



Effect of heat treatment on the morphology of carbon fibers doped with Co₂P nanoparticles

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Abstract

Carbon fibers (CFs) decorated by Co₂P nanoparticles and carbon nanotubes were prepared via needle-less electrospinning technique. Formation of catalytically active Co₂P nanoparticles and growth of carbon nanotubes were monitored in open and closed sintering environment at different sintering exposure times. Higher porosity, important in the catalytic reaction for easier penetration of electrolyte into the CFs, was achieved by mixing two immiscible polymers with sodium dodecyl sulfate and subsequent heat treatment process. Structure and morphology of the prepared modified carbon fibers were characterized by XRD, SEM and TEM. The time of heat exposure at the sintering temperature of 1200 °C and closure of the sintering space showed distinct effect on the growth and shape of carbon nanotubes. SEM and Raman spectroscopy revealed that closure of the system led to the formation of carbon nanotubes with smaller diameters and less structural disorder. Comparing of as-prepared CFs revealed that CFs with Co₂P sintered in the closed system exhibited the best electrocatalytic activity for hydrogen evolution reaction due to lower overpotential and smaller Tafel slope in acidic solution.

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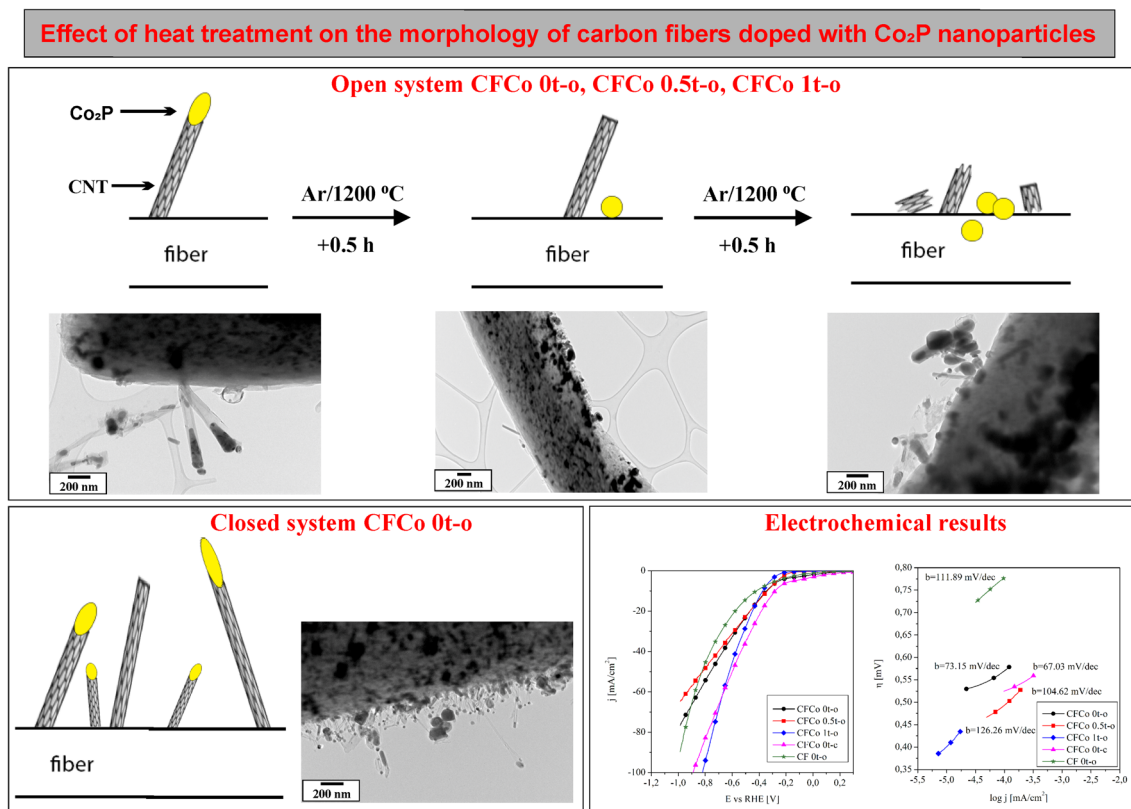
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Graphic abstract



Keywords Carbon fibers · Cobalt phosphide · Heat treatment · HER

Introduction

The high costs of coal and gas, together with negative impact of fossil fuels on the environment and human life, have led to increased interest in the use of renewable energy. Moreover, according to the “Global Energy Review” published in April 2020 by the International Energy Agency, as one of many global world’s associations for shaping, secure and sustainable energy for future, in the, renewable energy has so far been the energy source most resilient to Covid-19 lockdown measures. According to this review, which examines the impact of the Covid-19 crisis on overall energy demand and CO₂ emissions, global use of renewable energy in all sectors increased in the first quarter of 2020 by about 1.5% relative to the same quarter of 2019 (emissions 2020). Despite supply chain disruptions that have paused or delayed activity in many regions, the expansion of solar, wind and hydropower was expected to help renewable electricity generation to rise by nearly 5% in 2020 (emissions 2020). Unfortunately, fossil fuels still cover almost 60% of global electricity supply, so water, wind and solar energy as alternative energy sources present new challenges in the field of safety of energy

supply, independent distribution, production, storage and use of surplus energy (Lisserre et al. 2010; Stigka et al. 2014). Although renewable energy is often seen as a new technology, the use of natural power has long been used for heating, transportation, lighting and more.

Hydrogen is considered a suitable energy carrier for the future due to its high energy density, high incidence and positive impact on the environment in terms of greenhouse gas emissions (Streckova et al. 2020). Water is an abundant source of hydrogen, and therefore, much effort is aimed at extracting hydrogen from water. Electrolysis of water using a direct current derived from sustainable energy is highly eco-friendly method for production of hydrogen, as it uses renewable H₂O and produces only pure oxygen as a by-product. Unfortunately, due to economic issues, only 4% of hydrogen can be currently obtained by electrolysis of water. However, water electrolysis supported by alternative energy sources (e.g., solar energy) and alternative use of other catalysts instead of noble metals could be even more economically advantageous at present. According to the experimentally measured electrochemical parameters of HER (exchange current density, overpotential, electron

transfer kinetics, etc.) and DFT (density functional theory) calculated from the Gibbs free energy ΔG_H of the adsorbed atomic hydrogen, three most perspective alternatives for catalysts are: metals and alloys, transition metal compounds, and carbonaceous compounds (Nørskov et al. 2005; Greeley and Nørskov 2005; Wang and Johnson 2009; Zhang et al. 2018; Andrews and Citra 2002; Zeng and Li 2015; Kibsgaard et al. 2015).

Distinctive properties of carbon materials, such as wide operating potential, large variability of modifications by shaping or easy incorporation into other materials, low cost and inertness against most electrolytes, make them suitable catalysts for HER. However, amorphous carbon is an electroactive inert and possesses poor catalytic activity (Streckova et al. 2020). Therefore, it is necessary to supplement other catalysts into the carbonaceous materials that increase their electrocatalytic activity. Alternatively, one may design more suitable carbon structures such as carbon fibers (CFs), carbon nanotubes (CNTs) single-walled CNTs (SWCNTs), graphene, or glassy paper. It is known that the incorporation of transition metal nanoparticles (TMN), such as Co, Cu, Ni, Fe, significantly increases the electrocatalytic activity, the efficiency of catalysts (Zhou et al. 2016; Shi and Zhang 2016; Kong et al. 2013), and also initiates the growth of carbon nanotubes (CNTs) (Streckova et al. 2020). Carbon with tubular nanostructures such as CNTs and SWNTs has attracted considerable attention not only because of its fascinating, unique physicochemical properties, but also due to its promising applications in transistors (Bachtold et al. 2001), field-emission tips (Iijima 1991), sensors (Zhu et al. 2002), supercapacitors (An et al. 2001), and bio-medical fields (Zanello et al. 2006). Various established methods are used to prepare carbon fibers and CNTs, such as electric-arc discharge, laser ablation synthesis techniques and catalytic pyrolysis of carbon-containing gases via catalytic chemical vapor deposition (CCVD) (Hernadi et al. 2000; Lyu et al. 2004; Bonard et al. 2002; Wei et al. 2003).

Needle electrostatic spinning (NLE) is the most suitable method for production of fibers from polymer solutions (Jirak and Petrik 2012). The principle of electrospinning is based on the generation of free charges on the surface of a polymer solution with a high voltage potential (in tens of kV) (Garg and Bowlin 2011; Inagaki et al. 2012). A Taylor cone is formed at the end of the needle, from which fibers are formed after overcoming the surface tension and are trapped on the collecting counter electrode. By this technique, the continual fibers can be created and modified by nanoparticles directly in the spinning process. Needle-less electrospinning (NLES) or electrospinning of free liquid surfaces can simultaneously generate a number of streams from the solution surface, which increases the amount of fibers prepared by an order of magnitude, thus significantly reducing the time required to produce fibers (Niu et al. 2019).

NLES is a promising technology for mass-scale production of nanofibers. Although there are some issues with the practical application, NLES is still the first choice for industrial production of nanofibers, considering its advantages and potential for continual improvement. Thanks to close collaboration between electrospinning, chemists, designers, machinery, electricians and automation controls engineers, NLES will produce high-quality nanofibers in a well-controlled way at low cost and be beneficial to our daily life, health care and various industrial sectors (Ding et al. 2019).

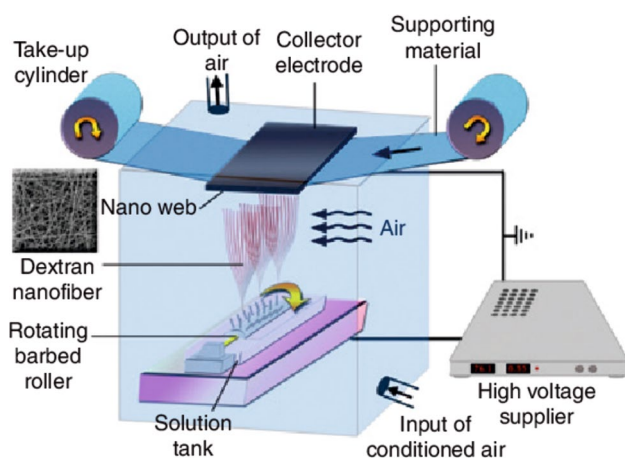
According to the literature and authors' knowledge, three main factors affect the catalytic behavior of carbon fibers prepared by needle-less electrospinning: i. structure of CF (Heckova et al. 2020) morphology and porosity of the fibers (size and diameters of CF, defect-less structure, distribution of incorporated nanoparticles, etc.); ii. modification of CF (Streckova et al. 2020)—surface or interior modification by additional nanoparticles or additives (CNT growth on the surface, type of TMN, etc.); iii. final heat treatment, which can influence the building of the carbon matrix and affect the distribution of TMN and growth of CNTs (Streckova et al. 2016).

The type and dispersion of the catalyst are key factors for the formation and controlling the growth of CNTs. Therefore, a great effort has been paid to explore the optimal catalysts to enhance reactivity and efficiency for HER. It was found that the growth and structure of CNTs was greatly influenced by the form and occurrence of Co in the substrate structure (Xiang et al. 2009). The role of cobalt catalyst in the Co-MCM-41 material at different pretreatment stages of the SWCNT synthesis was investigated by Chen et al. (Chen et al. 2004a, 2004b, 2006). During the SWCNT growth process, cobalt in Co-MCM-41 substrate becomes more active after partial reduction by hydrogen prior to CO exposure. A narrow distribution of SWNT diameter can be achieved by optimized prereduction (500–600 °C) and reaction temperature (750–800 °C) and by control of cobalt cluster sizes (Chen et al. 2004a). The experimental results showed that the effect of CO pressure and reaction time controls synthesis of SWCNTs depending on three competitive processes: reduction in cobalt, nucleation of the reduced cobalt atoms into clusters, and initiation and growth of the carbon nanotubes (Chen et al. 2004b). The concentration and size of Co nanoparticles also affected the growth of SWCNTs. The uniform SWCNT diameters were achieved at a cobalt cluster size of approximately 6–8 Å with concentration of Co catalytic nanoparticles of 3 wt.% (Chen et al. 2006).

In this article, the main goal is the investigation of the effect of heat treatment on the morphology and electrocatalytic activity of CFs modified by cobalt phosphide nanoparticles (CoP, Co₂P) for HER. Polymeric fibers containing the source compounds for Co₂P nanoparticles formation have been produced by NLES. These compounds are represented

Table 1 Composition and exposure times at 1200 °C for the precursor fiber solutions

Sample	DMF [ml]	PAN [g]	PVP [g]	H ₃ PO ₄ [μl]	SDS [g]	Co(NO ₃) ₂ [g]	exposure time [h]
CF 0t-o	10	1	1	100	0.3	0	0
CF 0t-c	10	1	1	100	0.3	0	0
CFCo 0t-o	10	1	1	100	0.3	1	0
CFCo 0t-c	10	1	1	100	0.3	1	0
CFCo 0.5t-o	10	1	1	100	0.3	1	0.5
CFCo 0.5t-c	10	1	1	100	0.3	1	0.5
CFCo 1t-o	10	1	1	100	0.3	1	1

**Fig. 1** Schematic diagram of the needle-less electrospinning system (Yalcinkaya et al. 2016)

by cobalt nitride and phosphoric acid. The further main step of modified carbon fibers preparation is sintering at 1200 °C in Ar atmosphere. The heat treatment process is responsible for carbon ladder matrix formation with incorporated Co₂P nanoparticles (Streckova et al. 2018). Cobalt phosphate nanoparticles have been incorporated into the carbon matrix in situ in the heat treatment process and act as the catalysts for CNT growth during sintering. The growth of CNT was performed according to the “Tip growth model” (Streckova et al. 2020).

No additional surface modification was necessary. The used sintering conditions had a great influence on the morphology, porosity and growth of CNTs on the CF surface, which had a positive effect on the active surface area for HER. CNT growth was catalyzed by the presence of CoP nanoparticles. Fiber morphology and CNT growth were observed by SEM and TEM. The type of phosphides was determined by XRD, and the amount of Co in the fibers was determined by atomic absorption spectroscopy (AAS). The electrocatalytic activity of fibers and the effect on hydrogen

evolution were analyzed using linear sweep voltammetry and Tafel slopes. The turnover frequencies (TOF) were calculated to further assess the electrocatalytic activity of two most active samples.

Experimental

Materials

The solutions used to prepare the precursor fibers were composed of two polymers: polyacrylonitrile (PAN, Mw = 150 000 g.mol⁻¹, Sigma-Aldrich) and polyvinylpyrrolidone (PVP, Mw = 55000 g.mol⁻¹, Sigma-Aldrich). Phosphoric acid (H₃PO₄, Merck, 85%) was employed to enhance the viscosity and conductivity and hence, the spinnability of the precursor solution. Sodium dodecyl sulfate (SDS, Merck, 90%) was used as a surfactant to improve porosity. N,N-dimethylformamide (DMF, Acros Organic, 99.8%) was used as a solvent. Cobalt nitrate hexahydrate [Co(NO₃)₂·6H₂O, Sigma-Aldrich, p.a.] was added as a source of Co nanoparticles. Electrochemical measurements were performed in 0.5 M sulfuric acid (H₂SO₄, Centralchem, 96%).

Methods

Fibers preparation

Precursor solutions were prepared by dissolving PAN, PVP, SDS and Co(NO₃)₂ in DMF for two hours at 80 °C. Then, H₃PO₄ was added dropwise, and the solutions were stirred vigorously for another 1 h at 90 °C until homogenization was complete. The notation of the as-prepared CF samples with the composition of the precursor polymer solutions and the sintering exposure times at 1200 °C is given in Table 1

Carbon fibers (CFs) were prepared by needle-less electrospinning (NLES) using Nanospider™ NS LAB (Elmarco, Czech Republic) from precursor solutions. The precursor solutions were poured into a spinning bath (30 ml vessel and a rotating electrode), and the spinning distance between the electrodes was kept at 180 – 190 mm. The rotation speed

of the spinning electrode was adjusted to 8 rpm, and a high voltage of 50–60 kV was applied to the surface of the spinning drum. A schematic diagram of the needle-less electrospinning system is depicted in Fig. 1 (Yalcinkaya et al. 2016).

The heat treatment optimization of PAN-based fibers is the key factor, especially in such composite systems as modified CFs with cobalt phosphide nanoparticles (Rahaman et al. 2007). The sintering of the CFs was carried out at 1200 °C in Ar atmosphere in an open system (referred to as -o), which means free access of Ar into the sample with the possibility of releasing gases to the environment. Another way was sintering in a closed system (referred to as -c) where the gases originated during sintering process (H_2 , CO_2 , CO and Ar) stayed in the system and were prevented from escaping. The sintering time at the maximum temperature is referred to as “-t” (e.g., -0t means no exposure time, -1t means 1 h). The complete notation of the samples sintered in a closed and open system is summarized in (Table 1).

Fibers characterization

Structural characterization of the prepared fibrous samples was done by XRD analysis (PhilipsX' PertPro, $CuK\alpha$

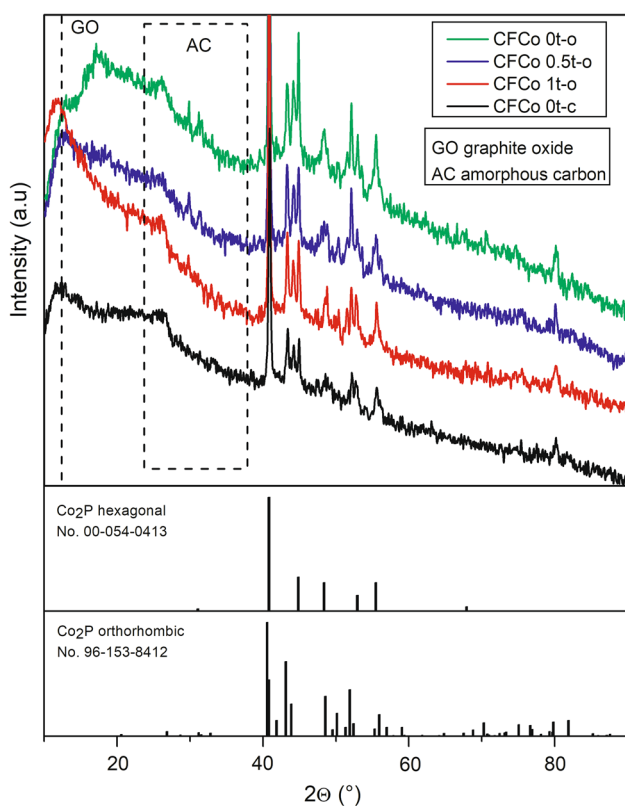


Fig. 2 XRD diffraction pattern for CFCo 0t-o, CFCo 0.5t-o, CFCo 1t-o, CFCo 0t-c

radiation). The morphology, size and shape of modified CFs were visualized by scanning electron microscope (SEM, JSM-7000F, JEOL, Japan) and transmission electron microscope (TEM, JEM-2100F, JEOL, Japan). Active surface area for HER was determined from adsorption/desorption isotherms using the NOVA 1200e Surface Area & Pore Size Analyzer (Quantachrome Instruments, USA). The values of specific surface area (S_{BET}) were determined by the Brunauer–Emmett–Teller method (S_p). Atomic absorption spectroscopy (AAS, Atomic Absorption Spectrometer PerkinElmer) was used to determine the cobalt content in the fibers. The Raman spectra were measured by the Raman spectroscopy (HORIBA -BX41TF).

The electrocatalytic HER performance was evaluated by linear sweep voltammetry (LSV) method in a three-electrode cell configuration connected to the Autolab potentiostat (PGSTAT302N, Metrohm, Switzerland) at ambient conditions. LSV measurements were performed in 0.5 M H_2SO_4 aqueous solution at a scan rate of 1 mVs^{-1} in the potential range from +600 mV to -1200 mV (vs. $Ag/AgCl/3\text{ M KCl}$). A silver chloride electrode ($Ag/AgCl/3\text{ M KCl}$) and a large area platinum foil were used as the reference and the counter electrode, respectively. All potentials were calibrated to a reversible hydrogen electrode (RHE). The carbon fibers in the form of carbon foam, used as compact electrode for electrochemical measurement, were weighted and directly used as working electrodes without further modification. All polarization curves were measured without IR correction.

Results

XRD characterization

The phase structure and crystallinity of carbon fibers modified by cobalt phosphides CFCo 0t-o, CFCo 0.5t-o, CFCo 1t-o, CFCo 0t-c with different heat treatment conditions were first identified by XRD. The obtained XRD pattern is shown in Fig. 2. Two different structures of cobalt phosphide were present in all forms of CFs. Two types of Co_2P nanoparticles were observed: i. hexagonal, characterized by diffraction peaks $\sim 2\theta = 40.836^\circ, 44.856^\circ, 48.376^\circ, 52.946^\circ, 55.477^\circ$ and indexed to (111), (021), (120), (002), (030) planes, and ii. orthorhombic, with diffraction peaks at $\sim 2\theta = 40.573^\circ, 40.837^\circ, 43.144^\circ, 43.888^\circ, 48.560^\circ, 50.145^\circ, 51.911^\circ, 79.808^\circ, 81.889^\circ$ indexed as (112), (210), (211), (103), (013), (301), (020), (322), (314) planes. The characteristic diffraction of graphite/graphene oxide (GO) peak at $\sim 2\theta = 12.4^\circ$ corresponding to the oxidation of the carbon structure and the formation of an oxygenated functional groups was identified (Johra et al. 2017; Kartick et al. 2013). Increasing GO peak intensity was detected with increasing exposure time at high temperature (the order, starting from

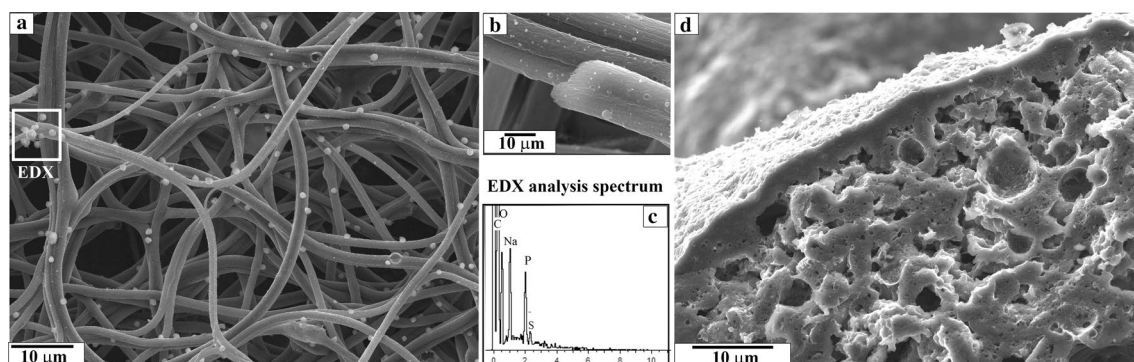


Fig. 3 SEM image of bare CFs, thermally treated in Ar atmosphere at 1200 °C: (a) fibrous structure of the CF 0t-o sintered in an open system, (b) detail of the morphology of CF 0t-o, (c) EDX analysis of CF 0t-o, (d) deteriorated structure of CF 0t-c sintered in a closed system

the lowest intensity, was CFCo 0t-o, CFCo 0.5t-o and CFCo 1t-o). Similarly, GO peak intensity decreased in the closed system CFCo 0t-c, where the presence of released hydrogen led to suppression of oxidation reactions. The small peak at $\sim 2\theta = 24.3^\circ$ with the additional broadening at the bottom of the peaks around $\sim 2\theta = 43.2^\circ$ can be identified as amorphous carbon (AC) in all observed samples. Generally, the broadening around the peak $2\theta \approx 20^\circ$ in XRD diffraction pattern can be assigned to the amorphous carbon what is typical for carbonaceous materials.

Morphological characterization

Bare carbon fibers

Bare carbon fibers were prepared from a mixture of PAN and PVP polymers in combination with SDS as a surfactant according to Table 1. The use of these two immiscible polymers in combination with SDS led to the creation of micelle structure (Heckova et al. 2020). Ionic surfactants play an important role in the fabrication of carbon-based fibers, because they facilitate the processability of carbon precursors and allow controlled alteration of carbon properties. It has been found that the microporosity of the electrospun carbon nanofibers is promoted in the presence of anionic sodium dodecyl sulfate (SDS) surfactant, regardless of whether metal particles are presented or not (Aykut et al. 2016). According to our previous studies (Heckova et al. 2020), heat treatment and extraction of the micelles from the porous structure of PAN matrix is suitable for better penetration of the electrolyte during the HER process; therefore, the formed micelles were extracted at high carbonization temperatures.

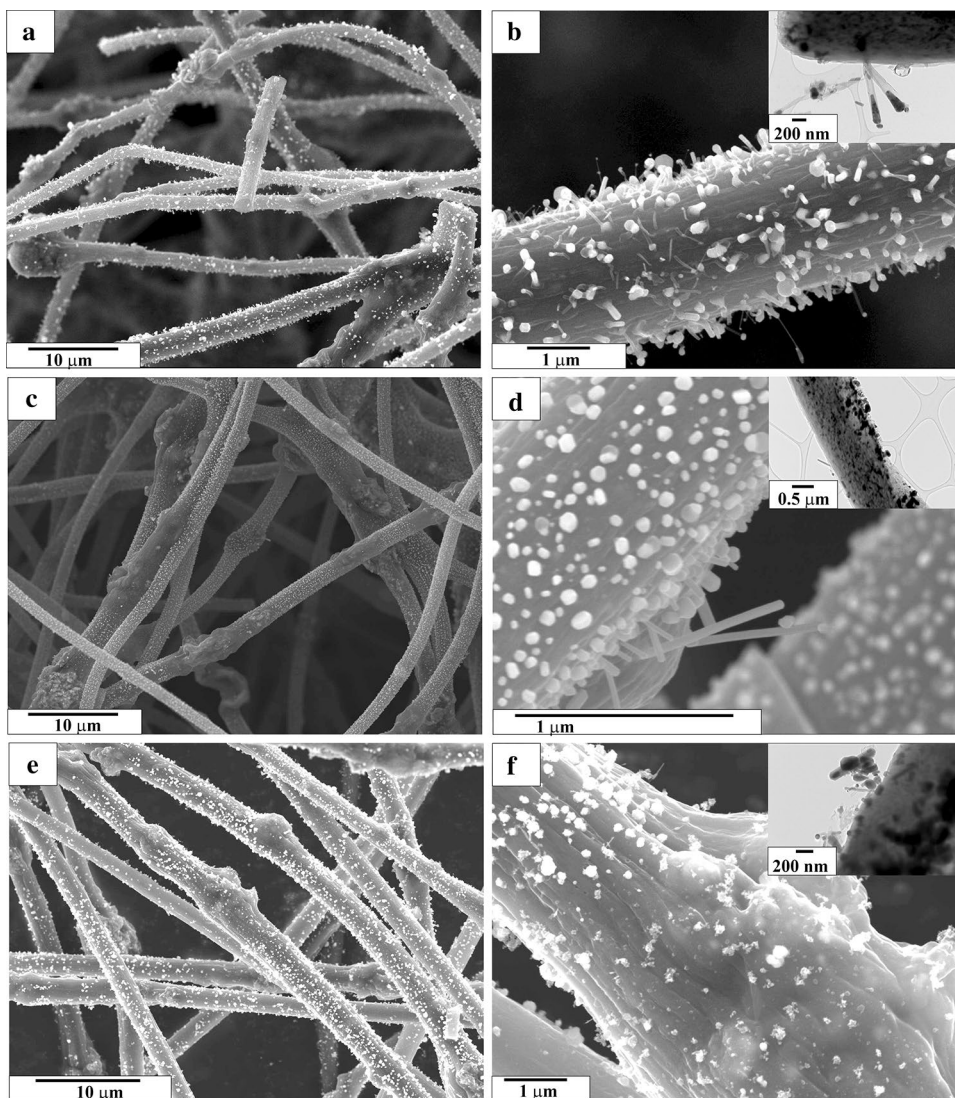
Regarding the evolution of hydrogen during the polymer ladder building, two different ways of the carbonization process were studied. The first was the sintering process in an open system (CF 0t-o), where the precursor fibers were loosely carbonized and all by-products could evolve into the

environment. The second one was realized in a closed system (CF 0t-c), where the fibers were carbonized in closed crucibles to prevent the release of hydrogen from the system, which also prevented the surface of the fibers being disturbed and thus, the porosity was increased. A comparison of the morphological characteristics of the two pure carbon fibers CF 0t-o and CF 0t-c calcined in an Ar atmosphere at 1200 °C is shown in the SEM images in Fig. 3. The continuous fibrous structure was observed for CF 0t-o sample with a diameter of 1.2 to 2.5 μm with a fine wrinkled surface (Fig. 3a,b). A lot of small spherical particles were dispersed outside and inside the carbon fibers, which were identified by EDX analysis as sodium phosphate particles created after the heat treatment process (Fig. 3c). Figure 3d shows the structure of the sample sintered in the closed system. As evident, the accumulation of the heat and hydrogen (as a by-product of the PAN pyrolysis) in the fibrous environment led to the deterioration and collapse of the fibrous structure (Fig. 3d). The closure of the system assumed accumulation and re-spilling of hydrogen into the carbon structure, which is expected to increase the porosity. However, the effect of hydrogen in a closed system is destructive and does not produce the desired fiber properties; in the case of bare carbon fibers, this is reflected a specific surface area as low as 13.5 m^2/g .

Carbon fibers modified by cobalt phosphide nanoparticles

Co_2P phosphides belong to the family of materials, which enhance the electrocatalysis of hydrogen from water. CNTs increase the active surface area and support the conductivity of the final material. Therefore, the prepared bare CFs were subsequently modified with both Co_2P nanoparticles and CNTs. It should be noted that Co_2P nanoparticles were formed in carbon fibers spontaneously at high temperature and in an inert atmosphere from polymer precursor fibers containing $\text{Co}(\text{NO}_3)_2$. Effect of the sintering temperature and time of exposure in the open system on the microstructure of

Fig. 4 SEM and TEM images of carbon fibers with incorporated Co_2P nanoparticles thermally treated in Ar atmosphere in open system with prolonged time of exposure at $1200\text{ }^\circ\text{C}$: (**a, b**) CFCo 0t-o without exposure (**c, d**) CFCo 0.5t-o with exposure of 0.5 h (**e, f**) CFCo 1t-o with exposure of 1 h



CFCo 0t-o, CFCo 0.5t-o, CFCo 1t-o samples was monitored by SEM and TEM (Fig. 4). As can be seen, the continuous fibrous structure was maintained in all cases of different exposure times. The rough and wrinkled surface with a fiber diameter of $0.8 - 3.9\ \mu\text{m}$ contained Co_2P nanoparticles densely distributed outside as well as inside the carbon fiber matrix (Fig. 4a,c,e). SEM images at larger magnifications confirmed a homogeneous distribution of CNTs and Co_2P nanoparticles (Fig. 4b,d,f). Moreover, the presence of CNTs was observed on the surface of the samples, and their growth and amount were influenced by the sintering exposure time. As shown in Fig. 4, sintering of CFs upon thermal exposure led to a reduction in the amount of CNTs and metal phosphide nanoparticles. A significant amount of CNTs was observed on the carbon surface matrix of the CFCo 0t-o sample (Fig. 4a,b). TEM proved the growth of CNTs initiated by Co_2P nanoparticles (zoom in Fig. 4b). The prolonged time of exposure at high temperatures resulted in decrease

CNT growth (Fig. 4c,d). Due to the long exposure of heat treatment, the CNTs disintegrate and disappear, whereas agglomeration of Co_2P occurs (Fig. 4e,f).

The effect of the closed sintering system on the morphology of CFs modified by Co_2P nanoparticles was also observed (Fig. 5). By comparing of carbon fibers sintered in the open system, both samples exhibited fibrous structure, but the surface of CFCo0t-c was more wrinkled with irregular cylindrical structure and distribution of soft CNTs. TEM images confirmed the higher amount of thinner and finer CNTs (Fig. 5c). Extending the time of heat treatment at $1200\text{ }^\circ\text{C}$ to 0.5 h led to the extinction of fine CNT fibers; therefore, it was not appropriate to further prolong the sintering time. The deteriorated fibrous structure indicated the possibility of an overall collapse of the carbon fibers as in the case of pure carbon fibers (Fig. 3). A reduced amount of carbon nanotubes with extended sintering time is indicated in Fig. 5d.

Fig. 5 SEM and TEM images of carbon fibers with incorporated Co_2P nanoparticles thermally treated in Ar atmosphere at $1200\text{ }^\circ\text{C}$ in closed system: **(a)** CFCo 0t-c, **(b)** detail on CNTs of CFCo 0t-c sample **(c)** TEM of CFCo 0t-c, **(d)** TEM of CFCo 0.5t-c

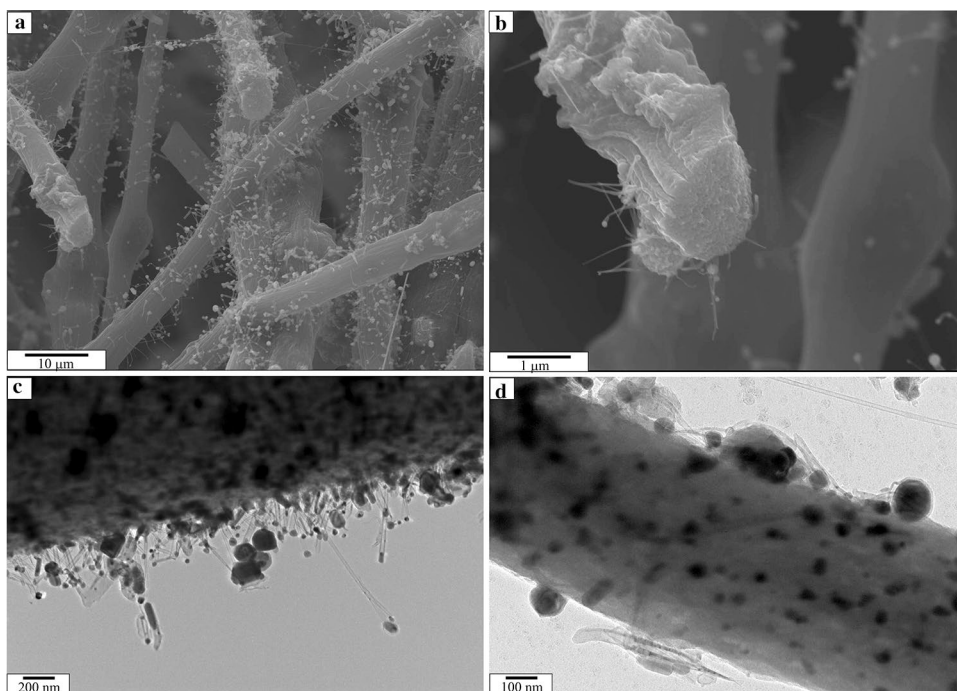
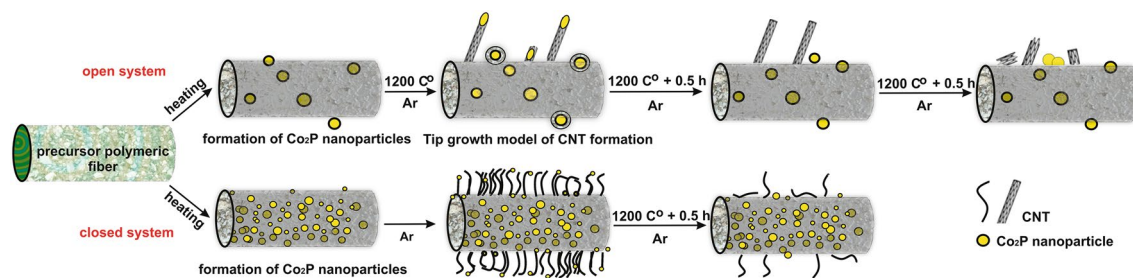
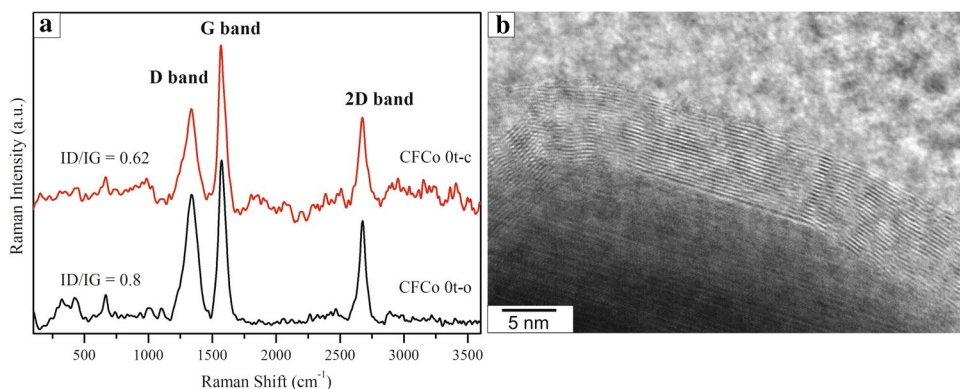


Fig. 6 **a** Raman Spectrum of CFCo 0t in closed and open systems, **b** HRTEM of CFCo 0t-o: multiwalled CNT growth on the carbon matrix initiated by the Co_2P nanoparticles



Scheme 1 Scheme of growth and subsequent degradation of CNTs due to the prolonged exposure time at $1200\text{ }^\circ\text{C}$ in Ar atmosphere

The CNT structure in CFCo 0t-o and CFCo 0t-c was confirmed by the Raman spectra (Fig. 6a,b). Both CFCo 0t-o and CFCo 0t-c samples had D-, G- and 2D peaks characteristic of graphene-based materials, which confirms the

presence of CNTs. Relatively high defect-activated Raman D peak indicates more disordered structure of CNTs. In addition, the small width of the 2D peak indicates the multilayer configuration of Co-initiated CNTs, as confirmed by

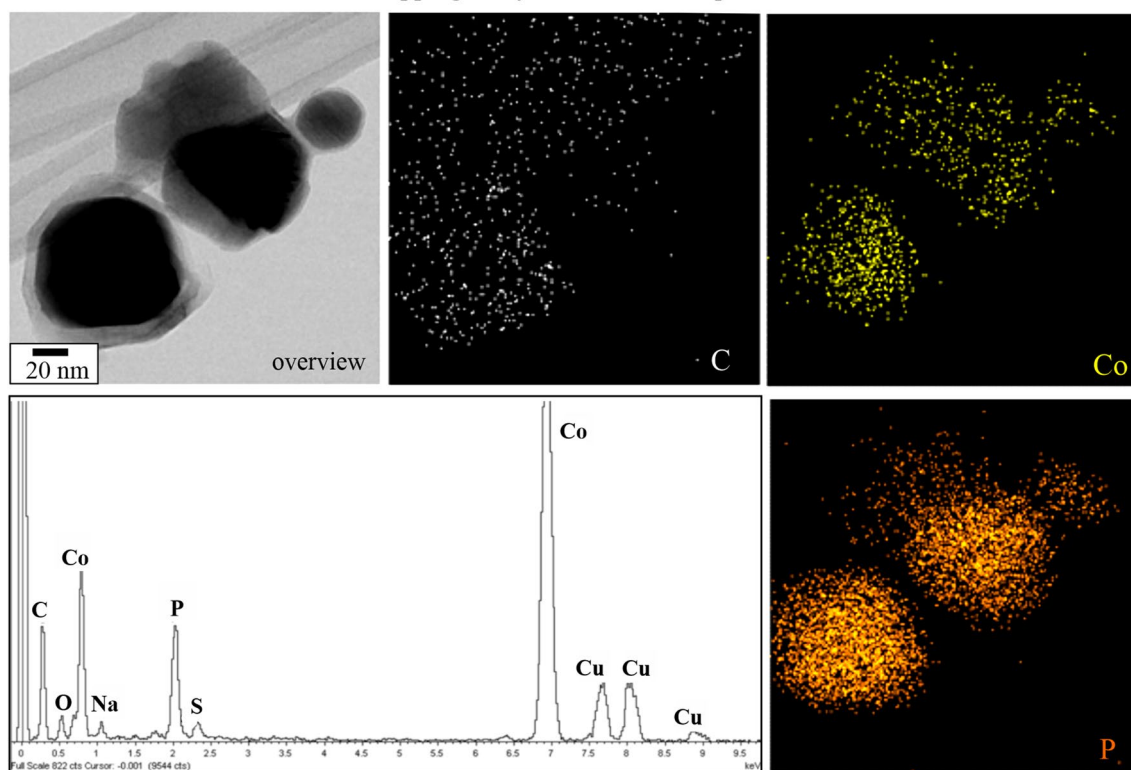
EDX mapping analysis of Co₂P nanoparticles

Fig. 7 Elemental mapping and EDX analysis of Co₂P nanoparticles

Table 2 Values of electrocatalytic activity parameters of carbon fibers with Co₂P nanoparticles for HER at different heat treatment conditions

Sample	m [mg]	S _{BET} [m ² /g]	S _t [m ² /g]	E _{onset} [mV]	η ₁₀ [mV]	η ₂₀ [mV]	B [mV/dec]	j ₀ [mA/cm ²]	Metal content %
CFCo 0t-o	4.1	349.6	57.75	-218.66	-345.07	-479.56	73.15	2.219	11.6
CFCo 0.5t-o	4.4	273.2	93.27	-226.81	-339.11	-482.45	107.62	0.268	13.1
CFCo 1t-o	4.1	55	/	-288.64	-372.67	-450.83	126.26	0.48	14.0
CFCo 0t-c	4.9	122.5	43.73	-157.16	-280.86	-380.99	67.03	1.754	9.67
CF 0t-o	9.1	13.5	/	-443.79	-421.80	-578.49	111.89	0.126	0

HRTEM (Fig. 6b). The D to G peak intensity ratio (ID/IG) is often employed to evaluate the level of disorder in CNTs. The lower ID/IG value (0.62) for the CFCo 0t-c sample indicates that the closure of the system led to a more ordered structure with higher amount of thinner CNTs. The mechanism of CNTs growth has been previously discussed by many other authors; however, the upraise of CNTs in situ by affecting temperature is very rare (Kumar and Ando 2010; Pérez-Cabero et al. 2004; Zhao et al. 2006; Yuan et al. 2003; Mo et al. 2001; Vinciguerra et al. 2003). As mentioned in the authors' previous work, Co and Co₂P nanoparticles can initiate the growth of CNTs as described in the “tip growth model” (Streckova et al. 2020). The suggested mechanism of growth and deterioration of carbon nanotubes affected by

the prolongation of exposure times at 1200 °C is depicted in Scheme 1. First, the high temperature causes the growth of CNTs that has been catalyzed by Co₂P nanoparticles immersed on the surface of CFs. The prolonged exposure time causes the falling out of Co₂P nanoparticles and degradation of carbon nanotubes. Finally, long exposure at the high temperature leads to the aggregation of Co₂P nanoparticles on the surface of carbon matrix and the subsequent penetration of Co₂P nanoparticles into the interior of the fibers. The collapse of CNTs is evident.

The chemical composition of Co₂P nanoparticles encapsulated in the carbon matrix was proved by the EDX mapping analysis shown in Fig. 7. It is evident that the

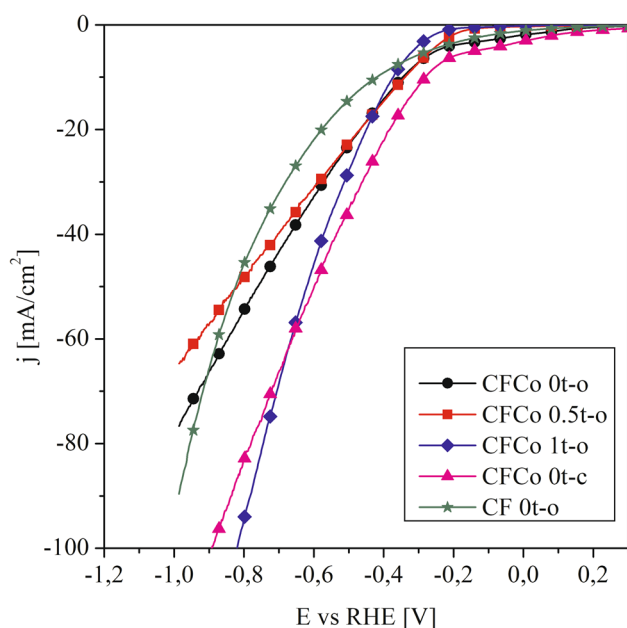


Fig. 8 LSV curves of CF with/without Co_2P nanoparticles thermally treated in Ar atmosphere at different exposure times

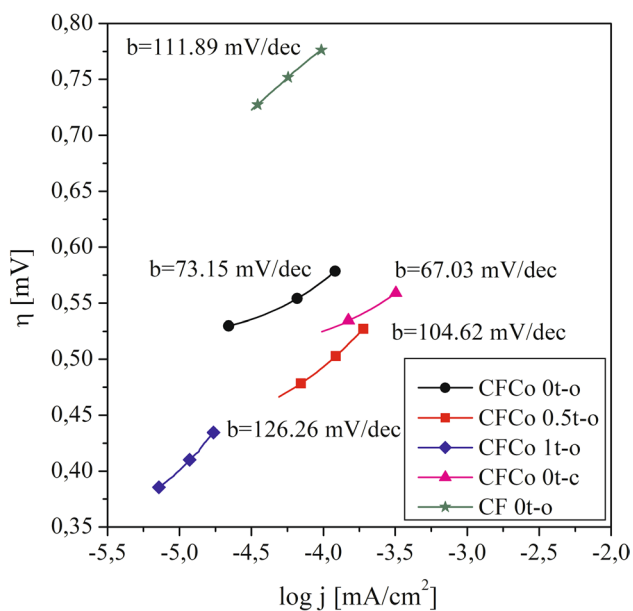


Fig. 9 Tafel slopes with/without Co_2P nanoparticles thermally treated in Ar atmosphere at different exposure times

homogeneous distribution of Co and P elements at the same spots was confirmed.

Electrochemical characterization

The HER activity measurements were carried out in an acidic medium of $0.5 \text{ mol} \cdot \text{dm}^{-3} \text{ H}_2\text{SO}_4$ using a

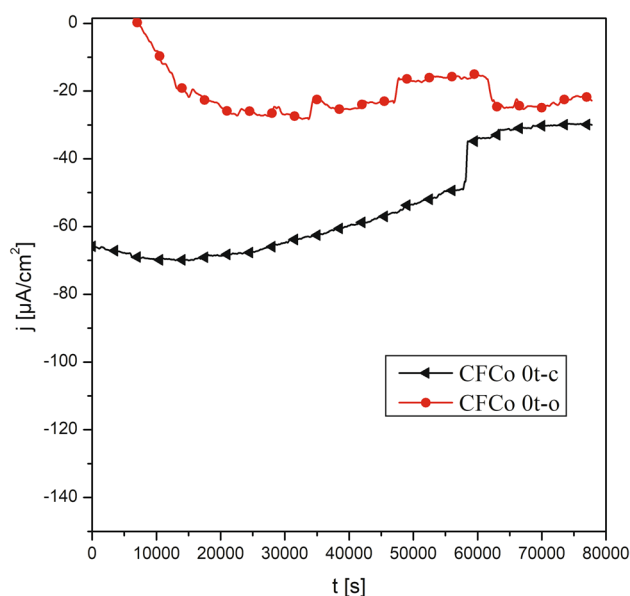


Fig. 10 Plots of current density vs. time held at a constant overpotential of 200 mV for 80,000 s for the two most efficient samples: CFCo 0t-c and CFCo 0t-o

three-electrode system. The electrocatalytic activity of carbon fibers was evaluated electrochemically on a compact fibrous sample. The fibers were not affected in any way prior to measurements to preserve the porosity, structure of CNTs and active sites for HER. The structure of fibers was maintained during the measurements. The HER activity of all studied fibrous samples was derived from the values of electrocatalytic activity parameters, such as onset potential (E_{onset}), overpotentials to achieve cathodic current density of 10 mAcm^{-2} (η_{10}) and 20 mAcm^{-2} (η_{20}), exchange current density (j_0) and Tafel slope (b) calculated from LSV curves. The data are summarized in Table 2. As can be seen in Fig. 8., the polarization curves revealed that the catalyst CFCo 0t-c produced in the closed sintering system without exposure exhibits the highest HER activity with the overpotential of -280.86 mV at the cathodic current density of $10 \text{ mA} \cdot \text{cm}^{-2}$ (η_{10}) and -380.99 mV at the cathodic current density of $20 \text{ mA} \cdot \text{cm}^{-2}$ (η_{20}) (Fig. 8). In contrast, catalysts prepared in the open system exhibit higher overpotentials with prolonged time of exposure at high temperatures. The worse HER performance of the samples sintered in the open system at longer exposure times could be associated with agglomeration of Co_2P nanoparticles and degradation of CNTs, resulting in lower active surface area (only $55 \text{ m}^2/\text{g}$ for the CFCo 1t-o sample). The catalyst without active cobalt phosphide nanoparticles CF 0t-o exhibits the lowest electrocatalytic activity among all studied samples, which is in good agreement with previous studies (Streckova et al. 2020). The poor HER activity of bare CFs indicates that incorporation of Co_2P nanoparticles into samples

significantly enhances the catalytic activity, which may be related to the fact that the Co_2P nanoparticles present in the fibers not only catalyze HER but also initiate the growth of CNTs. Tafel slopes (Fig. 9) were calculated using Tafel plots derived from the polarization curves. Smaller Tafel slope usually corresponds to a steep rise of the electrocatalytic current density and indicates a fast reaction rate. The lowest values of Tafel slopes were obtained for catalysts fabricated in the closed system without exposure at high temperature ($b = 67.03$ and 73.15 for CFCo 0t-c and CFCo 0t-o, respectively). This indicates that HER took place via Volmer and Heyrovsky pathway.

A long-term stability test was performed to assess the stability and durability of the best-performing CFCo fibers produced with zero time of exposure in closed and open systems (CFCo 0t-c, CFCo 0t-o). The undeniable stability of these samples can be seen from the time-dependent current density curves (Fig. 10) for a static overpotential of 200 mV and a catalytic activity sustained for over 22 h.

The turnover frequencies (TOF) were calculated to further assess the electrocatalytic activity of two the most active samples (CFCo 0t-o and CFCo 0t-c). TOF estimation for the HER in 0.5 M H_2SO_4 was performed conceding that each individual Co and P atom on the outermost surface layer is a potentially active site due to the vagueness of the hydrogen binding sites (Nørskov et al. 2005; Chen et al. 2014). Both theoretical and experimental surface areas were used for TOF calculations. Experimental surface areas were derived from S_{BET} values, supposing that the surface area of Co_2P nanoparticles represents 10% and 12% of experimentally determined surface area for CFCo 0t-o and CFCo 0t-c samples, respectively, based on the Co content (Table 2). Theoretical surface areas were calculated assuming 70 nm (d) spherical Co_2P particles. The TOF values for CFCo 0t-c sample calculated utilizing the theoretical surface area are 67.77 s^{-1} at $\eta = 100 \text{ mV}$ and 91.41 s^{-1} at $\eta = 200 \text{ mV}$. The TOF values estimated for this sample using experimental surface area are 73.73 s^{-1} at $\eta = 100 \text{ mV}$ and 99.45 s^{-1} at $\eta = 200 \text{ mV}$. The TOFs obtained by both methods are in good agreement. The TOF values for CFCo 0t-o sample determined by use of the theoretical surface area are 9.46 s^{-1} at $\eta = 100 \text{ mV}$ and 12.96 s^{-1} at $\eta = 200 \text{ mV}$, and TOFs calculated from experimental surface area are 3.87 s^{-1} at $\eta = 100 \text{ mV}$ and 4.11 s^{-1} at $\eta = 200 \text{ mV}$. The obtained TOF values represent only an approximation of the real TOFs, as the actual hydrogen binding sites are not exactly identified. However, it is possible to conclude that the intrinsic catalytic activity of the CFCo 0t-c sample is about 10 times higher than the activity of the CFCo 0t-o catalyst. The obtained TOF values proved the beneficial effect of the use of a closed sintering system during the production of the catalyst on its catalytic activity.

Conclusion

In summary, carbon micro-fibers modified with Co_2P nanoparticles and decorated by CNTs were prepared by needle-less electrospinning technique. As expected, after optimization of sintering conditions and the structural and compositional metrics, carbon fibers with cobalt phosphides sintered in closed systems exhibited better electrochemical performance for HER than bare carbon fibers or carbon fibers with cobalt phosphides sintered freely. On the other hand, carbon fibers sintered in open system without prolonged exposure time at the $1200 \text{ }^\circ\text{C}$ showed better electrochemical activity than the same carbon fibers held in the furnace at high temperature for longer time. More specifically, fibers CFCo 0t-c calcined in the closed system require an overpotential of -280 mV to afford a current density of $10 \text{ mA}\cdot\text{cm}^{-2}$ with a low Tafel slope of $67.03 \text{ mV}\cdot\text{dec}^{-1}$, while the samples CFCo 0t-o calcined in the open system showed the overpotential of -345.07 mV for the same current density with the Tafel slope of $73.15 \text{ mV}\cdot\text{dec}^{-1}$. Both electrodes exhibit excellent stability under operating conditions.

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