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Published online 30 June 2023 | https://doi.org/10.1007/s40843-023-2456-6 Sci China Mater 2023, 66(8): 3146-3154



Construction of ZnIn₂S₄/Sv-MoS₂ photocatalysts with subtle atomic-level intimate contacts: Enhancing interfacial interactions to improve photocatalytic H₂ evolution in visible light

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ABSTRACT The low charge separation efficiency between catalyst and cocatalyst severely limits the performance of photocatalysts. The strong interfacial interactions between catalyst and cocatalyst can improve the charge separation efficiency. Introducing interfacial chemical bonds to enhance the interfacial interaction between components is an effective way to improve photocatalytic performance. Herein, ZnIn₂S₄ (ZIS)/Sv-MoS₂ photocatalyst was synthesized. The binding effect between S atoms in ZIS and the uncoordinated Mo atoms in Sv-MoS₂ forms the interfacial Mo-S bond which greatly improves the photocatalytic activity of ZIS. The MoS₂xh was prepared by NaBH₄ etching for different times. This extraordinary photocatalytic activity is mainly due to the easy transfer of photogenerated electrons from ZIS to MoS₂ via the heterojunction interface tightly connected under the action of Mo-S bonds. Photoelectric measurements show that ZIS/ MoS₂-4h possesses effective charge transfer. This work reveals the effect of the introduction of interfacial chemical bonds on the photocatalytic activity of ZIS/MoS₂, and provides a simple and effective method for designing excellent cocatalysts through interface engineering.

Keywords: Sv-MoS₂, ZnIn₂S₄, NaBH₄ etching, interfacial chemical bond, photocatalytic hydrogen evolution

INTRODUCTION

With the industrialization and fast development of economy, the energy shortages and atmospheric pollution become serious issues needed to be addressed [1-3]. Photocatalytic hydrogen production is an effective way to convert solar energy into clean energy [4-9]. Sulfides are excellent candidates for photocatalytic hydrogen production because they have suitable band gaps and conduction band [10-14]. However, the photocatalytic activity of sulfides is usually limited by the rapid recombination of photogenerated carriers and the lack of corresponding active sites. To address these issues, considerable efforts have been implemented to boost the photocatalytic activity of photocatalyst, such as element doping, heterostructure construction and cocatalyst loading. Many studies show that metal and nonmetal element doping significantly affect the photocatalytic

activity of materials [15,16]. Recently, Ding's group [17] proposed that different doping depths of elements have great influence on charge transfer. It is difficult for a single-component photocatalyst to have a wide light absorption range, rapid charge separation and strong redox ability at the same time. By reasonably constructing heterojunction, the advantages of multiple components can be integrated [18–22]. Mao's group [23] designed a new photocatalyst (CdS/ZnS core-shell nanorods), in which CdS nanorods were modified by dodecylamine and then wrapped by intermittent ZnS shells. Due to the rapid transfer of photogenerated electrons from ZnS shell to CdS nanorods, the adsorption performance of H⁺ was improved after dodecylamine modification. The photocatalytic activity was remarkably improved. Supported cocatalyst has been proved to be an effective way to inhibit carrier recombination and reduce reaction overpotential. Precious metals like Au, Pt, Pd, Ag, Ru and their oxides such as RuO₂ are commonly used as cocatalysts for photocatalysis [24-27]. Sun's group [28] synthesized Au nanodots@thiol-UiO-66@ZnIn₂S₄ (ZIS) photocatalysts. Au nanodots were anchored in the pores of thiol-functionalized UiO-66, and ZIS nanosheets were wrapped on the metal-organic framework (MOF). In Au@thiol-UiO66@ZIS, photogenerated electrons were transferred from ZIS to MOF, and then to Au nanodots, and a smooth electron transport channel was established. But its high cost and scarcity are not suitable for practical applications. It is necessary to find functional materials with low toxicity, low cost and high efficiency to replace precious metals as cocatalysts for photocatalysis.

Recently, many non-noble metal cocatalysts such as metal nitrides, metal sulfides, metal phosphides and metal carbides have been developed. Among them, molybdenum disulfide has attracted widespread attention in the field of photocatalytic hydrogen evolution [29,30]. Theoretical calculation shows that the moderate adsorption capacity of intermediate hydrogen atoms on MoS_2 and Pt significantly reduces the activation energy of the reaction [31]. In other words, MoS_2 is a promising alternative to Pt. ZIS is widely used for photocatalytic hydrogen production due to its outstanding visible light trapping ability, photostability and low toxicity [32–35]. The combination of ZIS and MoS_2 can effectively promote the separation of electrons and holes. Wei *et al.* [36] prepared ZIS/MoS₂ composites by

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impregnating ZIS nanoflowers with (NH₄)₂MoS₄ aqueous solution, followed by high-temperature treatment in an H₂S stream to convert Mo(VI) to Mo(IV). This catalyst greatly improves the photocatalytic hydrogen performance of production $(141 \,\mu\text{mol}\,\text{h}^{-1})$ due to the increase of charge separation efficiency. Amorphous MoS₂ has been shown to promote photocatalytic activity for the first time. By using a one-pot solvothermal reaction, Zhang et al. [37] prepared ZIS/MoS₂ photocatalyst, which significantly improved the photocatalytic H₂ conversion rate of ZIS upon visible light irradiation due to its tight interfacial contact. The introduction of synergistic cocatalyst can further improve the photocatalytic performance of ZIS. Yuan et al. [38] synthesized MoS₂-graphene composites as synergistic cocatalyst to promote the photocatalytic activity of ZIS (4167 μ mol g⁻¹ h⁻¹). The outstanding catalytic activity is ascribed to the unique synergistic effect between MoS₂ and graphene, which function as a co-catalyst and electron transfer channel for hydrogen evolution reactions, respectively. Although MoS₂/ZIS heterojunction has been widely reported, the low charge transfer efficiency between the catalyst and cocatalyst seriously limits the performance of photocatalyst. At present, it is still lacking attempts to improve the efficiency of electron separation by enhancing the interface interaction between the catalyst and cocatalyst. Introducing interfacial chemical bonds to enhance the interfacial interaction between components is an effective means to improve photocatalytic hydrogen production. For example, Li's group [39] synthesized P-doped, MoP-modified g-C₃N₄ (PCN/MoP) heterojunction photocatalysts. The interfacial Mo-N bond serves as an efficiently functioning electron "transfer channel" between PCN and MoP, accelerating the migration and separation of the photogenerated carriers. Shen et al. [40] synthesized C_3N_4/WO_3 heterojunction. Theoretical and experimental results confirmed that N-W bond can be used as an atomic interface channel to accelerate the interface electron transfer.

Herein, ZIS/Sv-MoS₂ photocatalyst was synthesized. The binding effect between S atoms in ZIS and the uncoordinated Mo atoms in Sv-MoS₂ forms the interfacial Mo–S bond which greatly improves the photocatalytic activity of ZIS. The interfacial Mo–S bond serves as an efficiently functioning electron "transfer channel" between ZIS and MoS₂, accelerating the migration and separation of the photogenerated carrier. The photocatalytic performance of the optimized ZIS/MoS₂-4h composite was significantly improved. The optimized ZIS/MoS₂-4h composite shows a significant hydrogen production rate of 7.6 mmol g⁻¹ h⁻¹, which is approximately 4.75 times that of the original ZIS (1.6 mmol g⁻¹ h⁻¹) and 2.05 times that of ZIS/MoS₂-4h is 1.71% at 420 nm (5 mg catalyst).

EXPERIMENTAL SECTION

Preparation of MoS₂

In a typical synthetic procedure, ammonium molybdate tetrahydrate ((NH₄)₆Mo₇O₂₄·4H₂O) (1.236 g, 1 mmol) and thiourea (CH₄N₂S) (1.0657 g, 14 mmol) were added to 30 mL of deionized water. After stirring for 30 min at room temperature, the resulting solution was transferred to a 50-mL Teflon-lined stainless-steel autoclave and heated at 220°C for 18 h. When the solution was cooled to room temperature, the collected black product was centrifuged, washed several times with deionized water and ethanol and dried under vacuum at 80°C for 12 h.

Preparation of MoS₂-xh

The prepared MoS_2 (100 mg) was suspended in 10 mL of aqueous solution and sonicated for 2 h. Then 50 mg NaBH₄ was added and further stirred for a certain time (2, 4, and 6 h) at room temperature to obtain MoS_2 -*x*h (MoS_2 -2h, MoS_2 -4h, and MoS_2 -6h).

Preparation of ZIS

An aqueous solution of hydrochloric acid (8 mL, pH = 2.5) and glycerol (2 mL) were added into a 25-mL flask and sonicated. Then, after ultrasonic treatment for 30 min and stirring for 5 min, 54.4 mg ZnCl₂, 234.4 mg InCl₃·4H₂O, and 120 mg thioacetamide (TAA) were added. The resulting solution was stirred for 30 min, and then put into an oil bath at 80°C. After 2 h of reaction, the product was collected by centrifugation, washed three times with ethanol and deionized water, and dried in vacuum at 70°C.

Preparation of ZIS/MoS₂ and ZIS/MoS₂-xh

ZIS/MoS₂ and ZIS/MoS₂-xh were prepared with the same procedure as for ZIS, except that 10 mg of MoS₂ or MoS₂-xh was added to the beaker.

Photocatalytic H₂ evolution experiment

The photocatalytic reaction was carried out in a sealed Pyrex top irradiation reactor with a capacity of 180 mL. Photocatalyst (5 mg) was dispersed in 5 mL aqueous solution containing 1 mL of triethanolamine (TEOA) as the sacrificial reagent and continuously stirred until completion. Prior to irradiation, the reaction system was evacuated and purged three times with argon gas to remove all air. The temperature of the reaction was controlled by a circulating condensation system at 25°C. A 300-W Xe lamp with a 420-nm cut-off filter was used to irradiate the suspension. The gas chromatograph (Agilent 7820A) was used to determine the amount of H_2 . Usually, 1 mL gas sample was taken from the reaction vessel and injected into the gas chromatograph every one hour.

The apparent quantum efficiency (AQE) of the reaction system was measured using a Xe lamp with light band-pass filter (λ = 420, 450, and 520 nm). The incident monochromatic illumination intensity was measured using the ILT 950 spectro-radiometer. The following equation was used to calculate AQE.

$$AQE = \frac{2MN_Ahc}{AIt\lambda} \times 100\%,$$
(1)

where *M* is the molar amounts of H₂ during 2 h of irradiation, N_A is the Avogadro constant (6.022 × 10²³ mol⁻¹), *h* is the Planck constant (6.626 × 10⁻³⁴ J s), *c* is the light velocity (3 × 10⁸ m s⁻¹), *A* is the irradiation area (3.14 × 2.25 × 10⁻⁴ m²), *I* is the intensity of irradiation light (mW cm⁻²), *t* is the reaction time (7200 s), and λ is the wavelength of the monochromatic light (nm).

RESULTS AND DISCUSSION

Synthesis and characterization

The preparation process of ZIS/Sv-MoS₂ is depicted in Fig. 1. Firstly, nanoflower-like MoS_2 was synthesized by a hydrothermal method. Then, MoS_2 with rich S vacancies (Sv-MoS₂) was prepared by NaBH₄ etching at room temperature (Fig. S1). If the



Figure 1 Schematic diagram of the preparation process of ZIS/Sv-MoS₂ heterojunction.

internal structure of MoS_2 is seriously damaged, it will affect the electron transfer, and more defects will also affect the growth process of ZIS. Therefore, we adopted room temperature conditions to avoid more internal defects. Finally, Sv-MoS₂ was immersed in Zn²⁺, In³⁺, TAA, and glycerol solution to grow ZIS nanosheets through an oil-bath procedure.

X-ray diffraction (XRD) measurements were implemented to reveal the composition and phase structure of the as-prepared material. The diffraction peaks of bare MoS_2 at 2θ values of 13.94°, 33.36° and 57.53° can be ascribed to the (002), (100) and (110) crystal planes of hexagonal MoS₂ (JCPDS Card No. 87-2416) [41]. With the increase of NaBH₄ etching time, the XRD diffraction peak intensity of MoS2-xh decreases, which indicates that the structure becomes more unordered, because MoS₂-xh has produced more defects (Fig. 2a). The pure ZIS exhibits three main peaks at 21.48°, 27.90°, and 47.24°, corresponding to (006), (102), and (110) planes of the hexagonal ZIS (JCPDS card No.065-2023) [42]. The characteristic diffraction peaks of ZIS and MoS₂ are clearly shown in the XRD pattern of ZIS/MoS₂-*x*h, which indicates the successful synthesis of ZIS/MoS₂-xh composite (Fig. 2b). The microstructure of the samples was also studied by Raman spectroscopy. The pristine MoS₂ shows peaks at 379.3 and 406.0 cm⁻¹ attributing to the in-plane E¹_{2g} and outof-plane A_{1g} [43], which involve anti-parallel vibration of two S atoms relative to Mo atoms, as well as only the vibration of S atom in-plane and out-of-plane in the opposite direction [44]. Compared with MoS₂, the A_{1g} and E^{1}_{2g} peaks of MoS₂-xh are slightly red-shifted. For example, the A_{1g} and E¹_{2g} peaks of MoS₂-4h are shifted to 377.4 and 403.2 cm⁻¹, respectively (Fig. 2c). This phenomenon shows that there is lattice distortion in MoS₂*x*h, and implies that there are many defects in the catalyst due to a large number of hanging bonds on the surface [45]. This is consistent with the XRD results. In the synthesis of ZIS/MoS₂-4h, the S vacancy in MoS₂-4h will be occupied by the S atom in ZIS. If the interaction between MoS₂-4h and ZIS is weak, the Raman peak of MoS₂-4h in ZIS/ MoS₂-4h will be closer to MoS₂. However, as shown in Fig. 2d, e, the Raman peak of MoS₂ in ZIS/MoS₂-4h moves in the direction of the blue arrow. This shows that there is a strong interaction between MoS₂-4h and ZIS. Electron paramagnetic resonance (EPR) was used to confirm the presence of S vacancies (Fig. 2f). The pristine MoS_2 shows lower EPR intensity with a *g* value of 2.004, which was assigned to the S vacancies on the surface [46]. In contrast, MoS_2 -4h shows a sharply increased EPR signal, confirming the presence of a large number of S vacancies in MoS_2 -4h. In addition, it is worth noting that the EPR intensity of ZIS/ MoS_2 -4h significantly decreased compared with MoS_2 -4h. This should be due to the binding effect between the S atoms in ZIS and the uncoordinated Mo atoms in MoS_2 -4h, reducing the number of unpaired electrons, which forms the interfacial Mo–S bond [47]. As reported in literature, the defect MoS_2 -4h, which would eliminate the influence of defects on the photocatalytic activity.

Morphology and microstructure of the catalyst were characterized by scanning electron microscopy (SEM). As shown in Fig. 3a, MoS₂ has a nanoflower-like structure, in which the nanosheet size is about 200 nm. The morphology of MoS₂-xh has no obvious change compared with MoS₂ (Fig. S2). As shown in Fig. 3b, the pure ZIS exhibits a relatively characteristic nanoflower-like structure with an average diameter of 1–2 μ m, consisting mainly of staggered and curled nanosheets. Fig. 3c, d illustrate the SEM images of ZIS/MoS₂-4h, which can be observed that ZIS and MoS₂ are intimately entwined (Fig. S3). The tight interface between materials will promote the transfer and separation of photogenerated charges, thus improving the photocatalytic performance [49,50].

Transmission electron microscopy (TEM) and high-resolution TEM (HRTM) were used to further study the microstructure of the catalysts. When the crystal is defective, the lattice stripes will bend or break, and the defects can be reflected by observing the lattice. Compared with MoS₂, the lattice fringe of MoS₂-4h shows partial bending and fracture, which indicates the existence of defects in MoS₂-4h (Fig. 4a, b and Fig. S4). ZIS and MoS₂ are intimately entwined and a few ZIS nanosheets are located at the surface (Fig. 4c). In the atomic force microscopy image of ZIS/ MoS₂-4h, the ultrathin structure can be found (Fig. S5). In Fig. 4d, two different lattice spacings (*d*) are observed, which are 0.32 and 0.62 nm, respectively, ascribed to the (102) plane of hexagonal ZIS and the (002) plane of MoS₂ [51,52]. The energy



Figure 2 (a) XRD patterns of MoS_2-xh ; (b) XRD patterns of ZIS and ZIS/ MoS_2-xh ; (c-e) Raman spectra of the as-prepared samples; (f) room-temperature EPR spectra of the samples.



Figure 3 SEM images of (a) MoS₂, (b) ZIS, and (c, d) ZIS/MoS₂-4h.

dispersive X-ray spectroscopy (EDS) element distribution of ZIS/MoS₂-4h was characterized (Fig. 4e). As shown in Fig. 4f, Zn, In and S elements are evenly located in the composite, while Mo elements are largely distributed in the middle, indicating the presence of some ZIS nanosheets on the surface.

The surface composition and chemical state of the catalysts and the interaction between MoS_2 -xh and ZIS were studied by X-ray photoelectron spectroscopy (XPS). The survey spectrum reveals that Zn, In, S and Mo peaks coexist in ZIS/MoS₂-4h, which is in accordance with the EDS results (Fig. S6a). The Mo 3d signal in MoS₂-4h shows obvious convolution signs, sug-



Figure 4 HRTEM images of (a) MoS_2 and (b) MoS_2 -4h; (c) TEM image of ZIS/ MoS_2 -4h; (d) HRTEM image of ZIS/ MoS_2 -4h; (e, f) EDS mapping images of ZIS/ MoS_2 -4h.

gesting the unique chemical environment of Mo (Fig. S6b) [53]. The peak of Mo $3d_{3/2}$ for MoS₂-4h is blue shifted compared with



Figure 5 XPS spectra of the as-prepared samples. (a-c) Mo 3d; (d) Zn 2p; (e) In 3d; (f) S 2p.

that of MoS₂, which can be attributable to the decrease of the coordination number of Mo caused by the generation of S vacancy [47]. The Mo 3d signal in ZIS/MoS₂-4h shows similar convolution with MoS₂, which indicates that the combination of S atoms in ZIS and uncoordinated Mo atoms in MoS2-4h reduces S vacancies, thus forming an interface Mo-S bond. Compared with MoS₂, the XPS peak of Mo 3d_{3/2} for ZIS/MoS₂-4h shifts negatively. This shows that there is a strong interaction between ZIS and MoS₂ (Fig. 5a). As shown in Fig. 5b, the XPS peaks of the Mo 3d_{3/2} are negatively shifted for ZIS/MoS₂-4h as compared with those of ZIS/MoS₂. Meanwhile, the peak at 228.4 eV verifies the formation of Mo-S bonds as shown in Fig. 5b, c [47]. Because MoS₂ has some S vacancies (Fig. 2f), ZIS/ MoS₂ also shows a small peak at 228.2 eV, which is lower than that of ZIS/MoS₂-4h. The peaks of Zn $2p_{3/2}$ and Zn $2p_{1/2}$ for pure ZIS are 1022.46 and 1045.46 eV, respectively, which are consistent with numerous reported literature [54]. When ZIS/MoS₂ composites were formed, the characteristic signals of Zn 2p_{3/2} shifted slightly toward higher binding energy. The Zn 2p_{3/2} peak of ZIS/MoS₂ is moving to 1022.64 eV, which is 0.18 eV larger than that of the parent ZIS. The Zn $2p_{3/2}$ peak of ZIS/MoS₂-4h was further shifted by 0.12 to 1022.76 eV compared with that of ZIS/MoS₂ (Fig. 5d). The peaks of In $3d_{5/2}$ and In $3d_{3/2}$ for pure ZIS are 445.13 and 452.69 eV, respectively. The In 3d_{5/2} peak of ZIS/MoS₂ is moving to 445.49 eV, which is 0.36 eV larger than that of the parent ZIS. The In 3d_{5/2} peak of ZIS/MoS₂-4h was further shifted by 0.12 to 445.61 eV compared with that of ZIS/ MoS₂ (Fig. 5e). The XPS peaks of the S 2p were positively shifted for ZIS/MoS₂-4h as compared with the ZIS (Fig. 5f). These results confirm that there is a strong interaction between ZIS and MoS₂ in ZIS/MoS₂-4h. It is Mo-S bond that strengthens the connection between MoS₂ and ZIS to facilitate the electron-hole pair separation.

Photocatalytic H₂ evolution

Fig. 6a shows the photocatalytic hydrogen generation perfor-

mance of ZIS/MoS₂-xh using TEOA as a sacrifice agent. The flower-shaped porous ZIS microspheres have photocatalytic activity for hydrogen production, and the H₂ production rate is 1.6 mmol h^{-1} g⁻¹. The optimized ZIS/MoS₂-4h composite shows a significant hydrogen production rate of 7.6 mmol g⁻¹ h⁻¹, which is approximately 4.75 times that of the original ZIS $(1.6 \text{ mmol g}^{-1} \text{ h}^{-1})$ and 2.05 times that of the ZIS/MoS₂ $(3.7 \text{ mmol g}^{-1} \text{ h}^{-1})$. However, the decreased activity of ZIS/ MoS₂-6h may be due to excessive defects which are not conducive to electron hole separation. Fig. 6b illustrates the photocatalytic hydrogen production performance of ZIS compounded with different masses of MoS₂-4h (5, 10, 15 mg), in which TEOA is used as a hole scavenger. The bare MoS₂ sample is almost inactive. After compounding ZIS with different masses of MoS₂-4h, the optimized photocatalyst, 10 mg ZIS/MoS₂-4h, exhibits a high hydrogen evolution rate of 7.6 mmol $g^{-1} h^{-1}$. The influence of different sacrificial agents on the photocatalytic performance of ZIS/MoS2-4h was explored through photocatalysis experiments. As we can see from Fig. 6c, the best H_2 evolution efficiency was found in TEOA solution, indicating that ZIS/MoS₂-4h is highly selective for sacrificial reagents. The ZIS/ MoS₂-4h hybrid was found to release H₂ persistently over 10 h (Fig. 6d). The stability of ZIS/MoS₂-4h was assessed by allowing the reaction to proceed for 15 h and calculating the H₂ yield in each 5 h reaction. No obvious deactivation was observed in three sequential cycles (Fig. 6e), indicating the lasting stability of the photocatalyst. The XPS spectrum of the catalyst after the reaction proves the stability of ZIS/MoS2-4h. After photocatalytic reaction, the peak of Mo 3d XPS decreases, which may be due to the fact that part of the weakly bound MoS₂ is separated from the surface of ZIS and dissolves in the solution. However, the interfacial Mo-S bond (228.3 eV) still exists, which indicates the importance of interfacial Mo-S bond in improving photocatalytic activity. The XPS peaks of In 3d and S 2p shift slightly, which may be due to slight photo-corrosion of the catalyst (Fig. S6c-f). XRD and field emission SEM characterization of the



Figure 6 (a) Photocatalytic hydrogen generation performance of 5 mg ZIS/MoS₂-xh photocatalyst when ZIS are complexed with 10 mg MoS₂-xh; (b) photocatalytic H₂ evolution performance of ZIS compounded with different masses of MoS₂ (5, 10, 15 mg); (c) photocatalytic water splitting performance of ZIS/MoS₂-4h under visible light irradiation ($\lambda \ge 420$ nm) with different hole scavengers; (d, e) stability of photocatalytic H₂ production for ZIS/MoS₂-4h; (f) wavelength-depended AQE of ZIS/MoS₂-4h.

samples after the photocatalytic reaction further support the stability of ZIS/MoS₂-4h (Fig. S7). Fig. 6f shows that the variation of AQE of ZIS/MoS₂-4h is consistent with the absorption spectrum of ZIS, which indicates that ZIS is a photocatalyst and MoS₂ is a cocatalyst for accelerating electron transfer and providing reactive sites (see Table S1 for the specific AQE values of ZIS/MoS₂-4h). This extraordinary photocatalytic activity is mainly responsible for the photogenerated electrons readily shifted from ZIS to MoS₂ across the heterojunction interface between the two materials, which are intimately connected under the action of Mo–S bonds. The H₂ productivity of this work was compared with other recently reported studies (Table S2). This result demonstrates that well-designed co-catalysts have great advantages in improving photocatalytic performance [55,56].

Photoelectrochemical and photoluminescence (PL) measurements Ultraviolet-visible diffuse reflection absorption spectrum (UVvis DRS) was used to study the optical absorption characteristics of the samples. The absorption range of MoS₂ spans the entire visible region. The UV-vis DRS of MoS₂-*x*h does not change significantly. However, there are some protrusions in MoS₂-*x*h, which may be caused by Sv [36] (Fig. S8a). ZIS has a strong absorption in the visible region, with absorption edge at 525 nm. Compared with pure ZIS, the absorption of the composite in the visible region is significantly enhanced, which may be due to the absorption of light by MoS₂ (Fig. S8b) [57]. By extrapolating the linear part of $(\alpha h v)^{1/2} vs$. photon energy (where α is the absorbance index, *h* is a constant, and *v* is the light frequency), the band gap energy of ZIS is estimated to be 2.58 eV (Fig. S8c). In general, the higher the PL intensity, the higher the probability of photo-generated charge recombination. In order to investigate the transfer efficiency of interfacial electrons from ZIS to MoS_{2} , the PL spectroscopy was performed. Fig. 7a demonstrates that ZIS exhibits a PL emission peak at about 720 nm. PL intensity of ZIS/MoS₂ hybrids was decreased by the incorporation of MoS₂ nanosheets. Furthermore, PL intensity of the ZIS/MoS₂-4h was drastically reduced, verifying that the recombination of photoinduced electron-hole pairs was suppressed. Therefore, the time-resolved PL (TRPL) images were used to investigate the charge transfer capability of the as-prepared samples. The fluorescence lifetime of ZIS/MoS₂-4h (4.66 ns) is shorter than that of ZIS/MoS₂ (6.09 ns), indicating that the photoinduced electrons of ZIS can be rapidly transferred to MoS₂ in ZIS/MoS₂-4h for H₂ formation (Fig. 7b) [58]. Transient photocurrent responses of ZIS and ZIS/MoS₂-xh were compared under visible light irradiation by a 300-W Xe lamp equipped with a cutoff filter ($\lambda > 420$ nm). Fig. 7c shows the photocurrent responses of ZIS and ZIS/MoS₂-xh. Among these samples, ZIS presents the lowest photocurrent density due to its severe recombination of photogenerated electron-hole pairs. The ZIS/MoS₂-xh exhibits a higher photocurrent, demonstrating that loading cocatalysts can effectively accelerate charge transfer [59]. Notably, the photocurrent of the ZIS/MoS₂-4h composite is the highest, which indicates that there is a strong interface interaction between MoS₂ and ZIS in ZIS/MoS₂-4h. These results show that the photogenerated charge continuously migrates and separates from ZIS to MoS₂ through the heterojunction interface between the two materials, which are closely linked under the action of Mo-S bonds. The charge transfer of the synthetized catalysts was then further investigated by electrochemical impedance spectroscopy (EIS). As shown in Fig. 7d, the arc of ZIS/MoS₂-4h is



Figure 7 (a) PL spectra of ZIS, ZIS/MoS₂ and ZIS/MoS₂-4h; (b)TRPL spectra of ZIS/MoS₂ and ZIS/MoS₂-4h; (c) transient photocurrent responses; (d) Bode plots of ZIS and ZIS/MoS₂-xh.

smaller than that of ZIS and ZIS/MoS₂ [60]. These results show that the Mo–S bond can strengthen the interface interaction of heterojunction and improve the electron separation efficiency of ZIS.

Mechanism analysis

According to the above experimental results, the possible reaction mechanism of photocatalytic hydrogen production of ZIS/ MoS₂-4h is proposed in Fig. 8. Under visible light irradiation, the valence band electrons of ZIS are excited to the conduction band, and the flat band potentials of ZIS can be determined by Mott-Schottky measurement (Fig. S8d). According to the formulas E(NHE) = E(Ag/AgCI) + 0.197 and E(ZIS vs. Ag/AgCI) =-0.88 eV (*E*(NHE) is the standard hydrogen electrode potential) (Fig. S8d), the E_{CB} value of ZIS is estimated to be -0.68 eV. The $E_{\rm VB}$ value of ZIS was calculated to be 1.90 eV according to the formula $E_{\rm VB} = E_{\rm CB} + E_{\rm g}$ (where $E_{\rm VB}$ refers to valence band, $E_{\rm CB}$ refers to conduction band and E_g is the band gap) because of the low Fermi energy level of MoS₂ [46]. Thermodynamically, the photogenerated electrons are readily transferred from ZIS to MoS₂ through the heterojunction interface between the two materials, which are closely linked under the action of Mo-S bonds. MoS₂ acts as a photogenic electron acceptor and inhibits electron hole recombination. The electrons on MoS2 will reduce H₂O to H₂. In the meantime, the holes on ZIS are consumed by the sacrificial agent. Photoelectric measurements show that ZIS/ MoS₂-4h has effective charge transfer. The results show that the Mo-S bond can strengthen the interface interaction of heterojunction and improve the electron separation efficiency of ZIS.



Figure 8 Possible mechanism of photocatalytic hydrogen evolution of ZIS/ MoS₂-4h under visible light.

CONCLUSION

In summary, MoS_2 with rich S vacancies was prepared by $NaBH_4$ etching, which improved the photocatalytic H_2 evolution activity of ZIS. The binding effect between S in ZIS and the uncoordinated Mo in Sv-MoS₂ forms the interfacial Mo–S bond. Introducing interfacial chemical bonds to enhance the interfacial interaction between components is an effective means to improve photocatalytic hydrogen production. The optimized ZIS/MoS₂-4h composite shows a significant hydrogen production rate of 7.6 mmol g⁻¹ h⁻¹, which is approximately 4.75 times that of the original ZIS (1.6 mmol g⁻¹ h⁻¹) and 2.05 times that of

the ZIS/MoS₂ (3.7 mmol g⁻¹ h⁻¹) and its photocatalytic activity remains unchanged after three cycles for 15 h. The improvement of activity and stability of ZIS/MoS₂-4h is due to the fact that photogenerated electrons are easily transferred from ZIS to MoS₂ through the tightly connected heterojunction interface under the action of Mo–S bond. This work reveals the influence of interface chemical bonds on the photocatalytic activity of ZIS/ MoS₂, and provides a simple and effective method for designing excellent photocatalytic cocatalysts through interface engineering.

Received 14 January 2023; accepted 21 March 2023; published online 30 June 2023

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Acknowledgements This work was supported by the National Key Research and Development Program of China (2021YFA1501500), the National Natural Science Foundation of China (22033008, 22220102005, and 22171265), and Fujian Science & Technology Innovation Laboratory for Optoelectronic Information of China (2021ZZ103).

Author contributions Gao S and Ye S designed and engineered the experiments; Ye S performed the experiments and the characterizations; Ye S wrote the paper with support from Gao S and Cao R. All authors contributed to the general discussion.

Conflict of interest The authors declare that they have no conflict of interest.

Supplementary information Experimental details and supporting data are available in the online version of the paper.



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构建具有微妙的原子级紧密接触的ZnIn₂S₄/Sv-MoS₂ 光催化剂:增强界面相互作用以改善可见光下的光催 化产氢性能

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摘要 催化剂与助催化剂之间的低电荷分离效率严重限制了光催化性能. 催化剂与助催化剂之间的强界面相互作用可以提高电荷分离效率. 通过引入界面化学键增强组分间的界面相互作用是提高光催化性能的有效手段之一. 本文合成了ZnIn₂S₄ (ZIS)/Sv-MoS₂光催化剂, ZIS中的S原子与Sv-MoS₂中未配位Mo原子之间的键合作用形成了界面Mo-S键, 这极大地提高了ZIS的光催化活性. 采用不同的NaBH₄蚀刻时间制备了MoS₂-xh. 优化后的ZIS/MoS₂-4h复合材料的产氢速率为7.6 mmol g⁻¹ h⁻¹, 是原ZIS (1.6 mmol g⁻¹ h⁻¹)的4.75倍, 是ZIS/MoS₂ (3.7 mmol g⁻¹ h⁻¹)的2.05倍. 非凡的光催化活性可归因于光生电子在Mo-S键的作用下更容易从ZIS转移到MoS₂. 光电测量表明, ZIS/MoS₂-4h具有有效的电荷转移.本工作揭示了引入界面化学键对ZIS/MoS₂光催化活性的影响,为通过界面工程设计优良的助催化剂提供了一种简单有效的方法.