Selective Roasting of Nd–Fe–B Permanent Magnets as a Pretreatment Step for Intensified Leaching with an Ionic Liquid

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Abstract

Oxidative roasting of Nd–Fe–B permanent magnets prior to leaching improves the selectivity in the recovery of rare-earth elements over iron. However, the dissolution rate of oxidatively roasted Nd–Fe–B permanent magnets in acidic solutions is very slow, often longer than 24 h. Upon roasting in air at temperatures above 500 °C, the neodymium metal is not converted to Nd₂O₃, but rather to the ternary NdFeO₃ phase. NdFeO₃ is much more difficult to dissolve than Nd₂O₃. In this work, the formation of NdFeO₃ was avoided by roasting Nd–Fe–B permanent magnet production scrap in argon atmosphere, having an oxygen content of $p_{O_2} \leq 10^{-20}$ atm, with the addition of 5 wt% of carbon as an iron reducing agent. For all the non-oxidizing iron roasting conditions investigated, the iron in the Nd–Fe–B scrap formed a cobalt-containing metallic phase, clearly distinct from the rare-earth phase at microscopic level. The thermal treatment was optimized to obtain a clear phase separation of metallic iron and rare-earth phase also at the macroscopic level, to enable easy mechanical removal of iron prior to the leaching step. The sample roasted at the optimum conditions (i.e., 5 wt% carbon, no flux, no quenching step, roasting temperature of 1400 °C and roasting time of 2 h) was leached in the water-containing ionic liquid betainium bis(trifluoromethylsulfonyl)imide, [Hbet][Tf₂N]. A leaching time of only 20 min was sufficient to completely dissolve the rare-earth elements. The rare-earth elements/iron ratio in the leachate was about 50 times higher than the initial rare-earth elements/iron ratio in the Nd–Fe–B scrap. Therefore, roasting in argon with addition of a small amount of carbon is an efficient process step to avoid the formation of NdFeO₃ and to separate the rare-earth elements from the iron, resulting in selective leaching for the recovery of rare-earth elements from Nd–Fe–B permanent magnets.

Keywords Nd–Fe–B magnet scrap · Ionic liquid leaching · Iron roasting · EPMA

Introduction

Production scrap of Nd–Fe–B permanent magnets, or just Nd–Fe–B production scrap, is currently considered as the most valuable secondary resource of neodymium and dysprosium [1]. Several flowsheets have been developed for the recovery of rare-earth elements (REEs) from Nd–Fe–B

scrap, via direct or indirect recycling [1–8]. Roasting of Nd–Fe–B permanent magnets is known to improve the selectivity and the dissolution rate of the consequent leaching step [9–11]. However, when roasting is carried out in air at temperatures above 500 °C, neodymium is converted to a ternary neodymium-iron oxide phase, NdFeO₃, and not to the binary neodymium oxide phase, Nd₂O₃ [12, 13]. If properly carried out, roasting can convert neodymium to Nd₂O₃ and separate the rare-earth elements from the less valuable iron. The oxidation mechanism of Nd–Fe–B permanent magnets in air has recently been well described by Firdaus et al. [14]. The authors suggested NdFeO₃ as the preferred form to recover neodymium from the roasted magnets. This statement is surprising since NdFeO₃ has been observed to dissolve much slower in acidic lixiviants than the binary Nd₂O₃ [15, 16]. The dissolution of NdFeO₃ can take up to 72 h [12, 17, 18]. The phase stability of the products from the roasting of Nd–Fe–B permanent magnets depends on the

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composition of the magnet, the partial oxygen pressure (p_{O_2}) and the roasting temperature [13, 19]. The p_{O_2} at equilibrium can be controlled by the C/CO equilibrium, where CO is used as the oxidizing agent and carbon as the reducing agent [13]. In particular, the conditions favorable to avoid the formation of $NdFeO_3$ are (1) temperatures between 1350 °C and 2000 °C; and (2) a p_{O_2} between 10^{-15} and 10^{-25} atm.

To avoid the formation of $NdFeO_3$, Maroufi et al. [20] and Bian et al. [21] carried out a combined process of oxidation at 1000 °C and consequent reduction above 1400 °C of Nd–Fe–B permanent magnets. In both studies, the product was a mixed rare-earth oxide phase separated from a metal phase mainly containing transition metals (iron, cobalt, and nickel). Maroufi et al. used waste tire rubber-derived carbon as the reducing agent, in an attempt to make optimal use of this waste source [20], whereas Bian et al. employed conventional carbon reduction [21]. In these two studies, no details are disclosed about the method by which the REOs and the metal phases were separated. A different approach was employed by Yoon et al., who first converted neodymium into $Nd(OH)_3$ by grinding Nd–Fe–B scrap in the presence of NaOH and, subsequently, converted the $Nd(OH)_3$ into Nd_2O_3 by roasting in air at 400 °C [22].

In the present paper, a process for pyrometallurgical separation of REEs from iron by roasting milled Nd–Fe–B scrap in argon atmosphere at $p_{O_2} \leq 10^{-20}$ atm with 5 wt% carbon addition for 2 h at 1400 °C was carried out, showing that a first step of oxidation is not necessary to separate iron from the rare-earth elements and to obtain Nd_2O_3 instead of $NdFeO_3$. Furthermore, the positive effect of the improved roasting process on the leaching rate of neodymium was corroborated by leaching the roasted Nd–Fe–B scrap in the water-containing functionalized ionic liquid betainium bis(trifluoromethylsulfonyl)imide, [Hbet][Tf₂N] (Fig. 1). This ionic liquid has been found useful for recovery of metals from urban waste and industrial process residues [12, 17, 23–25]. The roasting in reductive atmosphere rather than in air can be seen as a form of process intensification since, in comparison to previous studies, the total duration of the roasting and leaching steps is reduced more than tenfold. Eliminating the oxidative roasting step and intensifying the leaching reduced the environmental footprint of the designed flowsheet for the recycling of Nd–Fe–B permanent magnets.

Fig. 1 Structure of the ionic liquid betainium bis(trifluoromethylsulfonyl)imide, [Hbet][Tf₂N]

Experimental

Materials

Bulk, demagnetized Nd–Fe–B permanent magnets scrap were provided by Magneti Ljubljana d. d. (Ljubljana, Slovenia). The fluxing agent, calcium carbonate, $CaCO_3$ (> 99%), and betainium chloride [Hbet]Cl (> 99%) were supplied by Acros Organics (Geel, Belgium), while lithium bis(trifluoromethylsulfonyl)imide $Li[Tf_2N]$ (> 99%) was obtained from Iolitec (Heilbronn, Germany). Carbon was purchased as a high density skeleton activated charcoal (≥ 99.8) from Sigma-Aldrich (Diegem, Belgium). Iron, neodymium, and dysprosium standard solutions ($1000 \mu\text{g mL}^{-1}$) were obtained from Chem-Lab (Zedelgem, Belgium). The EpoFix epoxy resin was purchased from Struers GmbH Netherlands (Massluis, Netherlands).

Solid Pretreatment and Roasting

The magnets were crushed to a size ≤ 5.0 mm by a hydraulic press and, subsequently, milled in a planetary ball mill (Fritsch Pulverisette 7 premium). The following milling conditions were applied: stainless steel pots and balls; milling medium: ethanol (97% grade); charge ratio balls-to-powder: 4 g g^{-1} ; balls diameter: 10.0 and 1.0 mm; milling time: 5 min; pause time: 5 min; cycles (milling + pause): 3. The milling pots were cooled down to room temperature before opening, to avoid spontaneous ignition of the hot milled pyrophoric powder, and the content was transferred to a rotary evaporator to remove the ethanol. The dried powder was stored in argon atmosphere to prevent oxygen capture by the powder itself. The procedure for the elemental analysis of the solid by inductively coupled plasma optical emission spectrometer (ICP-OES, PerkinElmer Optima 8300) has already been described in previous work [12]. Briefly: 50 mg of milled, non-roasted of Nd–Fe–B permanent magnets scrap was dissolved in 10 mL of 37% HCl at 70 °C for 60 min. Afterwards, the solutions were diluted 100 times in 2% v/v HNO_3 . Three calibration standards were prepared containing 0.5–5–50 mg L^{-1} Fe, Nd, B, Dy, Pr, Co, Cu, Ga, and Al in 2% v/v HNO_3 . Rh was selected as internal standard for Fe, Nd, and Dy; Sr was selected as internal standard for B, Pr, Co, Cu, Ga, and Al.

About 0.5 g (for the screening tests of the roasting conditions) or 2.0 g (for the consequent leaching tests) of milled Nd–Fe–B powder was mixed with 5 wt% of reagent grade carbon in a magnesia crucible (outer diameter: 30.0 mm, height: 30.0 mm). The roasting experiments were carried out in a vertical tube furnace (impervious

recrystallized alumina, 80 mm internal diameter, GERO HTRV 100-250/18) with MoSi₂ heating elements, under purified argon atmosphere (flowrate: 0.3 L min⁻¹, $p_{O_2} \leq 10^{-20}$ atm). A heating rate of 5 °C min⁻¹ was applied to reach the targeted temperature, while a cooling rate of 5 °C min⁻¹ was utilized at the end of the experiment to cool the sample to room temperature, when quenching in water was not applied. The investigated temperatures were 1200 °C, 1400 °C, and 1600 °C. The holding times at the desired temperatures were 2.0, 4.0, and 9.0 h. In selected experiments, CaCO₃ was added as a flux in 50 wt% concentration. The calcine was ground to powder by using a hammer and/or a mortar and pestle. Electron-probe micro-analysis (EPMA) with five wavelength dispersive spectrometers (JEOL JXA-8530F, JEOL Ltd.) was carried out employing a probe current of 50 nA and an acceleration voltage of 15 kV. Prior to the analysis, the samples were embedded in epoxy resin (EpoFix), following by 1 day of curing under vacuum. Conventional grinding and polishing techniques were used. The polished samples were coated with an approximately 1.0 nm thick platinum film.

Leaching

A first series of leaching experiments was performed with 0.075 g of ground calcine in the ionic liquid betainium bis(trifluoromethylsulfonyl)imide, [Hbet][Tf₂N], with addition of 5 or 10 wt% water, until complete dissolution of the solid was observed. The ionic liquid was prepared starting from [Hbet]Cl and Li[Tf₂N], as described in the literature [26]. The [Hbet][Tf₂N] was dried on a rotary evaporator for 3 h and subsequently transferred overnight on a Schlenk line to remove the remaining traces of water. After drying and solidifying the ionic liquid (at room temperature), 5 wt% or 10 wt% of water was added to it. Leaching conditions were temperature: 80 °C; lixiviant/solid (L/S) ratio: 20 mL g⁻¹; mixing speed: 600 rpm. Magnetic stirring bars in polytetrafluoroethylene were used to mix the leaching solutions. The leaching temperature was set at 80 °C to compare the results obtained in this work with results previously obtained in our group. The leaching selectivity S was calculated using Eq. (1):

$$S = \frac{\%E_{\text{REEs}}}{\%E_{\text{Fe}}}, \quad (1)$$

where $\%E_{\text{REEs}}$ and $\%E_{\text{Fe}}$ are the percentage of extraction, respectively, of rare-earth elements (neodymium + dysprosium) and of iron. The $\%E$ of any metal M is calculated as in Eq. (2):

$$\%E_M : \frac{[M]_f}{[M]_{\text{MAX}}}, \quad (2)$$

where $[M]_f$ is the metal concentration in the leachate, while $[M]_{\text{MAX}}$ is the maximum metal concentration in the leachate and corresponds to a total dissolution of the metal from the solid to the lixiviant. The best performing ionic liquid from this first test was further used to study the dissolution rate of the REEs. Temperature, L/S ratio, and mixing speed were kept the same as in the first leaching tests; uptakes were made of 50.0 µL of leachate at 5, 10, 20 min and of 20.0 µL at 30, 45, 60 min. from an initial volume of 10.0 mL of lixiviant. All the leaching samples were diluted 100 times in 2% v/v HNO₃ and analyzed via inductively coupled plasma optical emission spectrometry (ICP-OES) on a PerkinElmer Optima 8300 spectrometer. Calibration solutions were prepared with standards of iron, neodymium, and dysprosium in a concentration range of 0.1 to 25 mg L⁻¹; a quality control solution of 1 mg L⁻¹ was used as well. The same experimental and analytical conditions were applied for, respectively, the 5 wt% and 10 wt% water-containing [Hbet][Tf₂N] samples.

Results and Discussion

Solid Pretreatment and Roasting

The elemental composition is reported in Table 1, the Nd–Fe–B scrap has a dysprosium content, currently the most critical REE, higher (circa 8.5 wt%) than the usual dysprosium content in Nd–Fe–B permanent magnets (3–5 wt%). This confirms the value of the Nd–Fe–B scrap as a secondary REEs source. The elemental distribution of the milled, not-roasted Nd–Fe–B scrap is displayed in Fig. 2. From this analysis, the presence of a mixed iron–rare earths phase and of a rare-earth oxides phase are confirmed. The first roasting experiments were carried out on 0.5 g of milled Nd–Fe–B scrap with 0.025 g of carbon powder at 1400 °C for 2.0, 4.0, and 9.0 h. The reaction was stopped for all the three tests by quenching. It was assumed that an amount of carbon equal to 5 wt%

Table 1 Elemental composition of the Nd–Fe–B permanent magnet scrap (in wt%) as determined with ICP-OES

Element	Composition* (wt%)
Fe	62.52
Nd	23.40
Dy	8.37
Pr	0.61
Co	3.08
Cu	0.15
Ga	0.21
B	1.00

Fig. 2 BSE (top left) and EPMA element mapping of the sample before roasting. The scale is in color intensity, but differs from element to element (Color figure online)

was sufficient for the smelting process of iron. Any excess of carbon would remain unreacted, without influence on the metals, whereas an insufficient amount would lead to the formation of γ -Fe which is not liquid at the operating temperature [27]. The added carbon has two simultaneous effects on the iron–rare earths separation at the investigated working temperatures: (1) it forms a eutectic with iron, which can form a liquid phase and flow within the sample; (2) it reduces the iron, which may have been

partially oxidized to Fe_2O_3 (according to the reactions in Eqs. (3–6)), but not the rare earths [28].

The elemental distribution in the phases after 2, 4, and 9 h of thermal treatment, followed by quenching, was investigated by electron-probe microanalysis (EPMA). The results are shown in Figs. 3, 4, and 5 with back-scattered electron (BSE) images of the samples.

Two main observations emerged from the EPMA mapping, namely (1) the rare-earth elements are well separated from the iron, also after only 2 h of holding time (Fig. 3) contrarily to what happens in the not-roasted sample (Fig. 2), and (2) iron was in metallic form while the REEs formed a phase with boron. The present results are not in perfect agreement with the modeling performed by Tranell and Jung [13]. In fact, the authors suggested a phase of liquid metal with Nd_2O_3 as a solid phase; moreover, according to them, boron would form an Fe–B solution. However, Tranell and Jung carried out a qualitative study without quantifying how much Nd_2O_3 is present in the system after roasting at 1400°C at $p_{\text{O}_2} \leq 10^{-15}$ atm and a $p_{\text{CO}} \leq 10^{-1}$ atm. Due to the very low content of oxygen in the roasting atmosphere, it is very likely that such Nd_2O_3 does not form during the

roasting process but represents the Nd_2O_3 already present in the Nd–Fe–B scrap (*circa* 5 wt%). Contamination of the Nd–Fe–B scrap powder by oxygen during the operations of crushing and milling cannot be excluded either. Nevertheless, the scope of the roasting was to separate the neodymium from the iron and this was successfully achieved. It is important to clarify that we largely omit dysprosium in the discussion on the ternary oxides, since the diffractogram of the roasted Nd–Fe–B scrap in air only revealed the presence of NdFeO_3 and not of any dysprosium compound. Furthermore, the stability of LnFeO_3 has been reported to decrease, at high temperatures, with decreasing ionic radius and, thus, the presence of DyFeO_3 can be excluded [29].

Nevertheless, it was not possible to separate the metallic iron from the REEs phase, neither magnetically nor mechanically. The magnetic separation was tested first on the roasted sample just after crushing and, subsequently, on the same powder dispersed in ethanol, instead of water, to avoid iron hydrolysis. In both cases, the entire specimen (the metallic iron together with the REEs phase) was attracted to the magnet. The reason is that the iron phase and the REEs phase have not been liberated, and the fineness of

Fig. 3 BSE (top left) and EPMA element mapping of the sample roasted with 5 wt% C at 1400°C for 2 h in an argon atmosphere with $p_{\text{O}_2} \leq 10^{-20}$ atm. The scale is in color intensity, but differs from element to element (Color figure online)

Fig. 4 BSE (top left) and EPMA element mapping of the sample roasted with 5 wt% C at 1400 °C for 4 h in an argon atmosphere with $p_{O_2} \leq 10^{-20}$ atm. The scale is in color intensity, but differs from element to element (Color figure online)

the crushed roasted solid also affects the separation [30]. An example is shown in the back-scattered image (Fig. 3), where the iron phase is encapsulated by the REEs phase. Firdaus et al. also stated that the separation of metal from slag is a major challenge in high-temperature processes for the REEs recovery from Nd–Fe–B permanent magnets [31]. Borra et al. were, instead, able to collect and easily separate the iron in the form of a nugget at the bottom of a bauxite residue sample after reductive smelting [32]. A flux was used in that work, but the main difference is the composition of the two samples. In case of Nd–Fe–B scrap, the

rare-earth elements are about 33 wt% of the sample and their atomic weight is, on average, three times the atomic weight of the iron. On the contrary, bauxite residue comprises only 0.1 wt% of rare-earth elements and about 30 wt% of the solid is made of Al_2O_3 and CaO , which are about one and half and three times lighter than Fe_2O_3 , respectively. Hence, it is difficult to obtain an easy-to-remove iron nugget at the bottom of selectively roasted Nd–Fe–B scrap. Different parameters (addition of a flux, roasting temperature, and cooling mode) were investigated to promote the growth of iron agglomerates to collect either at the bottom or dispersed

Fig. 5 BSE (top left) and EPMA mapping of the sample roasted with 5 wt% C at 1400 °C for 9 h in an argon atmosphere with $p_{O_2} \leq 10^{-20}$ atm. The scale is in color intensity, but differs from element to element (Color figure online)

in the sample. CaCO_3 was used as a flux, added in 50 wt% to the Nd–Fe–B scrap. As CaCO_3 decomposes to CaO and CO_2 at temperatures below the operational temperatures in this study, the actual flux is CaO. In this case, a significant amount of oxygen is introduced into the system. The experiment was carried out at 1400 °C for 2 h, and quenching was once more applied. The sample R2 as it appeared at the end of the roasting, is shown in Table 2, while the EPMA

is presented in Fig. 6. The distribution of oxygen actually shows a band that is typical for when a large area is investigated. It is an artifact of the machine/spectrometer curvature. Nonetheless, oxygen is not present in the iron phase.

Although, the CaCO_3 flux did not affect the settling behavior of the REEs phase from the metallic iron, boron preferentially reported to the Ca–O phase, indicating that boron can be separated from both the rare-earth elements

Table 2 Summary of the experimental conditions for the optimization of the roasting parameters with photos of the corresponding samples at the end of test

Sample	R2	R3	R4	R5
Appearance at the end of the roasting test
Temperature (°C)	1400	1600	1400	1600
C (wt%)	5	5	5	5
CaCO ₃ flux (wt%)	50	50	None	None
Quenching	Yes	No	No	No
Holding time (h)	2	2	2	2
Cooling rate (°C min ⁻¹)	–	5	5	5

All the samples were roasted in argon atmosphere with $p_{O_2} \leq 10^{-20}$ atm

and iron. Both the B–Ca–O phase and the REEs phase did not contain iron. The effect of the cooling mode on the growth of iron agglomerates was investigated, by replacing the quenching with a slow cooling at -5 °C min⁻¹ in the furnace under inert atmosphere. A flux was again used. The resulting sample R3 is shown in Table 2. Small iron spheres were observed for the first time: it is assumed that the slow cooling provides sufficient time for the metallic iron to migrate within the sample and uniformly grow to iron agglomerates. It was decided to continue the roasting study without adding a flux, since the flux does not mix with the REEs phase to increase the settling of the iron phase and since, even worse, the calcium contaminates the leachate of the subsequent leaching step. Rather than a calcium-based fluxing agent, the use of a fluorine- or boron-containing fluxing agent could decrease the melting point of the slag, but this was not investigated further. In the next experiment, about 0.5 g of Nd–Fe–B scrap was roasted at 1400 °C for 2 h, without flux and without quenching (R4, in Table 2). Finally, iron spheres of about 1.0–3.0 mm diameter were obtained, which could be mechanically removed with ease, by sieving after crushing the specimen.

The effect of the roasting temperature on the iron behavior was evaluated as well. An experiment was performed with the same conditions of the R4 test, albeit with a roasting temperature of 1600 °C. It was expected that a higher roasting temperature would decrease the viscosity of the melt, allowing the iron particles to migrate more freely and to agglomerate at a higher extent in the melt. The resulting sample R5 is shown in Table 2. Although iron pearls were obtained, a higher temperature is not favorable to the process: when the liquid sample solidified, the sample ended up to be very hard and the slag solidified as a thin coating

on the crucible and on the iron particles. Therefore, the slag can neither be separated from the iron particles, nor from the magnesia crucible. In the previous experiments (pictures R2, R3, and R4 in Table 2), the samples had the appearance of a sintered powder, meaning that a liquid melt did not really form. At this stage, in the perspective of the absence of NdFeO₃ formation and a clear microscopic separation of the iron particles from the REEs phase, the optimum conditions for roasting were defined as 5 wt% carbon, no flux, no quenching, and a roasting temperature of 1400 °C of 2 h.

Leaching

Once the pyrometallurgical step was optimized, the effect of the separation of the rare-earth elements phase from the iron phase on the leaching was investigated. The leaching was carried out in the [Hbet][Tf₂N] ionic liquid and the yields as well as the rates of the process were investigated. Previous work by the present authors has already shown that the leaching efficiency of roasted Nd–Fe–B scrap is affected by (1) the water content in the ionic liquid, since water saturates the first coordination sphere of the metals, and (2) the presence of the ternary oxide NdFeO₃ [12]. About 2.0 g of powder was successfully roasted under optimal conditions (no flux and no quenching, a roasting temperature of 1400 °C and roasting time of 2 h). Most of the iron was removed manually, resulting in variations of iron content in samples from different experiments and affecting the calculated selectivity. 5 wt% or 10 wt% water content was added to the [Hbet][Tf₂N]. It is known that water-containing ionic liquid is a much better solvent for metal oxides than the dry ionic liquid [26]. Most of the solid was dissolved after only 60 min in [Hbet][Tf₂N]

Fig. 6 BSE (top left) and EPMA element mapping of the sample roasted with 5 wt% C and 50 wt% CaCO_3 at 1400 °C for 2 h in argon atmosphere with $p_{\text{O}_2} \leq 10^{-20}$ atm. The scale is in color intensity, but differs from element to element (Color figure online)

containing 10 wt% of water and about 90 min in [Hbet][Tf_2N] containing 5 wt% of water. The residue left behind in the leachate was undissolved metallic iron. The results of the leaching tests, in terms of selectivity S (Eq. 1), are summarized in Table 3. The selectivity S is approximately 50 for both ionic liquid samples. Furthermore, the water content in the ionic liquid has a significant influence on the leaching rate of the Nd–Fe–B permanent magnets in [Hbet][Tf_2N], which is in agreement with previous observations [12]. The ionic liquid [Hbet][Tf_2N] with 10 wt%

Table 3 Selectivity S of Nd–Fe–B scrap leaching in 5 wt% [Hbet][Tf_2N] and 10 wt% [Hbet][Tf_2N]

Ionic liquid	Water content	$([\text{REEs}]/[\text{Fe}])_{\text{f}}$	S	RSD (%)
[Hbet][Tf_2N]	5 wt%	17.3	46.7	2.8
[Hbet][Tf_2N]	10 wt%	20.1	54.3	1.9

The Nd–Fe–B scrap was roasted with 5 wt% C for 4 h at 1400 °C in argon atmosphere with $p_{\text{O}_2} \leq 10^{-20}$ atm and cooled to room temperature at 5 °C min^{-1}

Leaching conditions: $T = 80$ °C, $t = 60$ min, $r = 600$ rpm, $L/S = 20$ mL g^{-1}

Fig. 7 %E of Nd (filled triangle), Dy (filled square) and Fe (filled circle), after manual separation and after leaching, as a function of the time t , in (min). Leaching conditions: lixiviant: 10 wt% [Hbet][Tf₂N]; $T=80\text{ }^{\circ}\text{C}$, $t=60\text{ min}$, $r=600\text{ rpm}$, $L/S=20\text{ mL g}^{-1}$

water was selected as the most favorable solvent system for further studies.

Next, the leaching rates of neodymium, dysprosium, and iron from the roasted sample were investigated by determining the metal concentrations in the ionic liquid as a function of time (Fig. 7).

The parameter %E was calculated based on the initial amount of metal in the Nd–Fe–B scrap and, thus, it includes the results of both the roasting-manual separation step and of the leaching. A more appropriate definition of %E would be based on the actual material processed in the leaching, after the manual separation of iron. Nevertheless, the unavoidable losses of material in this former step made impossible to identify the actually leached material. On the other hand, the results so presented in Fig. 7 give a global overview of the performance of the recovery process. Nevertheless, the value of S after 20 min was 15.8, i.e., about three times lower than the values in Table 3, showing how crucial the step of mechanical iron removal is. Such step can be more easily optimized at industrial scale for larger volumes with an automated system for the mechanical treatment (crushing, sieving, magnetic or gravity separation) of the roasted product, compared to the process in the lab where the operations of crushing the roasted specimen and removing the iron were achieved manually. Furthermore, iron recovery is not the objective of this work and the sample losses during the mechanical removal of iron were not minimized.

The concentration of the REEs stabilized after 20 min, indicating that this period of time is sufficient to fully dissolve the REEs phase. The leaching rates obtained in this work on the roasted sample are significantly higher

than the leaching rates obtained in previous research [17]. Dupont and Binnemans were able to leach Nd–Fe–B permanent magnets in water-containing [Hbet][Tf₂N], but a complete dissolution of 6 μm particles took at least 24 h when using 10 wt% [Hbet][Tf₂N] and 72 h in 5 wt% [Hbet][Tf₂N] [17]. In that work, the permanent magnets were pretreated by roasting at 950 $^{\circ}\text{C}$ in air for 15 h, resulting in an optimal total treatment time of 39 h, compared to the 2.3 h treatment time required in this work. The improved leaching rate is explained by the absence of any mixed neodymium-iron phase due to the roasting in argon atmosphere at $p_{\text{O}_2} \leq 10^{-20}\text{ atm}$ with 5 wt% carbon, whereas roasting in air results in the conversion of neodymium mainly as ternary oxide NdFeO₃. In particular, the NdFeO₃ content of the sample Nd–Fe–B magnets roasted in air is about 25 wt%, and the Nd₂O₃ content is only 0.44 wt% [12]. The recovery of the rare-earth elements from the ionic liquid leachate was outside the scope of this work and can be done, according to methods previously published, by precipitation stripping of the REEs with an aqueous H₂C₂O₄ solution [17, 33]. A suggested flowsheet is presented in Fig. 8: the green lines represent the solvometallurgical steps, while the red lines are pyrometallurgical steps and

Fig. 8 Conceptual flowsheet for the combined pyrometallurgical and solvometallurgical recycling of Nd–Fe–B scrap. The green lines represent the solvometallurgical steps, the red lines the pyrometallurgical steps, and the blue lines the hydrometallurgical steps. The dashed lines represent steps which have not been tested but assumed from the literature (Color figure online)

the blue lines are hydrometallurgical steps. Moreover, dashed lines represent steps which have not been tested, but assumed from the literature: in particular, the 10 wt% [Hbet][Tf₂N] ionic liquid is assumed to be reused at the end of the process for many times since the contact with aqueous H₂C₂O₄ does not degrade it. The limited input of chemicals in the flowsheet is in agreement with the principles of Green Chemistry [34]. The flowsheet was simplified, considering only the main elements iron, neodymium, and dysprosium in the process; as for other elements, cobalt would be separated together with iron, with which it forms a phase (see Figs. 3, 4, and 5), boron and praseodymium would follow the route of the REEs, traces elements (copper, gallium) were not considered.

Conclusion

The leaching rate of Nd–Fe–B permanent magnets has been observed to be affected by the formation of the ternary oxide NdFeO₃. Roasting of Nd–Fe–B scrap in argon atmosphere at $p_{O_2} \leq 10^{-20}$ atm with 5 wt% carbon for 2 h at 1400 °C resulted in two phases, i.e., a metallic iron phase and a non-metallic B–Dy–Nd phase. The targeted separation of neodymium from iron in the roasting step was successfully achieved. The iron phase neatly separated from the REEs phase without the need of distinctly carrying out an oxidation and a reduction step. The most favorable roasting conditions were 5 wt% carbon; no flux; no cooling by quenching, a roasting temperature of 1400 °C and a roasting time of 2 h. The beneficial effect of the optimized roasting was corroborated by leaching with the water-containing ionic liquid [Hbet][Tf₂N]. The total process duration (roasting + leaching) was 17 times shorter compared to the optimum conditions of previous work in our group (2.3 h instead of 39 h) [17]. In conclusion, a proof-of-concept was given that the recycling of Nd–Fe–B permanent magnets can be made more sustainable by optimizing their pretreatment (roasting): in fact, in only one roasting step the iron is reduced, the formation of NdFeO₃ is avoided, and, hence, the consequent leaching process is intensified.

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Compliance with Ethical Standards

Conflict of interest On behalf of all authors, the corresponding author states that there is no conflict of interest.

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