**Research Article** 

# Effect of the addition of TEOS on the SiC fibers fabricated by electrospinning

Jiayan LI, Panpan CAO, Yi TAN<sup>\*</sup>, Lei ZHANG

School of Materials Science and Engineering, Dalian University of Technology, Dalian 116024, China

Received: April 02, 2013; Revised: April 24, 2013; Accepted: April 25, 2013 ©The Author(s) 2013. This article is published with open access at Springerlink.com

**Abstract:** Polycarbosilane (PCS) has been widely used to fabricate silicon carbide (SiC) fibers via pyrolysis. In this paper, for improving the morphology of SiC fibers, tetraethyl orthosilicate (TEOS, m=1 g, 3 g and 5 g, respectively) was added into the PCS precursor solution (containing 1.5 g PCS). The continuous fibers have been prepared by electrospinning, and then the SiC fibers were synthesized by calcination at 1300 °C, 1400 °C and 1600 °C for 4 h respectively with a heating rate of 10 °C/min in flowing nitrogen (N<sub>2</sub>). The morphologies of the fibers were investigated by the scanning electron microscope (SEM) and it could be seen that the crystallinity of the SiC fibers was lower, the length of the SiC fibers was increased, and the diameter was uniform with the increase of the addition amount of TEOS.

Keywords: silicon carbide (SiC); tetraethyl orthosilicate (TEOS); electrospinning; fibers

# **1** Introduction

In the past decade, one-dimensional (1D) nanosized semiconducting materials have attracted attention with their fascinating optical, electronic and chemical properties. The size and morphology can affect their applications as catalysts, solar cells, light-emitting diodes and biological labeling [1–3]. As a semiconducting material, silicon carbide (SiC) possesses excellent physical and electronic properties such as high mechanical strength, high thermal stability and high thermal conductivity [4]. It also has wide applications including field emission displays, nano-sensors and other nanoscale devices [5,6].

Recently, much more effort has been devoted to the synthesis process of SiC fibers with nanostructures,

including carbothermal reduction of silica [7–9], chemical vapor deposition [6,10–12], carbon nanotube-templated growth [13], etc. It has been demonstrated that polycarbosilane (PCS) is used to synthesize SiC fibers by melt-spinning, curing and pyrolysis. In this paper, SiC fibers could be synthesized by a high-temperature carbothermal reduction progress using electrospun PVP (polyvinyl pyrrolidone)/PCS composite fibers as precursor. In addition, tetraethyl orthosilicate (TEOS) was added to improve the morphology of SiC fibers. Compared with the reported synthetic methods, the technique used in this work possesses the virtues including simplicity, low cost and absence of template and catalyst.

## 2 **Experiment**

As a starting material, PCS polymer was used to fabricate SiC fibers due to its rich content of silicon.

<sup>\*</sup> Corresponding author.

E-mail: tanyi@dlut.edu.cn

For achieving PVP/PCS/TEOS composite, 1.5 g PCS and TEOS (1 g, 3 g and 5 g, respectively, Tianjin Kermel Chemical Reagent Ltd.) were dissolved into 11 ml solvent including 8 ml tetrahydrofuran (THF, Tianjin Kermel Chemical Reagent Ltd.) and 3 ml absolute alcohol ( $\geq$  99.7%, Shenyang Xinxing Chemical Reagent Ltd.). Then 0.8 g PVP (K90, Tianjin Bodi Chemical Co. Ltd.) was added in the solution to increase viscidity. Subsequently, the mixture viscous solution was magnetically stirred for 1 h at room temperature until the floccules were completely dissolved. The solution was transferred into a plastic syringe with conductive stainless steel needle and ejected with an appropriate voltage of 30 kV. The dense net of fibers was formed on the collector which was coated with an aluminium foil. The distance between the needle and the collector was about 16 cm. Whereafter, the fiber mat was immediately dried in air oven at 80  $^{\circ}$ C for 0.5 h, and then was heated at 190  $^{\circ}$ C for 6 h. The Si-H bonds in PCS were oxidized and the Si-O-Si crosslinked structure was formed. The melting of PCS was prevented and the morphology of precursor fibers was remained during the process of high-temperature pyrolysis [14,15]. Subsequently, the PCS precursor fibers were calcined respectively at 1300 °C, 1400 °C and 1600 °C for 4 h with a heating rate of 10 °C/min in flowing nitrogen (N<sub>2</sub>) followed by calcination at 600 °C for 4 h in air to remove the remaining carbon.

X-ray diffraction (XRD) patterns were collected on a Shimadzu/XRD-6000 diffractometer with Cu K $\alpha$ radiation. The morphologies of SiC fibers were characterized by JSM-5600LV scanning electron microscope (SEM, JEOL).

# 3 Results and discussion

The XRD patterns of the SiC fibers after calcination at 1600 °C for 4 h which are derived from different precursor fibers are shown in Fig. 1. The main peaks in the patterns match well with the cubic  $\beta$ -SiC phase (JCPDS 74-2307), and their peak positions at  $2\theta = 36^{\circ}$ ,  $60^{\circ}$  and  $72^{\circ}$  are respectively attributed to the (111), (220) and (311) planes of  $\beta$ -SiC. The SiC crystal is successfully synthesized at this temperature. The diffraction peaks of products become weak and broad which indicates that the crystallinity is lower and the grain size in SiC fibers is decreased with the increase



(d)

(c)

(b)

(a)

20

30

 $2\theta(^{\circ})$ Fig. 1 XRD patterns of SiC fibers after calcination at 1600 °C for 4 h derived from precursor fibers with different TEOS contents: (a) 0 g, (b) 1 g, (c) 3 g, (d) 5 g.

60

70

80

90

50

40

of TEOS content. There are no other obvious diffraction peaks, which imply that the main product is cubic SiC phase and no impurity content is obtained in the fibers.

The SEM images of the PCS/TEOS precursor fibers are shown in Fig. 2. It could be observed that the surface of the precursor fibers is very smooth, which is attributed to the fine particles and the amorphous nature of the fibers. Moreover, with the increase of addition amount of TEOS, the length of the fibers is longer and the diameter is more uniform, which can also be seen in Fig. 3.

The SEM images of the SiC fibers calcined at 1300  $^{\circ}$ C and the corresponding columnar section of the diameter uniformity with different amounts of TEOS



Fig. 2 SEM images of the PCS precursor fibers with different TEOS contents.



Fig. 3 SEM images of the SiC fibers calcined at 1300 °C and the corresponding columnar section of the diameter uniformity with different amounts of TEOS.

are shown in Fig. 3. We observe that the fibers with diameters in the range of  $1-9 \mu m$  are brittle and easy to fracture when there is no or 1 g TEOS addition. However, the fibers become longer and their diameters concentrate in the range of  $4-6 \mu m$  when the addition amounts of TEOS reach 3 g and 5 g.

The SEM images of SiC fibers derived from different precursor fibers are shown in Fig. 4. After calcinations at 1600 °C for 4 h, the SiC grains as shown on the fiber surface are averagely smaller than 500 nm, and the grains are closely connected to each other with clear grain boundaries. With the addition of TEOS, the surface of SiC fibers becomes smoother and the SiC grain sizes in the fibers are decreased, which also have been confirmed by the XRD patterns.

The SEM images of the PCS/TEOS precursor fibers and SiC fibers calcined at different temperatures with 5 g TEOS addition are shown in Fig. 5. The average diameter of the fibers is about  $3 \mu m$ , and the length could reach several millimeters. The surface of SiC



Fig. 4 SEM images of SiC fibers calcined at  $1600 \,^{\circ}$ C derived from different precursor fibers with different TEOS contents.



Fig. 5 SEM images of the precursor fibers and SiC fibers calcined at different temperatures.

fibers after calcined for 4 h becomes rough due to the removal of organic components and the crystallization of the SiC phase, while the surface of the precursor fibers (Fig. 5(a)) is very smooth. The fibers show a slightly decrease in diameter due to the burning-off of PVP, whereas the continuous microstructure of the fibers is still maintained.

## 3 Conclusions

In summary, direct electrospinning was used in this work to prepare the PCS/TEOS precursor fibers, and the SiC fibers were obtained successfully after carbonization. The addition of TEOS improved the morphology of SiC fibers effectively. With the increase of addition amount of TEOS, the crystallinity was lower and the grain size in SiC fibers was decreased. The length of the fibers was longer and the diameter was more uniform.

#### Acknowledgements

This work was financially supported by the Fundamental Research Funds for the Central Universities (No. DUT12LAB03), the National Natural Science Foundation of China (No. 51104028), and the Specialized Research Fund for the Doctoral Program of Higher Education (20110041120031).

**Open Access:** This article is distributed under the terms of the Creative Commons Attribution Noncommercial License which permits any noncommercial use, distribution, and reproduction in any medium, provided the original author(s) and source are credited.

### References

- Ahmadi TS, Wang ZL, Green TC, *et al.* Shape-controlled synthesis of colloidal platinum nanoparticles. *Science* 1996, 272: 1924–1925.
- [2] Huynh WU, Peng X, Alivisatos AP. CdSe nanocrystal rods/poly (3-hexylthiophene) composite photovoltaic devices. *Adv Mater* 1999, 11: 923–927.
- [3] Chan WCW, Nie S. Quantum dot bioconjugates for ultrasensitive nonisotopic detection. *Science* 1998, 281: 2016–2018.
- [4] Wang ZL, Dai ZR, Gao RP, *et al.* Side-by-side silicon carbide–silica biaxial nanowires: Synthesis, structure, and mechanical properties. *Appl Phys Lett*

2000, 77: 3349.

- [5] Zhang Y, Nishitani-Gamo M, Xiao C, et al. A novel synthesis method for aligned carbon nanotubes in organic liquids. Jpn J Appl Phys 2002, 41: L408–L411.
- [6] Choi H-J, Seong H-K, Lee J-C, et al. Growth and modulation of silicon carbide nanowires. J Cryst Growth 2004, 269: 472–478.
- [7] Bootsma GA, Knippenberg WF, Verspui G. Growth of SiC whiskers in the system SiO<sub>2</sub>--C-H<sub>2</sub> nucleated by iron. *J Cryst Growth* 1971, **11**: 297–309.
- [8] Wang L, Wada H, Allard LF. Synthesis and characterization of SiC whiskers. *J Mater Res* 1992, 7: 148–163.
- [9] Meng GW, Zhang LD, Mo CM, et al. Preparation of β-SiC nanorods with and without amorphous SiO<sub>2</sub> wrapping layers. J Mater Res 1998, 13: 2533–2538.
- [10] Zhou XT, Lai HL, Peng HY, et al. Thin β-SiC nanorods and their field emission properties. Chem Phys Lett 2000, 318: 58–62.
- [11] Leu I-C, Lu Y-M, Hon M-H. Substrate effect on the preparation of silicon carbide whiskers by chemical vapor deposition. *J Cryst Growth* 1996, 167: 607–611.
- [12] Motojima S, Hasegawa M, Hattori T. Chemical vapour growth of β-SiC whiskers from a gas mixture of Si<sub>2</sub>Cl<sub>6</sub>--CH<sub>4</sub>--H<sub>2</sub>--Ar. *J Cryst Growth* 1988, 87: 311-317.
- [13] Tang CC, Fan SS, Dang HY, *et al.* Growth of SiC nanorods prepared by carbon nanotubes-confined reaction. *J Cryst Growth* 2000, **210**: 595–599.
- [14] Cheng XZ, Xiao JY, Xie ZF, et al. Research on curing of polycarbosilane fiber and preparation of SiC fiber. J Mater Eng 2004, 1: 29–37 (in Chinese).
- [15] Zheng CM, Li XD, Wang H, et al. Thermal stability and curing kinetics of polycarbosilane fibers. Trans Nonferrous Met Soc China 2006, 16: 44–48.