

The influence of La_2O_3 -doping on structural, surface and catalytic properties of nano-sized cobalt–manganese mixed oxides

Abdelrahman A. Badawy¹ · Shaimaa M. Ibrahim^{2,3}

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Abstract Structural, textural and catalytic activity of $\text{CoO-Mn}_2\text{O}_3$ system as being influenced by La_2O_3 -doping (0.75–3 mol%) and calcination temperatures (300–500 °C) were investigated. The techniques employed were XRD, N_2 -adsorption–desorption at –196 °C, EDX and catalysis of H_2O_2 -decomposition in aqueous solution at 30–50 °C. The results revealed that the investigated system consisted of nano-sized Co_2MnO_4 as a major phase together with unreacted portion of Co_3O_4 and $\gamma\text{-Mn}_3\text{O}_4$. Doping with the smallest amount of La_2O_3 greatly increased the surface molar ratio of Mn/Co (88 %) for the solid calcined at 500 °C with subsequent increase of the catalytic activity more than 12-fold for solids calcined at 500 °C. The increase of La_2O_3 -dopant above 0.75 mol% decreased progressively the surface molar ratio of Mn/Co with subsequent decrease in the catalytic activity which still measured higher values than that measured for the un-doped catalyst.

Keywords Cobalt manganite · La_2O_3 -dopant · Coprecipitation · Catalytic decomposition of H_2O_2

Introduction

Manganese oxides are reported to be considered as environment-friendly materials. MnO_2 and Mn_3O_4 were found to be active and stable catalysts for the combustion of organic compounds [1, 2].

Nanocrystalline manganese oxide powders were synthesized in previous study by an inert gas condensation technique [3]. The manganese oxide, which is prepared, is a mixture of MnO and Mn_3O_4 . The particle size of manganese oxides is greatly dependent on their preparation conditions. Dimesso et al. [4] claimed that manganese oxides can be prepared by an inert gas condensation technique followed by annealing in air and oxygen at various temperatures. The predominant phase of MnO and Mn_3O_4 are obtained after annealing in air at 400 °C.

Mixed oxides containing transition metal oxides are used to design the catalytic materials to replace noble metal catalysts. Lahousse et al. [1] have found that $\gamma\text{-MnO}_2$ and Pt/TiO_2 catalysts measured high catalytic activities as compared to noble metals catalysts.

Mixing manganese with transition metal oxides in many catalytic systems modify the catalytic activity of individual components [5, 6]. Mixed oxide materials are active for oxidation–reduction reactions and combustion processes. For example, cobalt–zinc manganites, manganese– CeO_2 mixed oxides and Co-containing mixed oxides prepared from hydrotalcite-like precursors were active catalysts in the reduction of nitrous oxide [7–9]. Also, Co–Mn mixed oxides were found to be active catalysts for oxidation of ethanol [10] and conversion of synthesis gas to light olefins [11]. However, Ag–Mn, Ag–Co and Ag–Ce composite oxides supported on Al_2O_3 have been reported as catalysts for oxidation of volatile organic compounds [12]. In a previous study, Mn–Cu mixed oxides have been reported to

✉ Abdelrahman A. Badawy
aabadawy107@yahoo.com

¹ Physical Chemistry Department, National Research Centre, Dokki, Cairo, Egypt

² Chemistry Department, Faculty of Education, Ain Shams University, Cairo, Egypt

³ Present Address: Department of Chemistry, Faculty of Science, Qassim University, Buraidah, Saudi Arabia

be catalytically more active towards ethanol oxidation as compared to individual Mn_2O_3 and CuO [6]. Both copper and manganese mixed oxides catalysts were found to be more active catalysts in many industrial oxidation processes, such as CO oxidation by O_2 , combustion of toluene, methanol, ethylene, ammonia, NO_2 and other combustion reactions [13–16]. However, Mn–Zr mixed oxide samples are active towards dehydrogenation of isopropanol, giving rise to acetone with high selectivity at partial conversion [17].

Hydrogen peroxide and its solutions find use as antiseptic in medicine [18, 19], other applications such as bleach in the textile and paper/pulp industry, in treatment of waste water [20]. However, the literature survey reveals that mixed oxide catalyst is more active in H_2O_2 decomposition. These catalysts have attracted much attention of chemists due to their application as low-cost fuel cells, their stability and high activity [21, 22]. The decomposition of hydrogen peroxide in presence of some metal oxides as LaMnO_3 at room temperature and nanocrystalline LaCrO_3 was investigated by Khetre et al. [23]. They found that the catalytic activity was increased by increasing both the amount of the catalyst and pH. The probable reaction mechanism has been suggested in which an intermediate surface complex is thought to be responsible for the enhancement of the decomposition of hydrogen peroxide.

The present work aimed at studying the effect of La_2O_3 -doping of $\text{CoO}/\text{Mn}_2\text{O}_3$ system prepared by coprecipitation method on its structural, surface and catalytic properties. The techniques employed were XRD, EDX, N_2 -adsorption isotherms carried out at -196°C and catalytic decomposition of H_2O_2 in aqueous solution at 30 – 50°C .

Experimental

Materials

Equimolar proportions of $\text{CoO}/\text{Mn}_2\text{O}_3$ were prepared by coprecipitation method of their mixed hydroxides from their nitrates solution using 1 M NaOH solution at pH 8 and a temperature of 70°C . The carefully washed precipitate was dried at 110°C till constant weight, and then subjected to heating at 300 , 400 , and 500°C for 4 h. Three La_2O_3 -doped samples were prepared by impregnating a given dry weight of the mixed hydroxides with calculated amount of lanthanum nitrate dissolved in the least amount of distilled water sufficient to make pastes. The pastes were dried at 110°C and then calcined at 300 , 400 and 500°C for 4 h. The dopant concentrations in the calcined solids were 0.75, 1.5, and 3 mol% La_2O_3 .

Techniques

X-ray powder diffractograms of various investigated samples calcined at 300 , 400 and 500°C were determined using a Bruker diffractometer (Bruker D 8 advance target). The patterns were run with copper K_α with secondly monochromator ($\lambda = 1.5405 \text{ \AA}$) at 40 kV and 40 mA . The scanning rate was 0.8° in $2\theta \text{ min}^{-1}$ for phase identification and line broadening profile analysis, respectively. The crystallite size of the phases present in pure and variously La_2O_3 -doped solids was determined using the Scherrer equation [24]:

$$d = K \lambda / \beta_{1/2} \cos \theta,$$

where d is the mean crystallite diameter, λ is the X-ray wave length of the X-ray beam, K is the Scherrer constant (0.89), $\beta_{1/2}$ is the full width at half maximum (FWHM) of the main diffraction peaks of the investigated phases, in radian and θ is the diffraction angle.

Energy dispersive X-ray analysis (EDX) was carried out on a Hitachi S-800 electron microscope with a Kevex Delta system attached. The parameters were as follows: -15 kV accelerating voltage, 100 s accumulation time, 8 \mu m window width. The surface molar composition was determined by the Asa method (Zaf-correction, Gaussian approximation).

Different surface characteristics, namely specific surface area (S_{BET}), total pore volume (V_{p}), mean pore radius (r^-) and pore volume distribution curves ($\Delta v/\Delta r$) of various solids were determined from nitrogen adsorption–desorption isotherms measured at -196°C using NOVA Automated Gas sorbometer. Before undertaking such measurements, each sample was degassed under a reduced pressure of 10^{-5} Torr for 3 h at 200°C . The values of V_{p} were computed from the relation:

$$V_{\text{p}} = 15.45 \times 10^{-4} \times V_{\text{st}} \text{ cm}^3/\text{g},$$

where V_{st} is the volume of nitrogen adsorbed at P/P^0 tends to unity. The values of r^- were determined from the equation:

$$r^- = \frac{2V_{\text{p}}}{S_{\text{BET}}} \times 10^4 \text{ \AA}.$$

The catalytic activities of pure and variously La_2O_3 -doped solids were determined by studying the decomposition of H_2O_2 in their presence at temperatures within 30 – 50°C using 25 , 50 and 100 mg of a given catalyst sample with 0.5 ml volume of H_2O_2 of known concentration diluted to 20 ml with distilled water (initial concentration of $\text{H}_2\text{O}_2 = 0.01 \text{ mol/L}$). The reaction kinetics was monitored by measuring the volume of oxygen liberated at different time intervals until no further O_2 was liberated. The volume of the liberated oxygen was recalculated under STP.

Results and discussion

X-ray investigation of various solids

X-ray diffractograms of un-doped and variously La_2O_3 -doped solids calcined at 300–500 °C were determined from the recorded diffractograms of these solids are illustrated in Figs. 1, 2, 3 for the solids calcined at 300, 400 and 500 °C, respectively. The different structural characteristics of the solids investigated are given in Table 1. Table 1 includes the peak area of the main diffraction lines of different phases present and the crystallite size of Co_2MnO_4 phase formed calculated from the Scherrer equation.

Examination of Figs. 1, 2, 3 and Table 1 shows the following: (1) pure and variously La_2O_3 -doped solids calcined at 300–500 °C consisted of nano-sized cobalt manganite (Co_2MnO_4) (02-1061-JCPDS-ICDD, Copyright, 2001) as a major phase together with un-reacted Co_3O_4 (42-1467-JCPDS-ICDD, Copyright, 2001) and $\gamma\text{-Mn}_2\text{O}_3$ (18-0803-JCPDS-ICDD, Copyright, 2001) phases. (2) The peak area of the main diffraction lines of Co_2MnO_4 phase decreases progressively by increasing the amount of La_2O_3 -added in different solids. For example, the peak area of the main diffraction peak of cobalt manganite for the solids calcined at 500 °C measured 73.5, 55.1, 37.7 and 28.9 (a.u.) for pure sample and those treated with 0.75, 1.5 and 3 mol% La_2O_3 , respectively. (3) The crystallite size of the produced Co_2MnO_4 varies between 21.3 and 75.7 nm depending on the dopant concentration and calcination temperature. (4) The increase in calcination temperature of various solids investigated within 300–500 °C increased progressively the peak area of the main diffraction line corresponding to the produced Co_2MnO_4 . (5) No diffraction peak of lanthanum or lanthanum-manganite and

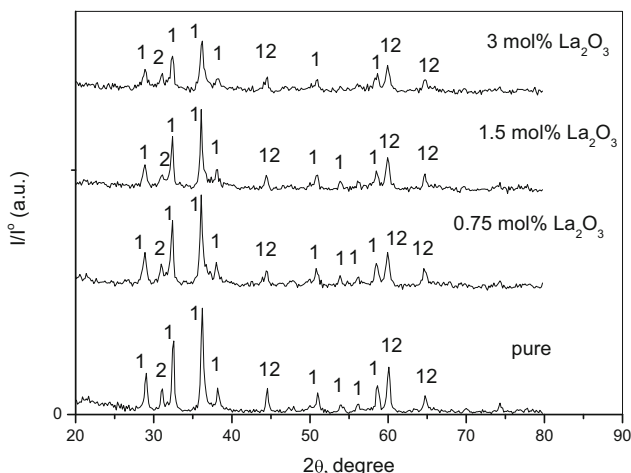


Fig. 1 X-ray diffractograms of pure and variously doped solids calcined at 300 °C. Lines 1 refer to Co_2MnO_4 , lines 2 refers to Co_3O_4

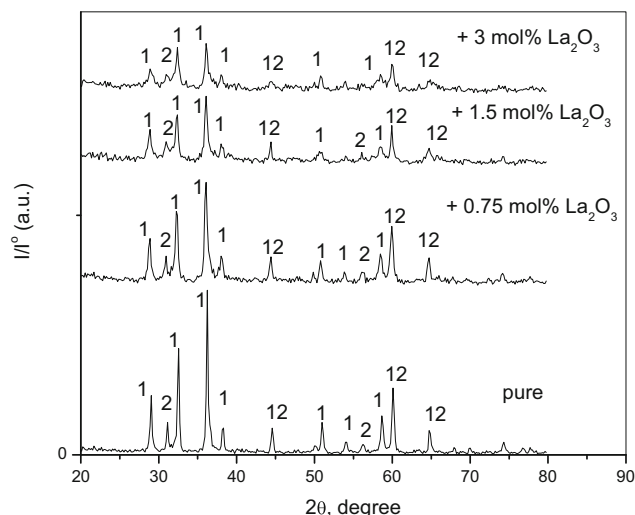


Fig. 2 X-ray diffractograms of pure and variously doped solids calcined at 400 °C. Lines 1 refer to Co_2MnO_4 , lines 2 refers to Co_3O_4

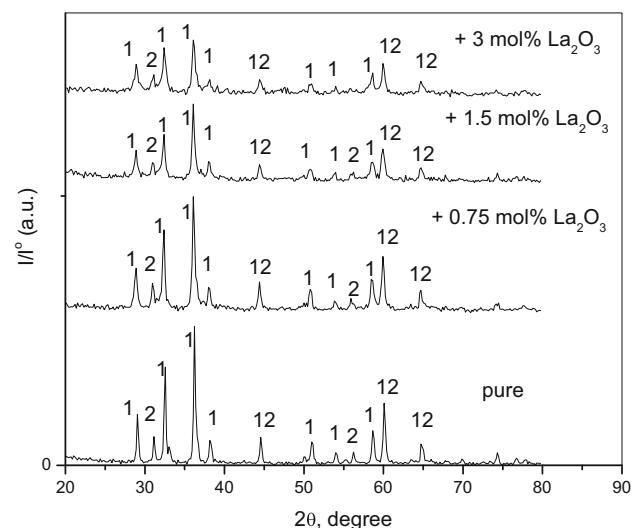
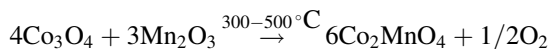


Fig. 3 X-ray diffractograms of pure and variously doped solids calcined at 500 °C. Lines 1 refer to Co_2MnO_4 , lines 2 refers to Co_3O_4

lanthanum-cobaltite composites were detected in the diffractograms. This finding suggested clearly that La_2O_3 acted only as a doping agent.

These results show clearly the role of La_2O_3 in hindering the solid–solid interaction between cobalt and manganese oxide to yielding cobalt manganite. The increase in calcination temperature within 300–500 °C stimulated the formation of Co_2MnO_4 . The formation of Co_2MnO_4 took place according to the reaction:

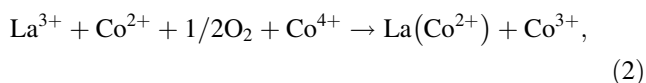
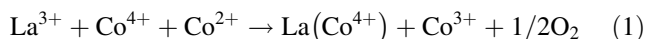


The addition of smallest amounts of La_2O_3 (0.75–3 mol%) followed by heating at 300–500 °C hindered Co_2MnO_4 formation to an extent proportional to its amount added.

Table 1 Peak area of different phases present in pure and variously doped Co₃O₄–Mn₃O₄ solids calcined at 300–500 °C and the crystallite size of Co₂MnO₄ phase

| Solids | Calcination temperature (°C) | Peak area (a.u.) of diffraction lines of | | | Crystallite size (nm) Co ₂ MnO ₄ |
|--|------------------------------|--|---|---|---|
| | | Co ₂ MnO ₄ phase 2.48 Å (100 %) | Co ₃ O ₄ phase 2.85 Å (34 %) | γ-Mn ₃ O ₄ phase 3.08 Å (90 %) | |
| Pure (0.5 mol CoO + 0.5 mol Mn ₂ O ₃) | 300 | 47.5 | 11.2 | 17.8 | 51.3 |
| | 400 | 70.3 | 13.9 | 27 | 59.7 |
| | 500 | 73.5 | 14.5 | 26.5 | 75.7 |
| +0.75 mol% La ₂ O ₃ | 300 | 39 | 7 | 10.8 | 26.3 |
| | 400 | 45.9 | 11.7 | 20.5 | 27.5 |
| | 500 | 55.1 | 14.3 | 21.3 | 31.5 |
| +1.5 mol% La ₂ O ₃ | 300 | 31.6 | 11.1 | 15.4 | 28.7 |
| | 400 | 31.7 | 10.8 | 15.4 | 30.4 |
| | 500 | 37.7 | 8.6 | 15.3 | 31.5 |
| +3 mol% La ₂ O ₃ | 300 | 25.2 | 9.3 | 11.6 | 21.3 |
| | 400 | 21.3 | 7.8 | 9.8 | 23.5 |
| | 500 | 28.9 | 9.8 | 15.2 | 30.8 |

The retardation effect of La₂O₃ might be attributed to dissolution of some of La₂O₃ (La³⁺) cation in the lattices of Co₃O₄ and/or Mn₃O₄ assuming that Co₃O₄ consisted of (2CoO and Co₂O₃, i.e., cobalt cations existed in di- and/or tetra-valent states. The dissolution of La₂O₃ in Co₃O₄ lattices could be simplified adopting Kröger's notations [25] according to:



where La (Co⁴⁺) is lanthanum cation located in the position of host tetravalent cobalt cation lattice and La (Co²⁺) is lanthanum cation located in the position of host divalent cobalt lattice cation in Co₃O₄. It is clear from Eqs. 1 and 2 that incorporation of La³⁺ in cobaltic oxide lattice decreased the concentrations of divalent and tetravalent cobalt cations converting them into Co³⁺ ions. The trivalent cobalt cations did not participate directly in the formation of CoMn₂O₄ which involved divalent cobalt and trivalent manganese cations. So, the incorporation of lanthanum cation in Co₃O₄ lattice according to mechanism 1 decreased the number of divalent cobalt involved in the formation of CoMn₂O₄ phase. Consequently, the retardation of La₂O₃ dopant in CoMn₂O₄ formation could be understood.

Surface characteristics of various prepared solids

Nitrogen adsorption–desorption isotherms measured at –196 °C were determined for pure and variously La₂O₃-doped Co₃O₄/Mn₃O₄ solids calcined at 300, 400 and

500 °C. All isotherms belong to type II of Branuer classification [26] having hysteresis loops of small areas closing at P/P^0 at about 0.5. Figure 4 depicts representative N₂-adsorption–desorption isotherms measured over pure and variously doped solids calcined at 500 °C. Figure 5 depicts $\Delta v/\Delta r$ curves of various solids calcined at 300 and 500 °C. Analysis of the recorded adsorption–desorption isotherms permitted us to calculate different surface characteristics, namely specific surface area (S_{BET}), total pore volume (V_{p}) and mean pore radius (r^-). The computed values of S_{BET} , V_{p} and r^- are given in Table 2.

Examination of Table 2 and Fig. 5 shows the following: (1) the S_{BET} and V_{p} values of pure and variously doped solids increased progressively by increasing the calcination temperature within 300–500 °C. The increase was, however, more pronounced for pure mixed solids which attained 113 and 295 % for S_{BET} and V_{p} , respectively [27]. (2) The r^- values cited in the last column of Table 2 show that the investigated solids are mesoporous adsorbents measuring (r^-) values varying between 30 and 88 Å. (3) La₂O₃-doping decreased effectively the S_{BET} to an extent directly proportional to its amount present. The doping process did not affect the r^- values which remained almost constant for heavily doped sample (3 mol% La₂O₃). (4) Most of the investigated solids exhibit bimodal pore volume distribution curves except the heavily doped sample (3 mol% La₂O₃) exhibited tri-modal distribution curves (c.f. Figure 5). The maximum hydraulic pore radius was located at 11.5 and 27.5 Å for pure solids and found at 10 and 23 Å for solids doped by 0.75 mol% La₂O₃ and 9.5 and 30 Å for solids doped by 1.5 mol% La₂O₃ and 10, 16 and 26 Å for solids doped by 3 mol% La₂O₃ being calcined at 500 °C.

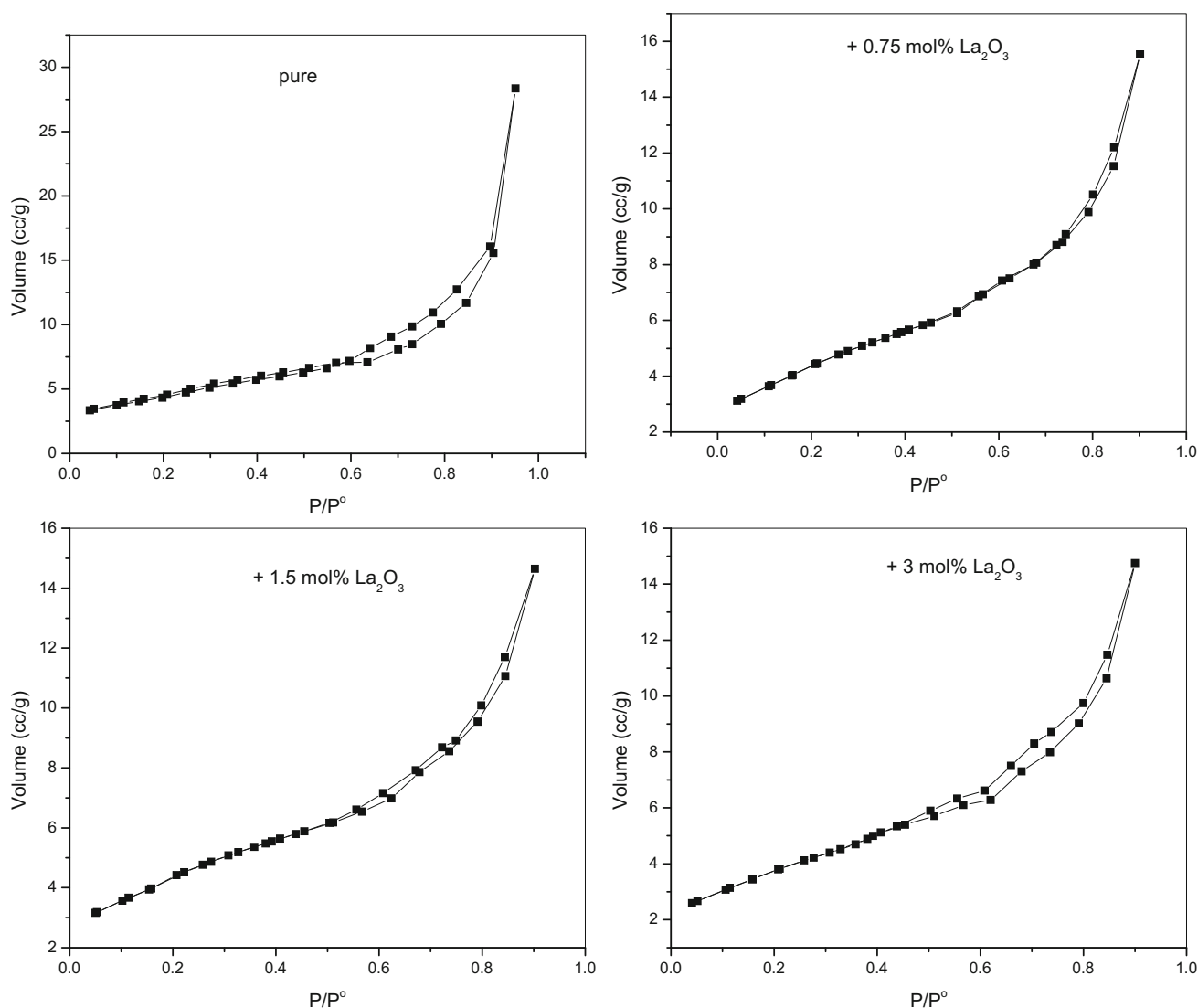


Fig. 4 N_2 -adsorption–desorption isotherms measured over pure and variously doped solids calcined at 500 °C

These results show clearly the role of La_2O_3 -doping in modifying the various surface characteristics of the system investigated. Comparison of (r^-) values of pure and variously doped solids calcined at 300–500 °C decreased effectively the calculated (r^-) values. This decrease might be followed by an increase in the S_{BET} values opposite to what was found. So, one might expect that the dopant process decreased the concentration of the narrowest pore located at 11.5 Å.

Energy dispersive X-ray analysis of various solids

EDX investigation of pure and doped solids calcined at 300–500 °C was determined. The relative atomic abundance of manganese, cobalt, oxygen and lanthanum species present in the uppermost surface layers of the calcined solids is given in Table 3. It is well known that EDX

technique supplies an accurate determination of relative atomic concentration of different elements present on their outermost surface layers [28–34]. In fact, this technique (EDX) has been successfully employed in determining the surface composition of a big variety of catalytic systems such as CuO/Mn_2O_3 [28], CuO/ZnO [29, 31, 33], TiO_2/Al_2O_3 [30], CuO/NiO [32] and Co_3O_4/Fe_2O_3 [34]. The thickness of these layers is bigger than those measured by using XPS technique. XPS is a well-known surface-sensitive technique that supplies very accurate relative atomic abundance of cationic and anionic species on the surfaces of investigated solids. Surface and bulk compositions of various solids are given in Table 3. Inspection of the results given in Table 3 reveals the following: (1) the surface composition of pure and variously doped solids is different from those of their bulk. (2) The surface concentration of cobalt species is always smaller than that presents in the

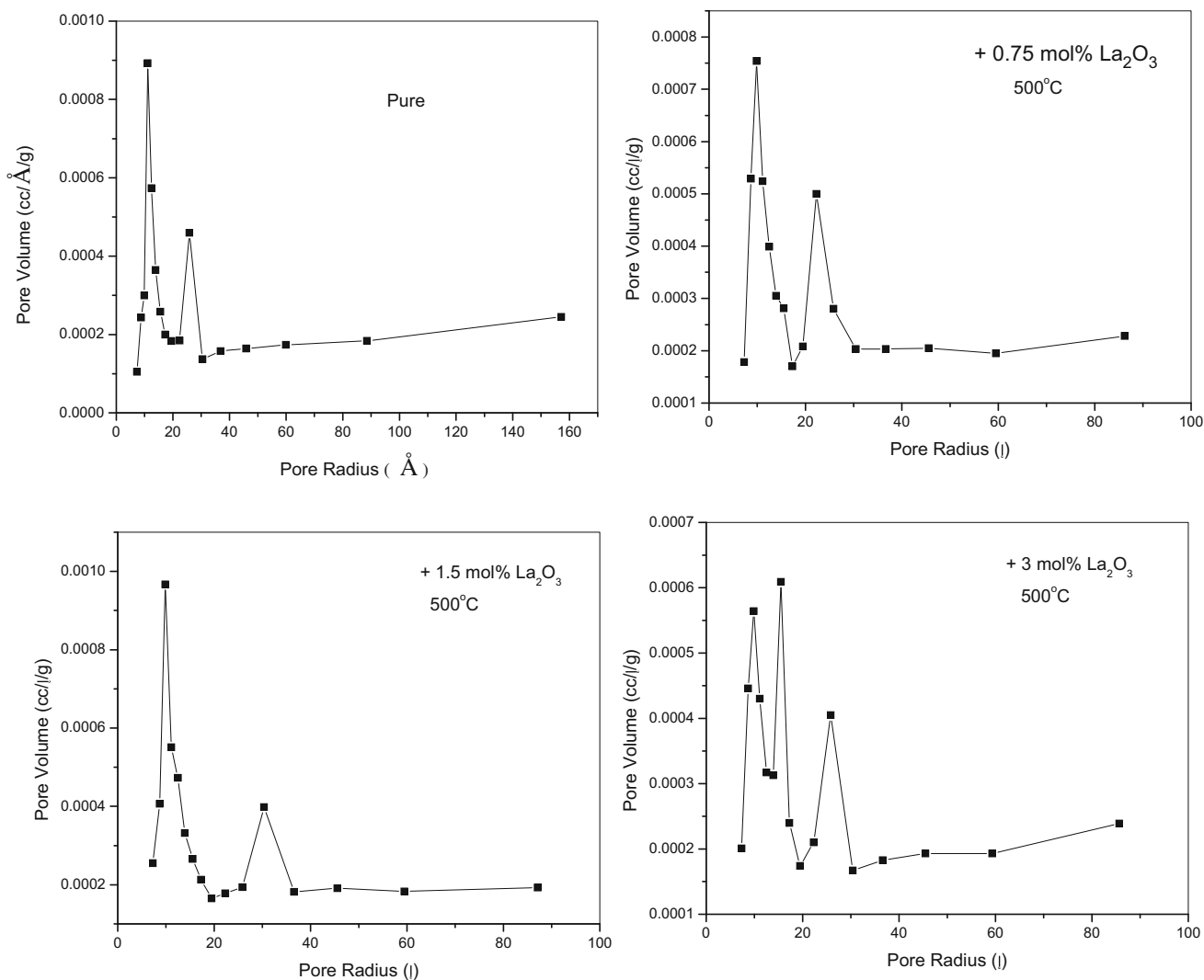


Fig. 5 Pore volume distribution curves $\Delta v/\Delta r$ for pure and variously doped solids calcined at 500 °C

Table 2 Surface characteristics of pure and variously treated solids being calcined at 300–500 °C

| Solids | Calcination temperature (°C) | S_{BET} (m^2/g) | Total pore volume V_p (cc/g) | Mean pore radius r^- (Å) |
|---|------------------------------|--|--------------------------------|----------------------------|
| Pure (0.5 mol CoO + 0.5 mol Mn_2O_3) | 300 | 16 | 0.038 | 48 |
| | 400 | 17 | 0.045 | 52 |
| | 500 | 34 | 0.15 | 88 |
| +0.75 mol% La_2O_3 | 300 | 13 | 0.021 | 32 |
| | 400 | 14 | 0.022 | 31 |
| | 500 | 16 | 0.024 | 30 |
| +1.5 mol% La_2O_3 | 300 | 12 | 0.011 | 18 |
| | 400 | 14 | 0.015 | 21 |
| | 500 | 15 | 0.016 | 21 |
| +3 mol% La_2O_3 | 300 | 10 | 0.016 | 32 |
| | 400 | 12 | 0.020 | 33 |
| | 500 | 14 | 0.023 | 33 |



Table 3 Surface molar composition of pure and variously treated solids determined by EDX

| Solid | Calcination temperature (°C) | Element | Atomic abundance (at.%) | | Mn/Co ratio (surface) |
|--|------------------------------|---------|-------------------------|---------|-----------------------|
| | | | Bulk | Surface | |
| Pure (0.5 mol CoO + 0.5 mol Mn ₂ O ₃) | 300 | Mn | 28.57 | 37.0 | 33.6 |
| | | Co | 14.29 | 1.1 | |
| | | O | 57.14 | 62.0 | |
| | 400 | Mn | 28.57 | 38.6 | 35.1 |
| | | Co | 14.29 | 1.1 | |
| | | O | 57.14 | 60.3 | |
| | 500 | Mn | 28.57 | 56.0 | 37.3 |
| | | Co | 14.29 | 1.5 | |
| | | O | 57.14 | 42.5 | |
| +0.75 mol% La ₂ O ₃ | 300 | Mn | 28.39 | 38.3 | 47.9 |
| | | Co | 14.19 | 0.8 | |
| | | O | 57.21 | 60.64 | |
| | 400 | Mn | 28.39 | 48.2 | 53.5 |
| | | Co | 14.19 | 0.9 | |
| | | O | 57.21 | 50.54 | |
| | 500 | Mn | 28.39 | 63.3 | 70.3 |
| | | Co | 14.19 | 0.9 | |
| | | O | 57.21 | 35.44 | |
| +1.5 mol% La ₂ O ₃ | 300 | Mn | 28.21 | 38.0 | 46.5 |
| | | Co | 14.11 | 0.8 | |
| | | O | 57.26 | 60.74 | |
| | 400 | Mn | 28.21 | 38.9 | 47.4 |
| | | Co | 14.11 | 0.82 | |
| | | O | 57.26 | 59.74 | |
| | 500 | Mn | 28.21 | 42.2 | 48.0 |
| | | Co | 14.11 | 0.88 | |
| | | O | 57.26 | 56.34 | |
| +3 mol% La ₂ O ₃ | 300 | Mn | 27.85 | 37.6 | 41.8 |
| | | Co | 13.93 | 0.9 | |
| | | O | 57.38 | 60.4 | |
| | 400 | Mn | 27.85 | 38.6 | 42.0 |
| | | Co | 13.93 | 0.92 | |
| | | O | 57.38 | 59.28 | |
| | 500 | Mn | 27.85 | 40.9 | 43.1 |
| | | Co | 13.93 | 0.95 | |
| | | O | 57.38 | 56.95 | |
| | | La | 0.84 | 1.2 | |



bulk of pure and variously doped solids. (3) The concentration of surface manganese was several folds that of cobalt. This finding might suggest that cobalt hydroxide was precipitated much earlier than manganese hydroxides. This conclusion seems logical because of the significant difference between the values of the solubility products of $\text{Co}(\text{OH})_3$ and $\text{Mn}(\text{OH})_2$ which measured 5.9×10^{-15} and 1.6×10^{-44} , respectively. (4) Lanthanum species present in the uppermost surface layers of all doped solids calcined at temperature within 300–500 °C is bigger than the amount present in the bulk of solids. This finding is expected because all doped solids were prepared by wet impregnation method [35, 36]. Furthermore, the surface concentration of lanthanum increases by increasing the calcination temperature of the doped solids. (5) The addition of the smallest amount of La_2O_3 (0.75 mol%) much increased the surface concentration of manganese present in all solids calcined at 300–500 °C. The increase in surface concentration of manganese species attained 4, 25 and 13 % for the solids calcined at 300, 400 and 500 °C, respectively. The increase of the dopant concentration above 0.75 mol% La_2O_3 decreased the surface concentration of manganese which still remained bigger than that measured for the un-doped samples calcined at the same temperatures.

It is well known that manganese species present in the uppermost surface layers in pure and doped solids are considered as the most catalytically active constituent involved in H_2O_2 -decomposition. This assumption is based on the possible presence of manganese cations in different oxidation states varying between di- and hepta-valence states leading to a significant increase in the concentration of manganese ion pairs participating in the catalytic decomposition process. So, the bigger the surface concentration of manganese the bigger will be the concentration of the most catalytically active constituent and the bigger the catalytic activity. This speculation will be confirmed in the next section of the present work dealing with the catalytic decomposition of H_2O_2 on pure and variously La_2O_3 -doped solids calcined at 300–500 °C.

Catalytic activity of pure and variously doped solids

The catalytic decomposition of H_2O_2 in aqueous solution was studied at 30, 40 and 50 °C over pure and variously doped solids precalcined at 300, 400 and 500 °C. First-order kinetics was observed in all cases. In fact, straight line was found upon plotting $\ln a/(a-x)$ against the time intervals t , where a is the initial concentration of H_2O_2 and $a-x$ is its concentration at time t . The slopes ($d \ln a/(a-x) / dt$) of plots determine the reaction rate constant (k) for the reaction conducted at a given temperature over a given catalyst sample. The reaction kinetics were monitored using

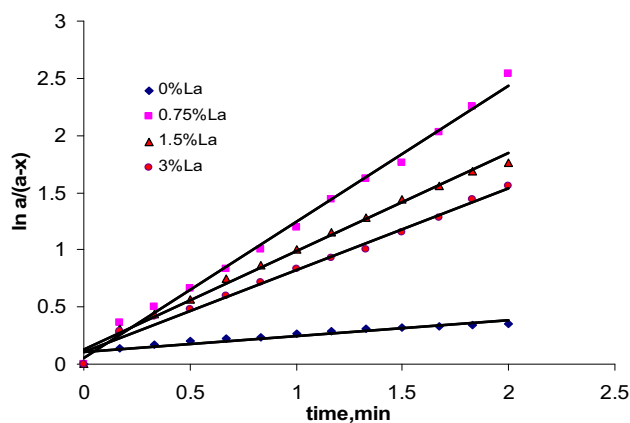


Fig. 6 First-order plots for catalytic decomposition of H_2O_2 at 50 °C, over pure and variously doped solids calcined at 500 °C

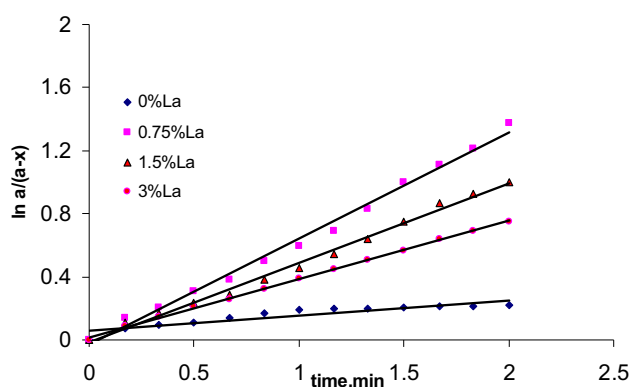


Fig. 7 First-order plots for catalytic decomposition of H_2O_2 at 50 °C, over pure and variously doped solids calcined at 300 °C

25, 50 and 100 mg of catalyst sample. The results, not given, showed that k value increases linearly by increasing the catalyst mass indicating the absence of any possible solid gas diffusion of liberated oxygen. Figures 6 and 7 depict representative first-order plots of the catalyzed reaction conducted at 30 and 50 °C over pure and variously La_2O_3 -doped solids pre-calcined at 300 and 500 °C, respectively. The values of the reaction rate constant per unit mass for the reaction carried out at 30, 40 and 50 °C were calculated from the slope of the first-order plots. Table 4 includes only the computed values of $k_{50 \text{ °C}}$. Inspection of Table 4 reveals the following: (1) the catalytic activity, expressed as reaction rate constant per unit mass, measured for pure and variously doped solids increased by increasing the calcination temperatures of all solids investigated calcined at 300–500 °C. (2) The presence of the smallest amount of La_2O_3 (0.75 mol%) increased considerably the (k) values. The increase reached about 10^3 -fold for the catalytic reaction carried out at 30 and 50 °C. So, the comparison between the role of ZrO_2 , as shown in our

Table 4 Reaction rate constant per gram catalyst, $k_{50^\circ\text{C}}^*$ ($\text{min}^{-1} \cdot \text{g}^{-1}$) $\times 10^3$ of catalytic decomposition of H_2O_2 , activation energy (ΔE) for the catalytic reaction carried out at 50°C over pure and variously La_2O_3 -doped solids calcined at $300\text{--}500^\circ\text{C}$

| Solid | Calcination temperature ($^\circ\text{C}$) | $k_{50^\circ\text{C}}^*$ ($\text{min}^{-1} \text{g}^{-1}$) | $\Delta E \times 10^3$ (kJ mol^{-1}) |
|---|--|--|---|
| Pure (0.5 mol CoO + 0.5 mol Mn_2O_3) | 300 | 1.24 | 5.2 |
| | 400 | 1.4 | |
| | 500 | 1.66 | |
| +0.75 mol% La_2O_3 | 300 | 13.6 | 1.5 |
| | 400 | 20.0 | |
| | 500 | 22.4 | |
| +1.5 mol% La_2O_3 | 300 | 10.2 | 2.0 |
| | 400 | 14.0 | |
| | 500 | 16.4 | |
| +3 mol% La_2O_3 | 300 | 7.2 | 2.7 |
| | 400 | 12.0 | |
| | 500 | 13.6 | |

previous studied [27], and La_2O_3 -doping of $\text{Co}_3\text{O}_4\text{--Mn}_3\text{O}_4$ system showed clearly that the increase in the catalytic activity due to the doping process was much more pronounced by doping with La_2O_3 as compared to ZrO_2 -doping [27]. (3) Increasing the dopant concentration above 0.75 mol% decreased the catalytic activity to an extent proportional to the amount of La_2O_3 added. (4) The observed increase in the catalytic activity of pure and variously doped solids by increasing their calcination temperature within $300\text{--}500^\circ\text{C}$ might be attributed to the observed increase in surface concentration of manganese species (considered as the most catalytically active constituent) and the observed increase in the specific surface areas (c.f. Tables 2, 3). (5) The observed significant increase in the catalytic activity due to doping with 0.75 mol% La_2O_3 could be also attributed to the observed decrease in the crystallite size of Co_2MnO_4 phase (c.f. Table 1). (6) The decrease in the catalytic activity of the heavily doped solids might result from the observed increase in the crystallite size of Co_2MnO_4 (the major phase present in pure and doped solids) and also due to the observed decrease in surface concentration of manganese species besides the significant decrease in the S_{BET} (c.f. Tables 1, 2, 3).

In order to throw more light about the role of both calcination temperature and dopant concentration (La_2O_3) in the mechanism of the catalyzed reaction the activation energy of which (ΔE) was determined for pure and doped solids calcined at $300\text{--}500^\circ\text{C}$. ΔE values were calculated from the values of k measured for the reaction carried out at 30, 40 and 50°C by direct application of the Arrhenius equation. The computed ΔE values are given in Table 4. Examination of Table 4 shows that ΔE values for pure and

variously doped solids decreased progressively as a function of dopant concentration. This trend ran parallel to observed increase in the catalytic activity, expressed as $k_{50^\circ\text{C}}$ values (c.f. Table 4). This finding expresses the observed increase in the catalytic activity of sample of 0.75 mol% La_2O_3 -doping. On the other hand, increasing the dopant concentration above this limit increased ΔE values which remained almost smaller than ΔE values measured for the pure catalyst samples.

Conclusions

The main conclusions derived from the results obtained can be summarized as follows:

1. The role of La_2O_3 -doping on structural, textural, surface composition and catalytic activity was investigated.
2. The prepared mixed solids consisted of nano-sized Co_2MnO_4 as a major phase together with un-reacted portion of Co_3O_4 and $\gamma\text{-Mn}_3\text{O}_4$.
3. Doping with La_2O_3 (0.75 mol%) much increased both surface molar ratio of Mn/Co with (0.75 mol%) about 88 % with subsequent increase in the catalytic activity more than 12-fold.
4. The apparent activation energy of the catalyzed reaction measured 5.2, 1.5, 2.0 and 2.7 kJ/mol for the un-doped and catalyst doped with 0.75, 1.5 and 3 mol% La_2O_3 .

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