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Face-to-face engineering of ultrathin Pd nanosheets on amorphous carbon nitride for efficient photocatalytic hydrogen production

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ABSTRACT Photocatalytic hydrogen production represents a promising strategy for clean, sustainable, and environmentfriendly energy supply. Up to now, great efforts have been devoted to designing the photocatalysts with noble metal as co-catalyst for visible-light-driven hydrogen evolution, while more efficient photocatalytic systems are still a major challenge. Herein, we report a facile strategy for synthesizing faceto-face ultrathin Pd nanosheets-amorphous carbon nitride (Pd NSs-ACN) structure with large contacting interface and short electronic transmission pathway, which can work as an efficient photocatalyst for hydrogen production. The synthesis starts with the growth of ultrathin Pd NSs, followed by assembly with the visible-light-response ACN through a simple stirring and annealing procedure. The resultant two dimensional face-to-face structures deliver an average hydrogen generation rate of 1.45 mmol h⁻¹ g⁻¹ at a temperature of 25°C, almost 2.6 times higher than that of Pd Nps-ACN with particle-to-face structural feature. The efficient photocatalytic activity is ascribed to the formation of high-density of active sites between ultrafine face-to-face contacted Pd NSs and the ACN, which cooperate more synergistically towards photocatalytic hydrogen production. The face-to-face engineered Pd NSs-ACN hybrids also offer a good stability revealed by photocatalytic hydrogen production measurements. The extraordinary performance highlights a powerful engineering model for designing other face-to-face contacting co-catalyst/ photocatalysts, which will be a great impetus to optimize new catalytic transformations.

Keywords: face-to-face engineering, ultrathin Pd nanosheets, amorphous carbon nitride, photocatalytic hydrogen production

INTRODUCTION

Photocatalytic water splitting to produce hydrogen is a promising means of storing solar energy to addressing the current energy and environmental demands. In past years, various inorganic and organic semiconductors have been developed as photocatalysts for hydrogen evolution driven by solar energy [1-7]. The undesirable recombination of photogenerated electrons-holes and the narrow light absorption range limit the efficiency of photocatalytic reactions and further application. To overcome these problems, many strategies have been developed, for instance, morphology control, element doping, metal deposition, and heterostructure formation in the aim of improving the performance of photocatalytic hydrogen production [8-10]. As well known, noble metals (Pt [11-12], Au [13-15], Pd [13] and Ag [16]) can not only trap the photogenerated electrons from the semiconductor to promote the electron-hole separation, but also serve as the catalytic active sites in the adsorption and reduction of H⁺. Currently, most of hydrogen production systems reported are noble metalloaded photocatalysts.

Much work so far has focused on metal-free polymeric graphitic carbon nitride $(g-C_3N_4)$ as photocatalyst, and transition metal as co-catalyst for photocatalytic hydrogen production, such as Fe [17], Co [18], Ni [19], Cu [20,21]. However, these noble-metal free/semiconductor systems suffer from a high recombination rate of photoexcited charge carriers, resulting in low photocatalytic activity. At this point, the noble metal/semiconductor photocatalyst systems, with their superior performance, is still the most likely to be industrialized in the future

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[17,21]. At present, it is necessary to design a noble metalco-catalyst/semiconductor photocatalytic system with high photoelectron mobility for efficiently catalyzing photocatalytic hydrogen production or photocatalytic oxygen production. For all of noble metal/semiconductor systems, the contact model between the noble metal and photocatalyst is particle-to-face [22-24]. Noble metal nanoparticles (NPs) can only provide very limited active sites and slow kinetics, leading to the low photocatalytic efficiency. Inspired by these challenging issues, the development of new methods for engineering noble metal co-catalyst, like nanostructured engineering [13,25], with maximum activity, stability, and minimum costs has become very important. Among various strategies, ultrathin two dimensional (2D) sheet-like structures with a single or few atoms thickness are of great interest due to their small lateral size, high electron mobility, and relatively high surface energy, which is a result of their high surface to volume ratio and high density of unsaturated atoms. Also, ultrathin 2D nanosheets (NSs) with most of atoms exposed for modifications can act as an ideal platform for

engineering their properties [26-28]. Herein, we demonstrate a face-to-face engineering fabrication of ultrathin Pd NSs-amorphous carbon nitride (Pd NSs-ACN) structure for photocatalytic hydrogen evolution. In this face-to-face hybrid structure, the ACN serves as light absorber for wide solar spectra harvesting (about below 672 nm), and the ultrathin Pd NSs act as an electron transfer relay at the photocatalyst/reaction solution interface to accelerate photoelectron output from photocatalyst to co-catalyst for fast photocatalytic H₂ production. The Pd NSs with thickness about 1.0 nm not only improves the separation of photoelectrons and holes, but also provides more active sites for photocatalytic reactions, which boosts rapid adsorption of hydrogen ions from water and further reduces them into the hydrogen gas. The optimal designed face-to-face Pd NSs-ACN shows almost 2.6 times higher hydrogen evolution activity under visible light than that of Pd NPs-ACN hybrid with particle-to-face structural feature. The ultrathin Pd NSs loaded on ACN for photocatalytic H₂ production has not been explored so far. Our work represents a new interface engineering strategy for the creation of more efficient interface between co-catalyst and photocatalyst for photocatalytic H₂ production driven by sunlight.

EXPERIMENTAL SECTION

Materials

Pd(II) acetylacetonate, poly(vinylpyrrolidone) (PVP), and

 $W(CO)_6$ were purchased from Aladdin Ltd. Citric acid (CA), *N*,*N*-dimethylformamide (DMF) and cetyltrimethyl ammonium bromide (CTAB) were obtained from Sigma-Aldrich. All chemicals were used as received without further purification. The water used throughout all experiments was purified through a Millipore system.

Preparation of ACN

The graphitic carbon nitride (GCN) photocatalyst was prepared according to the literature [29]. 5 g Melamine was heated at 500°C in a tube furnace under an air atmosphere for 4 h, with the ramping rate of 2°C min⁻¹, and then cooled to room temperature and ground to get GCN powder. The as-obtained GCN powder was heated at 620°C for 2 h under N₂ to get the ACN.

Preparation of Pd NSs

Pd NSs were synthesized according to the reported method [30]. In typical procedure, Pd(II) acetylacetonate (Pd(acac)₂, 8 mg), PVP (M_W =24,000, 15 mg), CA (75 mg), DMF (5 mL) and CTAB (30 mg) were mixed in a 25 mL flask and stirred for 1 h under N₂. When the color of the mixture changed from yellow to homogeneous orange-red, 50 mg of W(CO)₆ was added into the flask. Then, the flask was sealed and heated at 80°C for 1 h. Then the dark blue products were separated by centrifugation using a sufficient amount of acetone, and then washed with ethanol for several times. Finally, the Pd NSs were dispersed in 10 mL ethanol for later experiment.

Preparation of Pd NSs-ACN

Firstly, a certain amount of Pd NSs in ethanol solution and ACN (120 mg) were dispersed in 100 mL ethanol by sonication for 1 h, respectively. Next, the Pd NSs ethanol solution was added into ACN under stirring for 1 h, and then dried under vacuum at 50°C for 10 h. Finally, the sample was kept at 200°C for 1 h in a digital-type temperature-controlled oven and then naturally cooled to room temperature. To obtain different weight percent of Pd 0.05%, 0.12%, 0.36%, and 0.49% by the inductively coupled plasma mass spectrometry (ICP-MS), the amount of Pd NSs added was 0.5, 1.5, 3, and 5 mL, respectively.

Photocatalytic hydrogen production

The photocatalytic reactions of the composite photocatalysts were carried out in an outer irradiation-type photoreactor (glass) connected to a closed gas-airtight and a constant temperature system. 20 mg of the photocatalysts was dispersed in 90 mL ultrapure water and 10 mL triethanolamine (TEOA) with sonication. The suspension was thoroughly degassed to remove air and irradiated using a 300 W Xe-lamp (λ >420 nm). The photocatalytic H₂ evolution rate was analyzed using an online Agilent 7890B gas chromatograph (GC, TCD detector, 5 Å molecular sieve columns and Ar carrier). The controlled photo-deposition of 0.36 wt% Pd NPs as a reducing co-catalyst in pure ACN was prepared by dissolving K₂PdCl₄ in the reaction solution. All the photocatalytic hydrogen reaction underwent around 25°C.

RESULTS AND DISCUSSION

The face-to-face engineering fabrication of Pd NSs-ACN structure chooses ultrathin Pd NSs as a co-catalyst model, which was prepared through a classical method according to the previous literature [30]. The lateral size of the asprepared Pd NSs is about 10.5 nm (Fig. S1a and b), with an average thickness of about 1.0 nm (Fig. 1a). The percentage of top and bottom area to the whole surface is up to 90%. The large flat surface of the Pd NSs endows them with an ultrahigh specific surface area, making them highly favorable for surface active applications. Meanwhile, ACN has super-high specific surface area about 252.4 $m^2 g^{-1}$ (Fig. S2), which is conducive to the immobilization of Pd NSs. These advantages will lead to excellent catalytic performance in photocatalytic system of interface engineering between Pd NSs and ACN semiconductor.



Figure 1 Transmission electron microscopy (TEM) images of Pd NSs (a) and Pd NSs-ACN (c). (b) XRD patterns of the as-synthesized photocatalysts: ACN, Pd NSs-ACN, Pd NPs-ACN and Pd NSs. (d) High resolution TEM (HRTEM) image of the Pd NSs-ACN.

According to the X-ray diffraction (XRD) patterns (Fig. 1b), the Pd NSs-ACN, Pd NPs-ACN and ACN exhibit very similar profiles, where the characteristic peaks located at 27.2° can be well assigned to the (002) planes of GCN [29,31], due to the accumulation peak of the conjugated aromatic ring. The as-obtained Pd NSs are in the cubic structure (JCPDS 46-1043) [32]. The Pd-ACN did not show the apparent peak of Pd, due to its low Pd content (Fig. S3a). For GCN, two typical diffraction peaks at around 13.1° and 27.2° were sharp and intense, indicating its highly crystalline nature (Fig. S3b). However, these two sharp peaks disappear in the ACN pattern, and only one very near at 27.2° (Fig. 1b), suggesting the absence of long-range order in the atomic arrangements in ACN [33].

As shown in Fig. S4a, many pores with a relatively irregular pore size were formed by heating GCN in N₂, which can provide higher specific surface area. In addition, the high temperature treatment breaks the hydrogen bonds of layered carbon nitride, increasing visible light absorption range by creating abundant band tails. As indicated in Fig. 1c and Fig. S4b, both samples of Pd NSs and NPs have uniform dispersion on the surface of ACN nanosheet to form different types of Pd-ACN hybrid structures. It could be clearly seen that the Pd NPs and ACN contact with each other through the mode of particle-to-face. However, the Pd NSs and ACN contact with each other by another mode of face-to-face, in which the large interfacial areas formed between Pd NSs and ACN guarantee the smooth electron flow from ACN to Pd. The loading of Pd NSs on ACN can be tuned by changing the amount of Pd NSs-ethanol solution added (Fig. S5a-d). The high-resolution TEM (HRTEM) is further carried out to characterize the interface between Pd NSs and ACN (Fig. 1d). From Fig. 1d, the Pd NSs loaded on the surface of ACN exhibit an intimate contact between Pd NSs and ACN, forming the ultrafine Pd NSs-ACN interface. Fig. 1d reveals that the fringes with lattice spacings of 0.256 and 0.22 nm can be indexed to the Pd {200} and {111} [30], respectively. Note that the lattice fringes of ACN cannot be resolved by HRTEM, possibly due to its rapid degradation under irradiation of an electron beam.

The chemical states of carbon, nitrogen and palladium in the products (Pd NSs and Pd NPs (0.36%)-ACN) were investigated by X-ray photoelectron spectroscopy (XPS). The survey XPS spectrum (Fig. 2a) shows the C, N, Pd and O peaks in Pd-ACN. In Fig. 2b, the C 1s peak can be deconvoluted into three peaks at 288.1, 285.2 and 286.5 eV, corresponding to sp²-bonded aromatic struc-



Figure 2 XPS spectra of Pd NPs-ACN and Pd NSs-ACN. (a) Survey spectrum, (b) C 1s, (c) N 1s and (d) Pd 3d high-resolution spectra.

ture (N–C=N), graphitc carbon (C–C), and residual C–O, respectively. Meanwhile, high-resolution spectrum of N 1s (Fig. 2c) can also be divided into three peaks at 398.3, 399.7 and 400.8 eV, which are associated with nitrogen species in the N(C)₂, N(C)₃, and N(CH) groups of melon [34], respectively. The two peaks of Pd 3d (Fig. 2d) located at 335.5 and 339.2 eV are assigned to Pd $3d_{5/2}$ and Pd $3d_{3/2}$, respectively, which are in good agreement with the metal Pd. Hence, Pd NSs and Pd NPs belong to metal Pd [35].

The photocatalytic activity of all samples was evaluated by the H₂ evolution under Xe lamp irradiation (λ >420 nm), using TEOA as a scavenger. Fig. 3a shows the photocatalytic H2-production activity of Pd NSs-ACN (0.36 wt%), Pd NPs-ACN (0.36 wt%) and ACN catalysts. For ACN, the rate of H₂-production can be almost ignored because of the rapid recombination of photogenerated electron-holes and the quick reversible reaction (hydrogen and oxygen recombination into water). A small amount of Pd loading on ACN can obviously enhance the photocatalytic H₂ production. The average hydrogen generation rate of Pd NSs-ACN is 1.45 mmol h^{-1} g⁻¹, higher than that of the Pd NPs-ACN $(0.56 \text{ mmol h}^{-1} \text{ g}^{-1})$. The enhancement of H₂ production of ACN by ultrathin Pd NSs loading is attributed to the following three reasons: 1) the face-to-face contact between the ultrathin Pd NSs and ACN shortens the distance of photoelectron transport from ACN to Pd cocatalyst and inhibits the recombination of photogenerated electrons and holes; 2) the ultrathin Pd NSs provides a



Figure 3 (a) Photocatalytic H_2 evolution from the Pd NSs-ACN, Pd NPs-ACN and ACN. (b) Cycle stability test on Pd NSs-ACN photocatalytic H_2 evolution under visible light irradiation.

large number of surface active sites capturing the H^+ from the reaction solution for photocatalytic hydrogen production; 3) a large number of surface atoms exposed by ultrathin Pd NSs serve as an ideal platform for releasing hydrogen gas and further improve the photocatalytic performance. A series of control experiments and photocatalytic tests show the best weight percent (wt%) of Pd NSs to ACN is ~0.36% (Fig. S6).

In addition, to evaluate the stability of the Pd NSs-ACN, the recycling experiments for the H₂ evolution were conducted. The photocatalytic reaction was allowed to process for 1 h with intermittent evacuation every 5 h under visible light irradiation (λ >420 nm). As can be seen from Fig. 3b, the H₂ evolution rate remains fairly stable after 15 h irradiation, indicating the long-term photocatalytic stability of H₂ evolution by the face-to-face engineered Pd NSs-ACN hybrids, consistent with the TEM results of of Pd NSs-ACN after photocatalytic measurements (Fig. S7). It is worth noting that the excellent solar photocatalytic activity was obtained in the presence of very small amounts of co-catalysts (Pd NSs 0.36%), which suggests the striking potential of these Pd NSs-ACN complex structure for practical technology application.

To confirm the interface effect induced by the face-toface engineering between ultrathin Pd NS and ACN which is the main factor in enhancing the H_2 evolution activity, the UV-vis diffuse reflectance spectra (DRS), photoelectrochemical test, electrochemical impedance spectroscopy (EIS) measurements, photoluminescence (PL) and time-resolved PL (TRPL) spectra were performed. As show in Fig. 4, the UV-vis absorption of all samples shows a clear absorption edge at 400–600 nm, demonstrating that ACN has strong visible light absorbability. Furthermore, the Pd NSs-ACN exhibits a bit of stronger light absorption in the visible range in comparison with the Pd NPs-ACN control sample, due to the relatively strong surface plasmon resonance (SPR) effect of Pd NSs than in the Pd NPs. Given that the electron-



Figure 4 UV-vis DRS of Pd NSs-ACN, Pd NPs-ACN and ACN composites.

hole separation ability can be reflected by the photocurrent, the photoelectrochemical tests were carried out [36]. Fig. 5a reveals that the photocurrents of Pd NSs-ACN sample are much larger than those of Pd NPs-ACN and ACN, suggesting that Pd NSs can easily trap the electrons from ACN to promote the electron-hole separation in relative to Pd NPs. Moreover, after four lighton and -off cycles, the photoelectrochemical test shows no obvious decay for the Pd NSs-ACN, indicating that the separation and recombination of photogenerated charges maintains constant value.

The effect of face-to-face hybrid structures on the kinetics of interfacial charge immigration was further investigated by EIS analysis. The arc radius on the EIS Nyquist plot (Fig. 5b) reflects the reaction rate on the surface of the electrode. A smaller arc radius corresponds to a more effective separation of photogenerated electronhole pairs and a higher efficiency of charge immigration across the electrode/electrolyte interface. The arc radius of Pd NSs-ACN electrode is smaller than those of Pd NPs-ACN and ACN electrode, suggesting that the Pd NSs-ACN structure can make the separation and immigration of photogenerated electron-hole pairs be more efficient, in good accordance with the result of the photocurrent measurement.

Fluorescence spectroscopy is a powerful tool to determine the transfer and separation efficiency of the photogenerated charge carriers in excited semiconductors. The PL spectra of Pd NSs-ACN and Pd NPs-ACN excited at 330 nm are presented in Fig. 5c. Compared with the Pd NPs, the emissions in Pd NSs-ACN



Figure 5 (a) Photocurrent responses of Pd NSs-ACN, Pd NPs-ACN and ACN. (b) EIS Nyquist plots of Pd NSs-ACN, Pd NSs-ACN, and ACN under visible-light conditions. (c) PL spectra of Pd NPs-ACN and Pd NSs-ACN under photoexcitation at 330 nm. (d) The TRPL spectra of Pd NPs-ACN and Pd NSs-ACN and Pd NSs-ACN monitored under a 404 nm laser excitation.

were remarkably quenched, suggesting faster transfer of electrons from ACN to Pd NSs, which can suppress the electron-hole recombination and enhance the photocatalytic activity. The PL intensity of Pd NSs-ACN is lower than that of Pd NPs-ACN, which is also consistent with the former H_2 evolution activity results [37].

The TRPL spectra (Fig. 5d) monitored at emission peak of each product also prove that the lifetime of Pd NSs-ACN and Pd NPs-ACN is shortened gradually. The decay lifetime of Pd NSs-ACN and Pd NPs-ACN is 8.82 and 11.05 ns (Table 1), respectively. This lifetime can be calculated using the following equation [6]:

$$<\tau>=(f_1\tau_1^2+f_2\tau_2^2+f_3\tau_3^2)/(f_1\tau_1+f_2\tau_2+f_3\tau_3),$$

where τ refers to the average lifetime; τ_1 , τ_2 and τ_3 refer to the different excited states of radiative lifetimes; f_1 , f_2 and f_3 refer to the different relative percentages of charge carriers, respectively. The TRPL data further reveal that the lifetime of Pd NSs-ACN is much shorter than that of Pd NPs-ACN. Photocatalysts with co-catalyst materials are expected to have shorter lifetimes due to the efficient charge transfer from the photocatalyst to co-catalyst on the surface and suppression of the electron-hole recombination [38]. All the above results certify that the face-to-face engineering between ultrathin Pd NSs and ACN is more efficient to improve photocatalytic hydrogen evolution.

To understand the drastically enhanced photocatalytic activity of Pd NSs-ACN, we further proposed the photocatalytic mechanism of Pd NSs-ACN and Pd NPs-ACN,

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Sample	Decay time (ns)			Relative amplitude (%)			Average lifetime
	$ au_1$	$ au_2$	$ au_3$	f_1	f_2	f_3	$(\langle \tau \rangle, ns)$
Pd NSs-ACN	0.75	2.77	11.79	39.36	40.81	19.83	8.82
Pd NPs-ACN	0.76	2.89	14.35	30.03	43.72	26.25	11.05

Table 1 Summary of the photoluminescence decay time (τ) and their relative amplitude (f) in the Pd NSs-ACN and the Pd NPs-ACN



Figure 6 Proposed photocatalytic reaction mechanism of Pd NPs-ACN and Pd NSs-ACN.

respectively. As shown in Fig. 6, electrons in the valence band of the ACN are excited to the conduction band under visible light irradiation, and the photogenerated holes are accumulated at valance band. Because of the lower work function of ACN than Pd [39], the free electrons of ACN could easily transfer to Pd when contacting. It was obvious that ultrathin Pd NSs with lower thickness shorten the distance for electrons to transfer from photo-excited ACN. In addition, the face-to-face contacting in Pd NSs-ACN can provide the direct channel so that the photoelectrons can be transferred to the surface of Pd NSs from ACN for photocatalytic hydrogen reaction. However, due to the limited contact area in particle-to-face Pd NPs-ACN, the photoelectrons are gathered at the touch point and cannot be transmitted to Pd NPs efficiently, which have poor hydrogen production. Besides, the more rapid reduction of H⁺ in Pd NSs-ACN is another key factor for adsorption of new hydrogen ions from the solution after the H₂ desorption from the upsides of Pd NSs. Thus, the face-to-face contact between Pd NSs and ACN has more active sites and direct electron channel, revealing excellent photocatalytic performance than in the particle-to-face Pd NPs-ACN.

CONCLUSIONS

In summary, we demonstrate a feasible approach to

synthesize Pd NSs-ACN with distinctive face-to-face structure consisting of large-density active sites and largearea contacting interfaces. These features enable them impressive activity towards photocatalytic hydrogen production. The unique interface in Pd-NSs-ACN shows much superior photocatalytic activities for hydrogen generation than those in the Pd NPs-ACN and ACN. In the face-to-face interface structure, the ultrathin Pd NSs not only promote the rapid transfer of photoelectrons from ACN, but also expose outside surface as an ideal platform for capturing H⁺ to produce H₂. In the optimized Pd NSs-ACN (0.36%) with large contact interface and ultra-short electron transport distance, the hydrogen production activity is over 2.6 times of magnitude than that of particles-to-face Pd NPs-ACN. This work may open a new avenue for photocatalyst discovery and new catalytic optimizer with high solar-driven photocatalytic performance by engineering the interface between the cocatalyst and photocatalyst.

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Author contributions Li H and Guo S designed and engineered this work. Tang Y performed the experiments and wrote the paper with

Supplementary information Experimental data are available in the online version of the paper.



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超薄钯纳米片/非晶氮化碳共面复合结构的控制生长及其可见光驱动下的光催化产氢性能研究

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摘要 目前,能源与环境问题已经成为影响人类可持续发展的主要矛盾.为了实现人类社会的可持续发展,研究者们一直致力于开发新的 储能技术. 半导体光催化制氢凭借其清洁、可持续、环境友好的优势成为研究热点. 传统的光催化制氢体系以贵金属为助催化剂,宽带隙 半导体为光催化剂,这种光催化系统的太阳能转换效率难以满足实际需求. 在本文中,我们合成出一种具有较大接触界面和较短的电子传 递路径的共面型超薄钯纳米片/非晶氮化碳复合结构. 在室温25℃条件下,该结构平均氢气生成速率为1.45 mmol mg⁻¹ h⁻¹,是钯纳米颗粒-无定形碳化氮粒面型结构的2.6倍. 同时,该共面型光催化剂具有优良的产氢稳定性. 该催化剂既充分利用了钯纳米片表面高密度的活性位 点,又利用了无定型氮化碳宽的光谱响应. 本工作为可见光驱动的高效助催化剂和光催化剂界面设计提供了一种新的策略.