SPECIAL TOPIC: Graphene Oxides towards Practical Applications

Coordination polymer nanowires/reduced graphene oxide paper as flexible anode for sodium-ion batteries

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The growing demand for lithium-ion batteries (LIBs) will lead to the shortage and high cost of lithium resource in the foreseeable future. Thus, it is essential to develop alternative battery technologies which are reliable and cost-effective [1]. Sodium-ion batteries (SIBs) have been regarded as one of the most promising alternatives for LIBs due to the abundance of sodium [2]. However, the large radius of Na⁺ ion always brings about low kinetics and instability of batteries. Therefore, the major challenge in advancing SIB technology lies in finding suitable host materials for Na⁺ ions, especially high performance anode materials. Until now, the sodium storage capabilities of some inorganic (e.g., metal oxides and sulfides, alloying type metals) and organic (e.g., carboxylates, imides) materials have been tested [3–7]. Although metal-based inorganic electrodes exhibit high capacity in the storage of Na⁺ ions, large volume expansion and severe capacity fading arise during the conversion or alloying reactions [3–5]. Organic electrodes in SIBs mainly based on C=O, C=N bonds and doping reactions provide the possibility for stable accommodation of large Na⁺ ions [6]. Besides, organic electrode materials possess other advantages such as abundance, environmental friendliness, structural diversity and flexibility. However, the inherent poor conductivity and high solubility in non-aqueous electrolytes become the bottlenecks limiting their further developments [7].

Polymerization has been confirmed to be an efficient way to mitigate the solubility of organics in electrolyte. The large extended backbone of repeating units could largely improve the stability of organics. Thus, different kinds of polymers such as conductive polymers (e.g., polypyrrole, polyaniline); carbonyl polymers (e.g., polyimides, polyquinones), radical polymers (e.g., nitroxide-based polymer) and coordination polymers (CPs) (e.g.,

Prussian blue) have been synthesized and tested as cathode or anode materials for SIBs [6–8]. Among them, CPs, especially their porous type namely metal organic frameworks (MOFs), have attracted lots of attention due to their facile synthesis, structural versatility, large surface area and enormous channels [9]. The CPs can be treated as one-, two- or three dimensional (1D, 2D or 3D) networks resulting from the coordination of metal ions and organic ligands [10]. As one important kind of CPs, MOFs always have well defined crystalline and porous structures while CPs may not. The CPs-based materials have been used as electrode materials for SIBs. For example, Prussian blue and its analogues with a general chemical formula of A_xM[Fe(CN)₆]_y·zH₂O (A represents alkali metal ion, M_a and M_b are transition metal ions) have been treated as one of the most promising SIB cathodes as Na⁺ ions can frequently insert into and desert from their lattices in a relatively high voltage window [11]. The CPs can also store Na⁺ ions in low voltage potentials. For instance, Co- and Cr-based MOFs were synthesized by Dong *et al.* [12] through the coordination of 5-amino-isophthalic acid with Co²⁺ and Cr²⁺ ions, respectively, and tested as anode materials for SIBs. It has been confirmed that the redox reactions were carried out between Na⁺ ions and carboxyl/amine groups; nevertheless, the valences of Co²⁺ and Cr²⁺ were not changed during the charge and discharge processes. Recently, Co-hexaaminobenzene (Co-HAB) was synthesized by Park *et al.* [13], which could provide a three-electron redox reaction in a voltage window of 0.5–3.0 V, presenting a new promising anode material for SIBs.

Although, some good results have been achieved, there are still obstacles in designing and constructing CPs-based electrodes. One of the challenges is to improve their electrical conductivity and structural stability during

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electrochemical reactions. Slurry coating method is commonly used in constructing CP electrodes by using the current collector, conductive agent and polymeric binder, resulting in a complicated technology and low energy densities [14–16]. The fabrication of CPs/carbon hybrids is an efficient way to solve these problems because carbonaceous materials (e.g., graphene) can not only act as fast electron transport networks but also provide stable structural protections [17–19]. Moreover, the excellent mechanical properties of graphene make it suitable substrate to construct flexible electrodes. Herein, flexible, interconnected CP nanowires/reduced graphene oxide (CPNWs/rGO) paper was constructed through a facile assembly and reduction process with Fe-based CP nanowires (Fe-CPNWs) and GO as precursor. The Fe-CPNWs constructed by Fe^{2+} (metal ions) and nitrilotriacetic acid (organic ligands) have a high aspect ratio and a large amount of functional groups (e.g., carboxyl), which can not only act as building blocks for free-standing electrodes but also provide enormous active sites for Na^+ ion storage. The hybrid paper was used as a free-standing electrode for SIBs, showing a relatively high specific capacity, superior rate capability, and stable cycling performance.

As schematically illustrated in Fig. 1, the synthesis of Fe-CPNWs/rGO paper is facile with two steps. Firstly, the 1D Fe-CPNWs with Fe^{2+} ions on the surface are electrostatically attached to the 2D GO nanosheets containing negatively charged oxygen-containing functional groups, then assembled into paper structure *via* filtration. Secondly, GO is thermally reduced at a relatively low temperature of 300°C , which can improve the electrochemical conductivity of GO nanosheets without damaging the crystal structure of Fe-CPNWs.

As shown in Fig. 2a, the free-standing and flexible paper with a diameter of ~ 4 cm was obtained after the synthetic procedures. The paper structure exhibits good mechanical deformations such as bending, twisting and

folding (Fig. 2a and Fig. S1). Field-emission scanning electron microscopy (FESEM) and transmission electron microscopy (TEM) were used to investigate the morphology and microstructure of the Fe-CPNWs/rGO paper. The cross-section FESEM image (Fig. 2b) exhibits the thickness of paper structure is $\sim 46\ \mu\text{m}$. Meanwhile, the top-view image (Fig. 2c) depicts the hybrid paper is constructed by 1D Fe-CPNWs (with a diameter of ~ 150 nm) and 2D rGO nanosheets with enormous macro-pores. The magnified image (Fig. 2d) exhibits the Fe-CPNWs are closely attached to the surface of rGO nanosheets, consisting with the TEM image (Fig. 2e). Furthermore, the Fe-CPNWs consist of C, N, O and Fe elements confirmed by the elemental mapping images (Fig. 2f–i). Noteworthy, there is little morphology change of Fe-CPNWs before and after the thermal treatment (Fig. S2). As reported, the 1D/2D hybrid structure always endows materials with good mechanical and electrochemical properties due to their synergistic effect [20,21]. Moreover, the porous networks can provide enough channels for the soakage of electrolyte and diffusion of Na^+ ions [22,23]. Thus, outstanding electrochemical performance may be expected.

A series of measurements including thermal gravimetric analysis (TGA), X-ray diffraction (XRD), and X-ray photoelectron spectroscopic (XPS) have been taken to investigate the crystal structure and composition of Fe-CPNWs and their hybrids. As shown in Fig. 3a, the Fe-CPNWs have a high thermal stability with nearly no weight loss before 300°C under N_2 . Afterwards, there are two drastic weight decrements ranged in 400 – 458°C and 556 – 600°C , which can be ascribed to the decomposition of organic ligands and the following carbonization process. The Fe-CPNWs show less thermal stability in O_2 atmosphere as the sample begins to be oxidized at a lower temperature of $\sim 200^\circ\text{C}$. The final weight loss is 55.6 wt% due to the oxidation of Fe-CPNWs to iron oxide (Fe_2O_3). It can be inferred that the chemical formula of Fe-

Figure 1 Schematic illustration for the construction of Fe-CPNWs/rGO paper.

Figure 2 (a) Digital images, (b) cross-section and (c, d) top-view FESEM images, (e) TEM image and (f–i) corresponding elemental mapping images (C, N, O, Fe) of the Fe-CPNWs/rGO paper.

Figure 3 (a) TGA of Fe-CPNWs conducted in N₂ and O₂ atmospheres, respectively. (b) XRD patterns of pristine Fe-CPNWs and Fe-CPNWs/rGO paper. (c, d) XPS spectra (C 1s and Fe 2p) of the Fe-CPNWs/rGO paper.

CPNWs is $\text{Fe}_3(\text{NA})_2$ (NA represents nitrilotriacetic acid) based on the weight loss. According to the results of TGA, a suitable temperature of 300°C has been chosen to modestly reduce the functional groups attached on GO, and keep the intact structure of Fe-CPNWs. Fig. 3b shows that all the diffraction peaks of Fe-CPNWs in the hybrid paper are in accordance with the pristine samples, confirming there is no crystal structure change. However, the intensity of peaks becomes stronger, which can be attributed to the crystallization process during the thermal treatment.

XPS were analyzed to find out the surface chemical compositions of Fe-CPNWs/rGO. As shown in all survey spectrum (Fig. S3), there are C, O, N, Fe elements coexisting in the hybrid, which is in accordance with the elemental mapping. The C 1s spectrum can be de-convoluted into three peaks. The C–C peak located at 284.6 eV is assigned to the carbon framework of graphene. The other two peaks at 286.1 and 288.8 eV belong to C–N/C–O (amine/hydroxyl) and C=O (carboxyl and carbonyl) groups, deriving from the organic ligands (NA) and partial reduced graphene oxide nanosheets [24]. The Fe 2p spectrum (Fig. 3d) displays two couple peaks located at 710.6/724.0 eV and 714.4/727.9 eV, respectively. The former peaks can be attributed to Fe^{2+} ions coordinated with carboxyl (OCO–Fe) and amine (N–Fe) groups [24,25], while the latter Fe^{3+} ions come from the oxidation of surface Fe^{2+} ions due to their less stability in air.

Benefiting from the flexible mechanical property, the Fe-CPNWs/rGO paper is directly used as free-standing anode for SIBs. Fig. 4a shows the cyclic voltammetry (CV) curves of the paper electrode tested in a voltage window of 0.01–3.0 V at a scan rate of 0.1 mV s^{-1} . During the first cathodic scan, the broad reduction peak from 1.6 to 0.9 V can be ascribed to the organic ligand transformation (i.e., carboxylate to enolate conversion), which becomes stronger in the following cycles due to the activation process [26]. The sharp peak sloping down to 0.01 V is associated with the formation of solid electrolyte interface (SEI) film and the side reactions between the surface functional groups attached on polymers and Na^+ ions [12,27–29]. Two oxidation peaks centered at 1.79 V can be ascribed to the Na^+ desertion process. In the subsequent cycles, a new reduction peak at 0.93 V appears, due to the structure reconstruction in the first cycle. The CV curves are basically overlapped thereafter, indicating a good reversibility. To further investigate the electrochemical mechanism of electrodes, *ex-situ* XPS analyses of the electrode materials collected at the dis-

charge state after 5 cycles were done. As shown in the Fe 2p spectrum (Fig. S4a), there are only two peaks at 711.0 and 724.6 eV, which can be ascribed to the Fe^{2+} ions [25]. The absence of Fe(0) indicates the valence of centered metal atoms remains unchanged during the electrochemical reactions. However, the peak intensities and areas of C=O and C–N/C–O groups decrease significantly at the discharge state (C 1s spectrum, Fig. S4b). The above results point out that the organic moieties especially carboxylic groups play an essential role in the reversible redox reactions.

The initial discharge and charge capacities of the Fe-CPNWs/rGO paper are 787 and 454 mA h g^{-1} with a coulombic efficiency (CE) of 57.7% (Fig. 4b). The irreversible part can be attributed to the formation of SEI film and the side reactions [12,27–29]. The CE reaches over 96% for the 5th cycle and can be kept from then on. Fig. 4c shows the cycling performance of the Fe-CPNWs/rGO paper at a current density of 100 mA g^{-1} . The paper electrode demonstrates a very stable cycling performance as a discharge capacity of 319 mA h g^{-1} can still be kept after 200 cycles, which is superior to the Fe-CPNWs paper (7 mA h g^{-1}) without rGO addition (Fig. S5a). Evidently, the rGO substrate can not only improve the electrochemical conductivity of the hybrid electrodes (as confirmed by electrochemical impedance spectra, Fig. S6), but also help keep their structures intact (Figs S7 and S8). In addition to good cycling stability, the Fe-CPNWs/rGO paper also exhibits excellent rate capability, yielding average reversible capacities of 404, 308, 272, 220, 148 and 120 mA h g^{-1} at the current densities of 100, 300, 500, 1000, 2000 and 3000 mA g^{-1} , respectively. Meanwhile, the capacity can restore to 340 mA h g^{-1} after deep cycling at a high rate. In contrast, the Fe-CPNWs paper without rGO shows inferior rate performance as only 3 mA h g^{-1} can be obtained at 500 mA g^{-1} (Fig. S5b). To further confirm the rate capability and cyclic stability, long cycling performances of the Fe-CPNWs/rGO paper electrode under high rates of 500 and 1000 mA g^{-1} were tested (Fig. 4e). The capacity loss at the initial 500 cycles can be ascribed to the formation of unstable SEI layers at a high current density, which slow down the Na-ion transport behavior and degrade the electrochemical performance [16,22]. However, stable SEI layer is formed during the long cycling performance and capacities of 115 and 100 mA h g^{-1} can be achieved after 2000 and 4000 cycles with very low capacity decay rates of 0.03% (at 500 mA g^{-1}) and 0.02% (at 1000 mA g^{-1}) per cycle. The electrochemical performance of the Fe-CPNWs/rGO paper is superior to many organic anodes previously re-

Figure 4 Electrochemical performance of the Fe-CPNWs/rGO paper for SIBs. (a) CV curves (0.1 mV s^{-1}) between 0.01 and 3.0 V. (b) Galvanostatic charge and discharge voltage profiles (100 mA g^{-1}). (c) Comparative cycling performance (100 mA g^{-1}) of Fe-CPNWs/rGO paper and Fe-CPNWs paper without rGO. (d) Rate capability at various current rates from 100 to 3000 mA g^{-1} . (e) Long-term cycling performance at 500 mA g^{-1} and 1000 mA g^{-1} .

ported (Table S1) [12,18,30–34]. In addition, a full sodium ion cell was assembled with $\text{Na}_3\text{V}_2(\text{PO}_4)_3/\text{rGO}$ (NVP/rGO) as the cathode and Fe-CPNWs/rGO as the anode. A voltage window of 0.5–3.9 V has been chosen according to the voltage profiles and corresponding dQ/dV curves of NVP/rGO and Fe-CPNWs/rGO (Fig. S9a, b). The full cell delivers a relatively stable cycling performance as a charge capacity of 171 mA h g^{-1} can be kept after 100 cycles at 100 mA g^{-1} (Fig. S9c, d). Moreover, a flexible pouch-type SIB was also fabricated and the light emitting diode (LED) can be easily lightened under the normal and bending states (Fig. S10).

In conclusion, CP-based flexible SIB anodes consisting

of 1D Fe-CPNWs and 2D rGO are constructed *via* the assembly and controlled reduction process. The multi-dimensional hybrid electrodes exhibit excellent mechanical and electrochemical properties. When tested as free-standing paper anodes for SIBs, the hybrids exhibit high rate capability ($404\text{--}120 \text{ mA h g}^{-1}$ at a current density of $100\text{--}3000 \text{ mA g}^{-1}$) and long cyclic life (100 mA h g^{-1} can be kept after 4000 cycles at 1000 mA g^{-1}). The organic moieties (carboxyl and amine groups) are the redox centers in CPs. This work provides a new flexible anode candidate for SIBs.

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Supplementary information Experimental details and supporting data are available in the online version of the paper.

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配位聚合物纳米线/还原氧化石墨烯复合柔性薄膜电极的构筑及储钠性能研究

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摘要 构建基于有机材料的高性能柔性储钠电极面临诸多挑战. 本工作通过可控组装及还原的方式, 实现了铁基配位聚合物纳米线/还原氧化石墨烯柔性薄膜的构筑. 多维复合薄膜可直接用作钠离子电池自支撑负极, 且具有较高的储钠容量(200 mA g^{-1} 电流密度下可逆容量为 319 mA h g^{-1})和优异的倍率性能(3000 mA g^{-1} 大电流密度下比容量可保持在 $\sim 120 \text{ mA h g}^{-1}$). 研究表明有机配体(氨三乙酸)中的羧基及氨基为储钠活性位点, 而配位金属离子(Fe^{2+})不参与电化学反应.