Dynamic shrinkage of metal-oxygen bonds in atomic Co-doped nanoporous RuO₂ for acidic oxygen evolution

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ABSTRACT The design of highly active and stable catalysts for the oxygen evolution reaction (OER) in acidic media has become an attractive research area for the development of energy conversion and storage technologies. However, progress in this area has been limited by the poor understanding of the dynamic active structure of catalysts under realistic OER conditions. Here, an atomic Co-doped nanoporous RuO₂ electrocatalyst, which exhibited excellent OER activity and stability in acidic conditions, was synthesized through annealing and etching of a nanoporous Co-Ru alloy. *Operando* X-ray absorption spectroscopy results confirmed that the etching strategy produced abundant oxygen vacancies around the metal centers in the atomic Co-doped nanoporous RuO₂ electrocatalyst. These vacancies created contracted metal-oxygen ligand bonds under realistic OER conditions. The dynamic structural evolution of the synthesized electrocatalyst allowed them to experience lower kinetic barriers during OER catalysis, resulting in enhanced catalytic activity and stability. This study also provided atomic details on the active structure of the electrocatalyst and the influence of their structural evolution on OER activity.

Keywords: acidic oxygen evolution reaction, nanoporous catalysts, *operando* X-ray absorption spectroscopy, dynamic structural evolution

INTRODUCTION

The oxygen evolution reaction (OER) plays a critical role in the development of energy conversion and storage technologies, such as electrochemical water splitting, carbon dioxide reduction, and rechargeable metal-air batteries [1–4]. In contrast to alkaline OER, the acidic OER is more attractive due to the higher ionic conductivity and higher-grade purity of the products [5–7]. However, there are many challenges associated with the development of highly active and durable OER electrocatalysts that can operate in acidic media [8–10]. To date, ruthenium oxide (RuO₂)-based materials are one of the most promising acidic OER catalysts. Unfortunately, the poor intrinsic activity and sluggish kinetics of OERs still hinder their practical applications [11–15]. Therefore, rational design of

catalysts is required to enhance the OER performance.

A powerful strategy to increase the OER activity of metal oxide catalysts is the introduction of heteroatoms to alter the electronic and geometrical structure of the active sites of the catalyst [16–23]. Particularly, the introduction of heteroatoms into the host lattice generates oxygen vacancies, thus tuning the electronic density of the host element. However, the effects of heteroatoms doped on the catalyst's surface and the nature of the actual catalytic sites remain vague, especially under realistic OER conditions. Therefore, it is highly desirable to develop robust metal-doped catalysts for acidic OER and understand the dynamic behavior of heteroatoms and host atoms in the catalysts.

Here, we rationally designed an atomic Co-doped nanoporous RuO₂ as an acidic OER electrocatalyst, which was prepared by an annealing and electrochemical etching strategy. The atomic Co-doped nanoporous RuO₂ catalyst was endowed with an abundance of oxygen vacancies, as well as three-dimensional (3D) accessibility for reactant molecules. The prepared catalyst exhibited a small overpotential of 169 mV at a current density of 10 mA cm^{−2}, a low Tafel slope of 46 mV dec^{−1}, and excellent durability. These characteristics suggest that the prepared catalyst showed one of the best performances when compared with all reported acidic OER electrocatalysts [16–20]. *Operando* X-ray absorption spectroscopy analyses and density functional theory (DFT) calculations revealed that the dynamic shrinkage of metal-oxygen ligand bonds in the atomic Co-doped RuO₂ not only accelerated the OER kinetics but also led to the formation of the stable surface structure of the catalyst during the OER.

RESULTS AND DISCUSSION

The nanoporous Co₂₀Ru₈₀ alloy (denoted as np-Co₂₀Ru₈₀) was prepared through electrochemical dealloying of a single-phase Co₉₅Ru₅ precursor in an HCl solution (see Method and Figs S1–3 in Supplementary information) [10]. The np-Co₂₀Ru₈₀ was then annealed at various temperatures for 4 h in air. The products obtained with treatment under 150, 250, and 350°C were denoted as Co/np-RuO₂-150, Co/np-RuO₂-250, and Co/np-RuO₂-350, respectively. Finally, the cyclic voltammetry (CV) was conducted in the 0.5 mol L^{−1} H₂SO₄ electrolyte to further etch the Co element (Figs S4–6). The corresponding final pro-

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ducts (catalysts) were denoted as $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-150}$, $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$, and $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-350}$. X-ray diffraction (XRD) patterns of the samples annealed at different temperatures only showed the peaks for RuO_2 and metal Ru (at 150°C) with low crystallinity (Fig. 1a). Meanwhile, annealing at 150°C was not sufficient to completely oxidize Ru, resulting in the presence of metal Ru in the $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-150}$. Scanning electron microscopy (SEM) image revealed a nanorod morphology for $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ (Fig. S7). Transmission electron microscopy (TEM) image of $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ revealed the formation of ultrafine ligament and nanoporous structures with a pore size of less than 10 nm (Fig. 1b and Fig. S8). Furthermore, the lattice fringe spaces of rutile RuO_2 (110) were observed (Fig. 1c, d). The selected area electron diffraction (SAED) characterization of $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ showed continuous Debye rings, which can be indexed to the (110), (101), and (200) planes of RuO_2 (Fig. S8). The energy-dispersive X-ray spectroscopy (EDS) elemental mapping demonstrated that Co and Ru elements were uniformly distributed in the catalyst (Fig. 1e). In addition, N_2 adsorption-desorption measurements verified that $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ possessed a large Brunauer-Emmett-Teller (BET) surface area ($116.27\text{ m}^2\text{ g}^{-1}$ with an average nanopore size of ca. 5 nm) (Fig. 1f, g).

X-ray photoelectron spectroscopy (XPS) analysis was carried out to assess the chemical state at the surface of the different catalysts. The $\text{Ru } 3p_{3/2}$ spectra of $\text{np-Co}_{20}\text{Ru}_{80}$ showed two peaks that were assigned to Ru^0 (461.1 eV) and Ru^{4+} (462.7 eV) (Fig. 2a). The peaks corresponding to the $\text{Ru } 3p_{3/2}$ peaks of $\text{Co}/$

$\text{np-RuO}_2\text{-250}$ and $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ were assigned to Ru^{4+} and Ru^{3+} (464.2 eV) [23]. These results suggested the oxidation of Ru after annealing treatment and CV etching. The Co 2p spectra of $\text{np-Co}_{20}\text{Ru}_{80}$, $\text{Co}/\text{np-RuO}_2\text{-250}$, and $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ showed the oxidation of Co after the annealing treatment and CV etching, as evidenced by the positive shift of Co 2p peaks for $\text{Co}/\text{np-RuO}_2\text{-250}$ and $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ (Fig. S9). The O 1s spectra of $\text{Co}/\text{np-RuO}_2\text{-250}$ and $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ can be divided into four peaks, which corresponded to the lattice O (529.2 eV), OH^- (530.5 eV), oxygen vacancy (O_V) species (531.9 eV), and adsorbed H_2O molecules (533.0 eV) (Fig. 2b) [16]. It was found that the O_V species peak for $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ was much stronger compared with the O_V species peak for $\text{Co}/\text{np-RuO}_2\text{-250}$. This indicated the increase in the O_V due to CV etching, which may be explained by the electroneutrality principle (i.e., the decrease in the number of cations (Co cation leaching) required a decrease in the number of anions (the loss of O^{2-}) to appropriately balance the charge of catalysts) [16].

X-ray absorption spectroscopy (XAS) was performed to further probe the electronic and atomic structures of the catalysts. Fig. 2c shows the X-ray absorption near-edge structure (XANES) spectra at Ru k -edge of $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$, together with $\text{Co}/\text{np-RuO}_2\text{-250}$, Ru foil, and RuO_2 . The absorption edge of $\text{Co}/\text{np-RuO}_2\text{-250}$ was located between those of Ru foil and RuO_2 , indicating that the average valence state of Ru in $\text{Co}/\text{np-RuO}_2\text{-250}$ was between 0 and 4 (see Fig. S10). After CV etching, the Ru species in $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ were further oxidized, as evidenced by the positive shift of absorption edge. Fig. 2d exhibits

Figure 1 Morphological and structural characterizations of the prepared catalysts: $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-150}$, $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$, and $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-350}$. (a) XRD patterns of the different catalysts. (b) TEM image of $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$. (c) High-resolution TEM image of $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$. (d) Magnified HRTEM image and the corresponding intensity line profile of $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$ from (c). (e) EDS mapping profiles of $\text{Co}_{\text{CV}}/\text{np-RuO}_2\text{-250}$. (f) BET analyses of the different catalysts. (g) Pore size distributions of the different catalysts. Scale bars: 10 nm (b), 5 nm (c), 1 nm (d), and 10 nm (e).

Figure 2 XPS and XAS characterizations of the prepared catalysts. XPS spectra of (a) Ru 3p_{3/2} and (b) O 1s regions for np-Co₂₀Ru₈₀, Co/np-RuO₂-250, and Co_{Cv}/np-RuO₂-250. Ru *k*-edge (c) XANES spectra and (d) the corresponding FT-EXAFS spectra for different catalysts. Co *k*-edge (e) XANES spectra and (f) the corresponding FT-EXAFS spectra for different catalysts. (g) Schematic illustration of the Co leaching and the formation of vacancies during the CV etching process.

the Fourier transforms of extended X-ray absorption fine structure (FT-EXAFS) spectra at Ru *k*-edge of Co_{Cv}/np-RuO₂-250 and Co/np-RuO₂-250. The peaks of Co_{Cv}/np-RuO₂-250 and Co/np-RuO₂-250 located at ~1.5 Å were assigned to the Ru–O scattering feature. The intensity of Ru–O peak for both Co_{Cv}/np-RuO₂-250 and Co/np-RuO₂-250 was much lower than that of RuO₂, which was attributed to the coordination deficiency of Ru [16]. It is worth noting that the Ru–O peak of Co_{Cv}/np-RuO₂-250 showed higher peak intensity than that of Co/np-RuO₂-250. This suggested further bulk oxidation during the CV process, which led to the increase in the average Ru–O coordination number. The peak located at ~2.3 Å in the Ru *k*-edge FT-EXAFS spectra was assigned to the Ru–Ru scattering feature. The increase in the Ru–Ru peak intensity after CV etching also indicated a change of the coordination structure for residual alloy clusters due to oxidation. In addition, the Co *k*-edge XANES spectrum of Co/np-RuO₂-250 showed a higher energy absorption edge than that of Co foil, indicating the oxidation of Co species in Co/np-RuO₂-250 (Fig. 2e). The absorption edge of

Co_{Cv}/np-RuO₂-250 also shifted to higher energy, and this was accompanied by the increase of the white line peak, suggesting the oxidation of residual Co species. The corresponding Co *k*-edge FT-EXAFS spectrum of Co_{Cv}/np-RuO₂-250 showed similar scattering features with that of Ru *k*-edge FT-EXAFS spectrum, suggesting the atomic doping of Co into RuO₂ (Fig. 2f and Fig. S11). Meanwhile, the Co *k*-edge FT-EXAFS spectrum of Co_{Cv}/np-RuO₂-250 showed a decrease of Co–O peak compared with that of Co/np-RuO₂-250 (Fig. 2f). This confirmed that the oxygen vacancies were created around the residual Co centers after the leaching of Co (Fig. 2g). This not only led to the creation of more active sites but also optimized the electronic structure of RuO₂ [16].

The OER activity of the catalysts was investigated in O₂-saturated 0.5 mol L^{−1} H₂SO₄ by using a typical three-electrode system. Fig. 3a shows the linear sweep voltammetry (LSV) curves of Co_{Cv}/np-RuO₂-150, Co_{Cv}/np-RuO₂-250, and Co_{Cv}/np-RuO₂-350, together with commercial RuO₂ as the benchmark. A notable “decrease-increase” trend in the overpotentials of the

Figure 3 OER performance of the studied catalysts. (a) OER polarization curves of $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -150, $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250, $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -350, and RuO_2 in $0.5 \text{ mol L}^{-1} \text{ H}_2\text{SO}_4$. (b) The corresponding Tafel plots from (a). (c) Double-layer capacitances (C_{dl}) of different catalysts. (d) The corresponding ECSA-normalized polarization curves from (a). (e) Nyquist plots of different catalysts. (f) The comparisons of kinetics (Tafel slope) and activity (the overpotential required to achieve 10 mA cm^{-2}). (g) Time-dependent current density curve under static overpotential of $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 in acidic electrolyte.

catalysts was observed with the increase of annealing temperature from 150 to 350°C . The $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 showed the best OER performance in terms of the extremely low overpotential of 169 mV at the applied current density of 10 mA cm^{-2} . This value was 31, 61, and 101 mV lower than those obtained from $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -150, $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -350, and RuO_2 , respectively. A small Tafel slope of 46.1 mV dec^{-1} was derived for $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250. This value was significantly lower than those obtained for $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -150, $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -350, and RuO_2 (Fig. 3b). The electrochemical double-layer capacitance (C_{dl}) measurements showed the largest C_{dl} value for $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 when compared with the other prepared catalyst (Fig. 3c and Fig. S12), suggesting that more accessible active sites were constructed on $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250. After normalizing the current density to the electrochemical active surface area (ECSA), the $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 still produced the highest current density among the prepared catalysts, emphasizing the high intrinsic activity of $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 (Fig. 3d). Furthermore, the electrochemical impedance spectroscopy (EIS) measurements revealed the smallest charge transfer resistance obtained with $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 compared with the other prepared catalysts (Fig. 3e). According to the above results, the $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 also exhibited superior OER performance when compared with

the available Ru-based and Ir-based OER catalysts in acidic solution (Fig. 3f and Table S1) [4,7,19,24–27]. Moreover, the chronoamperometry durability test of $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 revealed that it was stable for 20 h in $0.5 \text{ mol L}^{-1} \text{ H}_2\text{SO}_4$ (Fig. 3g).

To understand the high OER activity and stability of the $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 catalyst, *operando* XAS analyses were performed on this catalyst under realistic OER and open-circuit voltage (OCV) conditions, which is an attractive technique for exploring the dynamic structure of catalysts [21,28,29]. As shown in Fig. 4a, the shift to the higher energy of the absorption edge of Ru *k*-edge XANES spectra of the catalyst was higher in the OER than in the OCV conditions. This suggested the increase of the oxidation state for Ru species in the OER conditions. Meanwhile, the Ru–O peaks in the corresponding FT-EXAFS spectra displayed a low-*R* shift in the OER conditions, which indicated a shrinkage of Ru–O bonds (Fig. 4b). The *operando* Co *k*-edge XANES spectra of $\text{Co}_{\text{CV}}/\text{np-RuO}_2$ -250 obtained under realistic OER conditions showed similar changes to that of the *operando* Ru *k*-edge XANES spectra (Fig. 4c). This suggested the oxidation of Co during OER. In addition, the spectra in the inset of Fig. 4c show no change in peak A but a decrease in the peak intensity of peak B under higher applied voltage (1.5 V vs. reversible

Figure 4 Operando XAS measurements and DFT calculations for the CoCV/np-RuO₂-250 catalyst. Operando Ru *k*-edge XANES spectra (a) and the corresponding FT-EXAFS spectra (b) collected under OCV, 1.4 V vs. RHE, and 1.5 V vs. RHE. Operando Co *k*-edge XANES spectra (c) and the corresponding FT-EXAFS spectra (d) collected under OCV, 1.4 V vs. RHE, and 1.5 V vs. RHE. (e) Schematic illustration of the OER mechanism determined by the operando XAS analyses. (f) FT-EXAFS fitting results from (d). (g) The calculated free energy diagrams of the LOM mechanism.

hydrogen electrode (RHE)). This suggested a change in the coordination structure of the catalyst. This can be further confirmed by the corresponding FT-EXAFS spectra (Fig. 4d), which show a sharp shrinkage of the Co-O bonds under the applied voltage of 1.5 V vs. RHE. It should also be noted that the Co-O bonds showed greater shrinkage than the Ru-O bonds under realistic OER conditions. This was attributed to the different properties of different metal-oxygen ligand bonds and the more stable Ru-O ligand bonds in the host lattice compared with the Co-O ligand bonds of Co dopant atoms. Based on the *ex-situ* results (see Fig. 2g), the Co leaching during the CV etching produced abundant holes on the oxygen ligands of Co and Ru. The resulting metal-oxygen ligand bonds are most likely shortened under realistic OER conditions, which may play a crucial role in the excellent activity of catalysts (Fig. 4e) [30]. In addition,

the stable surface structure resulted from the shortening of metal-oxygen ligand bonds is beneficial to enhancing the stability of the catalyst during the OER conditions [31]. To further study the structure-activity relationships, the FT-EXAFS fitting was performed (Fig. 4f). The results suggested the shrinkage of Co-O bonds from 1.97 to 1.90 Å with the increase of applied voltage (Fig. S13 and Table S2). Based on these changes, DFT calculations were employed to assess the role of bond shrinkage in the OER process (see Methods in Supplementary information). The Co/RuO₂ models with and without bond shrinkage were established according to the FT-EXAFS fitting results. The O vacancies (O_v) were identified as the active sites, and the lattice oxygen oxidation mechanism (LOM) was calculated [16,32,33] (Fig. 4g). The formation of (OH)-O_v-(OH) after the second H₂O attack was identified as the rate-determining step

(RDS). The shortening of metal-oxygen ligand bonds not only reduced the free energy barrier of RDS but also lowered the energy barrier of O-O coupling (the formation of OOH*). These suggested a more favorable OER kinetics under high applied voltage conditions.

CONCLUSION

In conclusion, we developed an atomic Co-doped nanoporous RuO₂ catalyst with abundant oxygen vacancies around the Co centers for OER in acidic media. The prepared Co_{CV}/np-RuO₂-250 catalyst exhibited remarkable catalytic activity and stability toward OER in acidic solutions, outperforming most of the reported electrocatalysts, including IrO₂-based and RuO₂-based materials. *Operando* XAS studies showed the dynamic shortening of the metal-oxygen ligand bonds in Co_{CV}/np-RuO₂-250 under realistic OER conditions. DFT calculations confirmed that the shorter metal-oxygen ligand bonds were probably responsible for the higher intrinsic activity and stability of the prepared catalyst. The shorter ligand bonds reduced the free energy barrier of RDS, thus creating a favorable OER environment. The proposed mechanisms in this work could significantly help in the rational design and synthesis of high-performance acidic OER electrocatalysts.

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Conflict of interest The authors declare no conflict of interest.

Supplementary information Experimental details and supporting data are available in the online version of the paper.

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原子级Co掺杂纳米多孔RuO₂催化剂中金属-氧配体键的动态收缩加速酸性氧析出反应

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摘要 设计在酸性介质中具有高活性和高稳定性的氧析出反应(OER)催化剂对能量转换和储存技术的发展具有重要意义。然而,在实际OER条件下催化剂的原子结构会发生变化,且目前对其动态活性结构的认识仍然不足。本文中,我们通过退火和蚀刻纳米多孔Co-Ru合金合成了具有优异酸性OER活性和稳定性的原子级Co掺杂纳米多孔RuO₂催化剂。原位X射线吸收光谱证实:蚀刻策略可以在原子级Co掺杂纳米多孔RuO₂的金属中心周围产生丰富的氧空位,从而在实际OER条件下产生收缩的金属-氧配体键。这种动态结构演变降低了催化活性位点在OER过程中的动力学势垒,因而催化剂的催化活性和稳定性大幅提高。本研究结果揭示了催化剂活性结构的原子细节以及它们的结构演化对催化活性的影响。