



# Breaking the periodic arrangement of atoms for the enhanced electrochemical reduction of nitrogen and water oxidation

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**ABSTRACT** The development of cost-effective and high-performance electrocatalysts has been increasingly studied to mitigate upcoming energy and environmental challenges. Amorphization and heterointerface engineering have played significant roles in the rational design of electrocatalysts and modulation of their electrocatalytic activities. However, the synergistic effect between amorphization and heterointerfaces has been scarcely reported. As a proof-of-concept attempt, we develop amorphous FeMo (a-FeMo) electrocatalysts with an abundance of heterointerfaces that are composed of amorphous components and evaluate their electrocatalytic performances toward the nitrogen reduction reaction and oxygen evolution reaction (OER). Benefitting from the synergistic effect between the amorphous nature of the a-FeMo electrocatalysts, which offer a high density of active sites, and significant electron redistribution at the heterointerfaces, the electrocatalysts exhibit a high Faradaic efficiency of 29.15%, an elevated yield rate of  $71.78 \mu\text{g}_{\text{NH}_3} \text{mg}_{\text{cat.}}^{-1} \text{h}^{-1}$  with long-term stability at a potential of  $-0.1 \text{ V}$  vs. reversible hydrogen electrode and excellent electrocatalytic activity toward the OER. This study provides a promising and effective method for the rational design of low-cost heterogeneous catalysts with desirable efficiency, selectivity, and stability.

**Keywords:** amorphism and heterointerface, nitrogen reduction reaction, OER, electrocatalyst design, electrocatalytic conversion

## INTRODUCTION

The world currently relies heavily on fossil fuels for energy production and chemical industries, such as the ammonia industry, which consumes approximately 1%–2% of anthropogenic energy, while also producing ~400 Mt CO<sub>2</sub> and causing serious environmental impacts [1,2]. To mitigate the energy crisis and potential ecological catastrophe, the search for alternative energy resources and economic storage and conversion systems has grown to be one of the focuses of scientific society [3–5]. As an essential part of sustainable energy systems, electrochemical energy conversion devices enable the synthesis of value-added chemicals, including but not limited to hydrogen,

hydrocarbons, oxygenates, and ammonia, from the abundant feedstock of H<sub>2</sub>O, CO<sub>2</sub>, and N<sub>2</sub>, thereby offering a promising strategy for the sustainable development of the world [6–10]. The principal component of electrochemical devices is electrocatalyst that enable energy conversion from one to another, and whose performance is significantly influenced by its chemical composition, size, and morphology along with the type of electrolyte [11,12].

Aiming for the preparation of electrocatalysts with high Faradaic efficiency, long-term stability, and sufficient yield rate, various approaches that aim to maximize the density of active sites, enhance the conductivity and improve the intrinsic activity of the active sites have been applied in the preparation of electrocatalysts [13–17]. Among them, defect and heterointerface engineering have been increasingly studied due to their feasible production, apparent influence on the electrocatalytic performances of electrocatalysts, and low cost; additionally, electrocatalyst engineered in this way has demonstrated advantages in the hydrogen evolution reaction (HER) [18–20], oxygen evolution reaction (OER) [21–26], CO<sub>2</sub> reduction reaction (CRR) [27–29], oxygen reduction reaction (ORR) [30–32], and nitrogen reduction reaction (NRR) [33–35]. Benefitting from the higher density of active sites due to the short-range ordering and relatively crumbly structure of these electrocatalysts, amorphization has arisen as an efficient way to modulate them [36–40]. On the other hand, interfaces including crystal boundaries that can stabilize active surface catalytic sites and enable their synergistic effects have also been proven important in tailoring the electrocatalytic properties of electrocatalysts [25,27,29]. Despite the intensive investigation of amorphization and interfaces, their simultaneous combination has rarely been reported, which may trigger interesting electrocatalyst properties.

Herein, as a proof-of-concept attempt, we report the synthesis of amorphous FeMo (a-FeMo) electrocatalysts with an abundance of interfaces and an optimized Fe/Mo ratio. We evaluate their electrocatalytic performances with regard to the NRR and OER. Owing to the synergistic effects of the amorphous nature of the a-FeMo electrocatalysts with a high density of active sites and significant electron redistribution at the heterointerfaces, the electrocatalysts exhibit a high Faradaic efficiency of 29.15%,

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an elevated yield rate of  $71.78 \mu\text{g}_{\text{NH}_3} \text{mg}_{\text{cat.}}^{-1} \text{h}^{-1}$  with long-term stability at a potential of  $-0.1 \text{ V}$  vs. reversible hydrogen electrode (RHE) and excellent electrocatalytic performance toward the OER. On the basis of the experimental results, we demonstrate the important role that the synergy between the atomic disorder and heterointerfaces plays in activating and enhancing the catalytic performance of electrocatalysts.

## EXPERIMENTAL SECTION

### Synthesis of Fe-Mo electrocatalysts with different Fe/Mo ratios

The Fe-Mo electrocatalysts were prepared *via* a simple hydrothermal method. In a typical process, 10 mL of ultrapure water containing  $\text{NaMoO}_4 \cdot 2\text{H}_2\text{O}$  (1.96 mmol, 164 mg),  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  (0.28 mmol, 110 mg), urea (0.25 mmol, 150 mg), and  $\text{NH}_4\text{F}$  (1 mmol, 37.5 mg) in a 30.0-mL vial was treated at  $50^\circ\text{C}$  for 30 min in an oil bath with continuous stirring until the reactants completely dissolved. Then, the temperature was quickly increased to  $120^\circ\text{C}$  and maintained for 12 h. The obtained products were washed several times with ethanol and water after being cooled to room temperature. Finally, they were dried at  $60^\circ\text{C}$  for 12 h for further characterization. Fe-Mo electrocatalysts with different Fe/Mo ratios were synthesized in a similar method by changing the amounts of  $\text{NaMoO}_4 \cdot 2\text{H}_2\text{O}$  and  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ . For simplicity, the Fe-Mo electrocatalysts with different Fe/Mo ratios are denoted as a-FeMo- $x$ , where  $x$  is the Fe/Mo ratio in moles.

### Synthesis of the crystalline FeMo-1/7 electrocatalyst

The crystalline FeMo-1/7 (c-FeMo-1/7) electrocatalyst was prepared *via* a simple annealing process. Typically, 50 mg of the a-FeMo-1/7 electrocatalyst was placed into a tube furnace and heated at  $285^\circ\text{C}$  for 4 h in nitrogen with a ramp speed of  $5^\circ\text{C min}^{-1}$ . After being naturally cooled to room temperature, c-FeMo-1/7 was obtained and ready for further characterization.

### Synthesis of the $\text{Fe}_3\text{O}_4$ and $\text{MoO}_3$ electrocatalysts

For the synthesis of  $\text{Fe}_3\text{O}_4$ , 20 mL of ultrapure water containing  $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$  (130.5 mg),  $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$  (269 mg), and NaOH (8 mg) was first stirred for 30 min and then transferred to a 40-mL Teflon tube and sealed into an autoclave, which was heated to  $160^\circ\text{C}$  and maintained for 12 h. The obtained products were washed several times with ethanol and water after being cooled to room temperature. Finally, they were dried at  $60^\circ\text{C}$  for 12 h before further characterization.

For the synthesis of  $\text{MoO}_3$ , 2 mL of water containing 486 mg of  $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24} \cdot 4\text{H}_2\text{O}$  was stirred until the solute was fully dissolved, after which concentrated  $\text{HNO}_3$  (0.6 mL) was dropwise added and further stirred for another 30 min. Then, the solution was transferred to a 20-mL Teflon tube and sealed into the autoclave, which was heated to  $90^\circ\text{C}$  and maintained for 3 h. The obtained products were washed several times with ethanol and water after being cooled to room temperature. Finally, they were dried at  $60^\circ\text{C}$  for 12 h for further characterization.

### Characterization methods

High-resolution high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) and tomography images were obtained on a Cs-corrected FEI Titan 80/300 kV TEM/STEM microscope. Powder X-ray diffraction (XRD) patterns were recorded using an X-ray diffractometer (SmartLab

(3), Rigaku) operated at 3 kW. X-ray photoelectron spectroscopy (XPS) analyses were performed using a Kratos Ultra DLD spectrometer equipped with a monochromatic Al K radiation source. The binding-energy scales were calibrated using the C 1s peak at 284.6 eV, which was from C contamination. The morphologies of the samples were examined with TEMc (JEOL H-7000).

### Electrochemical NRR performance test

The electrochemical NRR performance tests were carried out in a standard three-electrode system with an electrochemical workstation (CHI660E). Typically, 3.6 mg of the as-obtained electrocatalysts and 1.0 mg carbon black were dispersed in 0.4 mL ethanol-isopropyl alcohol solution with a volume ratio of 1:3. Then, 24  $\mu\text{L}$  of Nafion (5 wt%) was added, and the solution was further sonicated for 60 min. Next, 10  $\mu\text{L}$  of the homogeneous dispersion was drop-cast onto a glassy carbon electrode with a diameter of 5.0 mm. The prepared electrodes were dried at room temperature for further tests. The mass loading of the electrocatalysts was determined to be  $\sim 0.43 \text{ mg cm}^{-2}$ .

The NRR was performed using a two-compartment H-type cell that was separated by a pretreated Nafion 117 membrane. Pt wire and Ag/AgCl electrodes were used as the counter electrode and the reference electrode, respectively. All potentials were calibrated to the RHE reference scale by  $E$  (vs. RHE) =  $E$  (vs. Ag/AgCl) + 0.21 +  $0.0593 \times \text{pH}$ . The NRR tests were conducted in an  $\text{N}_2$ -saturated  $0.1 \text{ mol L}^{-1}$  KOH solution at room temperature under atmospheric pressure. Before the NRR test,  $\text{N}_2$  was purged into the electrolyte for at least 30 min to remove the residual air, and controlled potential electrocatalysis was performed at different potentials for 2 h.

### Determination of $\text{NH}_3$

The concentration of the produced  $\text{NH}_3$  was determined by the indophenol blue method. First, 5 mL of the solution was removed from the reaction vessel. Then, 2 mL of NaOH ( $1 \text{ mol L}^{-1}$  with 5 wt% salicylic acid and 5 wt% sodium citrate), 1 mL of NaClO ( $0.05 \text{ mol L}^{-1}$ ), and 0.2 mL of an aqueous solution of 1 wt% sodium nitroferrocyanide were added to the solution. After 2 h at room temperature, the absorption spectrum was measured using a UV-vis spectrophotometer at a wavelength of 655 nm.

### Determination of $\text{N}_2\text{H}_4$

The  $\text{N}_2\text{H}_4$  produced during the electrocatalysis process was estimated by the Watt and Chrisp method. A mixture of para-(dimethylamino) benzaldehyde (5.99 g), HCl (concentrated, 30 mL), and ethanol (300 mL) was used as the color reagent. Briefly, 2 mL of the electrolyte after electrolysis was mixed with 2 mL of the color reactor. The absorbance of the resulting solution was measured at 455 nm. The absolute calibration of this method was achieved using hydrazine monohydrate solutions of known concentrations as standards.

### Calculation of the yield rate and Faradaic efficiency

The Faradaic efficiency (FE) of the products was calculated using the following equation:

$$\text{FE} = c_{\text{NH}_3} \times V \times N \times F / Q,$$

where  $c_{\text{NH}_3}$  is the measured  $\text{NH}_3$  concentration,  $V$  is the volume of the electrolyte,  $N$  is the number of electrons transferred for

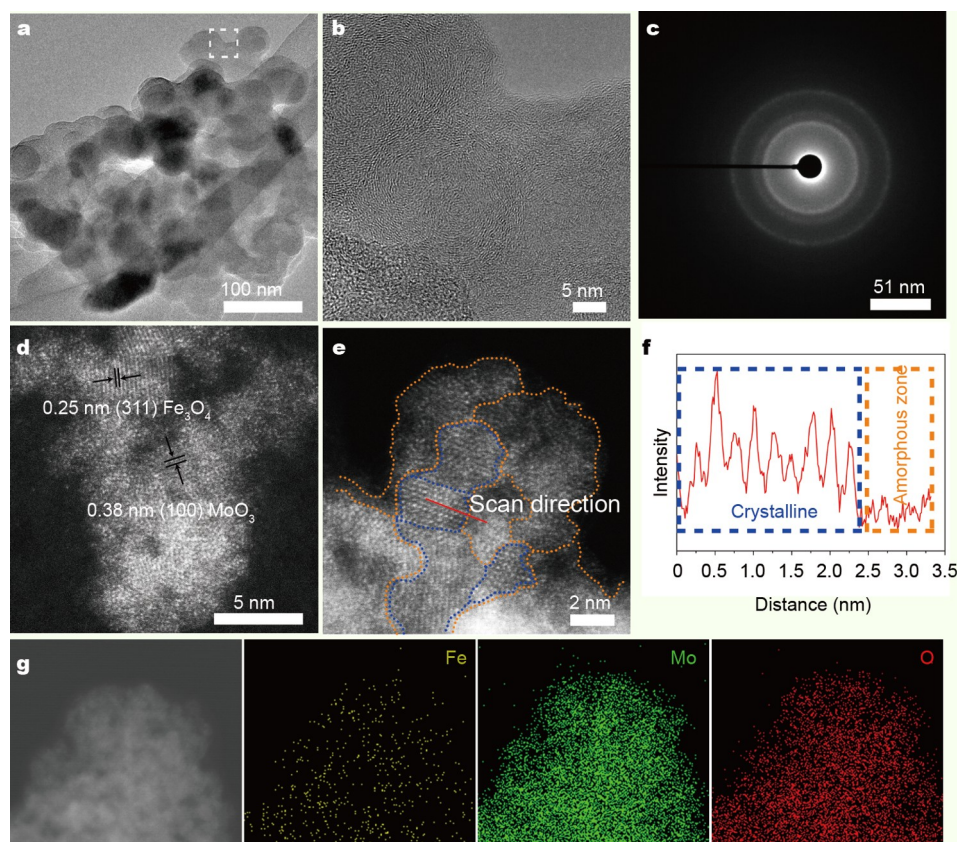
product formation,  $F$  is the Faraday constant ( $96,485 \text{ C mol}^{-1}$ ), and  $Q$  is the quantity of electric charge integrated by the current-time ( $i-t$ ) curve.

## RESULTS AND DISCUSSION

### Synthesis and structural characterization

For the synthesis of a-FeMo catalysts with different Fe/Mo ratios,  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  and  $\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$  with desired ratios were first dissolved in water, and then urea and ammonium fluoride were subsequently added (Fig. S1). After being heated in an oil bath for a certain time, the light orange precipitates were collected and washed with ultrapure water and ethanol several times and dried under vacuum for further characterization (details in the Supplementary information). The microstructure of the product obtained was analyzed with XRD, TEM, and high-resolution TEM (HRTEM). As indicated by the low-magnification TEM images, the a-FeMo electrocatalyst (a-FeMo-1/7) with a Fe/Mo ratio of 1/7 consists of uniformly distributed nanoparticles with a radius of  $\sim 50 \text{ nm}$  (Fig. 1a). The other a-FeMo electrocatalysts show similar particle sizes even though they are synthesized with different Fe/Mo ratios (Fig. S2). The fringes of the nanoparticles with HRTEM shows the disordered arrangement of the atoms, indicating the amorphous nature of the a-FeMo-1/7 electrocatalyst (Fig. 1b). The corresponding selected area electron diffraction (SAED) pattern shows no

obvious lattice arrangement, again confirming the amorphous nature of the a-FeMo-1/7 electrocatalyst (Fig. 1c). The XRD patterns of the a-FeMo electrocatalysts show no diffraction peaks, as indicated in Fig. S3, implying there is an atomic disorder in the a-FeMo electrocatalysts, regardless of the Fe/Mo ratio. The detailed structural information of a-FeMo-1/7 was further revealed by HAADF-STEM. As shown in Fig. 1d, lattice fringes with distinct  $d$ -spacing values can be readily observed, corresponding to the interplanar distance of (311) in  $\text{Fe}_3\text{O}_4$  and (100) in  $\text{MoO}_3$  [41,42], thus indicating the composition of the a-FeMo-1/7 electrocatalyst. More detailed observations with HAADF-STEM indicate the apparent domains of the crystalline zone and the amorphous zone of the a-FeMo-1/7 electrocatalyst, showing a well-ordered local structure for the crystalline zone and a disordered lattice structure with a random atom arrangement in the amorphous zone (Fig. 1e). The interface between the crystalline-amorphous phase boundaries has been proven to be beneficial for electrocatalytic processes by offering numerous adsorption sites for intermediates during electrocatalytic reactions [23,43]. The corresponding intensity profile recorded in Fig. 1e also demonstrates the existence of the crystalline-amorphous phase boundary and the irregular arrangement of atoms in the amorphous zone (Fig. 1f). The corresponding elemental mapping images of a-FeMo-1/7 reveal that Fe, Mo, and O are uniformly distributed (Fig. 1g). The atomic ratio of Fe/Mo is estimated to be 1/6.79 based on energy



**Figure 1** (a) Low-magnification TEM image of the a-FeMo-1/7 electrocatalyst; (b) enlarged HRTEM image of the a-FeMo-1/7 electrocatalyst marked in (a), indicating the amorphous nature of the electrocatalyst; (c) SAED pattern of the a-FeMo-1/7 electrocatalyst; (d) atomic resolution HAADF-STEM image of the a-FeMo-1/7 electrocatalyst with clear lattice fringes; (e) atomic resolution HAADF-STEM image of the a-FeMo-1/7 electrocatalyst, indicating the boundary between the amorphous region and the crystalline region; (f) intensity profile recorded from the line in (e); (g) EDS mapping of the a-FeMo-1/7 electrocatalyst.

dispersive spectroscopy (EDS), implying the successful preparation of the a-FeMo electrocatalyst with a Fe/Mo ratio of 1/7.

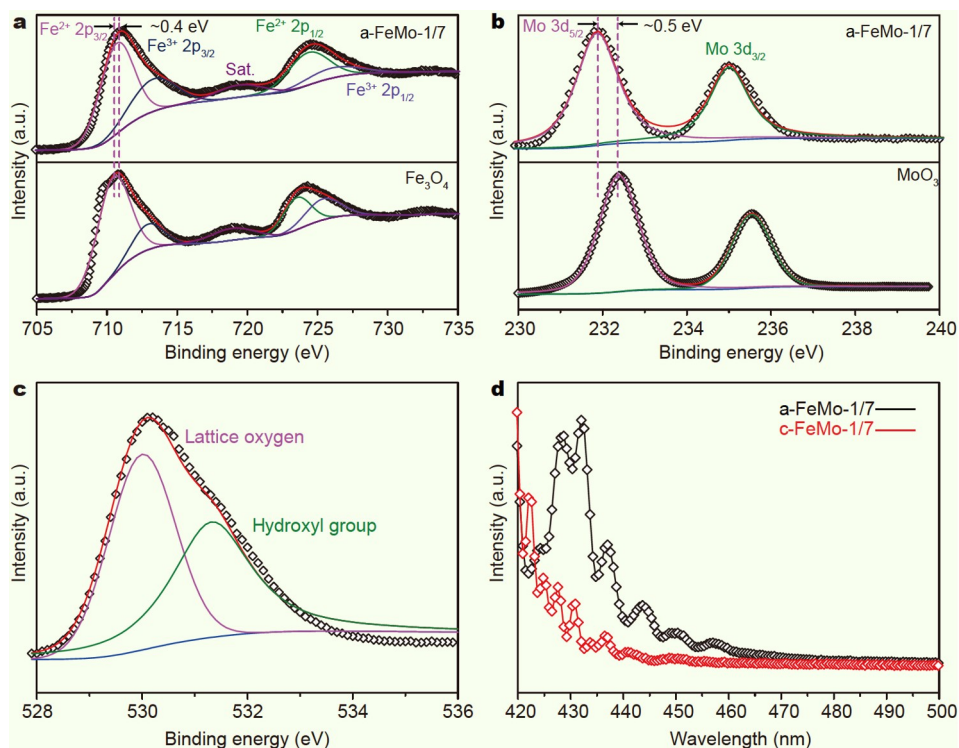
### Chemical and electronic configuration

To probe the amorphism-induced change in the electronic structure of the a-FeMo-1/7 electrocatalyst, XPS and photoluminescence (PL) spectra were collected. First, crystalline  $\text{Fe}_3\text{O}_4$  and  $\text{MoO}_3$  were prepared *via* a traditional solvothermal method and used as references. By annealing a-FeMo-1/7 at a high temperature for 4 h, the c-FeMo-1/7 electrocatalyst can be obtained (Fig. S4, with details in the Supplementary information). The XRD patterns of the as-obtained  $\text{Fe}_3\text{O}_4$  and  $\text{MoO}_3$  indicate the pure phase of these nanomaterials with high crystallinity and no obvious impurities (Fig. S5). As shown in Fig. 2a, the binding energies at 710.4, 723.5, 712.9, 725.4, and 718.9 eV can be assigned to  $\text{Fe}^{2+} 2p_{3/2}$ ,  $\text{Fe}^{2+} 2p_{1/2}$ ,  $\text{Fe}^{3+} 2p_{3/2}$ ,  $\text{Fe}^{3+} 2p_{1/2}$  and the satellite peak of Fe 2p in pure  $\text{Fe}_3\text{O}_4$ , respectively [44–46]. The Fe 2p peaks in the a-FeMo-1/7 shift positively to higher energies by approximately 0.4 eV, indicating the significant alteration of the electron distributions. The a-FeMo electrocatalysts with different Fe/Mo ratios demonstrate similar energy shifts, as shown in Fig. S6, validating the fact that the amorphous nature of the electrocatalysts shows a similar influence on their electronic structure. However, the crystalline FeMo electrocatalyst with a Fe/Mo ratio of 1/7 shows a negligible energy shift compared with pure  $\text{Fe}_3\text{O}_4$  (Fig. S7), indicating the slight effect of the crystalline interface on the electron distributions. The binding energies at 232.2 and 235.6 eV from the Mo 3d spectrum can be assigned to  $\text{Mo}^{6+} 3d_{5/2}$  and  $\text{Mo}^{6+} 3d_{3/2}$  in pure  $\text{MoO}_3$  [21,47] as shown in Fig. 2b. Unsurprisingly, the Mo 3d peak of a-FeMo-1/7 shows a significantly negative shift towards lower binding energies, indicating electron transfer from the Fe to Mo

atoms. The Mo 3d peaks of other FeMo electrocatalysts with different Fe/Mo ratios show the same trend, again validating the significance of amorphism on the electron distribution of the electrocatalysts (Fig. S6). The Mo 3d spectrum of c-FeMo-1/7 shows little change in the peak positions, implying the limited electronic interaction between crystalline  $\text{Fe}_3\text{O}_4$  and  $\text{MoO}_3$  (Fig. S8). Different from the Fe 2p and Mo 3d peaks, the O 1s peak of the FeMo electrocatalysts is located at almost the same position with binding energies of 530.1 and 531.6 eV, which can be ascribed to the lattice oxygen and hydroxyl groups (Fig. 2c and Figs S6 and S7) [21,45]. Fig. 2d shows the PL spectra of a-FeMo-1/7 and c-FeMo-1/7, which display a strong PL emission peak at  $\sim 435$  nm that corresponds to the recombination of holes in the disordered area of the electrocatalyst [46]. The PL spectra of the FeMo electrocatalysts with different Fe/Mo ratios also show obvious PL emission peaks, as indicated in Fig. S9. Combined with the XPS and PL spectra, the strong electronic interaction between  $\text{Fe}_3\text{O}_4$  and  $\text{MoO}_3$  and the electron accumulation on the Mo sites can be validated, which implies the potential for enhanced electrocatalytic performance.

### NRR electrochemical evaluation

To validate the assumption that the amorphism and interface can serve as an advantage in the electrocatalytic process, the electrochemical activity of a-FeMo-1/7 was evaluated with regard to the NRR and OER. The electrocatalytic activities of the samples were carried out in an  $\text{N}_2$ -saturated  $0.1 \text{ mol L}^{-1}$  KOH solution in a typical H-type cell at room temperature and ambient pressure. The produced  $\text{NH}_3$  was spectrophotometrically determined by the indophenol blue method, and the possible production of  $\text{N}_2\text{H}_4$  was detected by spectrophotometry with dimethylaminobenzaldehyde [27,38,48]. The



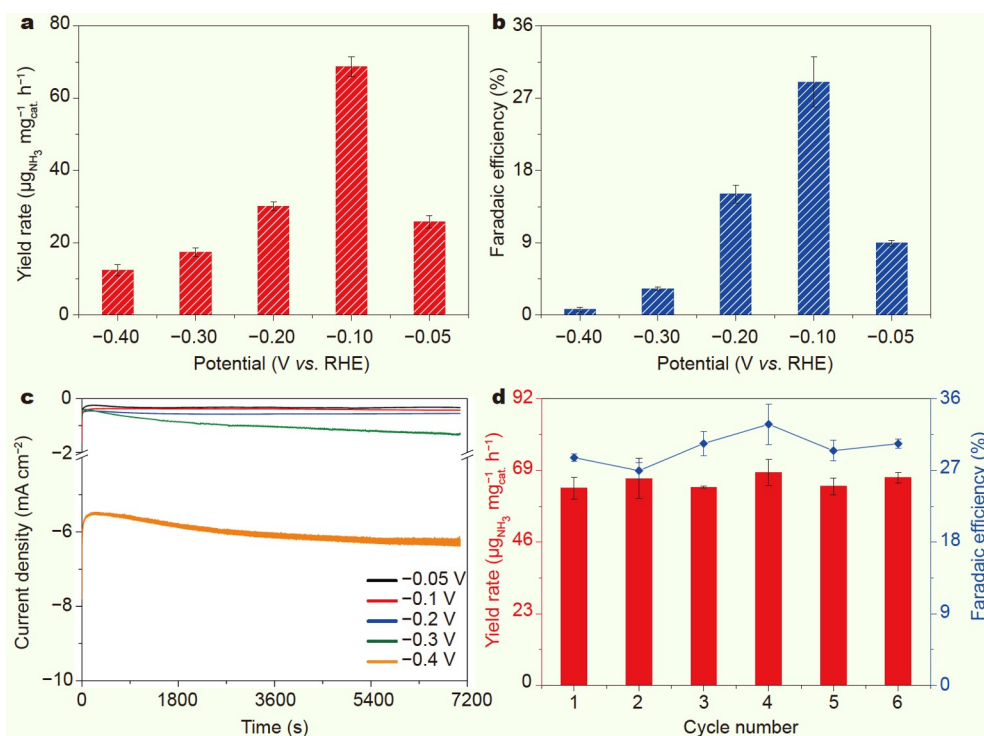
**Figure 2** (a) Fe 2p XPS spectra of the a-FeMo-1/7 electrocatalyst and  $\text{Fe}_3\text{O}_4$ ; (b) Mo 3d XPS spectra of the a-FeMo-1/7 electrocatalyst and  $\text{MoO}_3$ ; (c) O 1s XPS spectra of the a-FeMo-1/7 electrocatalyst; (d) PL spectra of the a-FeMo-1/7 and c-FeMo-1/7 electrocatalysts.

corresponding calibration curves are displayed in Figs S10 and S11. All potentials in this work were calibrated to the values *versus* RHE with no exceptions. The linear sweep voltammetry (LSV) curves of the FeMo electrocatalysts in Ar-saturated and N<sub>2</sub>-saturated electrolytes demonstrate different current densities, which indicates the obvious reduction of nitrogen (Fig. S12). The a-FeMo-1/7 shows a higher current density than the other a-FeMo electrocatalysts, implying its higher activity toward the NRR. Since the HER can be a major competing reaction with the NRR, a sequence of potentials was applied to achieve the optimum conditions for NH<sub>3</sub> production.

As shown in Fig. 3a, the highest average yield rate of NH<sub>3</sub> for a-FeMo-1/7 is 71.78 μg<sub>NH<sub>3</sub></sub> mg<sub>cat.</sub><sup>-1</sup> h<sup>-1</sup>, with a corresponding Faradaic efficiency of 29.15% (Fig. 3b) at -0.1 V *vs.* RHE; this electrocatalyst outperforms the a-FeMo electrocatalysts with different Fe/Mo ratios (Fig. S13). Thus, a-FeMo-1/7 is one of the best NRR electrocatalysts compared with other reported electrocatalysts (Table S1). To exclude artifacts, such as indoor ambient ammonia contamination and other sources of nitrogen, control experiments were carried out. The UV-vis spectra of the electrolyte after the NRR test in Ar and N<sub>2</sub> are shown in Fig. S14. The results indicate that a negligible amount of NH<sub>3</sub> is detected, thereby demonstrating that NH<sub>3</sub> mainly comes from the electrochemical reduction of N<sub>2</sub>. As indicated in Fig. 3c, the a-FeMo-1/7 further shows excellent durability during long-term electrolysis at different potentials, which is better than the a-FeMo electrocatalysts with other Fe/Mo ratios (Fig. S13). Further investigation of the stability is displayed in Fig. S15, with 36 h of continuous electrolysis at -0.1 V *vs.* RHE. The chronoamperometry curve shows no obvious drop in current density, which confirms its strong stability. Further characterizations reveal a negligible change in the morphology and XPS peaks

after the NRR durability test, illustrating its excellent stability (Fig. S16). Both the NH<sub>3</sub> yield rate and the FE values are also well maintained for 12 h (6 cycles) of the continuous electrochemical NRR test (Fig. 3d). To rationalize the excellent performance of the a-FeMo electrocatalyst during the NRR evaluation, the electrochemically active surface area (ECSA), which reveals the presence of active sites in electrocatalysts, was calculated by measuring the double-layer capacitance (*C<sub>dl</sub>*). The value of *C<sub>dl</sub>* for a-FeMo-1/7 is 1.78 mF cm<sup>-2</sup>, which is higher than that of the a-FeMo electrocatalysts with other Fe/Mo ratios (1.45 mF cm<sup>-2</sup> for a-FeMo-1/15, 1.69 mF cm<sup>-2</sup> for a-FeMo-1/3, and 1.51 mF cm<sup>-2</sup> for a-FeMo-1/1), as shown in Figs S17 and S18. This result reveals the higher density of active sites for a-FeMo-1/7. The largest ECSA value of the a-FeMo-1/7 electrocatalyst indicates the maximum number of active sites, which may originate from the disordered arrangement of atoms [46]. Despite the high density of active sites, the electrical transport properties revealed by electrochemical impedance spectroscopy (EIS) verify that the a-FeMo-1/7 electrocatalyst exhibits lower charge-transfer resistance and higher charge transport properties at different overpotentials (Fig. S19). Taken together, the a-FeMo-1/7 electrocatalyst demonstrates the most advanced NRR performance among various compositions, which results from its high density of active sites, better electronic conductivity, and smaller electron transport resistance.

To further address the importance of the synergy between the atomic disorder and heterointerfaces, the NRR activities of pure Fe<sub>3</sub>O<sub>4</sub>, MoO<sub>3</sub>, and c-FeMo-1/7 were also investigated and compared with the state-of-the-art a-FeMo-1/7 electrocatalyst. The LSV curves indicate a smaller difference in current density compared with a-FeMo-1/7, which implies lower NRR activity (Fig. S20). As shown in Fig. 4a and Fig. S21, pure Fe<sub>3</sub>O<sub>4</sub> and

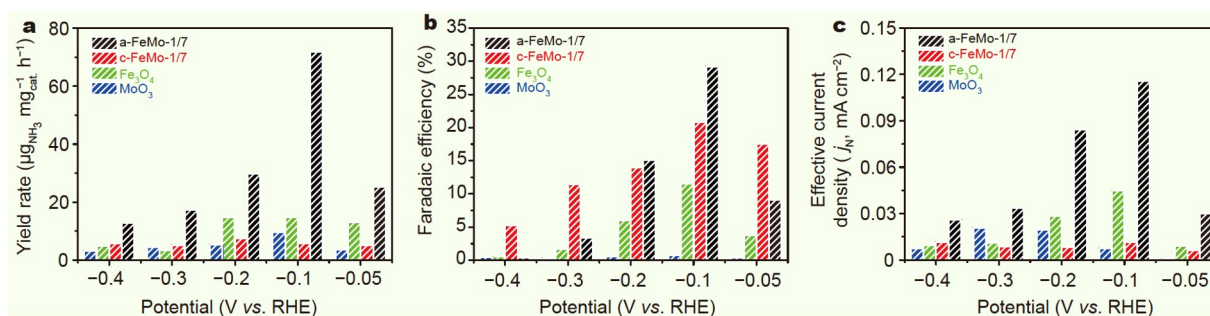


**Figure 3** (a) NH<sub>3</sub> yield rate of the a-FeMo-1/7 electrocatalyst at different potentials; (b) Faradaic efficiencies of the a-FeMo-1/7 electrocatalyst at different potentials; (c) chronoamperometric curves of the a-FeMo-1/7 electrocatalyst at different applied potentials; (d) recycling test of the a-FeMo-1/7 electrocatalyst at a potential of -0.1 V *vs.* RHE.

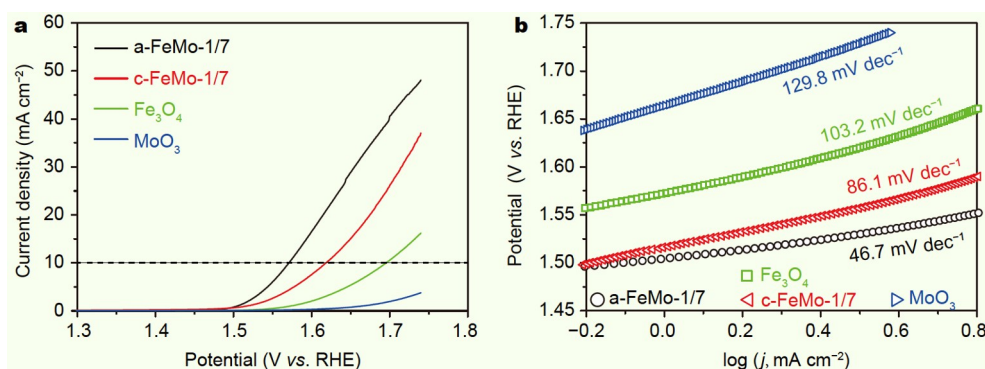
MoO<sub>3</sub> exhibit inferior NRR activity with yield rates of 14.72 and 9.72  $\mu\text{g}_{\text{NH}_3} \text{mg}_{\text{cat.}}^{-1} \text{h}^{-1}$  at potentials of  $-0.1 \text{ V vs. RHE}$ . These values are much lower than that of a-FeMo-1/7, indicating a weak interaction between nitrogen and these electrocatalysts. Notably, c-FeMo-1/7 only shows a yield rate of 5.77  $\mu\text{g}_{\text{NH}_3} \text{mg}_{\text{cat.}}^{-1} \text{h}^{-1}$  at a potential of  $-0.1 \text{ V vs. RHE}$ , while a-FeMo-1/7 shows a much higher yield rate (12.4 times higher), which highlights the much stronger electron interaction between N<sub>2</sub> molecules and the surface of a-FeMo-1/7 [43]. As expected, a-FeMo-1/7 delivered the highest Faradaic efficiency at a potential of  $-0.1 \text{ V vs. RHE}$  (Fig. 4b). The calculated effective density ( $j_{\text{N}}$ ) of the a-FeMo-1/7 electrocatalyst is 0.116  $\text{mA cm}^{-2}$ , which is 9.99 times higher than that of the c-FeMo-1/7 electrocatalyst (Fig. 4c), again demonstrating its superior NRR performance. Surprisingly, although the ECSA of the a-FeMo-1/7 electrocatalyst is still larger than that of c-FeMo-1/7, it is smaller than that of Fe<sub>3</sub>O<sub>4</sub> and MoO<sub>3</sub> (Figs S22 and S23), demonstrating the higher intrinsic activity of the active sites in the a-FeMo-1/7 electrocatalyst. Similarly, the EIS results at different potentials indicate that the a-FeMo-1/7 electrocatalyst exhibits the lowest charge transfer property but inferior electron transport behavior (Fig. S24). As commonly documented, the activation of the N≡N triple bond lies as the main obstacle for nitrogen reduction due to the high bond energy [27,48]; thus, activation of this bond requires a stronger interaction between N<sub>2</sub> and the surface of the electrocatalyst. By constructing a heterointerface between Fe<sub>3</sub>O<sub>4</sub> and MoO<sub>3</sub>, significant electron transfer from the Fe sites to the Mo sites is induced; thus, an excellent NRR electrocatalyst with high Faradaic efficiency of 29.15%, an elevated yield rate of 71.78  $\mu\text{g}_{\text{NH}_3} \text{mg}_{\text{cat.}}^{-1} \text{h}^{-1}$ , and long-term stability can be readily achieved.

### OER electrochemical evaluation

To further validate the advantage of the synergistic effect between the atomic disorder and heterointerfaces in the electrocatalysis process, a-FeMo-1/7, c-FeMo-1/7, Fe<sub>3</sub>O<sub>4</sub>, and MoO<sub>3</sub> were also subjected to OER investigations. The electrocatalytic activity of each sample was evaluated using a three-electrode system with a mass loading of  $\sim 0.36 \text{ mg cm}^{-2}$  in an alkaline electrolyte. The detailed experimental procedure for the electrochemical measurement can be found in EXPERIMENTAL SECTION. The LSV curves of a-FeMo-1/7, c-FeMo-1/7, Fe<sub>3</sub>O<sub>4</sub>, and MoO<sub>3</sub> are displayed in Fig. 5a, which shows that only a 338 mV overpotential ( $\eta$ ) is required for a-FeMo-1/7 to achieve an anodic current density of 10  $\text{mA cm}^{-2}$ . Importantly, the water oxidation activity of a-FeMo-1/7 outperforms that of c-FeMo-1/7 (390 mV) and Fe<sub>3</sub>O<sub>4</sub> (470 mV) under the same conditions. Further analysis demonstrates an enhancement of the mass activity of the a-FeMo-1/7 electrocatalyst in comparison with their counterparts (Fig. S25). The Tafel slopes of the samples are shown in Fig. 5b, indicating that a-FeMo-1/7 can be a potential electrocatalyst for electrolyzers with a smaller Tafel slope (46.7  $\text{mV dec}^{-1}$  for a-FeMo-1/7, 86.1  $\text{mV dec}^{-1}$  for c-FeMo-1/7, 103.2  $\text{mV dec}^{-1}$  for Fe<sub>3</sub>O<sub>4</sub>, and 129.8  $\text{mV dec}^{-1}$  for MoO<sub>3</sub>), thereby reducing power loss [46]. Besides, long-term stability can also be achieved for the a-FeMo-1/7 electrocatalyst for as long as 30 h (Fig. S26). The TEM characterization and the XPS patterns further validate its superior stability in both the morphology and composition (Fig. S27). The excellent OER activity of a-FeMo-1/7 once again proves the advantages of atomic disorder and heterointerfaces in the electrocatalytic process, which may be beneficial for further investigations of high-performance catalysts and provide insight for the design of highly



**Figure 4** (a) NRR production rate of the a-FeMo-1/7, c-FeMo-1/7, Fe<sub>3</sub>O<sub>4</sub>, and MoO<sub>3</sub> electrocatalysts; (b) NRR Faradaic efficiency of the a-FeMo-1/7, c-FeMo-1/7, Fe<sub>3</sub>O<sub>4</sub>, and MoO<sub>3</sub> electrocatalysts; (c) effective  $j_{\text{N}}$  of the a-FeMo-1/7, c-FeMo-1/7, Fe<sub>3</sub>O<sub>4</sub>, and MoO<sub>3</sub> electrocatalysts.



**Figure 5** Polarization curves (a) and corresponding Tafel plots (b) of the electrocatalysts during the electrocatalytic OER process.

efficient electrocatalysts *via* structural modulation.

## CONCLUSIONS

In summary, we demonstrated that amorphism and hetero-interfaces in electrocatalysts could serve as an efficient strategy for the rational design of high-performance electrocatalysts. We report the successful preparation of a-FeMo electrocatalysts with an optimized Fe/Mo ratio. The a-FeMo-1/7 electrocatalyst exhibited excellent NRR activity with a high Faradaic efficiency of 29.15%, an elevated yield rate of  $71.78 \mu\text{g}_{\text{NH}_3} \text{mg}_{\text{cat}}^{-1} \text{h}^{-1}$ , and long-term stability. These excellent properties originate from the amorphous nature of the electrocatalyst, which increases the density of active sites, and the existence of heterointerfaces that enables obvious electron redistribution between components. Further examination of the OER performance of the studied electrocatalysts again validates the argument. This proof-of-concept approach not only achieves an active and durable NRR and OER catalyst but also provides a promising and effective alternative method for the design of remarkable heterogeneous catalysts.

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## 非周期性原子排列实现电化学氮还原和析氧反应性能的增强

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**摘要** 制备低成本的高效电催化剂是缓解目前面临的环境能源危机的重要方式。通过无定型化和构建异质结构可以有效地改变催化剂的电子结构,从而实现电催化性能的稳步提升。但是对于两者之间协同作用的研究则相对较少。在本文中,我们构建了富含异质结构的无定型FeMo基(a-FeMo)电催化剂,并系统评估了该催化剂的氮气还原和析氧反应性能。得益于无定型结构引起的活性位点数量的增多和异质结构引发的电子的重新分布,a-FeMo催化剂表现出优异的电化学催化性能。在-0.1 V vs. RHE的电势下,a-FeMo催化剂表现出29.15%的法拉第效率和71.78 μg<sub>NH<sub>3</sub></sub> mg<sub>cat.</sub><sup>-1</sup> h<sup>-1</sup>的氨气产率。这为合理设计具有低成本优势、高活性和长时间稳定性的优异催化剂提供了有益的参考。