# All-SiC Fiber-Optic Sensor Based on Direct Wafer Bonding for High Temperature Pressure Sensing

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**Abstract:** This paper presents an all-SiC fiber-optic Fabry-Perot (FP) pressure sensor based on the hydrophilic direct bonding technology for the applications in the harsh environment. The operating principle, fabrication, interface characteristics, and pressure response test of the proposed all-SiC pressure sensor are discussed. The FP cavity is formed by hermetically direct bonding of two-layer SiC wafers, including a thinned SiC diaphragm and a SiC wafer with an etched cavity. White light interference is used for the detection and demodulation of the sensor pressure signals. Experimental results demonstrate the sensing capabilities for the pressure range up to 800 kPa. The all-SiC structure without any intermediate layer can avoid the sensor failure caused by the thermal expansion coefficient mismatch and therefore has a great potential for pressure measurement in high temperature environments.

Keywords: SiC; pressure sensor; optical-fiber; high temperature; direct bonding

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# 1. Introduction

High temperature pressure sensors that can precisely monitor static and dynamic pressures are of great significance for various industrial and aerospace systems under extremely harsh environments, such as turbine engines and high-speed aircraft [1]. For example, in a jet engine, the fuel efficiency and other performance of the jet engine can be improved by accurately monitoring the pressure in the combustion chamber in real time [2, 3]. Conventional silicon-based and even silicon-on-insulator (SOI)-based pressure sensors

cannot operate in environments with temperatures exceeding 500 °C due to high temperature deformation of the material [4, 5]. Silicon carbide (SiC) has been considered a better candidate than silicon for high-temperature applications due to its excellent mechanical robustness, chemical inertness, radiation resistance, and electrical stability at elevated temperatures [6-8]. SiC-based high-temperature pressure sensors based on piezoresistive and capacitive detection mechanisms have demonstrated sensing capabilities in the temperature range of 350 °C - 600 °C [9-12]. However, these sensors cannot perform well under

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harsh environments due to many inherent problems such as strong high temperature dependence, electromagnetic interference, and poor performance in long-term stability and repeatability.

Compared with the electrical sensors, fiber-optic pressure sensors are tolerant to extreme temperature and can offer immunity to electromagnetic interference (EMI), high sensitivity, electrical passivity, and good long-term stability [13–18]. diaphragm-based extrinsic Among them. Fabry-Perot interferometer (EFPI) sensors have been widely investigated for their high resolution and precision, ease of manufacture, and excellent survivability in harsh environments [19-24]. Pulliam et al. [25] proposed an all-SiC pressure sensor fabricated by a deposited SiC diaphragm and a single-crystal SiC substrate. However, the pressure sensor shows a nonlinear response due to the internal stress left in the membrane. Jiang et al. [26] reported an all-SiC pressure sensor fabricated via an ultrasonic vibration mill-grinding and a nickel diffusion bonding method. However, the leakage rate of the sensor cavity is significantly increased at elevated temperatures due to the imperfect bonding process.

In this paper, we propose an all-SiC EFPI pressure sensor based on the hydrophilic direct technology bonding for high temperature applications. The sensor head is formed by direct hermetically bonding of two-layer SiC wafers including a SiC diaphragm and a SiC wafer with an etched cavity. Scanning electron microscope (SEM) characterization and the tensile test are performed for evaluating the direct bonding quality. Since both the diaphragm and the substrate are composed of single crystal SiC and the bonding is performed without using any intermediate material, sensor failure attributed to thermal expansion mismatch and internal stress can be effectively avoided. In addition, the proposed all-SiC sensor can realize batch-production. Ultimately, the sensing capability of the as-fabricated sensor has been demonstrated and tested over a pressure range of 0kPa to 800kPa.

# 2. Design and principle of operation

Figure 1 schematically illustrates the configuration of the proposed SiC sensor and the transducing mechanism based on the EFPI sensing method. The size of the sensor head is  $8 \text{mm} \times 8 \text{mm} \times 0.555 \text{ mm}$ , and the multi-mode optical fiber is connected to the back of the sensor through an adhesive. The core and cladding of the multimode fiber have diameters of 62.5 µm and 125 µm,



Fig. 1 Sensor configuration and principle of operation: (a) schematic of the SiC fiber-optic pressure sensor and (b) FP cavity interference in the sensor and the interference spectrum.

respectively. The sensor head is formed by directly hermetical bonding of two SiC pieces, including a SiC diaphragm and a SiC substrate with an etched cavity with a diameter of 4mm. The bottom surface of the etched cavity (R1) and the inner surface of the SiC diaphragm (R2) form two reflective surfaces in the FP interference, as shown in Fig. 1(b). Light is passed through the optical fiber and partially reflected at the R1 reflective surface. The remaining light propagates through the air gap between the reflective surfaces of R1 and R2 and is reflected at the R2 reflective surface. The two reflections then propagate back through the same fiber and generate interference fringes. The resulting light signal is then transmitted into the detector where the signal is demodulated to determine the FP cavity length.

The intensity of the reflected light can be expressed as [17]

$$I(k) = I_0(k) \cdot \left[1 + \cos(2kL + \varphi_0)\right] \tag{1}$$

where  $k=2\pi/\lambda$  is the optical wave number,  $\lambda$  is the wavelength of light,  $I_0(k)$  is the background spectrum of the broadband source, *L* is the FP cavity length, and  $\varphi_0$  is the constant phase term related to the multimode excitation of the FP cavity. Based on (1), *L* can be extracted from the corresponding spectrum.

When a uniformly distributed external pressure

is applied to the diaphragm, the diaphragm deforms and the FP cavity length changes. According to the theory of elastic mechanics, the central deflection y of the sensitive diaphragm with an applied external pressure can be expressed as [16]

$$y = \frac{3Pr^4(1-v^2)}{16Et^3}$$
(2)

where P is the external pressure, E is the Young modulus, v is the Poisson's ratio, and r and tare the radius and thickness of the membrane, respectively. Therefore, the external pressure can be measured by detecting the change in the length of the FP cavity.

Finite element simulation of the proposed sensor is performed by the COMSOL software to understand deflection response of the SiC diaphragm over a range of pressures. The sensitive model unit size is  $8 \text{ mm} \times 8 \text{ mm}$ , the thickness of the SiC substrate is  $355 \mu m$ , the thickness of the SiC diaphragm is  $200 \mu m$ , and the diameter and depth of the FP cavity are 4 mm and 15  $\mu m$ , respectively. Figure 2 shows the displacement deflection of the SiC diaphragm at an external pressure of 2 MPa. Figure 2(b) clearly shows the change in displacement in the central axis of the the SiC sensitive diaphragm. It can be seen that the maximum deflection of about 2.2  $\mu m$  occurs at the center of the diaphragm.



Fig. 2 Simulation of the SiC sensor model: (a) displacement simulation of SiC diaphragm under external pressure and (b) displacement distribution in the central axis of the diaphragm.

# 3. Sensor fabrication

#### 3.1 SiC etching

A double-side polished 3-inch SiC wafer with the thickness of 355 µm and root-mean-square roughness of ~0.48 nm (Si surface) is employed in the SiC wafer etching process. The manufacturing process flow of the SiC cavity is schematically depicted in Fig. 3. To begin with the process, photoresist is spin-coated and photopatterned on the SiC substrate, as shown in Figs. 3(a) and 3(b). Subsequently, a 50nm Cr metal film and a 200nm Cu metal film are sputtered on the SiC wafer by using a magnetron sputtering device (Denton Vacuum, EXPLORER) as an adhesion layer and a seed layer, respectively [Figs. 3(c) and 3(d)]. After that, stripping is performed by immersing the SiC wafer in an acetone solution for sonication. In this way, the photoresist and the metal covering the photoresist can be removed to expose the area to be etched [Fig. 3(e)]. In order to etch a deeper cavity, a Ni metal layer with the thickness of 2.5 µm is electroplated on the SiC substrate as an etching mask by placing the wafer into the Ni plating solution, as exhibited in Fig. 3(f). Then, the SiC wafer is loaded into a magnetic neutral loop discharge (NLD) plasma device for etching. In the NLD etching step, the parameters are as follows:  $SF_6$  and  $O_2$  with gas flows of 120 sccm and 50 sccm, pressure of 0.67 Pa, RF power of 1250W, and bias power of 250W. After etching for 20min, a series of shallow cylindrical cavities with the 4mm diameter are etched on the SiC wafer [Fig. 3(g)]. Finally, Ni etching solution (HCl:HNO<sub>3</sub>:H<sub>2</sub>O=3:1:2), Cu etching solution (HF:HNO<sub>3</sub>=1:1), and Cr etching solution (22% ammonium cerium nitrate:8% acetic acid:70% H<sub>2</sub>O) are used to remove the remaining metal mask [Fig. 3(h)]. After removing the mask, the depths of the etched cavities are tested by a step meter to be  $14 \,\mu\text{m} - 15 \,\mu\text{m}$ .



Fig. 3 Etching process flow chart of the SiC cavity by NLD: (a) coating photoresist, (b) patterning by photolithography, (c) sputtering Cr (50 nm), (d) sputtering Cu (200 nm), (e) stripping, (f) electroplating Ni (2.5  $\mu$ m), (g) etching SiC cavity by NLD, and (h) remove remaining mask.

#### **3.2 SiC direct bonding**

Prior to direct bonding, the SiC wafer with etched cavities is cut into  $8 \text{ mm} \times 8 \text{ mm}$  square pieces in order to reduce surface warpage. A flat SiC wafer is thinned to 200 µm as a SiC diaphragm to

mate with the etched sheet. SiC direct wafer bonding mainly includes three main steps: (1) wafer surface cleaning, (2) wafer plasma activation and pre-bonding, and (3) wafer high temperature bonding. Since surface cleaning is critical to the success of SiC direct bonding and the corrosion resistance of SiC, the surface wet chemical cleaning is performed according to the following cleaning method. First, the samples are immersed in the sulfuric /peroxide mixture (SPM) (H<sub>2</sub>SO<sub>4</sub>:H<sub>2</sub>O<sub>2</sub>=3:1) at 120 °C for 20 min to remove the organic and metallic impurities, followed by soaking in a mixed solution (NH<sub>3</sub>H<sub>2</sub>O:H<sub>2</sub>O<sub>2</sub>:H<sub>2</sub>O=1:2:7) at 75 °C for 20 min to remove the particles. At the final stage of cleaning, the SiC wafers are dipped again in the mixture of H<sub>2</sub>SO<sub>4</sub> and H<sub>2</sub>O<sub>2</sub> for the hydrophilic treatment. After then, all the samples are rinsed using de-ionized water and blow-dried using high pure nitrogen.

For achieving hydrophilic pre-bonding of the SiC wafers, the cleaned SiC pieces are introduced to the plasma system (PVA Tepla IoN40) for surface activation by oxygen plasma. Parameters in this activation process are as follows, power of 200 W, chamber pressure of 200 mTorr, and activation duration of 45 s. Subsequently, the paired SiC pieces are dipped in methanol with a higher -OH concentration to increase the amount of hydroxyl groups on the wafers. Then, the paired SiC wafers are brought into contact immediately in air with their crystalline axes aligned. After the pre-bonding, the bonding strength is very low due to the paired SiC wafers being held together via hydrogen bonds and the polar -OH groups. To enhance the bonding strength, the pre-bonded samples are loaded into a graphite die, as shown in Fig.4, and introduced into a high-temperature hot-press furnace with an ultimate vacuum of  $5 \times 10^{-3}$  Pa for high-temperature bonding. The direct bonding temperature curve of the pre-bonded SiC pieces is illustrated in Fig. 5. The temperature is raised from room temperature to 200 °C at a heating rate of 10 °C/min and then heated to 1400 °C at a heating rate of 15 °C/min – 20 °C/min for 2 h. During the holding period, a pressure of 4 MPa is applied on the paired wafers. The SiC sample after high temperature bonding is shown in Fig. 5, and visual inspection reveals that all contacted areas of two SiC pieces are tightly bonded **Photonic Sensors** 

together with no sign of voids or cracks.

After direct bonding, the multimode fiber is cleaved and inserted into a silica ferrule, and vertically glued on the back surface of the SiC sensor by alumina ceramic adhesive.



Fig. 4 Photograph of the graphite die and the bonded SiC sample.



Fig. 5 SiC high temperature direct bonding curve.

#### 4. Results and discussion

#### 4.1 SEM characterization

In order to observe the internal structure of the SiC sample, SEM observation is performed after etching and direct bonding. Figure 6 shows the surface and cross section of the SiC wafer after etching. Figure 6(a) shows that the bottom surface of the etched cavity is flat and smooth, which can meet the requirements of optical flatness. It can be clearly seen from Figs. 6(c) and 6(d) that the sidewall of the etching cavity is smooth and has a very good verticality with an angle of about 92.2°. The depth

of the etched cavity is approximately  $14.83 \mu m$ , which is consistent with previous step measurements.

To further investigate the internal quality of the direct bonding interface, the bonded SiC sample is cut along the cross section using an ultraviolet laser and observed under an SEM, as shown in Fig. 7. Surprisingly, the bonding interface is not clearly visible, indicating that the two SiC sheets have been fused into one another after high temperature

bonding. Moreover, the FP cavity formed by SiC direct bonding is intact and still keeps at a height of about  $15 \,\mu$ m, as illustrated in Fig.7(b). In addition, the elements analysis of the direct bonding interface by energy dispersive spectrometer (EDS) attached to the SEM is performed, as shown in Fig. 8. The elemental composition is carbon, silicon, and gold, and the presence of the gold element is attributed to the sputtered gold film for the SEM cross section observation.



Fig. 6 SEM images of the SiC cavity by NLD.



Fig. 7 SEM images of the SiC direct bonding interface.



Fig. 8 EDS analysis of the SiC direct bonding interface.

#### 4.2 Bonding strength

For evaluating the bonding strength of the SiC bonded sample, a tensile measurement is performed

using the tensile tester (Instron 2710, 50 kN). As shown in Fig. 9(a), the bonded SiC samples are glued to the vertical jigs on both sides and then loaded into a tensile tester for testing. The fracture

position is located at the direct bonding interface, as exhibited in Fig. 9(c). And the bonding strength of 6.25 MPa is achieved for the SiC direct bonded sample. This value exceeds the minimum bond strength (4 MPa – 5 MPa) required for micro electromechanical system (MEMS) device fabrication. Furthermore, the bonding strength and sealing performance of the bonded SiC sample have been successfully demonstrated by applying them to a pressure test at room temperature in the following section.



Fig. 9 Tensile test of SiC direct bonded sample.

#### 4.3 Pressure response test

To demonstrate pressure measurement, the white-light interference system for sensor interrogation constructed is by using the configuration shown in Fig. 10. The optical interrogation system is briefly configured by a halogen lamp light source, a 3dB fiber coupler, and an ocean optics spectrometer. The broadband light from the halogen lamp propagates along the fiber and delivers to the FP cavity of the sensor, and the interference signal is routed back to the ocean optics spectrometer for measurement via a 3 dB fiber coupler. Spectral interrogation and frequencyestimation-based signal-processing algorithm [27] are used to process the interference signal to demodulate the FP cavity length. During testing, the sensor is placed in a sealed pressure chamber through a fiber optic adapter. The pressure in the chamber is controlled by an air pump and a pressure regulator. When the pressure in the pressure chamber increases, the cavity length of the proposed sensor reduces. Consequently, the pressure can be measured by monitoring the cavity length change.



Fig. 10 Schematic of the pressure measurement system.

Figure 11 depicts the sensor performance at room temperature and under high temperature environments. Figure 11(a) shows the curves of the FP cavity length response to external pressure during three loading pressure experiments at room temperature under 0kPa - 800kPa. It can be noted that the pressure response curves exhibit a good repeatability and the error is approximately 1.42%. Figure 11(b) shows sensor pressure response curves for loading and unloading pressure conditions over three cycles. It can be observed that the cavity length changes linearly with the pressure over the entire test range and exhibits a good recoverability. The sensor returns to the same initial cavity length of 14.766  $\pm$  0.010 µm (maximum variation) at approximately 0kPa after each cycle. The pressure sensitivities of three tests are 1.84 nm/kPa, 1.83 nm/kPa, and 1.83 nm/kPa in the pressure range of 0 kPa - 800 kPa. And the maximum hysteresis error is 0.41%. In addition, the sensor is retained at 800 kPa for 15 h, and its cavity length shows no significant drift. This long-term test of sensing cavity leakage, coupled with the phenomenon that the sensor returns to the initial cavity length after each test cycle, indicates that the bonded cavity is tightly sealed. In addition, the high temperature performance of the sensor is verified under temperature-pressure composite environments. Figure 11(c) shows the sensor pressure response curves at 23  $^{\circ}$ C – 400  $^{\circ}$ C under 0kPa – 800kPa. The results show that the cavity length changes linearly with pressure at different temperatures. The sensitivity of the sensor remains approximately

unchanged, with a value of 1.84 nm/kPa. Also, the initial cavity length exhibits temperature dependence, plotted against temperature, as shown in Fig. 11(d). We can observe that the initial cavity length of the sensor increases with increasing temperature, and the cavity length changes approximately linearly with the temperature with a coefficient of

0.617 nm/°C, which is mainly caused by the thermal expansion of the SiC material. Finally, in order to explore the stability of the sensor under high temperature environments, the sensor is retained at 400 °C and 0 kPa for 30 min. The curve of the cavity length versus time is shown in the inset of Fig. 11(d), and the cavity length has no significant change.



Fig. 11 Test results of the fabricated sensor: (a) curves of the FP cavity length response to external pressure during three loading pressure experiments, (b) sensor pressure response curves for loading and unloading pressure conditions over three cycles, (c) sensor pressure response curves at  $23^{\circ}$ C - 400 °C temperature and under 0 kPa - 800 kPa pressure, and (d) FP cavity length as a function of temperature under 0 kPa pressure.

#### 5. Conclusions

In this study, an all-SiC sensor structure for pressure test is proposed and demonstrated. The sealed pressure cavity is fabricated by a combination of NLD etching of the SiC wafer and SiC direct wafer boding. SEM characterization indicates that the sidewall of the etching cavity is smooth and has a very good verticality. In addition, the cavity structure after direct wafer bonding remains intact and the interface has a good bonding quality. The tensile test indicates that the bonding strength of SiC wafer direct boding reaches 6.25 MPa. The white-light interference system for sensor

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interrogation is constructed, and the pressure sensor displays a good linear response in the range of 0 kPa - 800 kPa, with a repeatability error of 1.42% at room temperature. The high-temperature test results of the SiC sensor show that the sensor can work stably under high temperature environments up to 400 °C . In the future, the non-contact measurement method will be used to avoid the influence of high-temperature ceramic adhesive on the temperature resistance of the sensor, thereby increasing the operating temperature of the SiC sensor.

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