



Coexistence of large positive and negative magnetoresistance in Cr₂Si₂Te₆ ferromagnetic semiconductor

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ABSTRACT Magnetoresistance (MR) phenomenon couples the electron transport with magnetic field, which has been at the forefront of condensed matter physics and materials science. Large-MR behaviors are of particularly importance for magnetic sensor and information memory applications, and their scarcity has aroused intensive research. Moreover, due to the different physical origins, combination of large positive and negative MR (pMR and nMR) in one single compound has rarely been reported. In present work, we achieved a coexistence of large pMR and nMR in Cr₂Si₂Te₆ ferromagnetic semiconductor single crystal with different field configurations. Specifically, a large nMR of about -60% was obtained under the in-plane field, while a large pMR higher than 1000% took over in the out-of-plane direction. We attribute this field direction-sensitive dualistic large MR behavior to the competition and cooperation effect from the ferromagnetic interaction, orbital scattering and electronic correlation that coexist in Cr₂Si₂Te₆, which contribute to nMR, pMR, and nMR, respectively, in dominated temperature and field ranges, and show different weights under different field directions. The elucidated multiple MR mechanism in this ferromagnetic semiconductor will shed light on the pursuit of coexistence of large pMR and nMR for field-sensitive device applications.

Keywords: large pMR and nMR, dualistic magnetoresistance, Cr₂Si₂Te₆, ferromagnetic semiconductor, single crystal

INTRODUCTION

Magnetoresistance (MR) describes the tendency of a material to change the value of its electrical resistance in an externally applied magnetic field, which couples the electron transport with magnetic field [1], and has always been at the cutting edge of condensed matter physics, solid-state chemistry and materials science [2,3]. In terms of application, materials that exhibit large MR (whether positive or negative), like giant MR (GMR) and colossal MR (CMR) [4–7], have already found use in devices such as magnetic sensors and memories [8,9], as well as hard drives [10]. However, large-MR phenomena and materials are uncommon, which motivates us to explore new and potential candidates for further progress in this field. Moreover, con-

sidering the field-sensitive device applications, hunting for material systems that combine large positive and negative MR (pMR and nMR) together is of great significance. However, due to the different origins of large pMR and nMR, coexistence of them in a single compound has rarely been reported. Large pMR mainly comes from the compensation effect or ultrahigh carrier mobility in nonmagnetic semimetals [11,12], while large nMR is usually found in artificially ferromagnetic (FM) multilayer systems with magnetic coupling interaction [4,5], or manganite perovskites arising from the interaction between the conduction and localized electron spins in different orbitals [6,13,14]. Thus, the different physical and chemical origins, and the corresponding material systems make it quite challenging to acquire both large pMR and nMR.

Intrinsically FM semiconductors (FMS's) with both ferromagnetism and semiconductor features hold great promise for fundamental physics and future applications in spintronics [15,16]. Among them, van der Waals (vdW) FMS's, like Cr₂X₂Te₆ (X = Si, Ge) and XI₃ (X = Cr, V) have received intense concern due to their 2D vdW lattice coupled with ferromagnetism [17–23], which is beneficial to studying the origin and implication of magnetic interaction in few/mono-layer crystals, as well as novel spintronic vdW heterostructures [24]. More importantly, due to the crystal anisotropy, layered FMS's can exhibit magneto-crystalline anisotropy, which may offer an opportunity for achieving both large pMR and nMR. Therefore, it is rewarding to explore the MR behavior in vdW FMS for the sustainability of this burgeoning field. As a typical vdW FMS, bulk Cr₂Si₂Te₆ shows 2D Ising-like FM behavior with the Curie temperature (*T_c*) of 32–34 K [25,26]. It has a layered structure which contains [Te–Cr_{2/3}(Si₂)_{1/3}–Te] sandwich layers stacking along *c* axis (see Fig. 1a). The infrequent [Si₂Te₆] ethane-like groups locate at the center of the hexagons formed by [CrTe₆] octahedron, which is reported to be responsible for Cr₂Si₂Te₆'s molecular-like sharp and localized electronic density of states (DOS) [27]. In the meantime, Cr₂Si₂Te₆ also exhibits magneto-crystalline anisotropy due to its crystal anisotropy, which spurs us to learn more about its anisotropic magnetotransport properties.

Here, we uncover the magnetotransport behavior of Cr₂Si₂Te₆ single crystal with strong MR anisotropy between the in-plane

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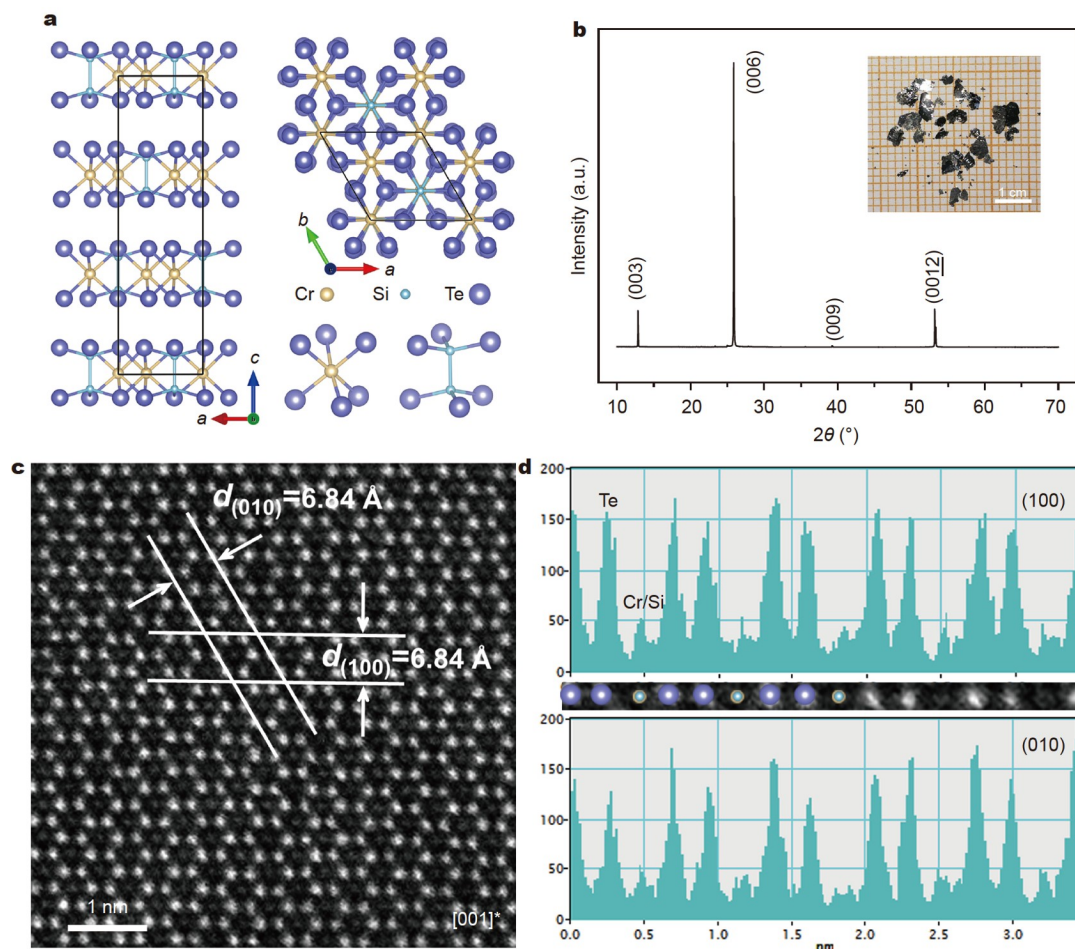


Figure 1 Structure and phase characterizations. (a) Schematic crystal structure of $\text{Cr}_2\text{Si}_2\text{Te}_6$ along the $[010]$ and $[001]$ directions, with the $[\text{CrTe}_6]$ octahedron and ethane-like $[\text{Si}_2\text{Te}_6]$ species. (b) XRD patterns for a piece of $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal. Inset: the optical photograph. (c) HAADF-STEM image of $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal obtained along the $[001]$ direction. (d) Intensity profiles along the $[100]$ and $[010]$ axes, respectively.

($H\parallel ab$) and out-of-plane ($H\perp ab$) magnetic field directions, and acquire the coexistence of large pMR and nMR under different field configurations. Specifically, a large nMR of about -60% is obtained below 22.5 K under $\sim 1\text{ T}$ and $H\parallel$ field. Nevertheless, as soon as the field is rotated to $H\perp$ direction, this large nMR is overwhelmed by the emergence of a pronounced pMR, and a large pMR higher than 1000% takes over at temperatures around T_c . A competition and cooperation mechanism among FM, orbital and electronic correlation is proposed to explain the unusual and intriguing MR behavior observed in $\text{Cr}_2\text{Si}_2\text{Te}_6$. FM, orbital and electronic correlation contribute nMR, pMR, and nMR, respectively, in dominated temperature and field ranges, and show different weighted contribution of positive and negative sources under different field directions. This multiple MR mechanism elucidated in present work will deepen our understanding of the microscopic magneto-transport process and shed light on the pursuit of qualified materials for field-sensitive device applications.

EXPERIMENTAL SECTION

Materials and synthesis

Cr powder (3N), Si powder (3N), and I_2 (4N) were purchased

from Sinopharm Chemical Reagent Co., Ltd.; Te powder (4N) was purchased from Shanghai Aladdin Biochemical Technology Co., Ltd. All chemicals were used as received without further purification.

$\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystals were grown by a chemical vapor transport (CVT) method. A stoichiometric mixture of Cr, Si, Te powders was sealed in evacuated silica tubes with I_2 as the transport agent according to the amount of 5 mg cm^{-3} . The starting materials were then placed inside a two-zone furnace to react over a period of 7 days between 923 and 1123 K , and then naturally cooled to room temperature. After the reaction, a fine dark-grey crystalline powder was obtained, within which some small lamellar crystals with silver metallic luster could be found.

Characterizations

The X-ray diffraction (XRD) data were taken with $\text{Cu K}\alpha$ ($\lambda = 0.15418\text{ nm}$) radiation using a Rigaku Dmax diffraction system. The microstructure and element analysis were performed on a JEOL JSM-6700F scanning electron microscopy (SEM) system with energy-dispersive X-ray spectroscopy (EDS), confirming a stoichiometric $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal. High-resolution transmission electron microscopy (HRTEM) images and high-angle annular dark-field scanning TEM (HAADF-STEM) images were

collected on a spherical aberration-corrected JEOL ARM-200F field-emission transmission electron microscope.

Physical property measurements

The magnetic measurements were performed on a Quantum Design MPMS-XL3 system in a temperature range from 4 to 200 K with a temperature step of 2 K and magnetic fields up to 5 T. The magnetotransport properties were performed on a Quantum Design PPMS system (PPMS[®]DynaCoolTM) using Horizontal Rotator Option in the temperature range of 20–300 K with the magnetic field up to 14 T. “Direct current (DC) resistivity” mode was used in the high-temperature range (above 50 K) while “electrical transport option (ETO)” mode was used in the low-temperature range (below 50 K), because the resistance increases sharply at low temperatures and exceeds the test range when using four-point “DC resistivity” mode. Electrical contacts were prepared by indium welding with a two or four-probe configuration depending on the electrical transport measurement modes. Single crystals with typical dimension of $2.5 \times 1.5 \times 0.3 \text{ mm}^3$ were used in magnetotransport properties. Repeated experiments on different crystals showed similar behavior, confirming the results presented in this paper.

ARPES measurements

Angle resolved photoemission spectroscopy (ARPES) experiments were performed with 20-eV photons at the beam line 13U of the National Synchrotron Radiation Laboratory (NSRL) in Hefei, China, using a Scienta R4000 electron spectrometer. The angular resolution was 0.3° , and the combined instrumental energy resolution was higher than 20 meV. All samples were cleaved *in situ* and measured under a vacuum higher than 8×10^{-11} mbar.

RESULTS AND DISCUSSION

Structure and composition characterizations

Single crystals of $\text{Cr}_2\text{Si}_2\text{Te}_6$ were obtained by the CVT method, which are small mirror-like shining platelets with lateral dimensions of several millimeters (inset in Fig. 1b). The clear and sharp (00 l) peaks observed from XRD patterns (Fig. 1b) indicate that the surface of the crystal is normal to the c axis. EDS analysis of the platelet gives an element composition of 2.2:0.1:6.04 (see Table S1), very close to the stoichiometry of $\text{Cr}_2\text{Si}_2\text{Te}_6$. Element mappings of $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal (see Fig. S1) show homogeneous distribution of Cr, Si, and Te elements. The crystal structure viewed along [001] is illustrated by HAADF-STEM image in Fig. 1c. According to the Z contrast principle, the brighter points in Fig. 1c correspond to Te atom columns, which build up a hexagonal close packing, while the “-Cr-Cr-Si₂-” atom columns are inserted perpendicularly at each center position of the hexagons. Intensity profiles along the [100] and [010] directions (Fig. 1d) clearly feature alternate intense and weak peaks due to local variation of elements along the lines, as labeled in Fig. 1d. Lattice parameter with $a = b = 6.84 \text{ \AA}$ can be easily obtained from the relative distance between the corresponding atoms, consistent with the theoretical values [25]. These results suggest that we have obtained high-quality $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystals.

Electrical and magnetic property characterizations

Fig. 2a presents the band structure of $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystals

measured by ARPES at different temperatures. The identical dispersive features at 30, 50 and 100 K indicate that the valence band structure of $\text{Cr}_2\text{Si}_2\text{Te}_6$ remains unchanged with temperature variation and keeps consistency from the FM to paramagnetic (PM) phase, although there is a slight rigid shift to deeper binding energy in the spectrum taken at 30 K due to the charging effect. From the broad range dispersion along the Γ - K direction, we could identify both dispersive feature near the valence band maximum (VBM) and rather flat bands at $\sim 1.5 \text{ eV}$ below the VBM, consistent with the previous report [28]. This flat band leads to a sharp and localized electronic DOS, as indicated by the energy distribution curve (EDC) intensity derived from 30 K ARPES data (Fig. 2b), which is attributed to the $[\text{Si}_2\text{Te}_6]^{6-}$ -like anion species as stated before [27]. The resistance of $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal (sample 1, marked as #S1) was first investigated using four-probe “DC resistivity option” (Fig. 2c, black hollow data points), which increases monotonically as the temperature decreases and quickly exceeds the measuring range of instrument below 50 K. The data between 150 and 300 K can be fitted by the Arrhenius’s equation, suggesting a thermal excitation electrical transport mechanism and the thermal activation energy E_{a1} is about 140 meV. Then, a high-impedance two-probe configuration (Fig. S2a) was further employed on the same single crystal with “ETO” (Fig. 2c, red solid data points). The resistance value obtained by ETO is in good agreement with the former below 70 K, and finally exceeds the measuring range again below $\sim 20 \text{ K}$. Therefore, in present work, electronic transport data obtained by four-probe were adopted at temperatures above 50 K, while at temperatures between 20 and 50 K, data measured by two-probe were adopted. Hall coefficient measurement (Fig. S2b) was taken to determine the carrier concentration (n) and mobility (μ). However, due to quite obvious background of MR in the low-temperature range (as discussed below), only n data above 100 K are obtained and shown in Fig. 2d, which also follows the Arrhenius’s law with a thermal activation energy E_{a2} of 141 meV, well consistent with the value of E_{a1} . The deduced μ is shown in Fig. S3, which follows a $T^{-3/2}$ dependence in the high-temperature range, suggesting the acoustic phonon scattering [29,30].

The magnetic property characterization results of $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal are displayed in Figs S4 and S5. Fig. S4a, b show the temperature-dependent zero-field cooling and field-cooling magnetization (M - T) curves of the $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal with H_{\parallel} and H_{\perp} field of 0.05 T. The T_c defined by the minima of the dM/dT curves (insets in the figures) is $\sim 34 \text{ K}$. Due to the Cr-Te-Cr superexchange interaction at low temperatures [31], there is an apparent bifurcation in the M signal below T_c . Meanwhile, the large discrepancy on M value between H_{\parallel} and H_{\perp} directions signifies its large magnetic anisotropy and the easy c -axis [25,26,32]. In M - T curves of both H_{\parallel} and H_{\perp} field directions (see Fig. S4c, d), M increases with H , and is nearly isotropic at high fields while significantly anisotropic at low fields, which corresponds to the saturation behavior of magnetic moment at higher fields. In the FM state, the maximum value of magnetic moment reaches 2.925 and 2.997 μ_B per Cr atom at 4 K (see Fig. S5a, b), which is in excellent agreement with the 3.0 μ_B value expected for Cr^{3+} cations. The saturation field $H_s \sim 0.20 \text{ T}$ with H_{\perp} field is much smaller than the $H_s \sim 1.25 \text{ T}$ with H_{\parallel} field (see Fig. S5c, d). These results confirm the easy c -axis and large magnetic anisotropy in vdW FM $\text{Cr}_2\text{Si}_2\text{Te}_6$.

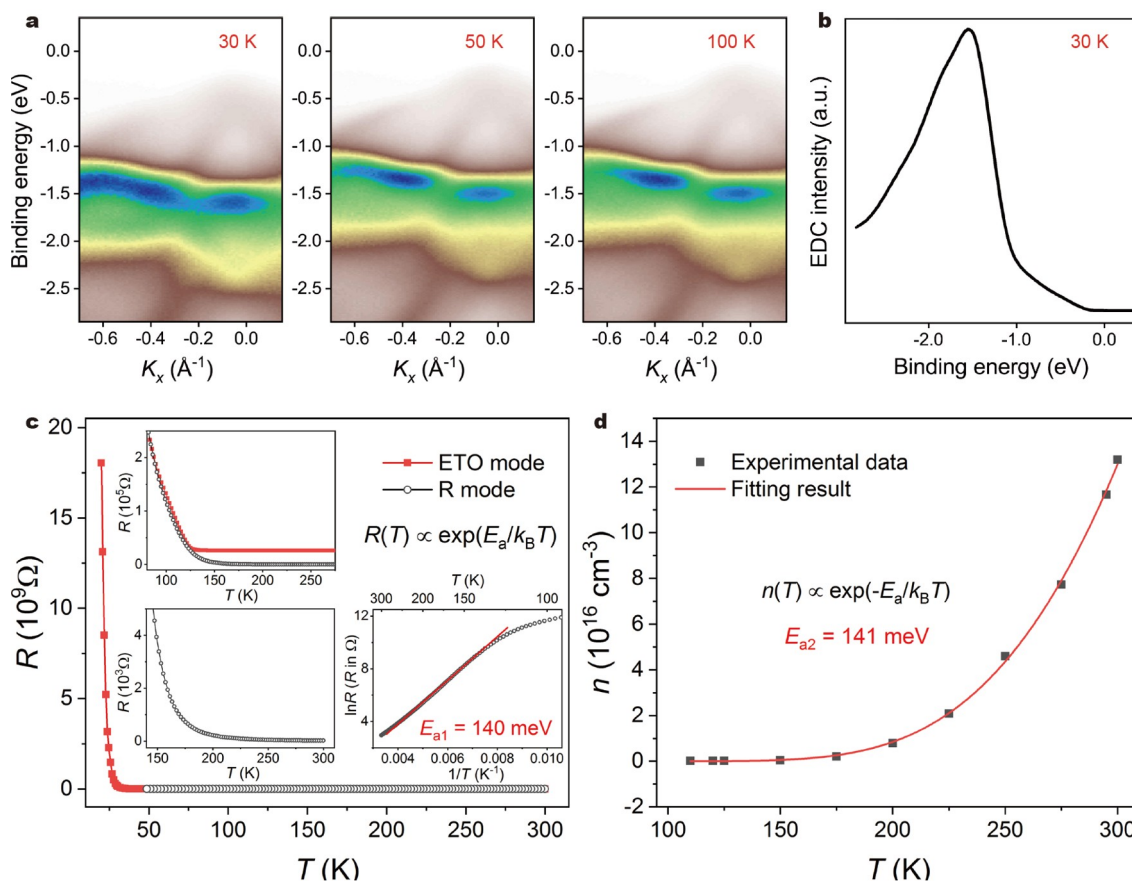


Figure 2 Electronic band structure and transport property characterizations. (a) Temperature-dependent band structures measured by ARPES along the Γ - K direction. (b) EDC intensity derived from the 30-K ARPES data. (c) Temperature-dependent resistance of $\text{Cr}_2\text{Si}_2\text{Te}_6$ measured by both two- and four-probe (ETO and DC resistivity) modes. Insets: enlarged displaying of the data and Arrhenius fitting of the data and Arrhenius fitting of the DC resistivity-mode in the higher-temperature range. (d) The temperature-dependent n with fitting using Arrhenius's law.

Dualistic magnetotransport behavior and coexistence of large pMR and nMR

Due to the magnetic anisotropy, we explored the anisotropic magnetotransport behavior of $\text{Cr}_2\text{Si}_2\text{Te}_6$ under different field configurations. The $H\parallel ab$ direction field-dependent MR data are shown in Fig. 3a–c and Fig. S6. Under $H\parallel ab$ field configuration, the field-dependent MR shows a roughly “W” shape and experiences a change of sign from negative below T_c to positive above T_c at high fields while always keeping a notable nMR at low fields. Below T_c , a very large nMR dominates in a wide field range, and shows extremum (the lowest point of “W” line) at 0.75–1.50 T (see Fig. S6a, b). Typically, at 22.5 K, a large nMR of about -60% is obtained at ~ 1.0 T (Fig. 3b, extracted from the orange-red line in Fig. 3a), which becomes smaller and upwarps in the high-field range when the temperature rises to T_c , but still persists very small negative values in the low-field range (Fig. S6c). The H values corresponding to the extremum remind us the saturation field H_s with $H\parallel ab$ field (Figs S4c and S5c), which is exactly in the same field range, indicating that this low-field nMR is related to FM interaction. As for the high-field range below T_c , the magnetotransport is dominated by ordinary orbit scattering which shows a pMR and causes the upwarp of the curve. Above T_c , with the increase of temperature, the absolute values of both pMR and nMR become smaller (Fig. 3c and Fig. S6d). Meanwhile, the nMR-to-pMR transition points extends to higher fields, which results from the contribution of

electronic correlation rather than FM interaction because the system goes into the PM state and only the electron interactions matter [28]. Finally, the signal disappears (covered by fluctuations) above 100 K. These results demonstrate that the FM interaction, orbital scattering and electronic correlation coexist in $\text{Cr}_2\text{Si}_2\text{Te}_6$, and contribute nMR, pMR, and nMR, respectively, in dominated temperature and magnetic field ranges. Among them, the more prominent FM interaction dominates below T_c in the low-field range and leads to the acquisition of larger nMR under $H\parallel ab$ field configuration.

When the magnetic field changes from $H\parallel ab$ to $H\perp ab$ configuration, field dependences of MR are shown in Fig. 3d–f and Fig. S7. At temperatures below 50 K, a pMR is demonstrated in the whole test temperature and field ranges. Its value rises first and then falls with increasing temperature, and reaches a maximum around T_c (see Fig. 3d and Fig. S7c, d). Typically, at 32.5 K, a large pMR higher than 1000% is obtained at 14 T and shows no sign of saturation (Fig. 3e, extracted from the tan line in Fig. 3d). Compared with that of $H\parallel ab$ direction, the MR behavior in the $H\perp ab$ direction seems to be quite different. However, when we look at the variation trend rather than absolute MR value, the MR mechanism behaves actually in the same way as in the $H\parallel ab$ direction. Specifically, below T_c , as shown in Fig. S7a, b, a “zero” MR phenomenon shows below the field of 0.2 T, which is exactly consistent with the saturation field H_s in the $H\perp ab$ magnetization direction (see Figs S4d and S5d).

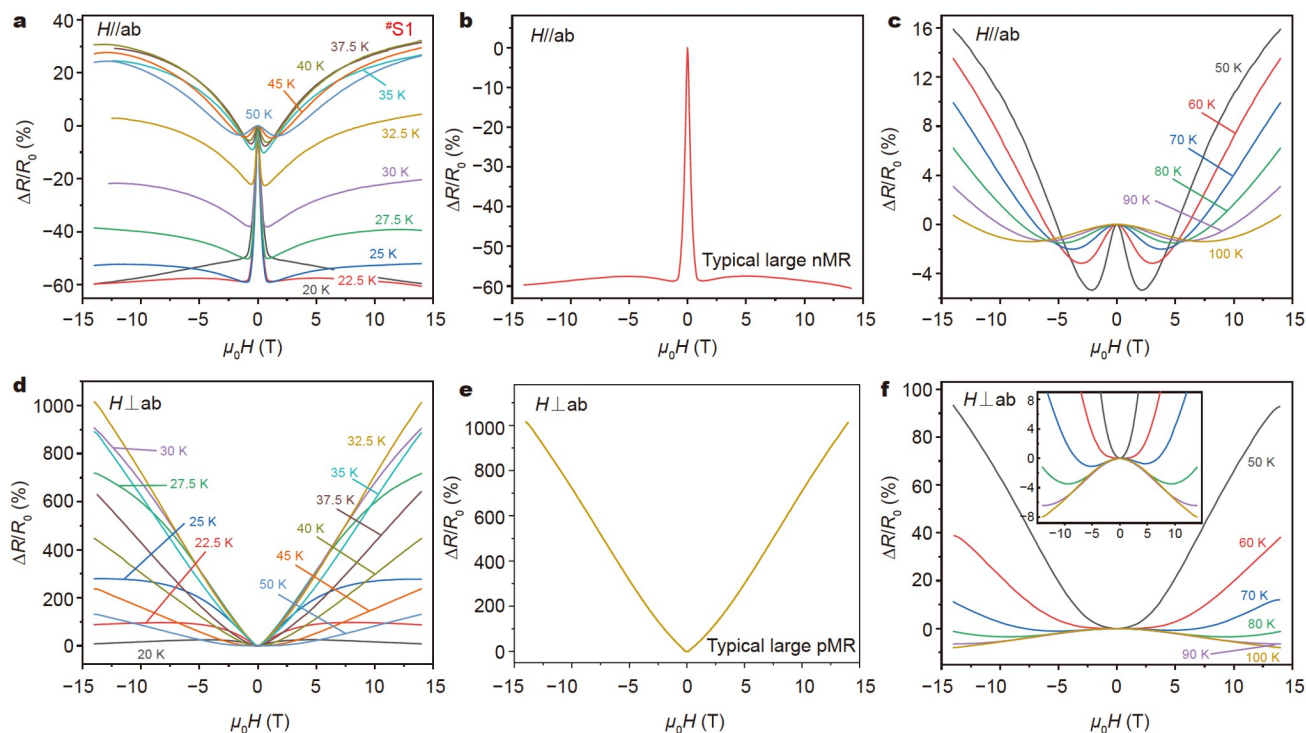


Figure 3 Magnetotransport behavior of $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal. (a) The $H\parallel ab$ direction MR measured below 50 K using ETO. (b) Typical large nMR obtained at 22.5 K. (c) The $H\parallel ab$ direction MR measured above 50 K using DC resistivity option. (d) The $H\perp ab$ direction MR measured below 50 K. (e) Typical large pMR obtained at 32.5 K. (f) The $H\perp ab$ direction MR measured above 50 K. Inset: enlarged displaying to highlight the nMR at higher temperatures.

This indicates that the FM-contributed nMR also exists at low fields (like that in the $H\parallel ab$ direction); however, it is almost completely covered by the pMR of orbit scattering [33]. The anisotropic orbit scattering is the dominant contribution of the large pMR which may come from the electron scattering by the orbit of the first valence band (as shown in Fig. 4e), and causes much larger magnitude of pMR in the $H\perp ab$ field than in the $H\parallel ab$ field. When the temperature rises above 50 K, the pMR continues to decrease and changes to nMR at 100 K (see Fig. 3f and inset). Besides, as the temperature increases, the nMR initially at relatively low magnetic field quickly extends to higher field and leads to a “W” shape MR, demonstrating that the weak localization caused by electronic correlation takes over, which is the same case in $H\parallel ab$ configuration. Finally, the MR signal also disappears above 100 K. These results suggest that MR mechanism in the $H\perp ab$ direction is the same as that of $H\parallel ab$ direction; however, the orbit scattering causes much larger magnitude of pMR than that of $H\parallel ab$ field, which covers the FM-contributed nMR and finally leads to the acquisition of large pMR under $H\perp ab$ field configuration.

Multiple competitive and cooperative MR mechanism

H - T color contour plots for the MR of $\text{Cr}_2\text{Si}_2\text{Te}_6$ are demonstrated in Fig. 4a, b, with data of the 70–100 K interval magnified in Fig. 4c, d. The similar color-scale maps between $H\parallel ab$ and $H\perp ab$ configurations confirm that they possess the same MR mechanisms but with different weighted contribution of positive and negative sources, as discussed above. As experimentally demonstrated above, $\text{Cr}_2\text{Si}_2\text{Te}_6$ is a vdW FMS. In addition, previous work further confirms the cooperation and interplay of electronic correlation and ferromagnetism in $\text{Cr}_2\text{Si}_2\text{Te}_6$ and its

decisive role in the formation of anisotropic MR behaviors in $\text{Cr}_2\text{Ge}_2\text{Te}_6$ homologue [28,34]. Therefore, the FM interaction and electronic correlation MR mechanisms are believed to exist in $\text{Cr}_2\text{Si}_2\text{Te}_6$ and take important roles in the dualistic large MR phenomenon. Together with the Lorentz force-induced ordinary MR (i.e., orbital scattering mechanism), we therefore propose the multiple MR mechanism consisting of FM interaction, orbital scattering and electronic correlation, as schemed in Fig. 4e, which shows the relatively dominant effect in MR as a function of temperature. As demonstrated in Fig. 4e, a competition and cooperation mechanism between FM interaction, orbital scattering and electronic correlation is proposed where the FM interaction contributes nMR and dominates at temperatures below T_c , while the electronic correlation also contributes nMR but dominates in much higher temperature range and disappears above 100 K. In the intermediate temperature range, the orbit scattering-contributed pMR is more prominent, especially at high magnetic fields. The competition and cooperation between the above MR mechanisms finally give birth to the intriguing and dual-sign MR behavior observed in $\text{Cr}_2\text{Si}_2\text{Te}_6$.

The angle-dependent behaviors of MR are shown in Fig. S8. As depicted in Fig. S8a, b, with the applied field changing from 90° , 60° , 30° to 0° to the ab plane, the MR behaviors of $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal at 25 and 40 K show a clear transition from nMR to pMR, confirming that $\text{Cr}_2\text{Si}_2\text{Te}_6$ is bestowed with an angle anisotropy in MR. Fig. S8c, d show the angular dependence of MR measured at 10 T. As shown in Fig. S8c, the absolute MR values decrease monotonically from 0° to 90° magnetic field and reach the minimum under 90° ($H\parallel ab$) field below 50 K. However, when the temperature is above 50 K (as shown in Fig. S8d), along with the increase of temperature, the variation trend of

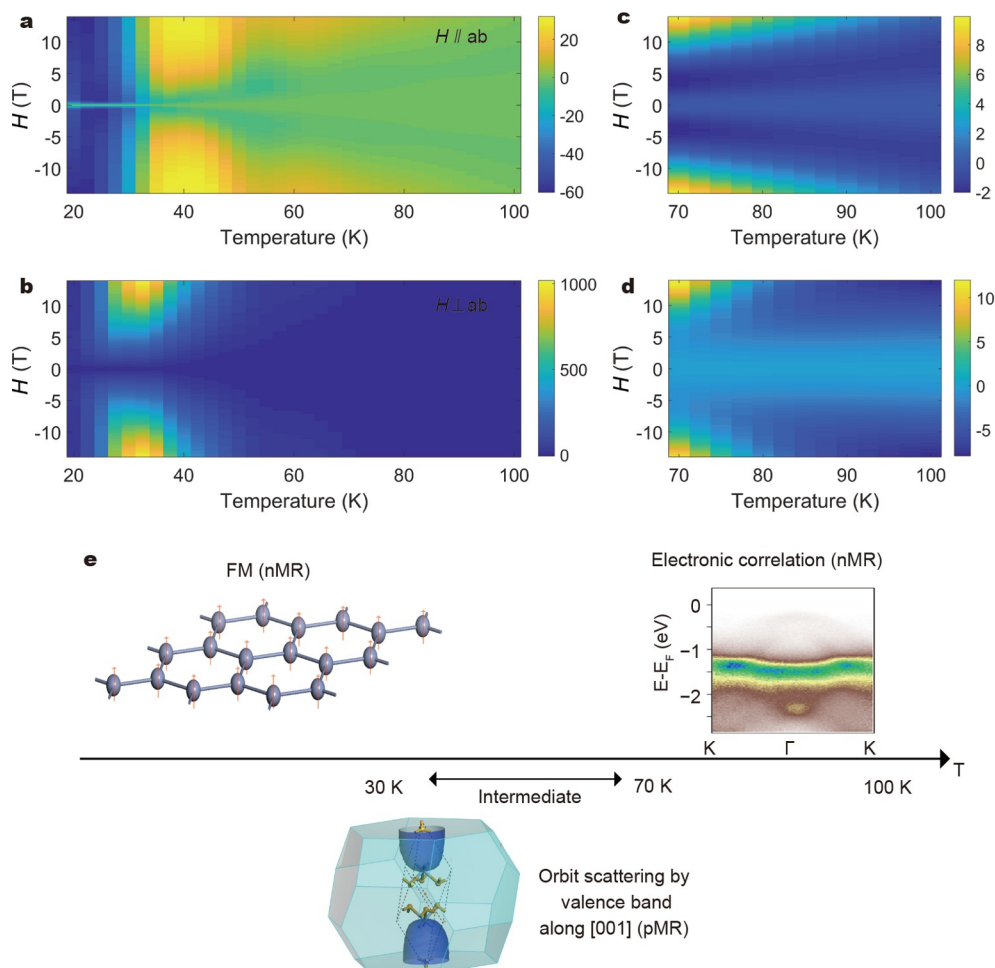


Figure 4 Mechanism for the MR effect in $\text{Cr}_2\text{Si}_2\text{Te}_6$. (a–d) H - T color contour plots for the MR of $\text{Cr}_2\text{Si}_2\text{Te}_6$ along $H \parallel ab$ (a, b) and $H \perp ab$ (c, d) directions. (e) Schematic diagram of the competitive effect between FM interaction, orbit scattering, and electronic correlation, which contribute nMR, pMR, and nMR to $\text{Cr}_2\text{Si}_2\text{Te}_6$, respectively.

MR value experiences a transition from monotonic decrease to monotonic increase with field changing from 0° to 90° , and finally reaches the minimum under $0^\circ/180^\circ$ ($H \perp ab$) field above 80 K (see in Fig. S8d). This can be understood by comparing the MR data in Fig. 3c, f, which should be attributed to the competition and cooperation between the multiple MR mechanisms discussed above. Worthy of note is that repeatability test of the magnetotransport behavior of $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal was made on separate samples #S2 (Fig. S9), which shows consistent results with sample #S1, proving the reliability of the data.

CONCLUSIONS

In summary, the disclosure of nMR of about -60% and pMR higher than 1000% in $\text{Cr}_2\text{Si}_2\text{Te}_6$ single crystal suggests that materials with vdW layered lattice, ferromagnetism and semiconductor natures are promising candidates for coexistence of large nMR and pMR. It is surprising that the acquisition of large and dual-sign MR is realized in this simple system. However, just like that there is a necessity in contingency, the diversity of material attributes (2D anisotropic lattice, ferromagnetism, and electronic correlation) coexisting in $\text{Cr}_2\text{Si}_2\text{Te}_6$ triggers multiple MR mechanisms, and thus fertilizes the emergence of simultaneous large pMR and nMR. Specifically, in $\text{Cr}_2\text{Si}_2\text{Te}_6$, the FM interaction, orbital scattering and electronic correlation con-

tribute nMR, pMR, and nMR in dominated temperature and magnetic field ranges, which comprehensively leads to the acquisition of both large pMR and nMR under different field configurations. This elucidated multiple MR mechanism and coexistence of large pMR and nMR in this vdW FMS can shed light on the pursuit of coexistence of large pMR and nMR for field-sensitive device applications.

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Supplementary information Supporting data are available in the online version of the paper.



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Cr₂Si₂Te₆铁磁半导体中大的正、负磁阻共存现象

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摘要 磁阻(MR)作为一种电子输运与磁场耦合的宏观现象,一直是凝聚态物理和材料科学的研究前沿.其中,大的磁阻现象及材料对磁传感器和信息存储应用尤为重要,而这种材料的稀缺也引起了人们的广泛关注.此外,由于物理起源不同,在一种化合物中同时存在大的正、负磁阻少见报道.本工作中,我们在Cr₂Si₂Te₆铁磁半导体单晶中实现了不同磁场方向下大的正磁阻(pMR)和负磁阻(nMR)共存.具体而言,在面内场方向获得了约-60%的大的nMR,在面外场方向获得了高于1000%的大的pMR.我们将这种磁场方向依赖的磁阻行为归因于Cr₂Si₂Te₆中的铁磁相互作用、轨道散射和电子关联作用的竞争和合作效应,它们在相应温度和磁场范围内分别使Cr₂Si₂Te₆表现出nMR, pMR和nMR.阐明这种铁磁半导体中的磁阻机制将为寻求大的pMR和nMR共存的磁场敏感器件提供指导.