SEM micrograph and the distribution histograms are shown in Fig. 4. A number of 100 nanowires per sample were taken into account to allow a meaningful comparison. The histogram of sample H1 presented an asymmetry towards values under 100 nm and the diameter around 60 to 95 nm. Sample H2 presented a narrow and symmetrical nanowire diameter distribution, with most of nanowires having diameters between 45 and 70 nm. Sample H2 showed nanowires less aggregated, with smaller diameter when compared to H1. In addition, most nanowires of sample H1 is parallel to one another and tightly bundled together. So, synthesis of β -AgVO₃ nanowires performed with higher Ag/V ratio produced β -AgVO₃ nanowires with narrow diameter distribution and less aggregates.

SEM micrographs confirmed the results observed in XRD and Raman spectroscopy. As expected, samples P1 and P2 presented more heterogeneous structures without regular shapes, which may be related to the formation of silver oxides that were identified by XRD. The one-dimensional structures present in the sample are smaller in size than in samples H1 and H2, as suggested by the broader peaks and bands of XRD and Raman. The hydrothermal treatment under higher pressures (1.7 MPa), temperatures (180 °C), and times (15 h) increases the particle size when compared to the precipitation reaction under lower temperatures (60 °C) and times (20 min). In addition, as indicated by Raman spectroscopy, the silver vanadate produced by the hydrothermal method presented more organized, crystalline and homogeneous β -AgVO₃ nanowires.

The stability of a colloidal suspension may be determined by measuring the zeta potential of particles, which is related to the surface particle charges of a dispersion.

Fig. 4 SEM histogram of the nanowire diameter distribution of H1 and H2



The particle charge of particles in dispersions is strongly influenced by the pH of the medium [41]. The zeta potentials of samples P1, P2, H1, and H2 are shown in Table 1. Particles with zeta potential greater than ± 25 mV are considered to form stable colloidal suspensions [42, 43].

Table 1 shows that samples P1, P2, H1 and H2 had negative zeta potentials, regardless the synthesis route and molar ratio of precursors. The negative values of zeta potentials are attributed to negatively charged hydroxyl groups present on the surface of particles, which lead to negatively charged particles [44, 45]. In the presence of water, the outermost surface oxygens of metal oxides particles hydrate, forming a high density of hydroxyl groups. These hydrated polar surface groups allow the formation of hydrogen bonds and other forces to the surfaces when in contact with polar liquids [46, 47]. As a result, a colloidal suspension is formed, which is showed by the potential zeta values (higher than 25 mV, in modulus). AgVO_3 nanowires synthesized by Artal et al. [48] presented zeta potential of -0.1 mV, confirming the negative behavior of the electric charges of these samples.

Sample H1 is a more stable colloidal dispersion since the higher (in modulus) value of zeta potential prevents the agglomeration of particles due to the reduction of intraparticle attractive forces [49]. Therefore, the presence of hydroxyl functional groups on the surface of AgVO_3 particles may aid the functionalization process with another chemical groups, expanding the applications of AgVO_3 particles as reinforcement agent in nanocomposites. $\beta\text{-AgVO}_3$ have great potential of applications as fillers in

biomedical prosthesis, especially in dentistry, due to their antibacterial properties [22], which is enhanced by the presence of metallic silver on $\beta\text{-AgVO}_3$ structure (showed by XRD). Also, electrode nanocomposites can also be produced, combining the high energy density and long-term stability of $\beta\text{-AgVO}_3$ with the high electrical conductivity of carbon materials [11].

In order to investigate the alterations of optical properties induced by different synthesis route, the samples were subject to analysis of UV-vis absorption spectra. The UV-vis spectra of samples P1, P2, H1 and H2 are shown in Fig. 5.

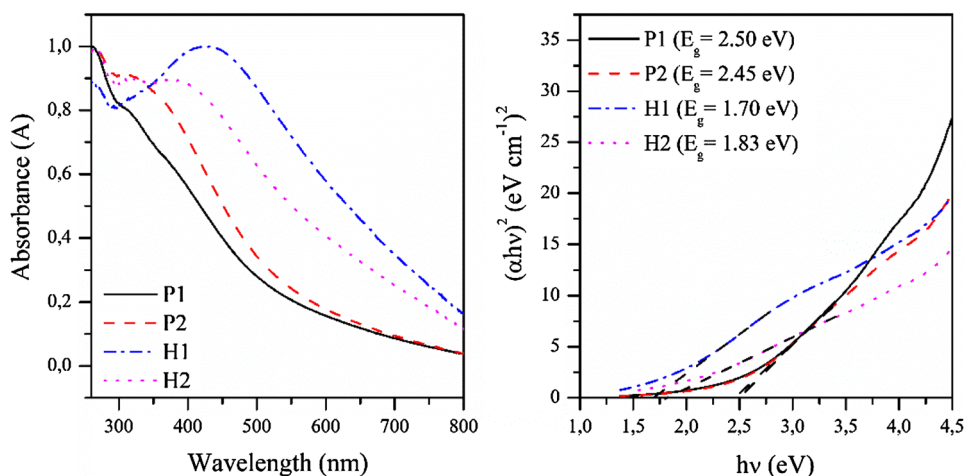
Samples P1, P2, H1 and H2 exhibited absorption in the UV and visible light region. The optical band gap energies (E_g) of each sample were estimated from absorption coefficients as function of wavelength using *Tauc relation*: $(\alpha h\nu)^n = A(h\nu - E_g)$, where A is the relative constant for the material, α is the absorption coefficient and $h\nu$ is the photon energy. The index n is a constant associated with electronic transitions ($n = 0.5$ for direct transitions, $n = 2$ for indirect transitions) [50]. Figure 4b shows the calculated E_g for samples P1, P2, H1 and H2 as 2.50 eV, 2.45 eV, 1.72 eV and 1.90 eV, respectively, which agree with the values obtained by Sivakumar et al. [50] and Zhang and Min [51].

The morphology and size of materials have important effects on optical properties. The band gap energy increase with the reduction of the particle size due to the quantum confinement in semiconductors nanoparticles. Moreover, the percentage of amorphous phase present in the semiconductors influences the values of E_g . In general, materials that are heat treated or synthesized at higher temperatures have more organized structures due to the reduction of structural defects and oxygen vacancies, resulting in reductions of band gap energy [28], justifying the band gap reduction for samples H1 and H2. Therefore, a reduction in E_g values of samples H1 and H2 were expected since the samples prepared by hydrothermal

Table 1 Zeta potential of samples P1, P2, H1 and H2

Zeta potential (mV)			
P1	P2	H1	H2
-30.31	-23.91	-35.56	-29.41

Fig. 5 **a** UV/Vis analysis of samples P1, P2, H1 and H2; **b** Plots of $(\alpha h\nu)^2$ versus photon energy ($h\nu$) of AgVO_3 prepared by different routes



(H1 and H2) route showed higher crystallinity than samples prepared by precipitation route (P1 and P2). Some works have been demonstrated that narrows band gaps are related to better photocatalytic activity [52]. Since β -AgVO₃ is an efficient photocatalyst under visible light radiation [53], the samples produced by hydrothermal routes (H1 and H2) is expected to present better photocatalytic activity than the samples synthesized by precipitation (P1 and P2).

5 Conclusions

A great understanding of the synthesis of silver vanadate and the comparison of two manufacturing processes, precipitation and hydrothermal method was reported. As shown by XRD analysis and Raman spectroscopy, both synthesis techniques produced the phase β -AgVO₃. However, the precipitation route produced β -AgVO₃ with lower purity, crystallinity, and a poor reproducible process. On the other hand, the use of hydrothermal method provided the adequate medium for the formation of β -AgVO₃ with higher purity, high crystallinity, homogeneous and wire-like morphology and, mainly, a process with high reproducibility. Although smaller particles are desired in many nanotechnological applications, the use materials with better crystallinity, higher purity and a high reproducible process is usually more important. In this context, the results showed that hydrothermal methodology is more indicated to produce β -AgVO₃ nanowires.

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Compliance with ethical standards

Conflict of interest The authors declare that they have no conflict of interest.

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