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### ARTICLE

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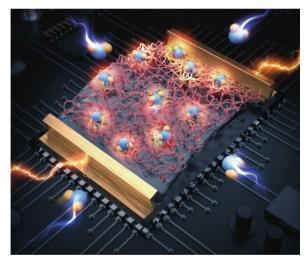
#### HIGHLIGHTS

## A Novel Artificial Neuron-Like Gas Sensor Constructed from CuS Quantum Dots/Bi<sub>2</sub>S<sub>3</sub> Nanosheets

Xinwei Chen<sup>1</sup>, Tao Wang<sup>1</sup>, Jia Shi<sup>1</sup>, Wen Lv<sup>1</sup>, Yutong Han<sup>1</sup>, Min Zeng<sup>1</sup>, Jianhua Yang<sup>1</sup>, Nantao Hu<sup>1</sup>, Yanjie Su<sup>1</sup>, Hao Wei<sup>1</sup>, Zhihua Zhou<sup>1</sup>, Zhi Yang<sup>1</sup>  $\bowtie$ , Yafei Zhang<sup>1</sup>  $\bowtie$ 

- An ultra-sensitive capture of NO<sub>2</sub> molecules and fast charge collection and transfer has been realized by constructing the model of artificial neuron-likegas sensing structure based on CuS quantum dots (QDs)/Bi<sub>2</sub>S<sub>3</sub> nanosheets (NSs)realizes.
- Simulation analysis revealed that CuS QDs and Bi<sub>2</sub>S<sub>3</sub>NSs can be used, respectively, as the main adsorption sites and charge transport pathways, thus leading to a greatly enhanced gas capture ability and charge conduction performance of NO<sub>2</sub>.

**ABSTRACT** Real-time rapid detection of toxic gases at room temperature is particularly important for public health and environmental monitoring. Gas sensors based on conventional bulk materials often suffer from their poor surface-sensitive sites, leading to a very low gas adsorption ability. Moreover, the charge transportation efficiency is usually inhibited by the low defect density of surface-sensitive area than that in the interior. In this work, a gas sensing structure model based on CuS quantum dots/Bi<sub>2</sub>S<sub>3</sub> nanosheets (CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs) inspired by artificial neuron network is constructed. Simulation analysis by density functional calculation revealed that CuS QDs and Bi<sub>2</sub>S<sub>3</sub> NSs can be used as the main adsorption sites and charge transport pathways, respectively. Thus, the high-sensitivity sensing of NO<sub>2</sub> can be realized by designing the artificial neuron-like sensor. The experimental results showed that the CuS QDs with a size of about 8 nm are highly adsorbable, which can enhance the NO<sub>2</sub> sensitivity due to the



rich sensitive sites and quantum size effect. The  $Bi_2S_3$  NSs can be used as a charge transfer network channel to achieve efficient charge collection and transmission. The neuron-like sensor that simulates biological smell shows a significantly enhanced response value (3.4), excellent responsiveness (18 s) and recovery rate (338 s), low theoretical detection limit of 78 ppb, and excellent selectivity for NO<sub>2</sub>. Furthermore, the developed wearable device can also realize the visual detection of NO<sub>2</sub> through real-time signal changes.

KEYWORDS Artificial neuron-like gas sensor; Heterostructure design; Nitrogen dioxide detection; Wearable device

Zhi Yang, zhiyang@sjtu.edu.cn; Yafei Zhang, yfzhang@sjtu.edu.cn

<sup>1</sup> Key Laboratory of Thin Film and Microfabrication (Ministry of Education), Department of Micro/Nano Electronics, School of Electronic Information and Electrical Engineering, Shanghai Jiao Tong University, Shanghai 200240, People's Republic of China

#### **1** Introduction

Over the last decades, the gradual development of industrialization and urbanization leads to the deterioration of environmental quality, especially the air condition [1, 2]. In order to cope with the timely detection of toxic and hazardous gases and to prevent the related diagnosis of respiratory injuries and diseases, the active development of low-cost, durable, high-performance gas sensors has become the focus of current research [3, 4]. With the development of the "Internet of Things" and artificial neural networks, the design of wearable gas sensors provides a promising strategy to broaden the practical applications by combining real-time monitoring and big data analysis, which allows the device applicable to the early detection of any health hazards [5-8]. Therefore, exploring wearable devices with improved room temperature detection ability ranks as the key to expand the field of gas sensing.

Similarities between biological structures and artificial neural-like networks have inspired the development of neuron-based mechanical sensing and optical sensing [9-11]. The olfactory neurons that exist in the olfactory bulb of higher animals play a key role in extracting olfactory signals through gas sensing [12]. When the human body's olfactory receptors receive a certain intensity of stimulation, it will cause changes in the cell membrane surface potential to trigger action potentials, convert chemical signals into electrical signals, and conduct signal analysis along the nerve fibers to the central system. A large number of olfactory sensory neurons and rapid potential changes are essential for high spiritual sensitivity. Inspired by this, it is very meaningful to construct an artificial neuron-like sensor model that is different from traditional bulk sensitive materials and contains a large number of adsorption sites and fast charge transport channels.

Among active sensing materials for artificial neurons, binary or heterogeneous structures constructed from twodimensional (2D) materials can achieve rapid response, which is of great significance for room temperature wearable sensing in practical applications. Bismuth sulfide ( $Bi_2S_3$ ) with the direct bandgap of 1.3 eV, high carrier mobility, and excellent environment-friendly characteristics received potential attention in sensing applications [13, 14]. As a member of the stibnite family, the layered structure of  $Bi_2S_3$ is projected along the *b*-axis direction by atomic-scale ribbons, which are fixed together by van der Waals (vdW) forces. Bulk Bi<sub>2</sub>S<sub>3</sub> can be separated into two-dimensional (2D) layer structures along the (010) crystal facets maintained by weak vdW forces [15, 16]. Unfortunately, like other similar 2D gas-sensitive materials, 2D-layered Bi<sub>2</sub>S<sub>3</sub> exhibits poor recovery properties at room temperature, which severely limits its widespread applications [17, 18]. In addition, recent studies have also proved that CuS as a p-type semiconductor material can be used alone or as a second-phase modifier to achieve rapid detection of trace gases at room temperature [19, 20]. Therefore, modifying the surface of 2D Bi<sub>2</sub>S<sub>3</sub> nanosheets (NSs)with OD CuS quantum dots (QDs) can realize the scattered construction of sensitive points on the neuron-like network and develop a neuron-like olfactory system that is different from the traditional inefficient adsorption detection-type sensors. This will effectively help to achieve rapid and sensitive detection at room temperature.

Herein, we first report a sensor model inspired by the biological olfactory neuron system, which realizes the efficient detection of NO<sub>2</sub> by uniformly loading CuS QDs on Bi<sub>2</sub>S<sub>3</sub> NSs. Density functional theory (DFT) reveals the charge distribution and transfer process of 0D-2D heterojunction neurons before and after contacting gas molecules. As a result, the 0D p-type CuS quantum dots with higher adsorption energy for NO<sub>2</sub> molecules are confirmed to be the main adsorption sites for the target gas, and the 2D n-type  $Bi_2S_3$ nanosheets provide the main transfer path for the charge carriers. Subsequent experiments confirmed that appropriately high-tuned 0D QDs were loaded on the carrier as secondphase particles. Attribute to the numerous active adsorption sites and a large amount of charge transfer, interface electronic interaction can be greatly enhanced, which facilitates the action of gas molecules on the material surface [21, 22]. Moreover, the design of this n-p heterojunction can effectively increase the rate of charge transfer and has realized the ultra-sensitive response of artificial neurons to the stimulation of gas molecules. Therefore, this study uses the synergistic effect between high adsorption energy and fast charge transfer to simulate the biological olfactory detection process by rationally constructing heterojunction neurons and further realizes the sensitive monitoring of NO<sub>2</sub>. Finally, the sensing materials have been further integrated into our self-developed wearable gas detection equipment. The realtime display and cloud storage of detection signals on smart device applications through data collection and Bluetooth wireless transmission has been fully achieved. Our work sheds light on the rational design of bio-simulation intelligence, wearable, and wireless sensing equipment toward the "neuron-like sensitive intelligence era."

#### **2** Experimental Section

#### 2.1 Synthesis of Bi<sub>2</sub>S<sub>3</sub> NSs

30 mg of commercial bulk  $Bi_2S_3$  powder was added to a 25 mL flask. 10 mL of aqueous ethanol in various volume ratios was added as a dispersion solvent. Ultrasonication was performed for 8 h. Then, the precipitate was removed by centrifugation treatment.

100 mg of  $Bi_2S_3$  powder and acetonitrile was ground with agate mortar for 2 h [23]. The samples were dried with a vacuum oven and then redistribute in a ethanol aqueous solution (70 vol%, 50 mL, Fig. S1). After ultrasonic treatment at 200 W for 3 h, the supernatant was extracted by centrifugation at 1500 rpm for 20 min to collect  $Bi_2S_3$  NSs [17, 24].

#### 2.2 Synthesis of CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs and CuS

0.0138 g of Cu(NO<sub>3</sub>)<sub>2</sub> and 0.01 g of Na<sub>2</sub>S•9H<sub>2</sub>O were dissolved in 20 mL of ethanol aqueous solution, respectively. Then, 0.1 g Bi<sub>2</sub>S<sub>3</sub> NSs were dispersed in 10 mL ethanol and stirred continuously with an ultrasonic instrument for 30 min. The dispersion was mixed with the above two solutions, respectively, for reaction, and finally, the prepared CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs were collected after centrifugation and washing [22]. Following the same procedure as above, bare CuS was obtained with the addition of only Cu(NO<sub>3</sub>)<sub>2</sub> and Na<sub>2</sub>S•9H<sub>2</sub>O. The as-prepared products were named as BC-2.5, BC-3, BC-4, BC-5, BC-6, BC-7.5, BC-10, and BC-20 corresponding to 2.5, 3, 4, 5, 6, 7.5, 10, and 20 wt% of CuS, respectively.

#### 2.3 Sensor Device Fabrication and Sensing Measurements

The sensor device uses a flexible electrode sputtered with Au fingers on polyimide (PI) substrate. Clean the electrode surface alternately with deionized water and ethanol to promote the contact between the sensing membrane and the electrode. To fabricate the gas sensor, the ethanol dispersion of CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs with a concentration of 70 vol% was sprayed onto the PI electrode using inkjet printing equipment. Finally, the as-prepared sensors were dried in the vacuum oven.

The details of the gas-sensitive test system can be found in our previous paper [25]. In simple terms, the chamber was purged with dry compressed air before the measurement to stabilize the baseline signal. Subsequently, the mass flow controller (MFC) was used to control the flow of NO<sub>2</sub> and dry compressed air with an initial concentration to obtain NO<sub>2</sub> dilution gas of different concentrations with a flow rate of 1 standard liter/min (SLM). The Agilent 4156C analyzer was utilized to detect current changes of gas sensors at room temperature (25 °C). The response value is defined as  $R_a/R_g$ , which are resistances of the sensor when exposed to compressed air and NO<sub>2</sub>, respectively. In addition, the response and recovery time ( $\tau_{res}$  and  $\tau_{rec}$ ) are the time required for the sensor to fully respond and recover to 90% after being exposed to NO<sub>2</sub>.

#### 2.4 Characterizations

Crystallinity and morphology of as-prepared CuS QDs/ Bi<sub>2</sub>S<sub>3</sub> NSs sample were analyzed with the X-ray diffractometry (XRD, D8 ADVANCE, Bruker), transmission electron microscopy (TEM, JEM-2100, JEOL, Japan), and scanning electron microscopy (SEM, Carl Zeiss Ultra Plus, Germany). The chemical components and band structures were studied by X-ray photoelectron spectroscopy and ultraviolet photoelectron spectroscopy (XPS and UPS, Kratos Axis Ultra<sup>DLD</sup>). Ultraviolet–visible (UV–vis) absorption spectra (Perkin-Elmer, USA) were utilized to characterize bandgaps of materials. The Raman spectra were acquired on the confocal Raman microscope (RENISHAW, England). Mott–Schottky plots measurements were performed with an electrochemical workstation (CHI 760E, Shanghai Chenhua).

#### **3 Results and Discussion**

#### 3.1 DFT Calculation of Neuron-like Sensor Network Inspired by Olfactory Sensory Neurons

It is of great significance to deeply understand the perception process of the olfactory organs of organisms to odor molecules and to compare the mechanism of artificial neuron gas sensors with it. As shown in Fig. 1a, when the human body's olfactory receptors receive a certain intensity of stimulation, it will cause changes in the surface potential of the cell membrane to trigger action potentials, convert chemical signals into electrical signals, and conduct analysis along nerve fibers to the central system and generate corresponding reaction [12]. A large number of sensory neurons in the biological olfactory system can realize super-sensitivity to gas molecules, and the timely conduction of electric potentials realizes the rapid transmission of signals. Inspired by this, the CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs-based neuron-like sensing model constructed by loading the 0D adsorption sites on the 2D semiconductor transmission channel has great potential to realize highly sensitive sensing of NO<sub>2</sub> molecules.

In order to reveal the role of artificial olfactory neuron sensing interface in the process of NO<sub>2</sub> sensing, DFT was used to analyze the adsorption and charge transfer processes of NO<sub>2</sub>. Four binding structures of NO<sub>2</sub> on CuS-Bi<sub>2</sub>S<sub>3</sub> are studied in our work, and the adsorption conformation is shown in Fig. 1d-g. The corresponding binding energies  $(E_b = E_{NO_2} + E_{CuO-Bi_2S_3} - E_{total})$  are 1.15, 1.70, 1.45, and 2.25 eV. The results showed that NO<sub>2</sub> prefers to bind to CuS-Bi<sub>2</sub>S<sub>3</sub> with forming N-Cu bonds, and the type of Bi-O binding structure has the smallest binding energy, and the length of M(Bi/Cu)-O bond is listed in Table S1. To explain the difference of binding property, charge density difference, Bader charge transfer, and M–O bond length are analyzed. Clearly, the binding structure with Cu–O bond structure is favorable with the highest binding. As displayed in Fig. 1b, c and Table S2, for Bi-O structure, the charge is mainly transferred between Bi and O atoms, while for Cu-O, it is between Cu and O atoms. Bader charge analysis shows that CuO and NO<sub>2</sub> have greater charge transfer, which may be beneficial to the binding of  $NO_2$ , which is consistent with the relationship between adsorption energy. This series of DFT results proved the conclusion that 0D CuS QDs are used as NO2 sensitive points, and 2D Bi2S3 NSs are used as a charge transport network to achieve an efficient gas sensing response. Therefore, in the sensing process of artificial olfactory neurons, the CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs heterostructure sensing unit can efficiently adsorb gas molecules and quickly transfer charge signals, which undoubtedly greatly improves the efficient sensing of NO<sub>2</sub>.

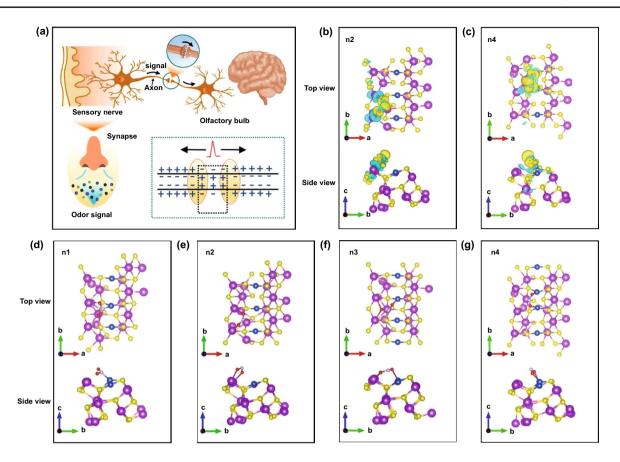
#### 3.2 CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs-Based Sensor Inspired by Biological Olfactory Neurons

Figure 2a shows the process of recognition and perception of the smell of the external environment by organisms. When the receptors located in the olfactory bulb of the human body are stimulated by a certain intensity of gas molecules, olfactory receptor cells release electrical signals. The electrical signal is then conducted in the olfactory glomerulus and finally sent to the brain area for perception and recognition [12].

Inspired by the sensory nervous system, we tried to prove that the sensory synapses of the artificial neuron structure have the same perceptual behavior as biological perception. The schematic diagram of olfactory artificial neuron is revealed in Fig. 2b. It includes an ultra-thin PI substrate deposited with interdigital electrodes and a sensitive detection unit based on CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs heterostructure. In the gas sensor, 0D-2D CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs are stacked on an ultra-thin PI substrate deposited with Au electrodes by inkjet printing (optical image in Fig. 2b–i). Besides, NO<sub>2</sub> molecules with strong electron deprivation properties are adsorbed on the surface of artificial neurons in the sensing process, which triggers changes in the surface charge distribution by depriving them of charges and finally leads to the generation of electrical signals (Fig. 2b-ii).

#### 3.3 Artificial Neuron-like Sensing Layer Based on CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs Heterostructure

To achieve ultra-sensitive detection, it is particularly important to construct a reasonable sensor layer of artificial olfactory neurons. As schematically shown in Fig. 3a, in brief, ultrasonic stripping of bulk  $Bi_2S_3$  was executed in a mixed solution of anhydrous ethanol and deionized water for the first time without any further complicated cleaning treatment. In stark contrast to commercial bulk  $Bi_2S_3$  (Fig. S2),



**Fig. 1** a Schematic diagram of charge change of biological olfactory system during stimulation. Sensing mechanism and the possible interaction energy of gases with configurations on the CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs using DFT calculations. **b** and **c** Diagrams of charge density difference of NO<sub>2</sub> on CuS-Bi<sub>2</sub>S<sub>3</sub> with n2 and n4 binding structure for side and top view, where the charge density isosurfaces of blue and yellow are 0.0015 and -0.0015 e Å<sup>-3</sup>, respectively. **d-g** Binding structures of NO<sub>2</sub> on CuS-Bi<sub>2</sub>S<sub>3</sub>

the lateral size and thickness of  $Bi_2S_3$  NSs were reduced from a few micrometers to the nanometer scale upon grinding and ultrasonic treatment.

Subsequently, a simple two-step chemical precipitation technique was designed to hybridize CuS QDs to the surface of  $Bi_2S_3$  NSs, and its chemical structure is shown in Fig. 3b. The composite structures of BC-5 (a series of CuS QDs modified  $Bi_2S_3$  NSs hybrid materials with different CuS content are numbered BC-n, where n is the mass percentage of CuS QDs in the hybrid material, n = 2.5, 3, 4, 5, 6, 7.5, 10, and 20) were studied by SEM and TEM. As shown in Fig. 3c, the morphology of BC-5 keeps the same as the bare  $Bi_2S_3$  NSs, which still exhibits a nanosheet appearance as the lateral size of ca. 200 nm without an observable change in morphology upon the loading of CuS. The thin 2D structure supplies a boosted surface area for the CuS QDs loading.

CuS QDs ( $\approx 8$  nm) uniformly distributed on the surface of Bi<sub>2</sub>S<sub>3</sub> NSs are observed in the TEM image (Fig. 3d).

Attribute to the higher surface energy at the edge positions of nanosheets and the individual distribution of quantum dots without agglomerates or clusters, the CuS QDs are prone to be attached to the edge positions of  $Bi_2S_3$  NSs. The intimate attachment of CuS QDs on  $Bi_2S_3$  NSs demonstrates the strong bonding and interacting heterogeneous structures at the interface, which is essential for their enhanced sensing performance [22, 26].

Based on the high-resolution TEM (HR-TEM), a 0.56 nm lattice spacing is found for  $Bi_2S_3$  NSs and BC-5 (Figs. S2d and 3e), which corresponds to the (020) facets of  $Bi_2S_3$  crystal and again supports the process of selective orientation stripping of  $Bi_2S_3$  NSs [13]. Additionally, a lattice fringe spacing of 0.304 nm is also found in Fig. 3e, which can be ascribed to the crystallographic facet (102) of CuS. The confirmation of these lattices further proves that the construction of CuS QDs and  $Bi_2S_3$  NSs forms a heterostructure rather than a simple physical mixing [27]. In addition, the

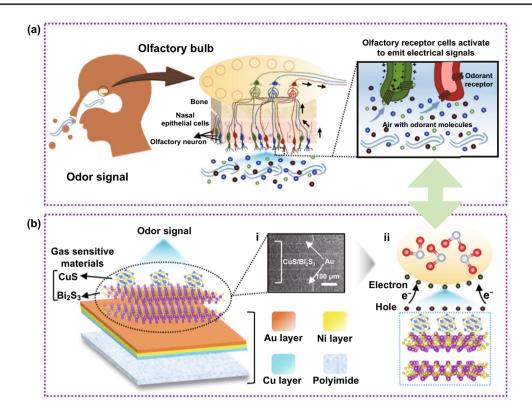
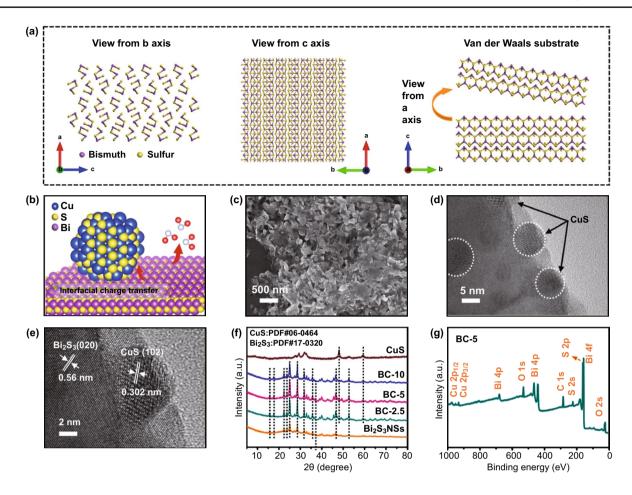


Fig. 2 a Schematic diagram of biological olfactory system. b Schematic diagram of artificial olfactory neuron sensing based on CuS  $QDs/Bi_2S_3$  NSs heterostructure

presence of Bi<sub>2</sub>S<sub>3</sub> NSs results in a more uniform loading of CuS QDs on its surface. The detailed microstructure of the irregular CuS without substrate nanosheets is indicated in Fig. S4c. The XRD confirmed the structural regularity of bulk material and the crystal planes of orthorhombic Bi<sub>2</sub>S<sub>3</sub>. Exfoliated Bi<sub>2</sub>S<sub>3</sub> NSs show preferential orientation along the (010) facet, which increases the intensity of the peaks including (130), (310) and (020) (Fig. S3a). In addition, a decrease in the overall peak intensity of the product sample has been observed, demonstrating that the exfoliation process distorts the crystal lattice [16, 28]. The similar Raman spectrum patterns of bulk material and thin layer are displayed in Fig. S3b, Supporting Information. The transverse in-plane A<sub>g</sub> vibration peaks occurring at 188 and  $240\ \mbox{cm}^{-1}$  and the longitudinal  $B_{1g}$  vibrations at 168 and  $265 \text{ cm}^{-1}$  coincide with the values reported previously [29]. It is noteworthy that Bi<sub>2</sub>S<sub>3</sub> NSs have a similar Raman signal as bulk material, further confirming that the properties of nanosheets remain unchanged after liquid-phase exfoliation process.

XRD pattern of the CuS QDs/ $Bi_2S_3$  NSs (Fig. 2f) is specifically indexed into orthorhombic  $Bi_2S_3$  and hexagonal CuS [27, 30]. As a blank control, bare CuS was prepared without the loading of  $Bi_2S_3$  NSs. The bare CuS has revealed a single-phase structure in XRD and Raman tests as expected in the pure hexagonal phase (Fig. S4) [31]. X-ray photoelectron spectroscopy (XPS) measurements have been further used to study the elements in CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs and other products. As displayed in Figs. 3g and S5, the full XPS spectrum of the heterostructure includes the corresponding peaks that are associated with  $Bi_2S_3$ , while the peak at 931.9 and 951.7 eV, respectively, supports the presence of Cu  $2p_{3/2}$  and  $2p_{1/2}$  in CuS [32]. This confirms the successful formation of CuS and  $Bi_2S_3$  in the composite product, which shows no difference with XRD results.

To consider the potential distribution of the resting state of the artificial neuron sensor layer before being stimulated by the olfactory sense, the interface interaction between the CuS QDs and  $Bi_2S_3$  NSs was further studied. The electronic structures of the composite products and the pure phase monomer constituent elements were studied by highresolution XPS measurement. As displayed in Fig. 4a, two separated peaks in the Bi 4f XPS spectrum at 158.3 and 163.6 eV are  $4f_{7/2}$  and  $4f_{5/2}$  hybridized orbitals of the Bi



**Fig. 3** CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs heterostructure and characterization of CuS and Bi<sub>2</sub>S<sub>3</sub> NSs. **a** Schematic diagram of crystal structure and fractureexfoliation process of Bi<sub>2</sub>S<sub>3</sub> bulk material with selective orientation. **b** Schematic representation of the atomic structure of CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs. **c-e** SEM, TEM and HR-TEM images of BC-5. **f** XRD patterns of CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs, CuS and Bi<sub>2</sub>S<sub>3</sub> NSs. **g** Full XPS survey spectrum of BC-5

4f orbital energy level for BC-5 sample, respectively [33]. Compared with bare  $Bi_2S_3$  NSs, the peak position shifts to the high binding energy, indicating that the electron density on the  $Bi_2S_3$  surface in BC-5 nanocomposites is reduced. Two peaks at 932.1 and 952.2 eV of Cu 2p (Fig. 4b) can be regarded as the Cu<sup>2+</sup> [30]. Compared to bare CuS, the peak position shifts about 1 eV toward the lower binding energy, indicating an elevated electron density on the Cu surface in the BC-5 nanocomposite. Meanwhile, the S 2p XPS spectra in Fig. 4a, c show similar results. The results of these peak shifts prove the formation of heterogeneous structures. The charge transfer between the interfaces is from  $Bi_2S_3$  to CuS. Moreover, the chemical bonds formed between CuS QDs and  $Bi_2S_3$  NSs lead to the rapid migration of interfacial charges and induce the appearance of strong polarization and

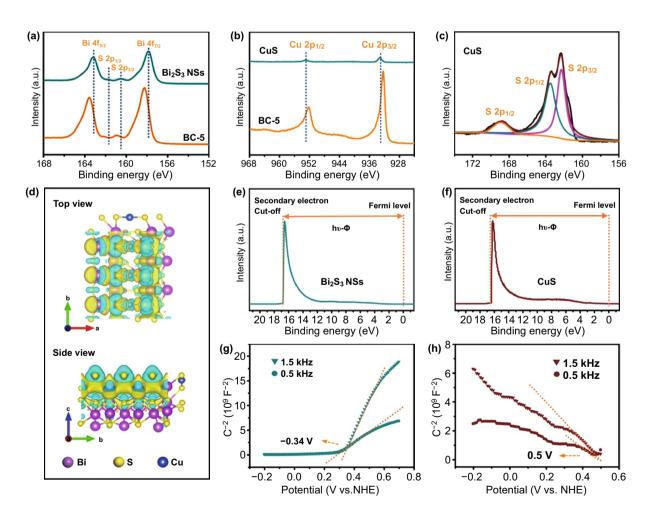
strong electric fields, which help to enhance the adsorption and desorption efficiency of  $NO_2$  [21, 26, 34].

To understand the charging behavior of the CuS QDs/ Bi<sub>2</sub>S<sub>3</sub> NSs, the charge density difference at the interface was studied using DFT calculation. As exhibited in Fig. S6 in Supporting Information, three Cu atoms in the CuS QDs/ Bi<sub>2</sub>S<sub>3</sub> NSs composite structure are bonded at the (010) crystal plane S–S bridge site. From the three-dimensional migration diagram of the CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs interfacial charge in Fig. 4d, the CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs interfacial charge shows a strong polarization, efficient electron–hole separation, and redistribution state. The differential charge density and bader charge transfer values are displayed in Table S3, showing a strong bonding formed between CuS and Bi<sub>2</sub>S<sub>3</sub>, and the bader charge transfer value is 3.371 e, which illustrates the correctness of this structural model and proves that the interfacial charge is transferred from  $Bi_2S_3$  to CuS.

Besides, to gain a deeper understanding of the redistribution of interfacial charges and to reveal the transition process of  $Bi_2S_3$  NSs upon CuS QDs loaded on the surface, the energy band structure of the interface materials was further characterized. Herein, ultraviolet photoelectron spectroscopy (UPS) measurements are employed to establish the specific energy band location of heterojunction, where the position of the conduction band (CB), the Fermi energy level ( $E_f$ ), and the valence band maximum (VB) are determined by  $E_f = 21.22 \text{ eV} - E_{\text{cutoff}}$  ( $E_{\text{cutoff}}$  is the energy at which the secondary light emission starts),  $E_V = 21.22 \text{ eV} - [E_{\text{cutoff}} - (E_f - E_{VM})]$  ( $E_{VM}$  is the valence band maximum). Then, the bandgap values of  $Bi_2S_3$  and CuS

were, respectively, determined to be 1.3 and 1.8 eV from the UV–Vis diffuse reflectance spectra and the corresponding Tauc plots (Fig. S8) [35]. In addition, the Mott–Schottky technique is employed to study the semiconductor properties of interfacial materials, as well as flat-band potentials. As shown in Fig. 4g, h, the positive and negative slope indicates the n-type and p-type behavior of  $Bi_2S_3$  and CuS, while its flat-band potential is calculated as -0.34 V, as well as 0.5 V, respectively [34, 36].

Moreover, the specific energy band locations of heterojunction materials are described in Fig. S9 with a corresponding model to explain the interfacial charge redistribution. When the two different semiconductors at the interface come into contact, due to the distance between the  $E_f$  of  $Bi_2S_3$  and the vacuum electron energy level ( $E_{vac}$ ) of



**Fig. 4** Analysis of interface charge state of the heterostructure. **a-c** High-resolution XPS spectra of Bi 4f, Cu 2p and S 2p of the BC-5, CuS, and  $Bi_2S_3$  NSs. **d** Charge distribution at the interface of CuS QDs and  $Bi_2S_3$  NSs calculated by density functional theory. UPS spectra of **e**  $Bi_2S_3$  and **f** CuS. Mott – Schottky plots for **g**  $Bi_2S_3$  NSs and **h** CuS

-4.37 eV, driven by the thermodynamic difference between the work function of  $Bi_2S_3$  (-4.37 eV) and CuS (-4.79 eV), electrons will spontaneously transfer from  $Bi_2S_3$  to CuS across the interface, thus forming an electron depletion layer on the surface of  $Bi_2S_3$ . On the contrary, holes of p-type CuS near the interface tend to diffuse into n-type  $Bi_2S_3$ , thus establishing a hole depletion layer on the CuS surface. The electron–hole diffusion continues until the heterojunction establishes a uniform  $E_f$ , which conduces to the energy band bending at the interface. Furthermore, the CuS QDs/ $Bi_2S_3$ NSs p–n junction exhibits a type-II band structure, which is conducive to inhibit electron–hole recombination [37].

#### 3.4 Sensing Performance of the Artificial Olfactory Neuron-like Sensor

To assess the effect of interfacial electrons on materials after the construction of heterogeneous structures, the sensing material was sprayed on the electrode surface using inkjet printing technology, and the change of its gas-sensitive performance was studied in a homemade dynamic sensing system (Figs. S10 and S11). The variation in sensing performance of starting bulk materials, Bi<sub>2</sub>S<sub>3</sub> NSs, and CuS QDs/  $Bi_2S_3$  NSs with different weight ratios for 10 ppm NO<sub>2</sub> is displayed in Figs. 5a, S12 and S13. The exfoliated Bi<sub>2</sub>S<sub>3</sub> NSs showed greater electrical resistance compared to the bulk material, which was ascribed to the large number of defects introduced after liquid-phase exfoliation (Table S4) [17, 26]. The pristine Bi<sub>2</sub>S<sub>3</sub> displayed an increase in resistance immediately after being exposed to NO<sub>2</sub>, indicating an n-type semiconducting property as identical to the Mott-Schottky (M–S) test results. Meanwhile, the larger specific surface area after exfoliation enriches the number of the active sites on the samples, resulting in increased response and reduced recovery performance.

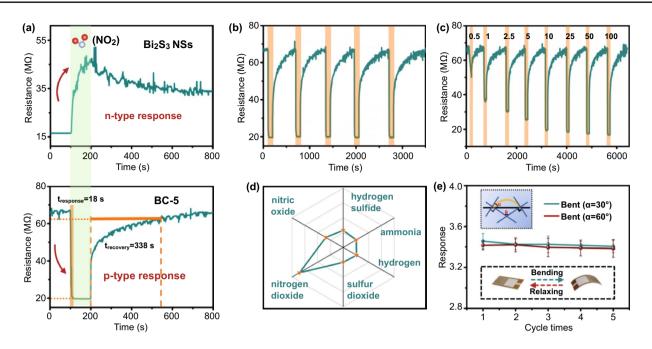
When CuS QDs are loaded at the lower concentrations (2.5 wt% and 3 wt%), an increase in the sample resistance, as well as a decrease in the response magnification, can be observed, which can be attributed to the recombination of electrons in  $Bi_2S_3$  NSs with holes from the p-type CuS. As the loading amount of CuS QDs further increases (beyond 4 wt%), the sensor exhibits a decreasing resistance change in the presence of NO<sub>2</sub>, indicating that the n-type  $Bi_2S_3$  has been completely transformed into a p-type cuS. This result suggests

that the CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs possess a complete transition of n- to p-type in semiconductor properties. When the CuS QDs loading on the surface of Bi<sub>2</sub>S<sub>3</sub> NSs is further increased (over 6 wt%, until it reaches bare CuS), a further decrease in the surface resistance of sensitive materials is observed. This phenomenon is attributed to the further increase in the majority carrier (hole) concentration of the p-type semiconductor composite product. The continuous improvement of the conductivity of sensitive materials makes the resistance change more difficult to be observed, which is manifested as a decrease in the response rate [38]. By comparing the response and recovery properties of the

heterojunction-based sensors with different CuS QDs loadings to 10 ppm NO<sub>2</sub>, BC-5 has been identified as the optimal device with a response amplification of 3.4 and ultra-fast response and recovery rates (18 and 338 s, respectively) without the assistance of any additional means. Our BC-5 has shown a grand advance in sensing performance over the previously reported works on NO<sub>2</sub> room temperature flexible sensors (Table S5). According to the previous DFT calculations, compared to the initial bulk Bi<sub>2</sub>S<sub>3</sub> material, the reason for the superior performance of the BC-5-based sensor attributes to the construction of many sensitive points and the rapid conduction of charges.

To reveal the importance of reversibility for BC-5-based gas sensors, the cyclic test is executed to 10 ppm NO<sub>2</sub>. The sensing performance was kept nearly unchanged after five cycle uses (Fig. 5b). In addition, Fig. 5c shows the response-recovery curves of the BC-5-based sensor exposed to NO<sub>2</sub> at concentrations of 500 ppb to 100 ppm. With the response change and error bar data summarized in Fig. S13, BC-5-based sensors exhibit excellent sensitivity and recovery capability. The theoretical limit of detection (LOD) of the device toward NO<sub>2</sub> was calculated as approximately 78 ppb by the least squares method using 3RMS<sub>noise</sub> divided by the slope of the low-concentration partial fit curve (Fig. S14b) [8, 39]. The fitted curve shows a dichotomous distribution due to the scattering effect of NO<sub>2</sub> molecules resulting in reduced carrier mobility [40].

To study the sensing performance of the device in the real environment, we have further investigated its long-term stability and selectivity. The long-term stability of BC-5-based sensor was estimated upon exposure to 10 ppm NO<sub>2</sub> for 2 months at an interval of 1 week (Fig. S14c). The response value of the device remained virtually unchanged for two months, indicating excellent long-term stability.



**Fig. 5** NO<sub>2</sub> sensing performances BC-5-based sensor. **a** Resistance variation for the device to NO<sub>2</sub> at 10 ppm concentration. **b** Response-recovery curve of the device for 5 cycles to 10 ppm NO<sub>2</sub>. **c** Transient response-recovery curve for the device to NO<sub>2</sub> with different concentrations. **d** Selectivity for the device to 10 ppm different target gases. **e** Response graph and response error bar for the device to 10 ppm NO<sub>2</sub> with different bending angles

This result indicates that the depletion layer formed at the interface by our heterostructure building process can act as a passivation layer, preventing environmental oxidation of sulfides and leading to reliable long-term stability [41, 42]. Furthermore, the selectivity of BC-5-based sensor for different gases was tested. Different analytes were investigated including 10 ppm NO<sub>2</sub>, H<sub>2</sub>S, NH<sub>3</sub>, SO<sub>2</sub>, H<sub>2</sub>, and NO. As shown in Figs. 5d and S15, the response value for NO<sub>2</sub> was significantly higher than other test gases, indicating that the device exhibited excellent selectivity. Besides, considering the influence of humidity, the sensor was exposed to 10 ppm  $NO_2$  with relative humidity ranging from 0 to 80% (Fig. S16). The response value decreases from 3.4 to 1.6 as the relative humidity increased, indicating that the sensor can still work normally under high humidity conditions. The negligible decrease in response is probably arising from the fact that some  $H_2O$  molecules occupy the sensing sites [43].

In addition, the illustration of Fig. 5e depicts a schematic diagram of the mechanical strain bending direction of the flexible device. The flexible sensor was bent at 30° and 60° followed by sensing tests in 10 ppm NO<sub>2</sub> for five cycles. The response results of the device clearly showed that the

bending of the flexible electrode will not reduce the property of the as-prepared device.

#### 3.5 Gas sensing Mechanisms of the Artificial Neuron-like Olfactory System

Based on the DFT calculations and the corresponding experimental results, we conducted an in-depth understanding and discussion on the mechanism of the artificial neuron-like sensor based on CuS QDs/ $Bi_2S_3$  NSs to enhance the NO<sub>2</sub> sensing performance.

First of all, the DFT calculation results prove that in the neuron-like sensor model constructed by the olfactory sensory neurons, a large number of dispersed OD CuS QDs can be used as  $NO_2$  sensitive points to efficiently capture gas molecules. The 2D  $Bi_2S_3$  NSs as a conductive network and a heterojunction substrate can achieve rapid charge transfer through synergy.

Secondly, the artificial olfactory neuron sensor based on the heterogeneous structure of CuS  $QDs/Bi_2S_3$  NSs has the same charge transfer method as the biological olfactory perception process in the resting state and after being stimulated by gas molecules. The charge distribution process of the CuS QDS/Bi<sub>2</sub>S<sub>3</sub> NSs heterostructure itself in the resting state was studied in the previous chapter. In brief, upon intimate contact, electrons move spontaneously from  $Bi_2S_3$  to CuS across the interface for the equilibrium of Fermi level, leading to the formation of the electron and hole depletion layer on the surface of  $Bi_2S_3$  and CuS, respectively. In addition, the formation of a built-in electric field and the bending of energy band also occur.

As shown in Fig. S17, when exposed to NO<sub>2</sub> (electron acceptor), gas molecules are adsorbed onto the CuS QDs surface and trap electrons, breaking the equilibrium of the built-in electric field formed previously. The additional holes return to CuS that change the width of the potential barrier until formed a new equilibrium. The presence of a potential barrier in the heterojunction at the interface can reflect small fluctuations in carrier concentration due to the adsorption of NO<sub>2</sub> molecules as large changes in resistance, which undoubtedly facilitates the response process [44]. In addition, the built-in electric field can also accelerate the charge carrier transfer efficiency, so as to improve the sensing performance [45, 46].

Furthermore, the switch from n-type to p-type character and the construction of surface heterostructures also affects the selectivity of the device. Compared with other gases, the adsorption of NO<sub>2</sub> with a higher charge density can significantly increase the heterojunction barrier and improve its sensing performance [37]. The design of this series of interfacial heterostructures and charge modulation successfully enhanced the room temperature sensing performance of the artificial olfactory neuron-like sensors for NO<sub>2</sub>, making it significantly better than the previously reported  $Bi_2S_3$ -based sensors and providing a new idea for the design of the heterogeneous interface of the biological simulation artificial olfactory neuron sensor.

#### 3.6 Wireless Wearable Devices and Real-time Monitoring of NO<sub>2</sub>

In the previous content, we correlated the process of biological olfactory perception with the electrical signal generation and transmission process of the artificial neuron-like sensor in the process of detecting gas. The result of the transmission of sensory signals in the organism is the reception, analysis and response of the central nervous system of the brain. This series of processes is consistent with the sensor's back-end information collection and real-time data extraction. Previously, many flexible gas sensors have been developed while their real-world applications in wireless and wearable electronics still confront a challenge and are rarely demonstrated [47]. Intending to design wearable applications in real time, we integrated CuS QDs/Bi<sub>2</sub>S<sub>3</sub> NSs-based artificial neuron-like sensor, circuit, and data acquisition to develop a portable NO<sub>2</sub>-monitoring system (Fig. 6a).

The hardware and software block diagram of the asprepared device system is illustrated in Fig. 6b. During the operation of the device, Bluetooth communication is first turned on, and the mobile device equipped with the selfdeveloped Android-based application scans the surrounding Bluetooth devices and verifies and connects after finding a matching sensing device. After the connection, the application will request the data acquisition circuitry for information such as the sensor's output voltage and real-time time. Once the data are obtained, the application can calibrate the ambient temperature and humidity, connect to a database to calculate the corresponding environmental pollution conditions, and display the relevant gas concentrations in real-time according to the requirements written. Moreover, by presetting the threshold value, the user can be warned in time when the pollution is aggravated and the gas concentration exceeds the standard.

For the working tests in this paper, a 200 mAh rechargeable lithium-polymer battery (TF402030) was used to provide the appropriate voltage for all circuits, and the actual voltage on the flexible sensing device was 3.3 V (device circuit schematic is displayed in Fig. S18). The double-sided copper-clad flexible PCB is connected to the PI flexible electrode deposited with the gold finger by flexible silver wires (Fig. S19a), ensuring the flexibility of the wearable device and the accuracy of data transmission. The entire wireless wearable sensing device was tested for NO<sub>2</sub> atmosphere in a fume hood at room temperature. In the actual test procedure, the signal connection and transmission from the Bluetooth device were first performed (Fig. S19b and Video S1), followed by a drop in the resistance value on the mobile application with the tube placed beside the device, i.e., the NO<sub>2</sub> signal was detected. As the gas tube is removed, the curve trajectory spontaneously

returns to the baseline (Fig. 6c). Also, given the fact that the interface-modulated CuS  $QDs/Bi_2S_3$  NSs were able to return the signal to baseline without external assistance, we tested a continuous response for 4 min and revealed it on a phone application to demonstrate the reproducibility (Fig. S19c and Video S2).

The results presented herein efficiently cross-fertilize the fields of biological neuron-like sensing unit design, mechanical circuit fabrication, and software development to successfully develop wearable sensing devices that are lightweight, visual, and multifunctional. The advantages of biomimetic association, cheap, and low power

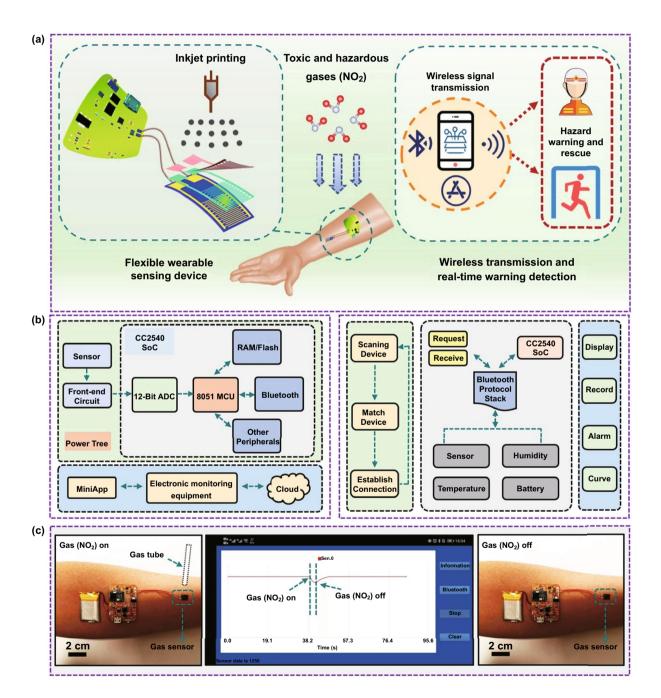


Fig. 6 Application process of the wireless wearable sensor system. a Schematic diagram of structure and application concept of wireless wearable devices. b Circuit hardware and software logic block diagram. c Photographs of the change in response to  $NO_2$  gas by smartphone application during the actual test

consumption allow for sustainable utilization, and the development of wireless data transmission coincides with the expansion of the modern Internet of Things.

#### **4** Conclusions

In summary, a self-designed bio-simulated neuron-like gas sensor that can drive wireless wearable devices for real-time monitoring of NO<sub>2</sub> has been reported. First, 0D p-type CuS QDs were successfully loaded onto the surface of 2D n-type Bi<sub>2</sub>S<sub>3</sub> NSs. Characterization analysis and DFT calculations were used for the first time to reveal that the charge transfer process of artificial neuron-like sensors in the resting state and when stimulated by gas is consistent with the process of biological olfactory perception. Moreover, the heterogeneous integration of artificial neuron-like sensor was unveiled that synergistically combines CuS QDs sites with high adsorption energy for target gas molecules and Bi<sub>2</sub>S<sub>3</sub> NSs as fast charge transport channels. Compared with traditional bulk sensing materials, the problem of high-efficiency detection at room temperature can be effectively solved by this type of artificial neuronlike sensor through many scattered sensitive points. This exhibits an ultra-fast response (18 s) and recovery (338 s), an exceptional theoretical detection limit (78 ppb), a tunable sensing mechanism, and excellent selectivity. Second, through the low-concentration NO2 ventilation analysis of the wearable gas sensing device on the arm of the human subject, the visualization and real-time observation of the target gas on the smartphone app is realized. This artificial neuron-like device establishes a promising platform for the next generation of simulating complex biological nervous systems and promoting the development of artificial intelligence.

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