Surface hydrophobicity: effect of alkyl chain length and network homogeneity

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Abstract Understanding the nature of hydrophobicity has fundamental importance in environmental applications. Using spherical silica nanoparticles (diameter = 369 ± 7 nm) as the model material, the current study investigates the relationship between the alkyl chain network and hydrophobicity. Two alkyl silanes with different chain length (triethoxymethylsilane (C1) vs. trimethoxy(octyl)silane (C8)) were utilised separately for the functionalisation of the nanoparticles. Water contact angle and inverse gas chromatography results show that the alkyl chain length is essential for controlling hydrophobicity, as the octyl-functionalised nanoparticles were highly hydrophobic (water contact angle = $150.6^\circ \pm 6.6^\circ$), whereas the methyl-functionalised nanoparticles were hydrophilic (i.e., water contact angle = 0° , similar to the pristine nanoparticles). The homogeneity of the octyl-chain network also has a significant effect on hydrophobicity, as the water contact angle was reduced significantly from $148.4^\circ \pm 3.5^\circ$ to $30.5^\circ \pm 1.0^\circ$ with a methyl-/octyl-silane mixture (ratio = 160:40 $\mu\text{L} \cdot \text{g}^{-1}$ nanoparticles).

Keywords hydrophobicity, surface energy, wettability, alkyl chain network, silica nanoparticle

1 Introduction

Due to their low cost, exceptional stability as well as controllable physiochemical properties, silica-based nanostructures have a wide range of environmental applications including oil recovery, carbon dioxide storage and environmental remediation [1–4]. Spherical nanoparticle prepared via the Stöber method is one of the most popular nanostructures due to its high monodispersity in size and shape as well as its versatility for chemical modification [5–13]. The diameter of Stöber nanoparticle

can cover the entire nanometre range (i.e., 10 nm–1 μm) by controlling the synthesis conditions such as pH and temperature [5,6]. Through methods such as co-condensation and grafting, different chemical groups can be attached to the surface of Stöber nanoparticles [14]. For instance, superhydrophobic Stöber nanoparticles can be obtained by attaching fluorinated copolymers onto their surface [15].

The surface chemistry of silica nanoparticles is of fundamental importance for colloidal stability as it affects the interaction among the nanoparticles themselves as well as between the nanoparticles and the surrounding [16]. In the case of Pickering emulsions, it was found that nanoparticles with an intermediate degree of hydrophobicity (i.e., a water contact angle $\sim 90^\circ$) could stabilise droplets in water-in-oil and oil-in-water systems [17,18]. Typically, the intermediate degree of hydrophobicity is achieved by grafting nanoparticles with both hydrophobic and hydrophilic silanes or a mixed polymer silane with similar nature [19–24].

In this study, we demonstrate the nature of hydrophobicity as the result of an interacting network among the hydrophobic chains on the surface of Stöber nanoparticles. Instead of using both hydrophobic and hydrophilic silanes as in the existing literature, two similar silanes with different alkyl chain lengths (i.e., C1 vs. C8) were mixed in various proportions and grafted onto Stöber nanoparticles. Various techniques including contact angle measurement, inverse gas chromatography (IGC), thermal gravimetric analysis and helium pycnometry were used to characterise the nanoparticles in order to analyse the effect of chemical modification.

2 Experimental

2.1 Materials

All chemicals were used without purification unless stated

otherwise. Tetraethyl orthosilicate (TEOS) (98%), trimethoxy(octyl)silane (96%), triethoxymethylsilane (99%) were purchased from Sigma Aldrich (Gillingham, UK). Ethanol ($\geq 99.8\%$), acetone ($\geq 99.8\%$) and sodium hydroxide were purchased from VWR (Lutterworth, UK). Cyclohexane ($\geq 99.8\%$) was purchased from Fisher Chemical (Loughborough, UK). De-ionised water was supplied by PURELAB Chorus 1 (ELGA LabWater) water purification system. More details can be found in Table S1 (cf. Electronic Supplementary Material, ESM).

For the characterisation of silica nanoparticles, carbon-coated copper transmission electron microscopy (TEM) grid was supplied by EM Resolutions (Sheffield, UK) and carbon conductive adhesive tape was supplied by Agar Scientific (Stansted, UK). Silanised glass wool was purchased from Sigma-Aldrich (Gillingham, UK).

2.2 Synthesis of silica nanoparticle

Silica nanoparticles were synthesised with the Stöber method [5,6]. TEOS (60 mL) was added into ethanol/de-ionised water mixture (960 mL/138 mL) at 40 °C, followed by 0.1 mol·L⁻¹ NaOH (42 mL). The final mixture was stirred overnight at 40 °C. The mixture was centrifuged (6000 r·min⁻¹, 10 min) to recover the synthesised silica nanoparticles, which were dried at room temperature prior to further modification.

2.3 Modification of silica nanoparticle

The dried silica nanoparticles (1.0 g) were added into cyclohexane (100 mL), followed by the target silanes (i.e., trimethoxy(octyl)silane and triethoxymethylsilane) (5–200 μ L). The mixture was sonicated at room temperature for 10 min and refluxed overnight (boiling point of cyclohexane = 81 °C). The mixture was cooled to room temperature and centrifuged (6000 r·min⁻¹, 10 min) to recover the modified silica nanoparticles. The recovered nanoparticles were washed repeatedly with fresh acetone and dried at room temperature and finally in vacuum oven (60 °C, 100 mbar) overnight.

2.4 Characterisation of silica nanoparticle

The shape and size of silica nanoparticles were confirmed with TEM and scanning electron microscopy (SEM). For TEM analysis, silica nanoparticles were sonicated in ethanol for proper dispersion. One drop of the sonicated mixture was placed onto a carbon-coated copper TEM grid to be dried naturally. The on-grid sample was analysed with a 2100 Plus transmission electron microscope (JEOL USA, Inc., USA). For SEM analysis, silica nanoparticles were spread on the carbon conductive adhesive tape and coated with 15 nm chromium in a Q150T turbomolecular-pumped coating system (Quorum Technologies Ltd., UK)

(sputter current = 120 mA, tooling factor = 2.43 and maximum sputtering time = 4 min). The chromium-coated sample was analysed with a Leo Gemini 1525 scanning electron microscope (Carl Zeiss AG, Germany) with an accelerating voltage of 5 kV.

The skeletal density of silica nanoparticles was measured at 25 °C by helium pycnometry with an AccuPyc II 1340 pycnometer (Micromeritics Instrument Corp., USA).

The hydrophobicity of silica nanoparticles was quantified by sessile drop contact angle and IGC measurements. For sessile drop contact angle measurement, silica nanoparticles were further dried in a vacuum oven (60 °C, 100 mbar) overnight and compressed into tablets using a R500 system (Gamlen Tableting Ltd., UK) (100 mg sample, target load = 200 kg, speed = 8 mm·min⁻¹). Contact angle was measured with a Model 590 goniometer/tensiometer (Ramé-hart Instrument Co., USA) at room temperature using de-ionised water as the test liquid. One drop of water (10 μ L) was placed on the tablet and the static contact angle was measured over 100 seconds with 1 second measurement interval.

All the IGC measurements were conducted at infinite dilution with an IGC surface energy analyser (Surface Measurement Systems, UK). The sample (~50 mg) was packed into a silanised glass column (length = 300 mm, internal diameter = 3 mm), which was plugged with silanised glass wool on both ends. A jolting voltameter (Surface Measurement Systems, UK) was used to provide mechanical tapping to the sample in order to ensure good powder packing in the column. The packed sample column was placed into the column oven. The carrier gas was helium at a flow rate of 10 mL·min⁻¹ and the reference gas was methane to determine the dead volume. Before each measurement, the sample column was conditioned at 30 °C and 0% relative humidity for 2 h at 10 mL·min⁻¹ carrier gas flow rate. All the measurements were performed at 30 °C and 0% relative humidity (i.e., same as the analysis conditions). Polar probes including ethyl acetate and dichloromethane, and *n*-alkane probes including decane, nonane, octane, heptane and hexane were used under infinite dilution of 3% probe coverage.

The retention behaviour of the probes was used to calculate the dispersive component (γ_d) of surface energy with the Schultz approach, whereas the acid-base (γ_{ab}) component of surface energy was determined using Della Volpe approach [25,26]. Firstly, V_N was determined with Eq. (1). The free energy of adsorption (ΔG_a) was determined from V_N with Eq. (2). The dispersive surface free energy (γ_s^d) is related to V_N of the non-polar alkane solute according to Eq. (3) [25]. To determine γ_s^d , a linear plot of $\ln V_N$ against $(\gamma_1^d)^{0.5}$ of the *n*-alkane probes was made and the slope of the best fit line was used to calculate γ_s^d .

$$V_N = F \frac{J}{m} t_R \frac{T}{273.15}, \quad (1)$$

where F is the carrier gas flow rate, J is the James and Martin pressure correction factor for gas compressibility, m is the mass of the sample, t_R is the retention time of the probe, T is the column temperature.

$$\Delta G_a = -RT \ln V_N + K, \quad (2)$$

where R is gas constant and K is a constant.

$$-\Delta G_a^d = RT \ln V_N = 2Na (\gamma_1^d)^{0.5} (\gamma_s^d)^{0.5} + K', \quad (3)$$

where N is Avogadro's number, a is the cross-sectional area of solute, γ_1^d is the Lifshitz-van der Waals dispersive surface free energy of the probe and K' is another constant that is different from the constant K in Eq. (2).

To determine the acid-base surface free energy (γ_s^{ab}), it was necessary to determine the acid-base component of the free energy of adsorption for polar solutes (ΔG^{ab}) from the difference between the alkane regression line and the polar probe according to Eq. (4). According to Das et al. [27], ΔG^{ab} of monopolar acid and monopolar base probes were used to calculate the acidic and basic parameters of the solid surface energy (γ_s^+ and γ_s^- respectively based on Della Volpe scale) based on the Van Oss-Good theory [28] with Eq. (5). γ_s^{ab} was determined with Eq. (6). The total surface energy (γ_s^{total}) was the sum of γ_s^d and γ_s^{ab} (Eq. (7)).

$$\Delta G_a = \Delta G_a^d + \Delta G_a^{ab}, \quad (4)$$

$$\Delta G_a^{ab} = 2Na \left((\gamma_1^+ \gamma_s^-)^{0.5} + (\gamma_1^- \gamma_s^+)^{0.5} \right), \quad (5)$$

$$\gamma_s^{ab} = 2(\gamma_s^+ \gamma_s^-)^{0.5}, \quad (6)$$

$$\gamma_s^{\text{total}} = \gamma_s^d + \gamma_s^{ab}. \quad (7)$$

3 Results and discussion

3.1 Surface hydrophobicity: effect of alkyl chain length

TEM analysis showed that the synthesised Stöber nanoparticles had an average diameter of 369 ± 7 nm

and a smooth surface (Fig. 1(a)), whereas SEM analysis (Fig. 1(b)) confirmed the high monodispersity of size and shape as expected from the highly robust Stöber method [5]. The skeletal density of the pristine Stöber nanoparticles measured by helium pycnometry was $2.03 \text{ g} \cdot \text{cm}^{-3}$, which was consistent with the value reported in literature [16].

Functionalising the nanoparticles with triethoxymethylsilane and trimethoxy(octyl)silane (Fig. 2) changed the hydrophobicity of nanoparticles to various extent. As shown in Fig. 3(a), the pristine nanoparticles had a 0° contact angle as the water droplet was instantly absorbed by the tablet prepared from the pristine nanoparticles. Similar results were obtained for the methyl-functionalised nanoparticles regardless of the quantity of triethoxymethylsilane per gram of nanoparticles during functionalisation. In contrast, the water contact angle of octyl-functionalised nanoparticles remained close to the superhydrophobic level (between $144.1^\circ \pm 0.4^\circ$ and $158.1^\circ \pm 0.2^\circ$) as the silane to nanoparticle ratio during functionalisation increased from 5 to $200 \mu\text{L} \cdot \text{g}^{-1}$. This demonstrates the effectiveness of the trimethoxy(octyl)silane in forming a hydrophobic coating on the hydrophilic silica nanoparticle.

With IGC, the surface energy of pristine and functionalised silica nanoparticles can be determined with higher accuracy and sensitivity than the contact angle method to provide a fundamental understanding about the effect of chemical modification on the wettability of the nanoparticle [29–33]. The calculated surface energy for pristine, methyl- and octyl-functionalised silica nanoparticles is presented in Fig. 4.

Firstly, the chemical modification with silanes resulted in an observable reduction (i.e., maximum 8%) in the total surface energy from $53.2 \text{ mJ} \cdot \text{m}^{-2}$ for the pristine nanoparticle to $48.7\text{--}49.6 \text{ mJ} \cdot \text{m}^{-2}$ for the methyl-functionalised nanoparticle, and to $49.0\text{--}51.5 \text{ mJ} \cdot \text{m}^{-2}$ for the octyl-functionalised nanoparticles. This provides evidence for the effectiveness of chemical modification, which is especially important for the methyl-functionalised nanoparticle, as water contact angle measurement is unable to distinguish it from the pristine nanoparticle. By separating the total surface energy into the dispersive and acid-base

Fig. 1 (a) TEM and (b) SEM images of Stöber nanoparticles.

Fig. 2 Triethoxymethylsilane (left) and trimethoxy(octyl)silane (right).

Fig. 3 Water contact angle for pristine (i.e., silane:silica NP = 0 $\mu\text{L}\cdot\text{g}^{-1}$) and modified nanoparticles (pictures showing water droplet on tablet or just tablet after the absorption of water droplet). Note: error bars are too small to be visible (e.g., $158.1^\circ \pm 0.2^\circ$).

components (i.e., γ_s^d and γ_s^{ab}), it is obvious that reduction of total surface energy was mainly attributed to the reduction in the dispersive surface energy, which was $43.3 \text{ mJ}\cdot\text{m}^{-2}$ for the pristine nanoparticle and $39.1\text{--}42.5 \text{ mJ}\cdot\text{m}^{-2}$ for the functionalised nanoparticle. Similar observation was previously made for the functionalisation of pharmaceutical excipient powder, where the reduction in the dispersive surface energy (γ_d) was attributed to the passivation of high energy sites by the covalently attached silanes [31].

Secondly, while the total surface energy of methyl- and octyl-functionalised nanoparticles was in the similar range (approximately $49 \text{ mJ}\cdot\text{m}^{-2}$), there was observable difference in the acid-base surface energy (γ_s^{ab}) of the functionalised nanoparticles, which resulted in the significant difference in water contact angle (Fig. 3). For the methyl-functionalised nanoparticle with different triethoxymethylsilane to nanoparticle ratios, the acid-base surface energy (γ_s^{ab}) was $9.1\text{--}9.6 \text{ mJ}\cdot\text{m}^{-2}$, slightly lower than the value of the pristine nanoparticle ($9.9 \text{ mJ}\cdot\text{m}^{-2}$). As the ratio of trimethoxy(octyl)silane to nanoparticle increased from 5 to $200 \mu\text{L}\cdot\text{g}^{-1}$, the acid-base surface energy (γ_{ab}) of the resulting octyl-functionalised nanoparticle decreased significantly from 9.0 to $7.8 \text{ mJ}\cdot\text{m}^{-2}$, even though the water contact angle remained relatively constant (between

Fig. 4 Surface energy of pristine (i.e., silane:silica NP = 0), (a) methyl- and (b) octyl-functionalised silica nanoparticles.

$144.1^\circ \pm 0.4^\circ$ and $158.1^\circ \pm 0.2^\circ$). The high water contact angle was probably due to the hydrophobic network formation by the octyl-chains as discussed in more detail below, showing the effectiveness of this hydrophobic network in shielding the nanoparticle surface from water.

As the particle diameter ($369 \pm 7 \text{ nm}$) is significantly larger than the length of the alkyl chains, the surface of nanoparticle can be treated as a flat surface with respect to the alkyl chains [21]. The hydrophilicity of pristine nanoparticles originates from the free silanol groups

(–SiOH) available on the surface as well as within the nanoparticles (Fig. 5(a)). As the water droplet was placed onto the tablet of pristine nanoparticles, the water readily wetted the surface of the nanoparticles and occupied the interparticle voids, resulting in the instant absorption or imbibition of water droplet.

In contrast, due to the length of the octyl chain (–(CH₂)₇–CH₃) (Fig. 5(c)), it is easier to form a hydrophobic network for shielding the nanoparticle from water, resulting in a high water contact angle. The thermal gravimetric analysis results of the pristine and functionalised nanoparticles (silane:silica ratio = 200 $\mu\text{L}\cdot\text{g}^{-1}$ for both silanes) show a similar trend for mass loss over a wide range of temperature (100 °C to 800 °C), which agree with previous studies on the thermal properties of pristine and functionalised silica nanoparticles [34–37]. The similar mass loss for both pristine and functionalised nanoparticles indicate that the mass of the methyl- and octyl-functional groups is negligible with respect to the mass of the nanoparticle (Fig. 6). In other words, even though the octyl-network is only a very thin layer on the surface of the nanoparticle, it is highly effective in preventing the contact of water with the nanoparticle. Unlike the octyl chain, the methyl chain (–CH₃) is much shorter and hence is unable to form a hydrophobic network despite the increase in

Fig. 6 Thermal gravimetric analysis results for pristine and modified nanoparticles (silane:silica nanoparticle ratio = 200 $\mu\text{L}\cdot\text{g}^{-1}$ for both silanes).

coverage density achieved with the increase in the silane to silica nanoparticle ratio during functionalisation (Fig. 7). As a result, water still readily wetted the methyl-functionalised nanoparticles due to the accessibility of the free silanol groups of the methyl-silane and nanoparticle surface (Fig. 5(b)).

3.2 Surface hydrophobicity: effect of alkyl chain network homogeneity

Besides the length of alkyl chain, the homogeneity of the functional group has an important effect on the surface hydrophobicity of the nanoparticle. Figure 8(a) shows the water contact angles of octyl-functionalised nanoparticles prepared with pure trimethoxy(octyl)silane and a mixture of triethoxymethylsilane and trimethoxy(octyl)silane (in the latter case, the total silane to nanoparticle ratio was kept constant at 200 $\mu\text{L}\cdot\text{g}^{-1}$ (e.g., 100 $\mu\text{L}\cdot\text{g}^{-1}$ trimethoxy(octyl)silane + 100 $\mu\text{L}\cdot\text{g}^{-1}$ triethoxymethylsilane)). The octyl-functionalised nanoparticles prepared with the pure trimethoxy(octyl)silane remained highly hydrophobic (i.e., water contact angle $\gg 90^\circ$) even with the silane to nanoparticle ratio as low as 5 $\mu\text{L}\cdot\text{g}^{-1}$ (Figs. S1–S8, cf. ESM). This is in stark contrast with the nanoparticles functionalised with the silane mixture, whose water contact angle decreased drastically from $154.1^\circ \pm 0.3^\circ$ to $30.5^\circ \pm 1.0^\circ$ as the trimethoxy(octyl)silane to nanoparticle ratio was reduced to 40 $\mu\text{L}\cdot\text{g}^{-1}$.

The surface energy of silica nanoparticles functionalised by mixtures of triethoxymethylsilane and trimethoxy(octyl)silane is shown in Fig. 9. Similar to the nanoparticles functionalised by pure trimethoxy(octyl)silane (Fig. 4(b)), the dispersive surface energy remained relatively constant regardless of the ratio of triethoxymethylsilane and trimethoxy(octyl)silane, whereas the

Fig. 5 The surface of (a) pristine, (b) methyl-functionalised and (c) octyl-functionalised silica nanoparticles.

Fig. 7 Surface of nanoparticle functionalised with the same coverage density of octyl- and methyl-functional groups (i.e., before and after complete substitution with methyl groups).

acid-base surface energy (γ_s^{ab}) had a general decreasing trend ($9.1 \text{ mJ}\cdot\text{m}^{-2}$ to $7.9 \pm 0.3 \text{ mJ}\cdot\text{m}^{-2}$) as the ratio of trimethoxy(octyl)silane to nanoparticle increased (Fig. 8(b)). The decrease in acid-base surface energy (γ_s^{ab}) corresponded to the extent of functionalisation and the reduction in free silanol group ($-\text{SiOH}$) that could interact favourably with water, leading to a general increasing trending for water contact angle.

The acid-base surface energy (γ_s^{ab}) data in Fig. 8(b) seem to suggest the presence of an additional factor that influenced the wettability phenomena, as the acid-base surface energy (γ_s^{ab}) for nanoparticles functionalised with silane mixtures (trimethoxy(octyl)silane:nanoparticle = $40\text{--}200 \mu\text{L}\cdot\text{g}^{-1}$) was in a similar range (i.e., $8.2 \text{ mJ}\cdot\text{m}^{-2}$ ($40 \mu\text{L}\cdot\text{g}^{-1}$) to $7.9 \pm 0.3 \text{ mJ}\cdot\text{m}^{-2}$ ($60\text{--}200 \mu\text{L}\cdot\text{g}^{-1}$)), but the water contact angle increased significantly from $30.5^\circ \pm 1.0^\circ$ ($40 \mu\text{L}\cdot\text{g}^{-1}$) to $148.4^\circ \pm 3.5^\circ$ ($60\text{--}200 \mu\text{L}\cdot\text{g}^{-1}$). This is further confirmed by comparing the water contact angle measurements for nanoparticle functionalised with pure trimethoxy(octyl)silane (silane:nanoparticle = $5 \mu\text{L}\cdot\text{g}^{-1}$) and those functionalised with silane mixture (trimethoxy(octyl)silane:nanoparticle = $40 \mu\text{L}\cdot\text{g}^{-1}$). Although the former had a significantly higher acid-base surface energy ($9.0 \text{ mJ}\cdot\text{m}^{-2}$) than the latter ($8.2 \text{ mJ}\cdot\text{m}^{-2}$) due to the presence of more free silanol group ($-\text{SiOH}$), its water contact angle ($144.1^\circ \pm 0.4^\circ$) was still significantly higher than the latter ($30.5^\circ \pm 1.0^\circ$).

We propose the homogeneity or extent of defects of the octyl-chain network as the additional factor. Although the chemical functionalisation method used was relatively

successful, it was highly unlikely that all the free silanol group ($-\text{SiOH}$) was eliminated. As discussed earlier in this article, the hydrophobicity of nanoparticle surface is the result of the octyl-chain network, whose density determines its effectiveness in shielding the residual silanol group ($-\text{SiOH}$) on the nanoparticle surface from water. Defect occurs when some octyl-chains are replaced with the much shorter methyl-chains, creating hydrophilic patches within this network (Fig. 7—before and after particle substitution with methyl groups). The effect of such defects was negligible for the trimethoxy(octyl)silane to triethoxymethylsilane ratio as low as 60:140 (v/v) (Fig. 8), suggesting the robustness of the hydrophobic octyl-chain network as seen from the high water contact angle ($144.1^\circ \pm 0.4^\circ$) of nanoparticle functionalised with pure trimethoxy(octyl)silane (silane:nanoparticle = $5 \mu\text{L}\cdot\text{g}^{-1}$). However, as the proportion of triethoxymethylsilane continued to increase in the silane mixture, the inhomogeneous or defect-rich octyl-chain network reached a tipping point where the hydrophilic patches could form an interconnected network, resulting in a significant decrease in water contact angle as the triethoxymethylsilane to nanoparticle ratio reached $160 \mu\text{L}\cdot\text{g}^{-1}$.

The results presented in this article are highly relevant for understanding the nature of hydrophobicity and the control strategy. For many of the existing studies that utilised mixtures of hydrophilic and hydrophobic silanes to reduce the hydrophobicity of the target surface, the current findings suggest that the reduction in hydrophobicity was partly due to the disruption of the hydrophobic network

Fig. 8 (a) Water contact angle and (b) acid-base surface energy (γ_{AB}) for modified nanoparticles (total silane:silica NP = 200 $\mu\text{L}\cdot\text{g}^{-1}$). Note: error bars are too small to be visible.

Fig. 9 Surface energy of silica nanoparticles functionalised with triethoxymethylsilane and trimethoxy(octyl)silane mixtures (total silane:silica NP = 200 $\mu\text{L}\cdot\text{g}^{-1}$).

and partly due to the presence of the hydrophilic groups. For studies focusing on the design of superhydrophobic surface, the current findings reveal the important design variables including alkyl chain length and homogeneity of the hydrophobic network.

4 Conclusions

In this study, spherical silica nanoparticles were functionalised with silanes that contain methyl- and octyl-functional groups. While the pristine silica nanoparticles were hydrophilic (water contact angle = 0°) due to the

presence of free silanol groups, the octyl-functionalised nanoparticles were highly hydrophobic (water contact angle between $144.1^\circ \pm 0.4^\circ$ and $158.1^\circ \pm 0.2^\circ$), demonstrating the effectiveness of the octyl-chain network in shielding the hydrophilic nanoparticle from water. In contrast, methyl-functionalised nanoparticles were hydrophilic (water contact angle = 0°) regardless of the triethoxymethylsilane/nanoparticle ratio, suggesting the importance of alkyl chain length in forming a hydrophobic network. More importantly, by functionalising the nanoparticles with a mixture of triethoxymethylsilane and trimethoxy(octyl)silane and hence incorporating both functional groups onto the surface of the nanoparticles, it was shown that the homogeneity of the octyl-chain network was crucial for hydrophobicity, as increasing the triethoxymethylsilane to trimethoxy(octyl)silane ratio to 160:40 (v/v) could reduce the water contact angle significantly from $148.4^\circ \pm 3.5^\circ$ (60–200 μL trimethoxy(octyl)silane $\cdot\text{g}^{-1}$) to $30.5^\circ \pm 1.0^\circ$ (40 μL trimethoxy(octyl)silane $\cdot\text{g}^{-1}$). The findings in this study are useful for the intelligent design of surface with tuneable hydrophobicity in environmentally relevant applications.

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