Glazed sgraffito ware from Torre Alemanna (Foggia, fifteenth to sixteenth century A.D.): technological aspects of a local production

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Abstract

The archaeometric investigation of 46 potsherds of “Torre Alemanna type” pottery aimed to define a compositional reference group and to understand the technological characteristics of its production. Principal component analysis applied to bulk chemical data (XRF) of the ceramic body showed a strong compositional homogeneity. Their comparison with local clays and 6 fragments of bricks sampled from the ceramic kiln, on the one hand, revealed the use of alluvial clays as raw material and, on the other hand, proved their fractionation for the production of pottery. The mineralogical assemblages detected by X-ray powder diffraction analysis inferred maximum firing temperatures between 750 and 1000 °C for the ceramic body. Polarising optical microscopy and scanning electron microscopy with energy dispersive spectroscopy (SEM–EDS) investigations on coating revealed the presence of quartz-rich white engobe covered with a high lead transparent glaze. The polychromatic decoration was analysed by colorimetry and EDS to compare the colour characteristics through all the potsherds. A temperature range between 780 and 950 °C of liquidus temperatures was inferred from the ternary phase diagram of PbO–Al₂O₃–SiO₂ system. Overlap of temperature ranges for sintering of the ceramic body and maturing glaze points to a single firing of the Torre Alemanna type ware. The results obtained define the reference compositional group and technology of the Torre Alemanna type ware, already attested in several archaeological contexts of southern Italy, and showed a technological continuity with the past about the use of local carbonate-rich clays to produce fine pottery.

Keywords Torre Alemanna · Lead glaze · Sgraffito · Colorimetry · Pottery · Apulia

Introduction

In the 1960s, thanks to the growing number of archaeological investigations on medieval contexts in Italy, the existing typological complexity of glazed ceramics began to come to light (Whitehouse 1978).

Although scholarly attention has mainly focused on the masterpieces of painterly maiolica of the Italian Renaissance (Liverani 1957; Whitehouse 1967; Whitehouse 1978; Goldthwaite 1989; Fabbri et al. 1990; Kingery 1993; Ruffini et al. 2005; Viti et al. 2003; Tite 1991, 2008; Antonelli et al. 2014), a multifaceted stylistic and technological world is emerging on the Italian ceramic production of that period

(Gardini and Mannoni 1995; Clark et al. 1997; Capelli et al. 2002; Comodi et al. 2004; Laganara Fabiano 2004; Ricci et al. 2005; Capelli and Cabella 2010; De Santis et al. 2012; Heimann et al. 2014).

From the end of the tenth century, there was a resumption of long-distance trade relations which allowed the circulation of coated types of ceramics from the Mediterranean area (Gelichi 1993). The strengthening of trade, banking and specialised manufacturing in the late Middle Ages, especially in central and northern Italy, determined the cultural and technological innovations that led to the definition of fineware maiolica and Medici porcelain (Kingery and Vandiver 1986; Goldthwaite 1989). The import of glazed ceramics from Maghreb and the eastern Mediterranean attests to changes in aesthetic taste and a growing demand for them. Byzantine pottery was a stylistic model for the widespread sgraffito technique in several Italian production centres, as well as Islamic pottery with painted ornaments on a white background and tin-glazed introduced aesthetic and technological aspects, which became typical of Italian maiolica.

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These influences, in a changing socio-economical context, turned some Italian villages from importers of glazed pottery to exporters within a short time span (Whitehouse 1986; Gelichi 1993; Arthur 2007).

In southern Italy, the political conditions for cultural and technological exchange occurred in the multi-ethnic and multi-religious Norman Kingdom of Sicily (1130–1194) (Randazzo 2019). Furthermore, Apulia and Sicily acquired a relevant political and economic role in the late Middle Ages also thanks to the Hohenstaufen dynasty (Buttigieg and Phillips 2016). Apulian *protomaiolica* (Waagé 1934), as well as other Italian pottery, was found in the Levant from the end of the thirteenth century as traded objects and/or part of the baggage of crusaders and pilgrims (Pringle 1982; Gelichi 1993).

The availability in Apulia of good fine clays, particularly from the Tavoliere plain (Balenzano et al. 1977; Dell'Anna and Laviano 1991; Dondi et al. 1992), and the early human civilisation of its entire territory implied a considerable production and exchange of ceramics, dating back to the Upper Neolithic (Levi et al. 1995; Laviano and Muntoni 2006; Muntoni and Laviano 2011) and developed in Roman times (De Benedetto et al. 2004; Eramo et al. 2004; Gliozzo et al. 2010; Gliozzo et al. 2018) and in the subsequent medieval period (Busto et al. 2001; Laganara Fabiano 2004; Favia 2012; Valenzano 2016).

In last decades, several papers have contributed to the knowledge of the production and circulation of glazed ceramics in southern Italy in the medieval and modern ages (Alaimo et al. 2004; dell'Aquila et al. 2006; Catalano et al. 2007; Giannossa et al. 2014; Caggiani et al. 2021).

Noteworthy in this context is the glazed sgraffito (incised) ware produced in the Torre Alemanna complex, where specific forms were produced, such as plates, two-handled cups, bowls and rare jugs. The characteristic decoration was made with a fine carving tool on a white engobe, revealing the light brown background. The decorative motifs are typically geometric and plant like, though some human, zoomorphic and heraldic motifs have also been attested, and colours are yellow–brown, purple and green; glaze drippings are also quite common. A double firing has been hypothesised to sinter the ceramic body and the painted engobe and then to mature the lead glaze (dell'Aquila 2015). Evidence of the diffusion of the Torre Alemanna type has been found in several archaeological contexts in the region (Bari, Canosa di Puglia, Bovino, Gravina, Trani, Laterza, Taranto, Lecce, Soleto) and also in neighbouring Campania (i.e. Montella, Salerno), Molise (Isernia) and Abruzzo (L'Aquila, Sulmona) regions (Salvatore 1980; Carbosiero and Magistrale 1994; Troiano and Verrocchio 2001; dell'Aquila 2015).

Further examples of “sgraffito” ceramics, very similar to the Torre Alemanna type, have been found in other centres in Italy, for example in several sites in Tuscany and in Udine

since the fifteenth century, where this type of production was attested in many artisanal kilns (Amato et al. 2006; Ricciardi et al. 2007).

The main objectives of this work are twofold: first, to confirm the archaeological hypothesis of local production and the technology used by the potters and, second, to define a compositional reference group for the Torre Alemanna type, to aid comparisons with other glazed pottery from the Mediterranean area. To achieve these objectives, all the samples were characterised from a mineralogical, petrographic and chemical point of view, and the data obtained with those of local clay sources available in the literature.

The building complex of Torre Alemanna

After the Teutonic Order's increased prestige following the Fifth Crusade (1218–1221), they obtained papal privileges and various donations to support the knights engaged in the Holy Land. The presence of the Teutonic Knights in Apulia was first established in Brindisi, one of the main ports on the Adriatic coast chosen by the crusaders, followed by Barletta and Bari and Torre Alemanna, located inland, and other minor houses (Buttigieg and Phillips 2016).

Torre Alemanna (Foggia, Apulia) is one of the most important medieval examples of the settlement of the Order of the Teutonic Knights in Southern Italy (thirteenth to the sixteenth century). Situated at the crossroads of two important transhumance routes in the Tavoliere plain, it became an extensive and rich farm. Its geographical position was strategic for producing and selling wheat, wine, oil and other products to ensure supplies and money to the Teutonic Knights engaged in the Holy Land. Until 1789, the fiefdom of Torre Alemanna, which was under the Church rule, was confiscated by the Kingdom of Naples (Busto 2000).

It is a building complex consisting of a square central tower, housing a church dating from the end of the thirteenth century, to which several residential buildings are attached. This central unit is surrounded by defensive walls and service and storage buildings (Busto 2008, 2012; Busto et al. 2015).

The site was abandoned for a long time and archaeological excavation only began in 1999 and covered the entire complex (Busto 2000). The archaeologists discovered and extracted many ceramic artefacts and other movable items from dump pits dated from the thirteenth to the seventeenth century (Busto 2008). The presence of a ceramic kiln, probably used to produce bricks and tiles, suggests a local pottery production. The oldest dump (thirteenth to fourteenth century) was found near the kiln and contained several open forms of glazed sgraffito ware, covered with green or colourless glaze and decorated with red and brown linear motifs.

The archaeological excavations conducted by the archaeological Authority of the Apulia region in 1999–2000 uncovered and extracted a number of imported maiolica pieces together with a large number of glazed sgraffito potsherds, dating from the fifteenth to the sixteenth century, found in a granary pit in the apartment house of the Abbot, reused as a dump and sealed during building alterations to the structure, dated to 1570 by an inscription. The maiolica pottery was marked with the symbol of Torre Alemanna production, then suggesting local glazed ware production (dell'Aquila 2015). Polychrome glazed pottery was previously unknown and its particular typological characteristics allowed the definition of “Torre Alemanna type” pottery (Busto et al. 2001).

Geological substratum

The Torre Alemanna building complex (DMS geographical coordinates: 41° 11' 08" N 15° 42' 46" E) is located in the Tavoliere di Puglia plain, a vast flat area bordering on the S with the Murgia plateau, on the W with the “Subappennino Dauno” (outer units of southern Apennine chain) and on the NE with the Gargano promontory. This area is part of the northern sector of the Bradanic foreland (Bradanic trough, Azzaroli et al., 1968), formed following the subduction of the Apulian platform under the Apennine chain (Middle Pliocene–Lower Pleistocene) (Ricchetti et al., 1988; Doglioni et al., 1994, 1996). The geological and geomorphological features of the Tavoliere are the result of two geodynamic phases that occurred from the Pliocene to the Holocene:

firstly, the accumulation of transgressive units on the Mesozoic limestone basement, from shallow marine carbonate deposits (Calcarene di Gravina Fm) to the silty–clayey hemipelagic deposits of the Argille subappennine Fm (Pieri et al., 1996; Tropeano et al., 2002), and secondly, the combination of glacio-eustatic sea-level fluctuations and polyphase uplift (0.1–0.5 mm/year), with the deposition of shallow marine and continental deposits on the Argille subappennine Fm, forming the Tavoliere di Puglia supersynthem from the Lower Pleistocene (Jacobacci et al., 1967; Merla et al., 1969; Caldara and Pennetta, 1991, 1993; De Santis et al., 2010; Moretti et al., 2010; Caldara et al. 2011; Gallicchio et al., 2014).

The available clayey raw materials in the Torre Alemanna area (Fig. 1) are essentially represented by the carbonate-rich Argille subappennine Fm (Lower Pleistocene) and alluvial deposits (Upper Pleistocene–Holocene) outcropping along river beds (Eramo et al. 2004; Gliozzo et al. 2005, 2018).

Sampling and analytical methods

All 46 potsherds analysed here were fragments of glazed sgraffito plates excavated from a dump in the apartment house of the Abbot, all datable from the late fifteenth to the sixteenth century (Table 1). The thickness of the walls is between 6 and 8 mm, while the maximum diameter of plates is 33 cm. A selection of the typical shapes and decorations is shown in Figures 2 and 3. Other 6 samples from the bricks (TAL1-6) of the ceramic kiln (Fig. 4), found in a building

Fig. 1 Geological map of the area of Torre Alemanna (De Santis et al. 2013, modified). Sampling points of alluvial clays (47: Ofanto valley; 22: Candela; 26, TFA: Carapelle valley; 27: clay quarry, Ascoli Satriano; 33: Faragola archaeological site; 34: Cappello Tosto;

30: Palazzo di Ascoli) and Argille subappennine Fm (24, 25, 28, 29, 39: Carapelle valley) considered reference for local clays (Gliozzo et al. 2018)

Table 1 The archaeological features of the analysed samples, including position in the vessel, thickness and colour coordinates (CIEL*a*b*) of the ceramic body. In parenthesis, the letter corresponds to the decorative motifs represented in Fig. 3. Abbreviations: TAt = Torre Alemanna type; ND = not determined

Sample	Typology	Position	Thickness (mm)	Body colour		
				L*	a*	b*
TA04	TAt	Rim (g)	6	70.9	10.2	20.1
TA05	TAt	Bottom (b)	5	67.3	10.5	31.5
TA06	TAt	Rim (d)	6	62.4	12.5	34.4
TA07	TAt	Wall (f)	7	65.2	10.7	20.0
TA08	TAt	Rim (d)	6	61.7	19.4	26.1
TA09	TAt	Rim (d, h)	9	71.3	8.0	22.8
TA10	TAt	Rim (d)	10	72.1	10.2	20.2
TA11	TAt	Rim (d)	7	72.6	12.4	16.9
TA12	TAt	Wall (h)	6	60.9	9.0	20.0
TA13	TAt	Rim (ND)	6	63.1	9.2	21.9
TA14	TAt	Bottom (d, h)	6	71.6	8.5	20.8
TA15	TAt	Rim (k)	5	68.6	10.4	19.1
TA16	TAt	Wall (h)	6	62.7	10.6	21.9
TA17	TAt	Wall (h)	6	69.9	11.1	24.1
TA18	TAt	Wall (h)	6	61.8	16.0	30.9
TA19	TAt	Rim (d, h)	7	62.5	16.8	20.0
TA20	TAt	Rim (d)	7	63.4	19.4	26.2
TA21	TAt	Rim (d)	6	65.2	12.1	20.0
TA22	TAt	Rim (ND)	4	60.3	4.3	11.5
TA23	TAt	Wall (d)	6	61.1	19.5	26.1
TA24	TAt	Rim (d)	7	62.1	8.2	15.0
TA25	TAt	Rim (ND)	7	30.8	8.1	30.0
TA26	TAt	Rim (ND)	6	64.8	10.8	20.0
TA27	TAt	Rim (ND)	4	61.9	10.3	21.9
TA28	TAt	Rim (ND)	6	63.0	10.8	18.7
TA29	TAt	Bottom (c)	6	41.2	15.0	17.6
TA30	TAt	Bottom (a)	6	65.8	16.0	30.9
TA31	TAt	Bottom (ND)	9	71.4	3.9	11.5
TA32	TAt	Wall (e)	6	64.2	12.8	17.0
TA33	TAt	Rim (ND)	7	65.2	19.4	26.7
TA34	TAt	Bottom (ND)	6	66.8	10.7	20.9
TA35	TAt	Rim (ND)	7	30.8	9.9	14.6
TA36	TAt	Bottom (ND)	7	62.9	8.5	22.5
TA37	TAt	Rim (l)	5	63.1	9.8	12.7
TA38	TAt	Rim (ND)	5	64.1	9.5	22.5
TA39	TAt	Rim (ND)	6	64.6	16.0	30.9
TA40	TAt	Rim (ND)	6	51.4	15.1	27.8
TA41	TAt	Rim (ND)	8	62.9	17.1	26.9
TA42	TAt	Rim (ND)	7	60.8	19.4	26.1
TA44	TAt	Rim (j)	6	41.2	15.0	17.6
TA45	TAt	Wall (ND)	8	62.8	11.0	20.3
TA46	TAt	Rim (ND)	9	59.2	10.4	17.9
TA47	TAt	Rim (h, i)	7	61.5	12.8	18.7
TA48	TAt	Rim (i)	6	41.2	15.0	17.7
TA49	TAt	Rim (i)	5	62.7	16.0	30.9
TA50	TAt	Rim (i)	6	63.6	19.4	26.1
TAL01	Brick	Firebox, wall (W sector, inside)	43	34.4	11.7	17.3
TAL02	Brick	Firebox, wall (W sector, inside)	49	38.9	15.5	18.5
TAL03	Brick	Firebox, domed roof (W sector, inside)	40	45.8	12.4	18.2

Table 1 (continued)

Sample	Typology	Position	Thickness (mm)	Body colour		
				L*	a*	b*
TAL04	Brick	Firebox, domed roof (W sector, outside)	42	30.5	14.4	24.6
TAL05	Brick	Firebox, domed roof (E sector, inside)	37	28.8	16.7	28.6
TAL06	Brick	Firebox, domed roof (E sector, outside)	38	35.4	14.2	21.3

Fig. 2 Representative samples of the Torre Alemanna type sherds found in the Abbot palace of Torre Alemanna (Cerignola, Foggia, Italy)

adjacent to the western defensive wall, were also analysed to compare their composition with that of the pottery (Busto 2000; Navarra 2008).

Colorimetry

Colorimetric analyses of the glaze were carried out with a Konica-Minolta CM-2600d spectrophotometer obtaining data in the CIEL*a*b* colour space. One hundred

and nineteen colour measurements were taken using the standard illuminant D65 (Xe flashlight source, UV included) over a measurement area of $\varnothing = 6$ mm (observer angle = 10°). Each measurement was acquired considering the specular component included (SCI) and was replicated three times to obtain the mean value. The measurements were performed on planar portions of the surface with homogeneous colour. A surface cleaning with 2% HCl water solution was conducted before colour measurements.

Fig. 3 Drawings of the decorative motifs identified on the analysed potsherds

Polarising optical microscope (POM)

Pottery fabric analysis was conducted on thin sections under a Carl Zeiss Axioskop 40 Pol polarising microscope (POM). The petrographic description included the compositional and textural aspects of the ceramic body, as well as those of the engobe and glaze, when present. The abundance of non-plastic inclusions (NPIs) and macro-porosity was obtained by visual estimation using comparison charts (Matthew et al. 1991), considering 15 μm as threshold

between NPIs and matrix (Maggetti 1982). Primary (pre-firing) and secondary (firing-induced) porosities were recognised under the POM. Following Eramo and Mangone (2019) and Eramo (2020), several aspects were recognised: birefringence, oxidation patterns, compositional features and clay pellets presence for the matrix; NPIs features in terms of nature, volume percentage and grain-size distribution for the ceramic body and the engobe; and vesiculation, presence of relic phases and alteration for the glaze.

Fig. 4 Position of the brick specimens sampled from the wall and roof of the ceramic kiln of Torre Alemanna

X-ray powder diffraction (XRPD)

X-ray powder diffraction analysis (XRPD) was performed by using a Philips X'Pert Pro X-Ray diffractometer under the following working conditions: Cu–K α Ni-filtered radiation, 40 kV, 40 mA, divergence slit 1°, anti-scatter slit 0.5°, receiving slit 0.2 mm, step angle of 0.02°, 2 θ from 2 to 65°, measuring time 1 s per step. The diffraction peaks of the XRPD spectra were identified by comparison with the Joint

Committee on Powder Diffraction International Centre for Diffraction Data (JCPDS–ICDD); diffraction chart and crystalline phases were detected.

Coatings and encrustations were mechanically removed from the analysed portions. The clean ceramic bodies were pulverised for 8 min at a frequency of 10 Hz in the vibratory ball mill Retsch MM 400, equipped with two WC grinding jars.

The degree of sintering ranges between “low-medium” and “medium-high”, according to their mineralogical characteristics and birefringence observed under POM. The range of the equivalent firing temperatures (EFTs, after Tite 1995) was estimated according to the thermal stability (disappearance/appearance) of the mineral phases and their peak ratios as detected by XRPD.

X-ray fluorescence (XRF)

Four grams of the sample powder (see above) were bound with Elvacite® dissolved in acetone and then dried for pill preparation. Pills were obtained by pressing the dry powder in small aluminium cups under a hydraulic press.

The determination of major and minor oxides (SiO₂, TiO₂, Al₂O₃, Fe₂O₃, MnO, MgO, CaO, Na₂O, K₂O and P₂O₅) and trace elements (Rb, Sr, Y, Zr, Nb) concentration was performed using an automatic spectrometer Panalytical AXIOS-Advanced, equipped with the X-ray tube X SST-mAX (Rh anode), following the analytical techniques outlined by Franzini et al. (1972, 1975) and Leoni and Saitta (1976a, b) and checking the accuracy with two international standards (AGV-1 of USGS-USA and NIMG of NIM-South Africa). Detection limit for major elements oxides was 0.01% wt. Loss on ignition (LOI) was determined by heating the samples at 1000 °C for 12 h.

Scanning electron microscopy (SEM–EDS)

The scanning electron microscope (SEM) used in this research was a 50XVP LEO, operated at 15 kV, 500 pA probe current, about 25,000 cps output as average count rate on the whole spectrum, counting time 50 s and 8.5 mm as working distance. Energy dispersive spectrometric (EDS) microanalyses were conducted using an X-Max N (80 mm²) SDD detector and AZtec software (Oxford Instruments) for X-ray maps and the correction of X-ray intensity was performed following Pouchou and Pichoir (1991). Different Micro-Analysis Consultants Ltd. (U.K.) mineral standards were used to check the accuracy of the analytical data. For the SEM investigation, 7 samples (TA09, TA29, TA38, TA42, TA44, TA45 and TA46) were selected after the macro- and microscopic analyses as representative of glaze colour variability and petrographic characteristics of

the coatings. Their thin sections were coated with graphite for SEM investigations.

Statistical analysis

Chemical data of TA (potsherds), obtained from XRF analysis, were compared with those of TAL (bricks) and with local clay sediments (Eramo et al. 2004, 2012, 2013; Gliozzo et al. 2018). To make these data, obtained in different laboratories, comparable, the concentrations of the major and minor oxides and trace elements were normalised to SiO₂ values. Multivariate statistical analysis of the chemical data was considered to verify the existence of correlations among variables of each dataset. Therefore, principal component analysis (PCA) was applied, after standardisation of normalised data to ensure the normal distribution and the same variance ($\mu=0$; $\sigma=1$) (Davis 1986, p. 517). The software Past 3 (Hammer et al. 2001) was used for the statistical processing and drawing of petrographic and chemical data.

Results

Macroscopic description

All the glazed potsherds belonged to open-shaped vessels, and on the concave surface, they showed signs of engraving that followed decorative motifs, from a geometric and vegetal repertoire, and yellow, brown, green and purple pictorial decorations, which usually followed the incised decorative motifs (sgraffito). Figure 3 shows the decorative motifs identified on the analysed potsherds. A thin layer of glaze was applied to all ceramic samples. Some appeared deteriorated due to devitrification and/or surface encrustation. The uncoated portions of the ceramic body showed signs of wheel throwing (Fig. 2). The thickness of fragments was 4–10 mm for rims, 6–8 mm for walls and 5–7 mm for bottoms. Some drippings due to the application of the engobe and glaze are recognisable on the back of the brim of the plates (Fig. 2). On the surface of some samples, black or grey burn shades were well visible, whereas no tripod marks were detected.

Colour analysis

The macroscopic observation was supported by colorimetric analysis of the ceramic body and glaze. Results, expressed in CIEL*a*b* chromatic space for each colour, are reported in Tables 1 and 2 and plotted in Fig. 5. Although colorimetric data of ceramic body referred to the predominant colour, in most of the samples of Torre Alemanna type a* and b* values gathered along the red range values and more sputtered along the yellow range values. Some samples showed a

colour more similar to that of the bricks (Fig. 5a). Referring to purple and green areas, data indicated a clear homogeneity of a* and b* values, as well as for colourless glaze on white engobe. In the yellow–brown cases, it appeared more dispersed along the yellow range values (Fig. 5b).

POM

Under the optical microscope, the 52 ceramic bodies showed 3 types of fabric. Full oxidation pattern (Str=O) is common to all the fabrics. A detailed report of the petrographic features is shown in Table 3.

The KQ_k fabric was observed in 39 potsherd samples (Fig. 6a) and 3 brick samples (Fig. 7a). They contained mainly about 20–30% vol of NPIs, which mostly included calcite and quartz, and in addition plagioclase, K-feldspar, muscovite, iron aggregates and polycrystalline quartz only in a few samples. The grain-size distribution was unimodal, with a mode of 32–63 μm for potsherds and 63–125 μm for bricks, although larger inclusions (up to 1000 μm) were observed. Matrix is carbonate-rich. The primary porosity ranges between 10 and 20% vol.

Fabric KQKa_k (Fig. 6b) is peculiar of 5 potsherd samples. The NPIs are about 20–30% vol in volume and were calcite, quartz and carbonate aggregates, with subordinate plagioclase, K-feldspar, muscovite and iron aggregates. Polycrystalline quartz was detected only in TA50. Unimodal distribution of NPIs has generally 16–32 or 32–63 μm as mode, even if fragments up to 800 μm were also identified. Matrix appeared marly. Primary porosity is about 10% vol.

Two samples of dishes (Fig. 6c) and 2 of bricks (Fig. 7b) show KQKa_kcp fabric. In those samples, the NPIs have unimodal distribution which was composed of calcite, quartz, carbonate fragments and following plagioclase, K-feldspar, muscovite and iron aggregates. NPIs reached size of 1400 μm . Matrix was marly and rich in clay pellets. Porosity was about 10–15% vol. In general, calcite grains show thermal alteration due to firing.

Petrographic investigation under POM identified the overlapping of white engobe and glaze layer on the ceramic body. The boundary between the engobe and the ceramic body is visible but not sharp (Figs. 6, 8). For all the observed samples, the engobe showed common characteristics—thickness of 100–200 μm , presence of monocrystalline quartz and minor feldspars with unimodal distribution (mainly 16–32 μm) and prevalent illite in the matrix—as evidenced by the K map in Fig. 8.

Lead glaze (“SEM–EDS” section) thickness ranges from 80 to 200 μm and is separated by a clear boundary from the engobe, except for TA04, TA34 and TA42, with a merging boundary. In most of the cases, traces of quartz relics and bubbles were observed (Table 3).

Table 2 Colour coordinates (CIEL*a*b*) of the polychromatic decorations of the coating

White (engobe)				Yellow–brown				Green				Purple			
Sample	L*	a*	b*	Sample	L*	a*	b*	Sample	L*	a*	b*	Sample	L*	a*	b*
TA04	80.2	−0.3	16.9	TA04	63.2	10.4	54.4	TA04	37.0	−12.9	9.9	TA04	26.8	6.1	3.2
TA04 bis	81.4	−2.7	15.4	TA05	55.5	11.2	40.6	TA05	53.2	−19.5	16.8	TA06	29.8	2.8	1.3
TA05	77.2	2.8	15.6	TA06	47.8	12.4	30.5	TA06	41.8	−13.9	13.2	TA08	27.3	1.0	−1.6
TA06	81.4	−0.4	21.9	TA07	62.9	4.5	42.9	TA07	42.8	−18.3	16.2	TA09	24.5	2.7	−3.6
TA06 bis	53.4	−1.1	14.5	TA08	54.8	17.0	50.5	TA08	47.6	−18.8	14.6	TA10	32.3	2.2	4.8
TA08	80.7	0.0	13.7	TA09	55.2	19.2	55.2	TA09	59.5	−16.8	17.7	TA11	30.4	0.1	−0.3
TA08 bis	82.4	−0.5	15.3	TA10	51.9	21.1	46.3	TA10	59.2	−14.0	20.2	TA12	31.1	4.9	4.9
TA09	79.2	3.8	17.8	TA11	53.5	17.4	45.9	TA11	47.9	−15.7	16.8	TA14	32.6	5.9	2.9
TA10	80.6	0.1	20.5	TA12	43.7	5.4	17.5	TA12	40.3	−18.0	10.2	TA14 bis	32.8	4.4	1.1
TA11	79.1	0.8	17.6	TA14	59.7	14.8	52.5	TA13	48.2	−16.8	22.0	TA16	16.2	0.3	−0.3
TA12	73.1	2.2	18.8	TA15	61.7	12.6	58.5	TA15	41.7	−19.2	15.2	TA21	40.2	3.1	9.5
TA13	79.8	1.0	16.8	TA16	53.6	16.2	43.0	TA16	46.2	−15.0	12.3	TA23	41.1	9.7	9.1
TA14	82.7	−1.8	22.2	TA19	54.3	10.2	30.5	TA19	51.4	−2.0	20.5	TA24	39.5	5.1	5.1
TA14 bis	78.7	0.7	19.4	TA21	65.8	9.0	25.4	TA19 bis	49.8	−6.2	21.0	TA25	34.9	4.5	1.6
TA15	84.9	0.4	16.4	TA23	60.5	14.8	47.4	TA21	51.2	−11.0	17.4	TA26	37.7	1.2	6.0
TA16	76.9	1.1	17.5	TA25	55.0	16.5	48.2	TA23	54.6	−5.3	18.3	TA28	33.7	2.4	5.1
TA20	72.7	5.3	21.5	TA26	55.0	9.6	27.7	TA25	52.2	−19.3	17.3	TA30	30.3	3.5	0.5
TA24	79.5	−2.0	22.2	TA27	43.5	10.3	20.1	TA26	42.0	−11.8	6.8	TA35	38.8	16.1	17.5
TA25	81.3	0.3	18.1	TA30	68.4	3.1	31.4	TA27	41.8	−15.6	10.5	TA41	35.2	1.8	4.5
TA32	74.5	2.4	14.3	TA32	57.9	11.2	41.2	TA30	56.9	−18.8	18.3	TA47	33.2	2.7	4.8
TA33	72.8	−0.2	12.0	TA35	43.8	15.6	29.6	TA30 bis	31.4	−5.3	3.8				
TA34	77.9	−2.5	18.8	TA41	58.6	9.1	27.8	TA32	67.7	−13.4	7.0				
TA35	78.0	1.3	16.6	TA44	59.1	13.4	41.1	TA35	44.9	−11.8	12.7				
TA36	74.9	−4.6	18.5	TA47	56.3	7.4	38.0	TA36	40.9	−13.6	14.0				
TA37	79.5	−2.5	20.2	TA48	61.1	4.4	33.2	TA37	36.2	−6.8	9.3				
TA38	81.2	3.7	13.9	TA49	61.3	6.9	39.8	TA40	47.8	−8.4	15.9				
TA39	66.9	3.4	20.6	TA50	66.5	5.6	38.0	TA41	37.0	−6.6	4.7				
TA42	69.5	6.3	17.1					TA42	45.0	−4.9	8.5				
TA47	52.4	2.8	19.2					TA44	49.5	−13.7	12.1				
TA48	79.0	−1.8	12.8					TA47	60.8	−13.4	18.9				
TA49	77.4	−1.6	17.9					TA47 bis	35.0	−3.6	5.7				
TA50	74.3	2.4	15.4					TA48	53.3	−15.1	12.7				
								TA49	51.9	−21.6	15.0				
								TA50	48.4	−18.4	14.8				

XRPD

Semi-quantitative results obtained by XRPD analysis (Table 4) revealed a somewhat compositional homogeneity of ceramic body of samples, which is predominantly composed of quartz, followed by calcite, illite/muscovite and K-feldspars as original minerals and gehlenite, diopside and hematite as new formed phases during firing. Plagioclase amounts account for original and new formed crystals.

XRF

The chemical analysis carried out by XRF technique allowed to investigate the concentration of major and traces oxides, reported in Table 5. In all the investigated TA samples, the most abundant oxide was SiO₂, with mean value of 51.3% wt. The second predominant oxide was Al₂O₃ ($\bar{x}$ = 16.1% wt), followed by CaO ($\bar{x}$ = 13.2% wt) and Fe₂O_{3tot} ($\bar{x}$ = 6.6% wt).

As for trace elements, XRF results revealed the presence of Sr (236–500 µg/g), Zr (117–186 µg/g), Rb (81–127 µg/g), Y (15–30 µg/g) and Nb (7–18 µg/g).

Fig. 5 Biplots of a^* and b^* values of CIEL*a*b* coordinates measured on cross-section (a) and on glaze (b)

SEM-EDS

SEM-EDS observations on the selected potsherd samples confirmed the low carbonates content of engobe and highlighted some glaze alteration (Pb leaching and formation of Pb and Ca carbonates) on the outer surface and/or along fractures (Fig. 8).

The EDS results (Table 6) showed that all the TA samples were covered by a high lead glaze. The most abundant oxides are PbO (49.9–67.5% wt), followed by SiO₂ (24.5–37% wt) and Al₂O₃ (0.9–8.5% wt). Minor oxides (on average lower than 1 wt.%) are FeO_{tot}, Na₂O, K₂O, MgO and CaO.

In general, the colourless glaze appears free of chromophores elements, except for TA09 and TA49, which respectively contain minor concentrations CuO and MnO due to the proximity to coloured areas. The same is for yellow

glazes which only in sample TA44 showed the presence of CuO. In the other samples, the green and the purple are respectively due to the presence of CuO (2.2–14.8% wt) and MnO (0.8–1.9% wt).

High magnification BSD images of the body-glaze interface revealed a different reactivity between the glaze and the body compared to the engobe (Fig. 9). While the engobe-glaze interface rarely shows the presence of a reaction zone with Pb-K feldspar formation (Fig. 9a), this is typically present between the ceramic body and the glaze (Fig. 9b). Its thickness is about 20 μ m and aggregates of Pb-K feldspar crystals are observed as liquidus phases, with the following mean weight percentages of major oxides (n. analysis=35): SiO₂ = 42.9 ± 3.8 ; PbO = 29.5 ± 7.6 ; Al₂O₃ = 14.4 ± 5.1 ; K₂O = 4.8 ± 3.0 ; CaO = 4.5 ± 2.6 ; FeO = 1.7 ± 0.9 ; Na₂O = 1.1 ± 0.4 .

Discussion

Ceramic body: clay provenance and processing

Microscopic observation and the compositional data of the ceramic body allowed to give insight of the type of clay used and the processing to obtain the ceramic paste.

XRPD analysis of the ceramic body (Table 4) confirmed the mineralogical content observed under the polarising microscope and identified the presence of gehlenite, diopside, hematite, clay minerals and micas, which were not detectable by the POM.

Overall, the petrographic analysis defined three different fabrics types (KQ_k, KQKa_k, KQKa_kcp), characterised by carbonate-rich clay, with more or less fine unimodal texture (Table 3).

The use of carbonate-rich clay to prepare the paste for pottery and bricks is coherent with the clays available in the Tavoliere, where the marly clays of the Argille subappennine Fm contribute to the finer component of the alluvial sediments. However, the silt size of NPIs (Table 3), as well as the bulk chemical features (Tab. 5), infers fractionation of the clay.

The comparison of the chemical composition of the ceramic body of the plates with the bricks of Torre Alemana and with the local clays of Argille subappennine Fm and alluvial clays (Gliozzo et al. 2018) showed significant correlations, as confirmed by PCA (Fig. 10).

Statistical data show the relative compositional homogeneity of TA ceramic body compared to the bricks (TAL) and also their difference. Furthermore, the chemical composition of bricks corresponds more closely to that of the alluvial deposits, than to that of the potsherds (Fig. 10). These aspects suggest that bricks were prepared using alluvial clays without a significant compositional modification, as

Table 3 Pottery fabrics and petrographic content. Key: *ext*, external; *int*, internal; *Bir*, birefringence; *i*, iron oxides-oxhydroxides; *k*, carbonates; *m*, micaceous; *cp*, clay pellets; *P1* and *P2*, primary and secondary porosity; *Txt*, texture; *D mode*, prevalent grain size class; *D max*, maximum grain size; *Qm*, monocrySTALLINE quartz; *Qp*, polycrystalline quartz; *Ch*, chert; *Pl*, plagioclase; *Kfs*, K-feldspars; *Ms*, muscovite; *Cpx*, clinopyroxene; *Ka*, carbonate sand; *Ka*, carbonate aggregates; *Ia*, ferruginous aggregates; *Bm*, microfossils; *B*, bioclasts; *Cal2*, secondary calcite; *O-3*, relative amounts; *thermally altered

Sample	Coating		Ceramic body																			B	Cal2														
	Pre- sence	Engobe	Boundary	Glaze		Matrix											Pores							Non-plastic inclusions													
				Bir	D	Body/ engobe	Engobe/ glaze	Rel- ics	Bub- bles	Fabric	i	k	m	cp	Bir	Str	% vol	P1	P2	% vol	Txt			D	mode (µm)	D max (µm)	Qm	Qp	Ch	Pl	Kfs	Ms	Cpx	K	Ka	Ia	Bm
int	ext	Bir	D	mode	Body/ engobe	Engobe/ glaze	Rel- ics	Bub- bles	Fabric	i	k	m	cp	Bir	Str	% vol	P1	P2	% vol	Txt	D	mode (µm)	D max (µm)	Qm	Qp	Ch	Pl	Kfs	Ms	Cpx	K	Ka	Ia	Bm			
TA04	1	0	1	16-32	2	1	0	1	KQKa_k	1	2	1	0	1	O	10	2	0	20	U	32-63	800	2	0	0	1	1	1	1	1	3*	2	2	1	0	0	
TA05	1	0	1	16-32	2	2	1	1	KQ_k	1	2	1	0	1	O	20	3	0	20	U	32-63	500	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	
TA06	1	0	1	32-63	2	2	0	1	KQ_k	1	2	1	0	1	O	10	2	0	20	U	32-63	400	2	0	0	1	1	1	0	3*	0	1	1	0	1	0	1
TA07	1	0	2	32-63	2	2	1	1	KQ_k	1	2	1	0	1	O	15	3	0	20	U	32-63	600	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA08	1	1	2	16-32	2	2	1	1	KQ_k	1	2	1	0	1	O	10	3	0	20	U	32-63	400	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA09	1	1	2	32-63	2	2	2	1	KQ_k	1	2	1	0	2	O	15	2	0	20	U	32-63	400	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA10	1	0	2	32-63	2	2	2	2	KQ_k	1	2	1	0	2	O	10	2	0	30	U	32-63	500	2	1	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA11	1	0	2	32-63	2	2	0	1	KQ_k	1	2	1	0	1	O	15	3	0	20	U	32-63	400	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA12	1	0	1	32-63	2	2	1	2	KQ_k	1	2	1	0	2	O	15	3	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA13	1	0	2	32-63	2	2	1	1	KQ_k	1	2	1	0	1	O	10	2	0	30	U	32-63	400	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA14	1	0	1	16-32	2	2	0	1	KQ_k	1	2	1	0	1	O	10	2	0	20	U	32-63	300	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA15	1	0	2	32-63	2	2	1	0	KQ_k	1	2	1	0	1	O	15	3	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA16	1	0	1	16-32	2	2	1	1	KQ_k	1	2	1	0	2	O	20	3	0	20	U	32-63	300	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA17	1	0	2	32-63	2	2	2	1	KQKa_k	1	2	1	1	2	O	10	2	0	20	U	32-63	300	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA18	1	0	1	16-32	2	2	1	1	KQ_k	1	2	1	0	2	O	15	3	0	20	U	32-63	300	2	0	0	1	1	1	0	3*	0	1	1	0	1	0	1
TA19	1	0	2	32-63	2	2	1	1	KQ_k	1	2	1	0	1	O	10	2	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	2
TA20	1	0	1	16-32	2	2	0	2	KQ_k	1	2	1	0	2	O	15	2	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	0	0	0	0	0
TA21	1	0	1	16-32	2	2	1	2	KQ_k	1	2	1	0	2	O	10	2	0	30	U	32-63	600	2	0	0	1	1	1	0	3*	0	1	1	0	0	1	0
TA22	1	1	1	16-32	2	2	0	3	KQKa_ kep	1	2	1	2	2	O	10	2	0	20	U	32-63	600	2	0	1	1	1	1	0	3*	2	1	1	0	0	0	0
TA23	1	0	1	32-63	2	2	0	1	KQ_k	1	2	1	0	1	O	20	3	0	30	U	32-63	300	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA24	1	0	1	32-63	2	2	1	1	KQ_k	1	2	1	1	2	O	20	3	0	20	U	32-63	500	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA25	1	0	1	32-63	2	2	1	1	KQ_k	1	2	1	0	1	O	10	2	0	20	U	32-63	400	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA26	1	0	1	16-32	2	2	0	1	KQ_k	1	2	1	0	1	O	10	2	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	1	0	0	1	0
TA27	1	0	2	32-63	2	2	1	2	KQ_k	1	2	1	0	1	O	15	3	0	20	U	32-63	400	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA28	1	0	2	16-32	2	2	1	2	KQ_k	1	2	1	0	1	O	15	3	0	20	U	32-63	500	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA29	1	0	1	32-63	2	2	1	1	KQ_k	1	2	1	1	2	O	15	3	0	20	U	32-63	1000	2	0	0	1	1	1	1	3*	1	1	1	0	1	0	1
TA30	1	0	1	32-63	2	2	1	1	KQ_k	1	2	1	0	2	O	15	3	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA31	1	0	1	16-32	2	2	0	1	KQ_k	1	2	1	0	1	O	15	3	0	20	U	32-63	300	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA32	1	1	1	32-63	2	2	1	1	KQ_k	1	2	1	0	1	O	10	2	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA33	1	0	1	<16	2	2	0	3	KQ_k	1	2	1	0	2	O	15	3	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	1	0	1	0	1
TA34	1	0	1	<16	2	1	0	1	KQ_k	1	2	1	0	1	O	10	2	0	20	U	32-63	500	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA35	1	0	2	32-63	2	2	1	1	KQ_k	1	2	1	0	2	O	15	3	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0
TA36	1	0	1	32-63	2	2	1	1	KQ_k	1	2	1	0	1	O	10	2	0	20	U	32-63	200	2	0	0	1	1	1	0	3*	0	1	1	0	0	0	0

Table 3 (continued)

Sample	Coating		Ceramic body																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																								
	Pres- ence	Engobe	Boundary	Glaze			Matrix			Pores			Non-plastic inclusions																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																														
				int	ext	Bir	D mode	Body/ engobe	Engobe/ glaze	Rel- ics	Bub- bles	Fabric	i	k	m	cp	Bir	Str	% vol	P1	P2	% vol	Txt	D mode	D max	Qm	Qp	Ch	Pl	Kfs	Ms	Cpx	K	Ka	Ia	Bm	B	Cal2																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																					
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Fig. 6 POM microphotographs of the three different fabrics of the Torre Alemanna type ware: KQ_k (a), KQKa_k (b) and KQKa_kcp (c)

Fig. 11 seems to confirm. Conversely, a positive correlation ($R^2=0.73$) between TA samples, Argille subappennine Fm and their relative clay fraction ($\varnothing < 2 \mu\text{m}$) shows that the composition of the ceramic body can be obtained from

Fig. 7 POM microphotographs of the two different fabrics of the kiln bricks: KQ_k (a) and KQKa_kcp (b)

fractionation of unsorted Argille subappennine Fm or alluvial clays (Eramo 2020).

Other than chemical arguments for fractionation of the unsorted clay used for bricks, such petrographic features as finer grain size mode, lower fraction of NPIs, fine muscovite, traces of foraminifers and carbonate aggregates (calcrete) support this interpretation (Table 3). The rare presence of volcanic clinopyroxene proves to be a further indication of the use of local alluvial clays as raw material.

Surface finishing and glazing

POM and SEM-EDS analyses revealed a double coating as surface treatment, composed of a white engobe overlaid with a glaze of varying thickness, depending on the viscosity reached during firing (Figs. 6, 8, 9). The engobe has a fine and unimodal texture. Monocrystalline quartz constitutes NPIs in a carbonate poor and illitic matrix. Such white slurry cannot be obtained by fractionation of Argille subappennine

Fig. 8 Layered elemental map (Ca, Al, Pb, Si) and K map obtained by SEM-EDS analysis of TA09, showing the composition and the microstructures of the coated ceramic body. Mineral abbreviations after Whitney and Evans (2010). Areas of purple dyed glaze (P1–P4) analysed with EDS

Fm or local alluvial clays and a primary clay must be considered raw material.

Following the macroscopic observation, it can be inferred that the engobe was applied to the green ceramic body by partial immersion of the plates in a white clay slurry and then incised.

The polychromatic decoration applied on the engobe was analysed by spectrophotometry, using CIEL*a*b* colour space (Table 2) and SEM-EDS (Table 6), to compare the colour characteristics and the chemical composition of the investigated areas. The coloured decoration was characterised by three main dyes, green, yellow–brown and purple, which partially filled the incised motifs on the inner surface of the plates. Colour coordinates show different degrees of

dispersion (Fig. 5b). The coloured areas were often irregular, with clear to merging borders. Pigments used for pictorial decorations on incised surfaces are Fe oxyhydroxides for yellow brown, MnO for purple and CuO for green. Colour strains and glaze drippings are very common, suggesting lower viscosity of glaze in these domains and vertical position of plates during firing.

Poor sharpness in pictorial decoration, complete dissolution of oxides pigments in the glaze and negligible presence of crystalline compounds at the interface (Pradell and Molera 2020) suggest brush application of coloured pigments on raw unfired glaze (inglaze), as in “Splashed and Slip-incised types of Lead-glazed class” (Mason 1997).

SEM-EDS analysis revealed that the composition of glazes is characterised by significant presence of PbO (50–68% wt), with an alkali content ($\text{Na}_2\text{O} + \text{K}_2\text{O}$) normally lower than 2% wt and Al_2O_3 content between 3 and 8% wt (Table 6). Relic quartz grains and minor amounts of alkali and alumina in glaze point to quartz-rich sand as component of the recipe for glaze preparation. PbO powder is considered probable source of lead. The results are consistent with the typical composition of transparent high lead type glazes (Tite et al. 1998).

Firing

The fabric and the mineralogical composition, as well as the chemical composition of the glaze, made it possible to understand the firing strategy adopted. Generally, for all the investigated samples, the POM analysis highlighted an oxidised homogeneous matrix (Table 3), medium–low birefringence and absence of shrinkage pores, then suggesting general oxidising firing conditions. The presence/absence of significant minerals revealed the equivalent firing temperatures (EFTs) (Tite 1995). The mineralogical composition was characterised by the absence (or traces) of the illite/muscovite and the presence of gehlenite, as metastable product of subsolidus reaction between phyllosilicates and carbonates (Maggetti 1982; Riccardi et al. 1999; Cultrone et al. 2001; Maritan et al. 2006; Allegrretta et al. 2016; Maggetti et al. 2011; Gliozzo 2020). Thermally altered calcite grains are visible under POM (Fig. 6).

The mineralogical data (XRPD) suggest EFTs between 750 and 1000 °C (Table 4) of the ceramic body.

In the case of engobe, birefringence of the matrix and SEM-BSD images show a sintering degree between medium and high and, in some cases, an extended digestion by glaze (Fig. 6).

In order to estimate the melting temperature of the glaze, the mean values of PbO, SiO_2 and Al_2O_3 of the colourless portions of glaze analysed with EDS were normalised to 100% and plotted on the ternary phase diagram of PbO– Al_2O_3 – SiO_2 system (Fig. 12). The corresponding liquidus

Table 4 Semi-quantitative mineral content by XRPD and estimated sintering degree and EFTs. Mineral abbreviations after Whitney and Evans (2010). Numbers indicate the abundance (0, absent; 1, traces; from 2 to 4, increasing presence)

Sample	Qz	Pl	Kfs	Cal	Ill/Ms	Gh	Di	Hem	Sintering	EFT (°C)
TA04	4	3	1	1	1	2	2	1	Medium	850–950
TA05	4	3	1	1	1	2	2	1	Medium	850–950
TA06	4	3	1	1	1	2	2	1	Medium	850–950
TA07	4	3	1	1	1	2	2	1	Medium	850–950
TA08	4	3	1	1	1	2	2	1	Medium	850–950
TA09	4	3	1	1	1	2	2	1	Medium	850–950
TA10	4	3	1	1	1	1	2	1	Medium	850–950
TA11	4	3	1	1	1	2	2	1	Medium	850–950
TA12	4	2	0	1	1	1	0	1	Medium	850–950
TA13	4	2	0	1	1	2	1	1	Medium	800–950
TA14	4	3	1	1	1	2	2	1	Medium	850–950
TA15	4	2	1	2	1	2	2	1	Low-medium	750–850
TA16	4	3	1	2	1	2	2	1	Low-medium	750–850
TA17	4	2	0	1	1	2	1	1	Medium	800–950
TA18	4	2	1	1	2	1	1	1	Medium	850–950
TA19	4	2	2	2	1	1	1	1	Low-medium	750–850
TA20	4	3	1	2	0	2	2	1	Low-medium	750–850
TA21	4	2	1	2	2	2	0	1	Low-medium	750–850
TA22	4	2	2	2	2	2	0	1	Low-medium	750–850
TA23	4	3	1	0	1	2	2	1	Medium–high	850–1000
TA24	4	2	0	2	2	2	1	1	Low-medium	750–850
TA25	4	3	0	1	1	2	2	1	Medium	850–950
TA26	4	2	1	2	1	2	2	1	Low-medium	750–850
TA27	4	2	1	2	1	2	2	1	Low-medium	750–850
TA28	4	3	0	1	0	2	2	1	Medium	850–950
TA29	4	3	1	2	1	2	0	1	Low-medium	750–850
TA30	4	2	2	2	1	2	1	1	Low-medium	750–850
TA31	4	2	0	1	1	2	2	1	Medium	850–950
TA32	4	2	0	2	1	2	1	1	Low-medium	750–850
TA33	4	3	0	2	1	2	2	1	Low-medium	750–850
TA34	4	3	1	2	1	2	2	1	Low-medium	750–850
TA35	4	2	0	2	1	1	1	1	Low-medium	750–850
TA36	4	3	1	2	1	2	2	1	Low-medium	750–850
TA37	4	2	2	1	1	1	1	1	Low-medium	750–850
TA38	4	2	0	2	1	2	2	1	Low-medium	750–850
TA39	4	3	1	1	1	2	2	1	Medium	850–950
TA40	4	2	0	2	2	1	1	1	Low-medium	750–850
TA41	4	3	1	1	1	1	2	1	Medium	850–950
TA42	4	3	1	2	1	2	2	1	Low-medium	750–850
TA44	4	2	0	2	2	1	1	1	Low-medium	750–850
TA45	4	3	1	0	1	2	2	1	Medium–high	850–1000
TA46	4	3	1	0	1	2	2	1	Medium–high	850–1000
TA47	4	2	1	2	1	2	2	1	Low-medium	750–850
TA48	4	2	0	2	1	2	2	1	Low-medium	750–850
TA49	4	2	0	2	1	2	2	1	Low-medium	750–850
TA50	4	2	1	2	1	2	0	1	Low-medium	750–850

temperatures range from 780 to 950 °C, although lower temperatures must be considered in portions of glaze where chromophore elements occur (i.e. Fe, Cu, Mn). Dispersed

compositional points in the plot are also consequence of an incomplete maturation of the glaze.

Table 5 Bulk chemical composition (XRF) of the ceramic bodies and bricks (oxides, % wt; trace elements, $\mu\text{m/g}$). Key: $\bar{x}$, mean; s , standard deviation; cv , coefficient of variation

	Sample	SiO ₂	TiO ₂	Al ₂ O ₃	Fe ₂ O _{3tot}	MnO	MgO	CaO	K ₂ O	Na ₂ O	P ₂ O ₅	LOI	Rb	Sr	Y	Zr	Nb
Dishes	TA04	51.69	0.78	16.01	6.65	0.11	3.28	13.96	2.82	0.91	0.21	3.58	115	302	25	158	16
	TA05	51.71	0.83	16.89	7.11	0.11	3.58	12.33	2.86	0.90	0.23	3.46	122	325	27	159	18
	TA06	49.26	0.74	15.15	6.17	0.10	3.72	15.19	2.51	0.87	0.19	6.09	111	333	25	159	15
	TA07	51.83	0.80	16.11	6.57	0.10	3.16	13.39	2.79	0.88	0.25	4.12	118	314	27	168	17
	TA08	51.78	0.83	17.00	7.18	0.11	3.67	12.20	2.92	0.79	0.23	3.29	126	306	26	150	17
	TA09	50.83	0.78	15.83	6.53	0.10	3.39	13.53	2.77	0.88	0.47	4.90	119	332	26	164	17
	TA10	51.09	0.80	16.44	6.95	0.10	3.55	13.57	2.78	0.80	0.22	3.69	121	322	26	152	17
	TA11	52.26	0.83	16.93	7.09	0.11	3.52	12.34	2.90	0.85	0.19	2.99	125	315	29	162	18
	TA12	51.20	0.80	16.26	6.71	0.11	3.10	13.50	2.86	0.81	0.19	4.47	119	307	26	160	17
	TA13	52.61	0.79	16.14	6.44	0.10	3.22	12.51	2.80	0.93	0.21	4.25	117	296	26	163	16
	TA14	51.37	0.81	16.54	6.93	0.10	3.59	13.13	2.84	0.82	0.22	3.65	122	324	28	160	17
	TA15	50.96	0.78	15.81	6.50	0.10	3.44	13.88	2.72	0.92	0.22	4.69	115	321	26	163	17
	TA16	52.09	0.82	16.85	7.05	0.10	3.56	12.21	2.88	0.84	0.23	3.37	124	313	27	158	18
	TA17	51.72	0.79	16.32	6.70	0.10	3.28	12.91	2.92	0.91	0.28	4.07	119	300	26	153	16
	TA18	52.94	0.80	15.96	6.62	0.10	3.16	13.24	2.82	0.84	0.28	3.24	119	313	27	172	17
	TA19	51.98	0.81	16.50	6.74	0.10	3.41	12.70	2.85	0.91	0.20	3.79	123	317	27	168	18
	TA20	49.77	0.77	15.95	6.57	0.10	3.48	13.33	2.89	0.80	0.23	6.10	121	327	28	162	17
	TA21	50.47	0.75	15.18	6.28	0.10	3.28	14.23	2.61	0.89	0.19	6.03	112	314	27	182	16
	TA22	52.37	0.78	16.10	6.66	0.10	3.21	12.40	3.17	0.95	0.25	4.00	118	314	27	160	17
	TA23	52.83	0.83	17.13	7.10	0.11	3.32	11.16	3.03	0.89	0.24	3.34	127	323	28	165	18
	TA24	51.45	0.81	16.68	6.94	0.10	3.23	12.52	2.92	0.80	0.20	4.34	124	312	28	164	18
	TA25	51.45	0.81	16.61	6.98	0.11	3.34	13.18	2.77	0.83	0.28	3.64	123	339	29	168	18
	TA26	51.44	0.79	16.07	6.63	0.10	3.39	12.91	2.83	0.83	0.21	4.81	120	316	28	166	17
	TA27	50.41	0.79	16.02	6.55	0.10	3.39	13.52	2.80	0.83	0.21	5.39	117	320	27	163	17
	TA28	52.41	0.80	16.37	6.72	0.10	3.45	12.73	2.75	0.94	0.20	3.54	122	323	30	179	17
	TA29	51.22	0.80	16.24	6.61	0.10	3.32	13.02	2.82	0.89	0.25	4.73	123	312	27	167	17
	TA30	51.09	0.77	15.87	6.52	0.10	3.19	13.26	2.79	0.88	0.30	5.22	118	315	29	179	17
	TA31	48.98	0.75	15.15	6.24	0.10	3.72	14.25	2.70	0.87	0.25	7.00	114	342	26	165	16
	TA32	50.72	0.78	16.01	6.55	0.10	3.06	13.68	2.80	0.87	0.22	5.21	120	326	29	171	17
	TA33	49.28	0.75	15.17	6.24	0.10	3.47	14.62	2.74	0.84	0.20	6.59	113	342	27	171	16
	TA34	49.39	0.74	15.18	6.17	0.10	3.78	14.98	2.51	0.86	0.18	6.09	113	355	28	172	16
	TA35	50.69	0.74	15.19	6.19	0.10	3.59	12.93	2.90	0.95	0.21	6.50	114	326	29	186	16
	TA36	50.85	0.76	15.51	6.32	0.10	3.38	14.13	2.61	0.89	0.21	5.24	114	337	28	172	17
	TA37	54.24	0.72	15.71	6.22	0.10	2.95	11.67	3.05	1.27	0.38	3.69	122	405	28	175	16
	TA38	51.66	0.79	16.15	6.65	0.10	3.07	12.89	2.86	0.84	0.47	4.53	121	313	29	163	17
	TA39	51.42	0.81	16.61	7.01	0.10	3.47	12.49	2.94	0.84	0.18	4.14	124	329	28	159	17
	TA40	49.44	0.70	14.46	5.90	0.10	3.47	13.66	3.06	0.85	0.28	8.06	101	335	26	181	15
	TA41	51.37	0.78	16.24	6.63	0.10	3.57	13.50	2.67	0.86	0.22	4.06	120	317	27	165	17
	TA42	51.38	0.77	15.71	6.37	0.10	2.38	14.33	2.66	0.80	0.15	5.36	118	301	27	155	17
	TA44	50.39	0.75	15.40	6.33	0.10	3.37	13.76	2.69	0.80	0.19	6.23	114	338	28	173	17
	TA45	51.95	0.81	16.49	6.85	0.11	3.37	12.91	2.74	0.83	0.24	3.69	122	341	29	178	18
	TA46	52.70	0.83	17.18	7.16	0.11	3.46	11.57	2.91	0.88	0.31	2.90	127	332	29	167	18
	TA47	51.27	0.81	16.61	6.89	0.10	3.48	12.40	2.92	0.81	0.21	4.49	124	312	29	162	18
	TA48	51.16	0.77	15.77	6.20	0.10	3.28	13.16	2.74	0.89	0.20	5.73	116	310	27	169	17
	TA49	51.05	0.76	15.60	6.12	0.10	3.19	13.37	2.72	0.88	0.22	6.00	116	318	28	173	17
	TA50	51.11	0.77	15.83	6.44	0.10	2.84	13.24	2.90	0.83	0.18	5.77	118	298	27	161	16
	$\bar{x}$	51.29	0.78	16.06	6.61	0.10	3.35	13.18	2.82	0.87	0.24	4.70	118.95	322.42	27.31	166.07	16.90
	s	1.06	0.03	0.61	0.32	0.00	0.25	0.85	0.13	0.07	0.06	1.21	5.01	18.13	1.19	8.11	0.78
	cv	0.02	0.04	0.04	0.05	0.04	0.07	0.06	0.05	0.08	0.27	0.26	0.04	0.06	0.04	0.05	0.05

Table 5 (continued)

	Sample	SiO ₂	TiO ₂	Al ₂ O ₃	Fe ₂ O _{3tot}	MnO	MgO	CaO	K ₂ O	Na ₂ O	P ₂ O ₅	LOI	Rb	Sr	Y	Zr	Nb
Bricks	TAL01	42.28	0.47	9.76	3.73	0.09	1.74	20.05	2.36	0.53	0.26	18.73	104	260	15	125	8
	TAL02	41.56	0.50	10.25	3.90	0.08	1.86	19.09	3.99	0.94	0.13	17.71	92	500	17	139	8
	TAL03	45.96	0.57	11.90	4.27	0.09	2.12	16.06	2.68	0.63	0.19	15.54	112	344	21	167	9
	TAL04	46.36	0.57	11.88	4.28	0.09	2.13	16.30	2.48	0.61	0.19	15.11	103	325	20	160	9
	TAL05	45.67	0.56	11.61	4.48	0.09	2.05	17.25	2.16	0.61	0.23	15.30	81	236	15	117	7
	TAL06	45.02	0.57	11.24	4.82	0.10	1.95	17.44	2.18	0.66	0.22	15.80	110	311	21	163	10
	$\bar{x}$	44.48	0.54	11.11	4.25	0.09	1.98	17.70	2.64	0.66	0.20	16.37	100.33	329.33	18.17	145.17	8.50
	<i>s</i>	2.04	0.04	0.90	0.39	0.01	0.15	1.57	0.69	0.14	0.04	1.49	11.78	92.97	2.86	21.23	1.05
	<i>cv</i>	0.05	0.08	0.08	0.09	0.07	0.08	0.09	0.26	0.21	0.22	0.09	0.12	0.28	0.16	0.15	0.12

Table 6 Chemical composition of coloured areas of glaze obtained by SEM–EDS. Key: *n*, number of measures; $\bar{x}$, mean; *s*, standard deviation

Sample	Colour	SiO ₂		PbO		Al ₂ O ₃		Na ₂ O		K ₂ O		MgO		CaO		FeO		MnO		CuO	
		$\bar{x}$	<i>s</i>	$\bar{x}$	<i>s</i>	$\bar{x}$	<i>s</i>	$\bar{x}$	<i>s</i>	$\bar{x}$	<i>s</i>	$\bar{x}$	<i>s</i>	$\bar{x}$	<i>s</i>	$\bar{x}$	<i>s</i>	$\bar{x}$	<i>s</i>	$\bar{x}$	<i>s</i>
TA09	Colourless (8)	34.2	2.6	58.8	4.5	2.9	1.6	0.4	0.2	0.6	0.3	0.6	0.1	1.8	0.8	0.3	0.4	-	-	0.4	0.5
	Yellow (4)	31.6	2.5	58.1	1.5	2.6	0.4	0.3	0.0	0.7	0.2	0.5	0.0	2.2	0.4	4.0	1.2	-	-	-	-
	Purple (4)	35.4	1.9	55.4	2.5	3.5	1.1	0.4	0.1	0.7	0.3	0.6	0.0	1.5	0.1	0.4	0.1	1.9	0.4	-	-
	Green (4)	30.5	1.2	63.8	1.3	1.5	0.4	0.2	0.1	0.3	0.1	0.4	0.0	1.0	0.1	0.1	0.2	-	-	2.2	0.4
TA29	Colourless (3)	31.9	0.7	56.8	3.2	8.5	2.5	0.2	0.2	0.7	0.2	0.7	0.0	1.2	0.1	-	-	-	-	-	-
	Yellow (3)	28.2	1.5	60.3	2.1	6.0	1.5	0.5	0.1	0.5	0.1	0.6	0.1	1.1	0.1	2.7	0.4	-	-	-	-
	Green (3)	27.3	0.3	60.6	1.3	5.0	0.5	0.9	0.1	0.4	0.1	0.5	0.0	1.2	0.2	1.1	0.3	-	-	3.2	0.5
TA38	Colourless (3)	26.7	1.7	62.3	2.4	4.4	0.8	0.5	0.0	0.6	0.1	0.5	0.1	1.2	0.2	3.8	3.1	-	-	-	-
	Yellow (3)	26.5	0.7	60.3	0.6	5.4	0.6	0.5	0.0	0.6	0.1	0.5	0.0	1.4	0.2	4.8	0.7	-	-	-	-
	Purple (3)	34.6	9.1	59.7	8.4	2.2	0.3	0.4	0.2	0.9	0.5	0.3	0.2	0.9	0.5	0.2	0.3	0.8	0.4	-	-
	Green (3)	27.8	1.6	58.6	0.7	6.1	0.2	1.2	0.1	0.8	0.2	0.6	0.1	1.7	0.4	0.7	0.4	-	-	2.6	0.4
TA42	Colourless (3)	37.0	1.3	49.9	2.8	7.9	0.6	0.6	0.1	1.2	0.3	0.6	0.1	1.8	1.6	1.0	0.7	-	-	-	-
	Green (3)	32.2	1.5	56.4	1.3	5.2	1.6	1.0	0.2	0.6	0.1	0.5	0.1	1.0	0.4	0.6	0.2	-	-	2.6	0.1
TA44	Colourless (5)	32.5	1.4	61.4	1.4	3.6	0.8	0.2	0.1	0.4	0.1	0.3	0.1	1.4	0.2	0.2	0.2	-	-	-	-
	Yellow (5)	31.2	1.5	59.2	1.4	3.7	1.1	0.2	0.1	0.5	0.1	0.3	0.0	1.2	0.1	3.5	0.9	-	-	0.3	0.6
	Green (5)	24.4	0.6	67.4	0.4	7.8	0.3	0.1	0.1	0.6	0.1	0.7	0.0	0.9	0.1	2.4	1.3	0.9	0.0	14.8	1.0
TA45	Colourless (4)	32.8	1.5	56.7	0.5	4.8	1.2	0.2	0.0	0.7	0.0	0.3	0.0	3.4	0.2	1.1	0.2	-	-	-	-
	Yellow (3)	34.1	1.4	59.7	1.3	2.3	0.4	0.2	0.0	0.5	0.1	0.6	0.1	0.7	0.1	2.0	0.4	-	-	-	-
	Green (4)	31.2	0.4	60.7	0.5	2.4	0.4	0.0	0.0	0.4	0.0	0.4	0.1	1.1	0.2	0.5	0.1	-	-	3.3	0.3
	Purple (4)	31.1	1.0	60.9	0.6	2.6	0.5	0.1	0.1	0.4	0.1	0.4	0.1	1.0	0.4	0.4	0.6	1.6	0.8	1.4	0.2
TA46	Colourless (6)	36.3	2.0	57.2	2.1	3.7	1.8	0.3	0.2	1.0	0.2	0.4	0.1	1.0	0.1	0.1	0.2	0.1	0.2	-	-
	Green (3)	34.0	2.3	58.7	3.0	1.8	0.6	0.2	0.1	0.8	0.1	0.3	0.1	0.7	0.1	0.1	0.2	-	-	3.3	0.1
	Purple (3)	31.1	0.9	64.7	0.4	0.9	0.6	0.1	0.0	0.3	0.0	0.2	0.0	0.8	0.0	0.1	0.1	1.8	0.0	-	-

For a uniform glaze without mineral relics and gas bubbles, the viscosity of 10^4 poise should be reached to mature the glaze. At the same firing temperature, lower liquidus temperatures for coloured glaze portions imply lower viscosity compared to colourless portions, and thus explain the colour glaze drippings and the partial digestion of the underlying engobe. Digestion of the engobe is less extended if compared to the ceramic body where the incisions occur

(Fig. 9). The preferential formation of Pb–K feldspars in the reaction zone between glaze and ceramic body agrees with the lower solidus of CaO rich clays compared to CaO poor clays (Heimann 1989). The composition of Pb–K feldspar microcrystals ($< 5 \mu\text{m}$) is similar to what measured by Maltoni et al. (2012) and Godet et al. (2019) for glazed pottery. The presence of quartz relics and bubbles points to subliquidus conditions comparable with those reported by

Fig. 9 Details (a, b) of the SEM-BSE image of the glaze-body interface of sample TA45 and elemental maps of Pb and Ca. In the table, the semi-quantitative composition (EDS) of the Pb-K feldspars in the frame b is reported

Molera et al. (2001) for low carbonate clays (K2) in contact with high lead glaze. Relatively higher viscosity associated with subliquidus conditions hinders chemical diffusion and bubble freeing, as well as body and engobe digestion. Moreover, the application of glaze by dipping, as observed on potsherds, is a further argument for reduced reactivity at glaze-engobe interface (Tite et al. 1998; Gualtieri et al. 2006). A comparison between the estimated temperatures for sintering of the ceramic body and maturing of glaze allows to discriminate between the double and single firings of the vessels. The overlap of the estimated temperature ranges for the ceramic body (750–1000 °C) and the glaze (780–950 °C) suggests a single firing, as also suggested by the coarse painted decorations. These temperature ranges match the maturing temperature range of high lead glaze (800–1050 °C), as for Torre Alemanna, reported by Tite et al. (1998).

Experimental work on the reactivity of the glaze-body can be used as a reference. Gualtieri et al. (2006) report a glaze-body interface thickness of 20–30 µm for ceramic tests with a composition very similar to that of Torre Alemanna after a single firing. Interface thickness of 20 µm was also observed by Ben Amara and Schvoerer (2006) in glazed ceramic test pieces fired once. However, the thickness of the glaze-body (or engobe) interface is a combination of several factors and increases with firing temperature and/or cooling time (Molera et al. 2001; Ben Amara et al. 2006). Setting aside the composition of the glaze and body and the firing temperatures, the cooling time can be considered short, if only the presence of Pb-K feldspars at the glaze-Ca-rich body interface is considered, without the formation of wollastonite as a second liquidus phase, as in Ben Amara and Schvoerer (2006). Moreover, two firings in the same temperature range seem unjustifiable to obtain poorly defined underglaze designs.

Conclusion

The results of the archaeometric investigation showed that the ceramic body of the Torre Alemanna type was prepared by fractioning a carbonate-rich clay from local alluvial

deposits. The decoration technique of the Torre Alemanna type production was based on the combination of sgraffito engraving with polychromatic coarse painting.

The surface treatment consisted of four consequential actions (Fig. 13) summarised as follows: after the fashioning and drying, the object was first covered by the engobe (action 1) and then the decorative patterns were incised to reveal the ceramic body beneath (action 2). A raw glaze was applied by dipping on the object surface (action 3), followed by the painted polychromatic decoration to highlight the decorative patterns (action 4).

A high lead glaze was used for all the vessels with some differences in composition and application. In the painted decorations, Cu, Mn or Fe pigments were added to obtain green, purple or yellow, respectively. Analytical results and aesthetic arguments point to a single firing to sinter the ceramic body and to mature the glaze at the same time. It is worth noting that glazed sgraffito ceramics are generally considered to be obtained through a double firing, although petrographic, mineralogical and chemical evidence for such an interpretation is not always provided. In this work, an independent estimate of the temperatures reached by both the ceramic body and the glaze is proposed as an additional tool for the technological analysis of the artefacts. The results obtained here refer to this ceramic type and cannot be generalised. The growing number of researches on the glazed ceramic production in the Middle Ages allows a more and more articulated vision of the technological choices and of the circulation of typological models and manufactured articles throughout Italy and the Mediterranean area.

As for the Tavoliere plain, the results obtained showed a technological continuity with the past regarding the use of local carbonate-rich clay to produce fine pottery and defined the reference compositional group of Torre Alemanna type ware, already attested in various archaeological contexts in southern Italy. The ceramic body showed strong compositional homogeneity, denoting a standardised use of raw materials.

a

b

Fig. 10 Biplots of PC1 vs PC2 (a) and PC1 vs PC3 (b) after principal component analysis of the bulk chemical data, including samples of Torre Alemanna type ware (TA), kiln bricks (TAL) and local clay sediments, both Argille subappennine Fm (SAC) and alluvial clays (Ac), presented in Gliozzo et al. (2018)

improved the quality of the paper, as well as to Alba Rosa Scattaglia for the English revision. While reviewing this article, one of the authors (AB) passed away after a serious illness and we dedicate this to his memory.

Author contribution Giovanna Fioretti: conceptualisation, investigation, visualisation, writing. Giacomo Eramo: conceptualisation, meth-

Fig. 11 Scatter plot showing the $\text{Al}_2\text{O}_3/\text{SiO}_2$ vs MgO/CaO ratio in samples of Torre Alemanna type ware (TA) and bricks (TAL), compared with unsorted (total) and $< 2 \mu\text{m}$ fraction of Argille subappennine Fm (SAC) and alluvial clays (Ac) collected in the surroundings (Eramo et al. 2004, 2012, 2013)

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Fig. 12 Ternary phase diagram of $\text{PbO}-\text{Al}_2\text{O}_3-\text{SiO}_2$ system (Chen et al. 2001, modified), indicating the normalised compositional points of the colourless glaze analysed by EDS (white circles) and the relative density map. Concentrations in % wt and temperatures in Kelvin

Fig. 13 The sequence of actions identified for the surface treatment before firing

(reviewing and editing), supervision. Alessandro Monno: investigation, writing. Austacio Busto: investigation, writing (original draft preparation). Rocco Laviano: resources, founding acquisition, editing.

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Declarations

Conflict of interest The authors declare no competing interests.

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