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Gel electrolytes based on polyacrylonitrile/thermoplastic polyurethane/polystyrene for lithium-ion batteries

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Abstract

Using an electrospinning method, high-performance gel polymer electrolytes (GPEs) comprising polyacrylonitrile/ thermoplastic polyurethane/polystyrene (PAN/TPU/PS) were prepared. Then, these electrolytes were used to assemble button cells. The GPEs exhibited a maximum ionic conductivity of 3.9×10^{-3} S cm⁻¹ at room temperature, with an electrochemical stability of 5.8 V. In addition, a Li/GPE/LiFePO₄ cell showed first charge and discharge capacities of 161.70 mAh g⁻¹ and 161.44 mAh g⁻¹, respectively, at a 0.1 C rate and a stable cycle performance (the capacity retention was nearly 94% after 50 cycles) with high coulombic efficiency. Additionally, the SEM images showed that the PAN/TPU/PS was not composed of beaded-fibres in a fibrous membrane but was composed of fibres that were slim with a uniform size distribution. The uptake behaviour and porosity of the PAN/TPU/PS fibres showed the best performance and superior physical properties of the tested samples (the elongation at break of the film was 115.15%, and the tensile strength was 9.86 MPa). Considering these results, the PAN/TPU/PS blend/composite film has broad prospects for use in lithium-ion batteries.

Keywords Electrospinning · Gel polymer electrolyte · Polyacrylonitrile · Thermoplastic polyurethane · Polystyrene

Introduction

With the consumption of non-renewable resources, such as petroleum and coal, lithium-ion batteries (LIBs) have received attention as the first of the new energy technologies. The electrolyte is an important component of LIBs, and its chemical stability and safety are important factors in guaranteeing a long cycle life and high security for the LIBs [1–4]. Gel polymer electrolytes (GPEs) [5–7] not only have the high ionic conductivities of liquid electrolytes [8–10] but also have a series of advantages, such as solving the difficult problem of electrolyte leakage, flammability, and explosivity [11]. However, the application of gel polymer electrolytes in LIBs is still difficult [12, 13] because of their problems such as low ionic conductivities and poor mechanical strengths at room temperature.

Polyacrylonitrile (PAN) [14–18] has outstanding properties, such as high thermal stability, high ionic conductivity, good compatibility with lithium electrodes, a wellformed morphology for electrolyte absorption, and the ability to minimise dendritic formation during the charge and discharge processes. PAN is superior to poly(vinylidene fluoride) (PVDF) [19, 20] because of its interfacial properties and minimization of dendritic growth [21]. The C≡N groups in PAN can interact with the lithium-ions and C=O groups of ethylene carbonate (EC), which makes it more versatile for a variety of applications [22]. However, there severe interfacial passivation occurs between PAN-based gel polymer films and lithium electrodes. Additionally, the mechanical properties of PAN decrease as the contents of the plasticiser increase [23]. Therefore, PAN cannot be used as a free-standing membrane in lithium-ion batteries. Therefore, to improve the mechanical stability, PAN has always been modified via copolymerization or blending with other materials.

Thermoplastic polyurethane (TPU) is an elastomer that has a high tensile strength, high elasticity, and low crystallinity [24, 25]. Having good mechanical properties of flexibility and rigidity is very important for LIB



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membranes. TPU has a two-phase microstructure with soft segments and hard segments [26-29]. The hardphase segments are connected to the soft-phase segments and retain their spatial stability due to π - π interactions and hydrogen bonding between the urethane groups. The soft segments, which possess with ether type of bond, can support ionic liquids, which is favourable for promoting the dissociation of alkali metal salts, thereby enhancing the ionic conductivity. The soft segments are physically cross-linked, ensuring good film-forming and good mechanical properties of the films. TPU is widely used in various industrial fields owing to its superior physical performance, chemical resistance, abrasion resistance, good adhesion, and ease of processing [30-34]. Therefore, TPU was selected as a matrix component in this report to ensure the stable electrochemical performance and good mechanical properties of the electrolyte.

Polystyrene (PS) is a traditional thermoplastic polymer that has the features of ease of processing, colourlessness, transparency, and high strength. Therefore, it is widely used in industrial production [35–37]. PS has excellent mechanical properties, such as high strength, fatigue resistance, and spatial stability. In addition, PS has a high glass transition temperature and a high breakdown field.

Therefore, PAN, TPU, and PS were used as matrix materials in this research to prepare PAN/TPU/PS, PAN/TPU, PAN/PS, and TPU/PS blend/composite films by an electrospinning technique. The physical and electrochemical properties of these films were investigated.

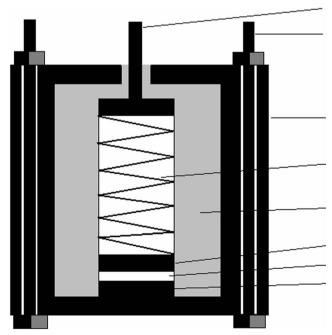


Fig. 1 The installation diagram of the assembled cell



Materials and methods

Materials

Polyacrylonitrile (PAN, average $M_{\rm w}=150,000$) was supplied by Sigma-Aldrich. Thermoplastic polyurethane (TPU) was provided by Wanhua Company, and polystyrene (PS) was purchased from BASF. All raw materials were dried at $60~^{\circ}$ C under vacuum for 12 h before use.

Preparation of the PAN/TPU/PS nanofibrous membrane

A certain amount of PAN, TPU, and PS (mass ratio of 5:5:1) were homogeneously dissolved in DMF to prepare a solution with a mass fraction of 10%. The solution was stirred for 12 h at 60 °C. PAN/TPU, PAN/PS, and TPU/PS solutions with 10% mass fractions were prepared by the same method. The mixture was transferred to a syringe and electrospun at a flow rate of 0.5 mL h⁻¹. The nozzle tip voltage was maintained at 24 kV, and the collector was at room temperature. The blend/composite films were dried at 60 °C under vacuum for 12 h.

Preparation of the gel polymer electrolytes

The dried blend/composite films were impregnated in a 1 mol L^{-1} LiPF₆–EC/DMC electrolyte to convert them into GPEs in a glove box with a high-purity argon atmosphere.

The physical properties of the membranes

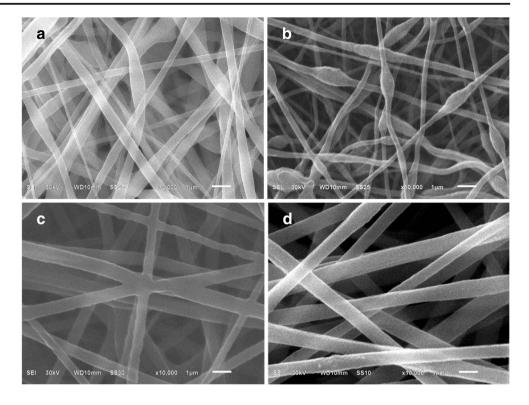
The morphologies of the membranes were studied by scanning electron microscopy (SEM) (Hitachi S-3500N, Japan). The material structures and chemical compositions were analysed by Fourier transform infrared spectroscopy (FTIR) (Spectrum One, PerkinElmer Instruments). The thermal properties of the fibrous films were tested by thermogravimetric analysis (model TGA Q50, TA Company, USA). Thermogravimetric analysis (TGA) measurements were carried out from 30 to 800 °C at a rate of 20 °C/min under a dry nitrogen atmosphere. The porosities of the films were calculated by formula (1):

$$P = \frac{W_w - W_d}{\rho_b V_p} \times 100\% \tag{1}$$

where W_w and W_d are the weights of the wet and dry separators, respectively, ρ_b is the density of n-butanol, and V_p is the volume of the dry film.

The absorption percentage of the electrolyte can be calculated by formula (2).

Fig. 2 SEM images of the electrospun films: a PAN/PS, b PAN/TPU, c TPU/PS, and d PAN/TPU/PS



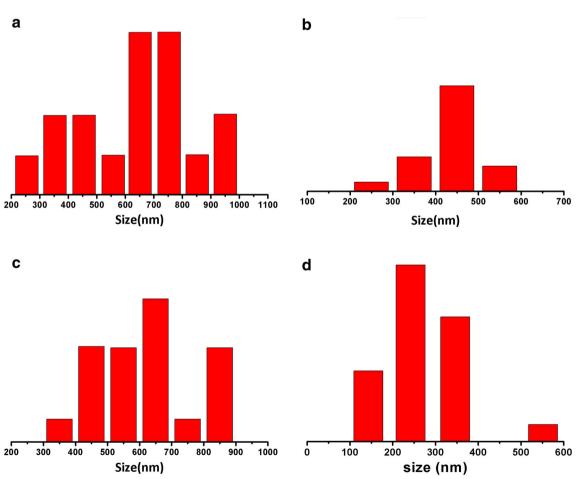


Fig. 3 Size of the electrospun films: a PAN/PS, b PAN/TPU, c TPU/PS, and d PAN/TPU/PS

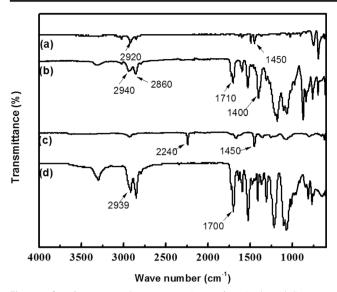


Fig. 4 Infrared spectra a PS, b TPU, c PAN, and d PAN/TPU/PS

Uptake (%) =
$$\frac{W - W_0}{W_0} \times 100\%$$
 (2)

In this equation, W_0 is the weight of the dry electrospun fibre membrane, and W is the weight of the wet film after the dry membrane is impregnated into the 1 mol/L LiPF₆-EC/DMC electrolyte.

The mechanical strengths of the films were tested at an elongation rate of 5 mm/min using an electronic tensile machine (Riegel, RGM-6005) after cutting the electrospun films into a width of 2 cm and a length of 5 cm and a thickness of $100-150~\mu m$. The elongation at break is calculated by the following formula:

$$Strain (\%) = \frac{L}{L_0} \times 100\% \tag{3}$$

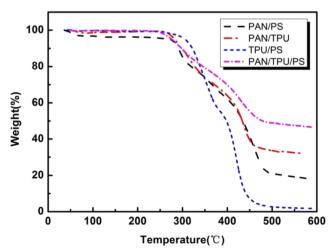


Fig. 5 TGA curves of the electrospun fibre films



where L_0 is the initial distance between the measuring moulds (mm).

The tensile strength is calculated by the following formula:

$$Stress = \frac{F}{S} \tag{4}$$

where S is the cross-sectional area (mm²).

Thermal shrinkage primarily refers to the thermoplastic material due to its inherent thermal expansion rate of volume change, which is the main reason for the contraction. In the test, the blend/composite films prepared in this experiment were pressed into a disc with a diameter of 19 mm, heated at 180 °C for 1 h, and the changes in their diameter and curl were measured.

Performance of the assembled cells

Figure 1 shows the installation diagram of a cell. The electrospun membranes were assembled into batteries according to this cell structure for the following experiments. Electrochemical impedance spectroscopy (EIS) was carried out by an electrochemical analyser (CHI660A, Shanghai Chenhua) with a frequency range from 10^{-2} to 10^{5} Hz. The ionic conductivity was then determined according to the following Eq. (5):

$$\sigma = \frac{h}{R_b S} \tag{5}$$

In this equation, σ is the ionic conductivity, R_b represents the bulk resistance of the membrane, h is the thickness, and S is the area. The charge and discharge tests of the button batteries were conducted by a Neware (model BTS-51, Shenzhen, China) battery tester with a test voltage of 2.0–4.2 V, and a cycle number of 50. Electrochemical stability was studied by linear sweep voltammetry (LSV). A Zahner Zhex electrochemical analyser was set to a scanning rate of 5 mVs⁻¹ and a voltage range of 0–7 V to study the coin cells.

Results and discussion

Morphology and structure

Figure 2 shows SEM images of the PAN/PS, PAN/TPU, TPU/PS, and PAN/TPU/PS electrospun films. The binary electrospun films (PAN/PS, PAN/TPU, and TPU/PS) have a variety of problems. The fibre diameter size of PAN/PS (Fig. 3a) has an uneven distribution, with an average fibre diameter of approximately 691 nm. The electrospun film of PAN/TPU (Figs. 2b and 3b) has beads, with diameter values ranging between 180 and

Table 1 Liquid absorptivity, ionic conductivity, and porosity of the different films

| rea (cm ²) | | Uptake behaviour (%) | Resistance (Ω) | Ionic conductivity (mS cm ⁻¹) | Porosity (%) |
|------------------------|----------------|-----------------------------------|---|--|--|
| .01 | 0.0155 | 250 | 3.69 | 2.08×10^{-3} | 57 |
| .01 | 0.014 | 148 | 2.77 | 2.51×10^{-3} | 62 |
| .00 | 0.015 | 201 | 2.43 | 3.08×10^{-3} | 67 |
| .01 | 0.015 | 315 | 1.76 | 4.24×10^{-3} | 82 |
| . 1 | 01 01 00 | 01 0.0155 01 0.014 00 0.015 | 01 0.0155 250 01 0.014 148 00 0.015 201 | 01 0.0155 250 3.69 01 0.014 148 2.77 00 0.015 201 2.43 | 01 0.0155 250 3.69 2.08×10^{-3} 01 0.014 148 2.77 2.51×10^{-3} 00 0.015 201 2.43 3.08×10^{-3} |

570 nm. Moreover, Fig. 2c shows that there is cross-linking in the electrospun film of TPU/PS, with the diameter values ranging from 380 to 870 nm (Fig. 3c). The average fibre diameter of TPU/PS is approximately 615 nm. Figure 2d shows that the fibres of the PAN/TPU/PS electrospun film are slim and uniform in diameter. The minimum diameter of the fibres is 282 nm, and the maximum diameter is 510 nm (Fig. 3d).

Under the same conditions (voltage, receiving distance, flow rate, and temperature), the viscosity and conductivity of the electrospinning solution affect the morphology of the electrospun film. Increasing the amounts of PAN and TPU causes an increase in the ionic conductivity of the mixture for electrospinning because of the high ionic conductivities of PAN and TPU. The increased conductivity of the solution allows the polymer droplets to be fully stretched under the influence of a high electric field force, forming more uniform fibres. As shown in Fig. 3b, the minimum diameter of the fibre is 180 nm. When the solution viscosity is low, beads appear in the fibres, such as those in Fig. 2b. This is because the molecular chains are not effectively stretched during spinning. Therefore, the fibres of PAN/TPU (Fig. 2b) have beads. With the addition of PS, the viscosity of the spinning solution increases, and the molecular chains are fully extended in the electrospinning process, such that the fibres have good

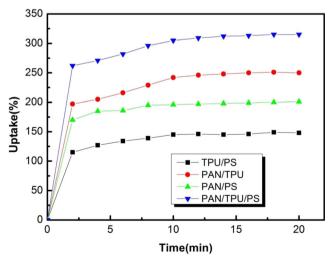


Fig. 6 Uptake behaviours of the four electrospun membranes

morphologies [38]. In addition, the ternary polymer solution greatly reduces the surface tension [39]. The mixed solution is thus completely stretched during the spinning process to form a smooth fibre. Therefore, PAN/PS, PAN/TPU, and TPU/PS electrospun films have certain visual defects, while the PAN/TPU/PS film has a very good appearance. We boldly speculate that the ternary blended PAN/TPU/PS electrospun membranes will have excellent electrochemical properties.

Infrared spectrum analysis

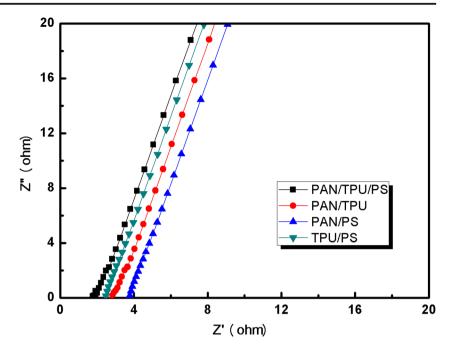
The FTIR spectra of PS, PAN/TPU/PS, PAN, and TPU are shown in Fig. 4. The characteristic absorption peaks of PS are observed at 2920 cm⁻¹ (stretching band of -CH₂- in the benzene ring) and 1450 cm⁻¹ (stretching band of C-C in the benzene ring). This is because the polystyrene polymerisation reduces the carbon-carbon double bonds into single bonds during intermolecular polymerisation. The characteristic absorption peaks of TPU can be identified at 2939 cm⁻¹ (stretching band of -NH₂ in the hard phase) and 1700 cm⁻¹ (stretching band of C=O). The typical peaks of PAN are at 2240 cm⁻¹ (stretching band of C=N) and 1450 cm⁻¹ (deformation vibration band of -CH₂-). However, in the case of the electrospun membrane of PAN/TPU/PS, the characteristic absorption peaks of TPU and PAN are shifted to 2940, 2860, 1710, and 1400 cm⁻¹. This originates from the repulsive dipolar interactions between the C=O groups of TPU and the C≡N groups of PAN. Furthermore, it is possible that the presence of PAN in the composites modified the hard-hard segment interactions of TPU, which consequently resulted in a new morphology of TPU with altered hydrogen-bonding characteristics between the hard-hard segments.

Thermal analysis

Thermograms of the different electrospun films are shown in Fig. 5. The PAN/TPU/PS film and the PAN/TPU film began to decompose at 260 °C. The PAN/PS began to decompose at 280 °C. The initial decomposition temperature of TPU/PS was 300 °C, but the quality of the fibre film rapidly declined between 300 and 450 °C, and the quality retention of TPU/PS



Fig. 7 Impedance spectra of the gel polymer electrolytes



was the lowest of the films. The retention percentages of the four blend/composite films (TPU/PS, PAN/PS, PAN/TPU, and PAN/TPU/PS) were 1.8%, 18%, 32%, and 47% at 600 °C, respectively. The TPU/PS film was nearly completely decomposed.

These results show that the PAN/TPU/PS membrane has better thermal stability than that of the other samples. In addition, when the temperature is lower than 260 °C, the four electrospun films hardly decompose thermally. Compared with the other films, a slight enhancement in the thermal stability of PAN/TPU/PS in the temperature range of 300 to 450 °C is observed, which is due to the introduction of PAN. The interaction between PAN and TPU promotes thermal stability. It is clear that the PAN/TPU/PS film is suitable for use in lithium-ion batteries as a separator.

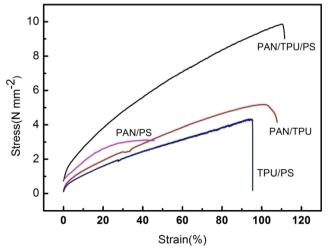


Fig. 8 Stress-strain curves of the four electrospun films



Porosity, electrolyte uptake, and ionic conductivity

Table 1 shows the liquid absorptivity, ionic conductivity, and porosity of the different films. Porosity is calculated from Eq. (1). These results verify the structure of the four electrospun films as shown in the SEM images. The porosities of the PAN/PS, PAN/TPU, TPU/PS, and PAN/TPU/PS fibre films are 57%, 67%, 62%, and 82%, respectively. It is obvious that the average fibre diameter of the ternary-blend PAN/TPU/PS film is the smallest, and its specific surface area and porosity are the largest.

The percentage of electrolyte uptake can be calculated according to Eq. (2). As shown in Fig. 6, the liquid absorption rate of the ternary-blend PAN/TPU/PS electrospun film is the largest among the four samples for the fluid uptake tests. The reason for this is that the material has a higher porosity than the other films. This structure is conducive to the rapid adsorption of the liquid electrolyte. The liquid absorbencies of the TPU/PS, PAN/PS, PAN/TPU, and PAN/TPU/PS fibre membranes are 140%, 190%, 245%, and 310%, respectively. The fastest liquid absorbency rates of all the electrospun films occurred within the first 2 min.

Figure 7 shows the electrochemical impedance spectroscopy (EIS) of the gel polymer electrolytes with different matrices. It can be observed from Fig. 8 that the bulk resistance (R_b) of the PAN/PS polymer electrolyte is 3.69 Ω . Correspondingly, the bulk resistance of the PAN/TPU and TPU/PS electrolytes is 2.43 Ω and 2.77 Ω , respectively. However, the bulk resistance (R_b) of the PAN/TPU/PS GPE is only 1.76 Ω . The ionic conductivities can be calculated with Eq. (5). The gel polymer electrolyte films have ionic conductivities of 2.08 ×

Table 2 Mechanical properties of all the electrospun fibrous membranes

| Sample | Surface area (cm ²) | Thickness (cm) | Tensile strength (MPa) | Elongation at break (%) |
|------------|---------------------------------|----------------|------------------------|-------------------------|
| PAN/PS | 1.99 | 0.016 | 3.10 ± 0.2 | 43.05 ± 0.2 |
| TPU/PS | 2.00 | 0.018 | 4.31 ± 0.2 | 95.26 ± 0.2 |
| PAN/TPU | 2.00 | 0.016 | 5.18 ± 0.2 | 100.58 ± 0.2 |
| PAN/TPU/PS | 1.99 | 0.018 | 9.86 ± 0.2 | 115.15 ± 0.2 |

 $10^{-3}~S~cm^{-1}$ (PAN/PS), $3.08 \times 10^{-3}~S~cm^{-1}$ (PAN/TPU), $2.51 \times 10^{-3}~S~cm^{-1}$ (TPU/PS), and $4.24 \times 10^{-3} S~cm^{-1}$ (PAN/TPU/PS).

The ionic conductivity is dependent on the porosity of the fibrous membrane. The ionic conductivity increases as the Li⁺ amount increases, which results from a higher electrolyte uptake and porosity of the film. Furthermore, it was found that the average diameter of the PAN/TPU/PS nanofibres was the smallest and most uniform of the different fibres based on the SEM measurements. Therefore, the electrolyte uptake and porosity of the PAN/TPU/PS GPE are great. This may be the reason for the resistance of the PAN/TPU/PS being the lowest of the samples. In summary, the PAN/TPU/PS GPE has the highest ionic conductivity of the different samples and should be suitable for application in lithium-ion batteries.

Mechanical performance

The mechanical properties of the electrospun films are shown in Fig. 8. The test results show that the mechanical properties of the membranes are improved by the inclusion of TPU and PS fillers. The PAN/PS membrane extended only 43.05% with a poor breaking tensile strength of 3.10 MPa. The PAN/TPU membrane without PS extended 100.58% with a poor breaking tensile strength of 5.18 MPa. However, the membrane

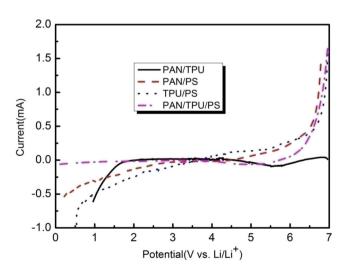
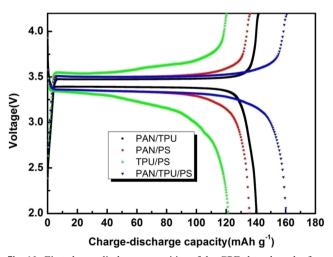


Fig. 9 Linear sweep voltammograms of the gel polymer electrolytes based on the electrospun membranes

composed of PAN/TPU/PS has 86.4% elongation and broke under a tensile strength of 10.8 MPa. All these data are included in Table 2. The increase in fracture strength and elongation at break of the PAN/TPU/PS film can be attributed to the addition of TPU and PS. TPU is an elastomer with a high tensile strength and elasticity, and PS has excellent mechanical properties, including high strength, fatigue resistance, and spatial stability. TPU and PS are included to maintaining the spatial stability of the PAN/TPU/PS film. Thus, the PAN/TPU/PS ternary-blend membrane shows excellent mechanical properties.

Electrochemical stability

Figure 9 shows the LSV curves of the gel polymer electrolytes. It can be seen that the electrochemical stability of the gel polymer electrolyte with the PAN/TPU/PS membrane is 5.8 V. The electrochemical stability plateaus of the PAN/TPU and PAN/PS fibrous polymer electrolytes occur at 4.2 V and 3.7 V, while the curve of the TPU/PS GPE continuously changes. As we can see from Fig. 9, the electrochemical stabilities are ranked as follows: the polymer film of PAN/TPU/PS(5.8 V) > PAN/TPU(4.2 V) > PAN/PS(3.7 V). It is clear that the gel polymer electrolyte comprising PAN/TPU/PS shows the best electrochemical stability, which is much higher than the requirements for practical application.



 $\begin{tabular}{ll} Figs. 10 First charge-discharge capacities of the GPEs based on the four different electrospun membranes \\ \end{tabular}$



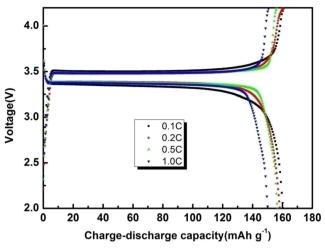
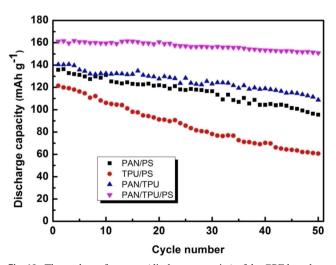


Fig. 11 First charge-discharge capacities of GPEs based on the electrospun PAN/TPU/PS membrane at different rates (1C = 170 mAh g^{-1})

There is almost no electrochemical reaction in the potential range from 2 to 5.8 V for the PAN/TPU/PS gel polymer electrolyte. When the electrode potential is greater than 5.8 V, the response current significantly increases, indicating that the polymer electrolyte experiences an electrochemical reaction and the onset of decomposition. The large surface area of the PAN/TPU/PS membrane significantly contributes to the increased stability of the electrolyte solution under electrochemical environments. In addition, the better anodic stability of the PAN/TPU/PS-based GPE is in part due to its excellent affinity for the carbonates (EC and DMC) of the liquid electrolyte. The interactions between the C=O groups of the carbonate molecules and the -NH groups of TPU or the -CN groups of PAN contribute significantly to the enhancement of the electrochemical stability. The above results demonstrate that the PAN/TPU/PS-based GPE has excellent



 $\begin{tabular}{ll} Fig.~12 & The~cycle~performance~(discharge~capacity)~of~the~GPE~based~on~the~electrospun~PAN/TPU/PS~membrane \end{tabular}$



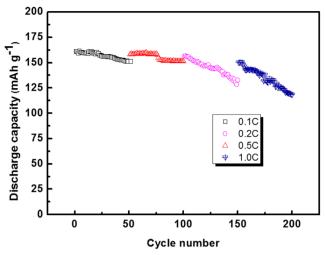


Fig. 13 Cycle performance of the PAN/TPU/PS GPE at different rates

electrochemical stability and is suitable for application in lithium-ion batteries.

Evaluation in Li/LiFePO₄ cells

As an active material, the theoretical specific capacity of LiFePO₄ is 170 mAh g⁻¹, which is typical of LiFePO₄ [40]. Figure 10 shows the first charge-discharge capacity curves of Li/GPE/LiFePO₄ using the different GPEs. The charge and discharge capacities of the PAN/TPU/PS GPE are 161.70 mAh g⁻¹ and 161.44 mAh g⁻¹, respectively, while the other three gel polymer electrolyte films deliver charge capacities of 141.79 mAh g⁻¹ (PAN/TPU GPE), 135.98 mAh g⁻¹ (PAN/PS GPE), and 121.37 mAh g⁻¹ (TPU/PS GPE) and discharge capacities of 140.53 mAh g⁻¹ (PAN/TPU GPE), 135.76 mAh g⁻¹ (PAN/PS GPE), and 121.33 mAh g⁻¹ (TPU/PS GPE). The gel polymer electrolyte film comprising PAN/TPU/PS has the best charge-discharge capacity, and a first charge-discharge efficiency of 95% of the theoretical capacity. This may be due to the large number of interpenetrating pores in the polymer electrolyte membrane, providing the necessary channels for ionic transport and facilitating rapid ionic transport. On the other hand, it may be a result of the compatibilities of PAN, TPU, and PS in the polymer matrix.

Additionally, as is shown in Fig. 11, we study the different rate capabilities of the PAN/TPU/PS GPE. At relatively high rates, the polymer film has a discharge capacity of 161.70 mAh g⁻¹ (0.1 C), 158.16 mAh g⁻¹ (0.2 C), 156.58 mAh g⁻¹ (0.5 C), and 150.07 mAh g⁻¹ (1.0 C). As the rate increases, the current intensity increases correspondingly, resulting in high polarisation of the electrode material. Therefore, the capacity of the battery experiences different degrees of attenuation.

We further evaluate the cycle stabilities of the four different GPEs under a rate of 0.1 C at 25 °C. It is obvious that at lower

current rates (0.1C), the button-cell battery based on the PAN/TPU/PS GPE retains 93.4% of its capacity after 50 cycles.

Figure 12 shows the cycle performances of polymer films at the 0.1 C rate. After 50 cycles, the capacity retention percentages of the PAN/PS, PAN/TPU, and TPU/PS button cells are only 80.1%, 68.0%, and 50.0%, respectively. The button-cell battery based on the PAN/TPU/PS GPE has the largest capacity retention of the different batteries at 93.4%. The button cell containing the PAN/TPU/PS GPE has the highest discharge capacity of the different batteries with a small capacitance decay, which clearly shows its superior cycle performance.

Figure 13 shows the cycle performance of the button cell based on the PAN/TPU/PS GPE at different rates. When the rates were 0.1 C, 0.2 C, 0.5 C, and 1.0 C, the first discharge capacities of the PAN/TPU/PS GPE were 161.44 mAh g⁻¹, 156.58 mAh g⁻¹, 158.21 mAh g⁻¹, and 150.17 mAh g⁻¹, respectively. After 50 cycles, the discharge capacities were 150.85 mAh g⁻¹, 132.83 mAh g⁻¹, 146.46 mAh g⁻¹, and 118.24 mAh g⁻¹, respectively. The capacity retention is thus 93.4%, 92.6%, 84.8%, and 78.7% for the respective rates. From the experimental results, it can be seen that the discharge capacity of the button cell based on the PAN/TPU/PS GPE decreases with increasing rate, especially at 1.0 C. This is mainly because a large current density increases the degree of polarisation of the battery.

Conclusion

Membranes based on PAN/TPU, TPU/PS, PAN/PS, and PAN/ TPU/PS are prepared by electrospinning. The PAN/TPU/PS electrospun membrane has the highest ionic conductivity of the different membranes at 3.83×10^{-3} S cm⁻¹, and an electrochemical stability of 5.8 V. The SEM images and porosity measurements showed that the electrospun membranes have a uniform and highly interconnected porous structure. Simultaneously, the blend/composite membranes have excellent mechanical properties (9.86 MPa) and elongation (115%). In addition, in battery tests, the Li/GPE/LiFePO₄ cell shows first charge and discharge capacities of 161.70 mAh g⁻¹ and $161.44 \text{ mAh g}^{-1}$ at the 0.1 C rate, which represents 95% of the theoretical value. After 50 cycles, the cycle performance of the button battery based on the PAN/TPU/PS GPE has the largest capacity retention of the different GPEs at 93.4% and a stable cycle performance. All of the results demonstrate that the PAN/TPU/PS-based GPE is an ideal choice for use in LIBs.

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