



Elaborated subloading surface model for accurate description of cyclic mobility in granular materials

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Abstract

The description of the cyclic mobility observed prior to the liquefaction in geomaterials requires the sophisticated constitutive formulation to describe the plastic deformation induced during the cyclic loading with the small stress amplitude inside the yield surface. This requirement is realized in the *subloading surface model*, in which the surface enclosing a purely elastic domain is not assumed, while a purely elastic domain is assumed in other elastoplasticity models. The subloading surface model has been applied widely to the monotonic/cyclic loading behaviors of metals, soils, rocks, concrete, etc., and the sufficient predictions have been attained to some extent. The subloading surface model will be elaborated so as to predict also the cyclic mobility accurately in this article. First, the rigorous translation rule of the similarity center of the normal yield and the subloading surfaces, i.e., elastic core, is formulated. Further, the mixed hardening rule in terms of volumetric and deviatoric plastic strain rates and the rotational hardening rule are formulated to describe the induced anisotropy of granular materials. In addition, the material functions for the elastic modulus, the yield function and the isotropic hardening/softening will be modified for the accurate description of the cyclic mobility. Then, the validity of the present formulation will be verified through comparisons with various test data of cyclic mobility.

Keywords Cyclic loading · Cyclic mobility · Constitutive equation · Elastoplastic deformation · Granular materials · Liquefaction

Abbreviations

a, b	Material constants for deviatoric hardening
b_r	Material constant for rotational hardening
\mathbf{c}	Elastic core (similarity center of normal-yield and subloading surfaces)
c_e	Material constant for evolution of elastic core

$C_n (\equiv \hat{\mathbf{n}}_c : \bar{\mathbf{n}})$	Scalar product of outward-normals of elastic-core and subloading surfaces
d_v^e	Elastic volumetric strain rate
d_v^p	Plastic volumetric strain rate
d_s^p	Plastic shear strain rate
\mathbf{d}	Strain rate tensor
\mathbf{d}^e	Elastic strain rate tensor
\mathbf{d}^p	Plastic strain rate tensor
\mathbf{E}	Elastic modulus tensor
E	Young's modulus
f	Yield stress function
F	Isotropic hardening function
G	Elastic shear modulus
H	Isotropic hardening variable
\mathbf{H}	Second-order tensor-valued internal variable
K	Elastic bulk modulus
\mathbf{l}	Velocity gradient tensor
M	Stress ratio in critical state
M_c	Stress ratio in critical state in triaxial compression state

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M_d	Stress ratio at boundary of deviatoric softening and hardening	$\vartheta (> \xi)$	Ratio of negative pressure to size of yield surface at which volume becomes infinite
M_r	Stress ratio for the rotational hardening limit surface	$\chi (< 1)$	Maximum value of \mathfrak{R}_c
\mathbb{M}^{ep}	Elastoplastic tangent modulus tensor		
\overline{M}^p	Plastic modulus		
n	Power number for pressure dependence in elastic shear modulus		
$\bar{\mathbf{n}}$	Normalized outward-normal tensor to subloading surface		
$\hat{\mathbf{n}}_c$	Normalized outward-normal tensor to elastic-core surface		
$\mathbf{0}$	Second-order zero tensor		
p	Pressure		
$R (\leq 1)$	Normal-yield ratio		
$\mathfrak{R}_c (\leq \chi)$	Elastic-core yield ratio		
u, \bar{u}, u_c	Material functions and constants regulating normal-yield ratio		
U	Function for evolution of normal-yield ratio		
\mathbf{w}	Continuum spin tensor		
$\bar{\alpha}$	Conjugate point in subloading surface to α in normal-yield surface		
$\alpha (= \mathbf{0})$	Reference point in normal-yield surface, which is fixed to the origin of the stress space		
β	Rotational hardening tensor		
ε	Accumulated strain		
ε^p	Accumulated plastic strain		
ϕ_c	Internal friction angle in triaxial compression state		
ϕ_d	Border angle of deviatoric hardening and softening		
ϕ_r	Limitation angle of rotational hardening		
$\xi (< 0.5)$	Shifting ratio of yield surface to negative pressure		
$\tilde{\kappa}$	Inclination of swelling line in both logarithmic plane of volume and pressure		
$\tilde{\lambda}$	Inclination of normal-consolidation line in both logarithmic plane of volume and pressure		
$\dot{\lambda}$	Plastic multiplier in terms of stress rate		
$\dot{\Lambda}$	Plastic multiplier in terms of strain rate		
ν	Poisson's ratio		
σ	Cauchy stress tensor		
σ_χ	Conjugate stress on limit elastic-core surface		
σ_y	Conjugate stress on normal-yield surface		
θ_σ	Lode's angle		

1 Introduction

It should be noticed that soils exhibit complex deformation behaviors, which are not observed in metals, e.g., the pressure dependence of the elastic moduli and the plastic deformation characteristics, the plastic compressibility, the plastic volumetric expansion induced by the deviatoric stress, i.e., the dilatancy and the rotational anisotropic hardening instead of the kinematic hardening. Moreover, granular materials such as sands exhibit not only volumetric but also deviatoric isotropic hardening, whereas metals and clays exhibit only the deviatoric isotropic hardening and volumetric isotropic hardening, respectively, usually. Above all, one of the most peculiar phenomena among various elastoplastic deformation behaviors in solids would be the *cyclic mobility* observed prior to the *liquefaction* induced in sandy ground during earthquakes. Although the permeability of sands is high in general, the shaking during earthquakes occurs at high frequencies so that the deformation occurs under approximately undrained condition. The butterfly-shaped stress loops in the mean effective and deviatoric stress plane, and the S-shaped deviatoric stress vs. axial strain loops with increasing strain amplitude are engendered in sands subject to cyclic loading of deviatoric stress under undrained condition. The cyclic mobility received the focus of attention after the Chile earthquake in May, 1960, the Alaska earthquake in March, 1964, and the Niigata earthquake in June, 1964, while the cyclic mobility is termed as *cyclic strain softening* in the definition by ASCE [6].

The description of cyclic mobility requires one of the most sophisticated formulations in the constitutive modeling. Numerous research reports have been published on the cyclic mobility, e.g., Zienkiewicz et al. [81], Mroz et al. [51], Prevost and Keane [59], Desai et al. [8], Elgamaal et al. [10], Fang et al. [11], Hashiguchi and Chen [34], Oka et al. [60], Zhang et al. [72], Hashiguchi [30], etc., within the framework of the elastoplasticity and Zienkiewicz et al. [82], Iai et al. [44], Akiyoshi et al. [1], Gerolymos and Gazetas [17], Zhang and Wang [73] by the empirical approaches. These extensive research efforts

have made significant contribution toward elucidating the fundamental mechanism and establishing accurate prediction methods for cyclic mobility.

The description of the cyclic mobility requires the sophisticated constitutive formulation to describe the plastic deformation during the cyclic loading with a small stress amplitude inside the yield surface. The formulation to fulfill this requirement would be able to be attained in the *subloading surface model* [19, 20, 28, 30], in which the surface enclosing a purely elastic domain is not assumed, while a purely elastic domain is assumed in the other elastoplasticity models. The *subloading surface*, which passes through the current stress and has a similar shape and direction to the yield surface (renamed the *normal-yield surface*), is assumed inside the normal-yield surface, and then it is postulated that the plastic strain rate develops as the stress approaches the yield surface, i.e., as the subloading surface expands. Therefore, the smooth transition from the elastic to the plastic state, i.e., the *smooth elastic–plastic transition* leading to the continuous variation of the tangent stiffness modulus tensor, is always described in this model. The subloading surface model has been applied to the descriptions of the elastoplastic deformation behaviors of various solids, e.g., metals [2–4, 12, 39, 41–43, 45–47, 52], soils [15, 16, 34, 35, 38, 53–56, 58, 66, 70–77, 79], rocks and sediments [13, 14, 78, 80] and friction phenomena [36, 37, 40]. Here, it should be noted that only the elastic deformation is induced in the unloading process and thus closed hysteresis loop cannot be described in the unloading–reloading process if the similarity center of the normal-yield and the subloading surfaces is fixed to the origin of the stress space. The similarity center is physically regarded as the *elastic-core* because the purely elastic deformation is induced when the stress coincides with the similarity center inside the yield surface. Therefore, the elastic core must be translated with the plastic deformation in order to describe the cyclic loading behavior. The subloading surface model in which the elastic-core is fixed is called the *initial subloading surface model* and further the model in which the elastic-core is translated is called the *extended subloading surface model*. The initial subloading surface model has been applied to the description of the cyclic mobility (e.g. [72]). However, the cyclic mobility can be described by the extended subloading surface model by incorporating the rigorous evolution rule of the elastic-core.

The extended subloading surface model will be elaborated so as to describe the cyclic mobility accurately in this article. First, the rigorous translation rule of the elastic-core is incorporated and the pertinent rotational hardening rule is formulated to describe the induced anisotropy of granular materials. In addition, the material functions for the elastic moduli, the yield function, the isotropic

hardening/softening, etc., will be extended for the accurate description of the cyclic mobility. Then, the validity of the present formulation will be verified through comparisons with various test data of cyclic mobility for three kinds of sands under various stress amplitudes and several loading cycles up to eighty cycles.

Throughout this paper, the signs of stress and strain are chosen positive for tension and extension, and all stresses signify the so-called *effective stress* excluding pore pressure. In addition, the direct notation $\mathbf{A} : \mathbf{B}$ for $A_{rs}B_{rs}$, \mathbf{AB} for $A_{ir}B_{rj}$, $\mathbf{\Gamma} : \mathbf{A}$ for $\Gamma_{ijrs}A_{rs}$ and $\mathbf{A} : \mathbf{\Gamma}$ for $A_{rs}\Gamma_{rsij}$ are used for arbitrary second-order tensors \mathbf{A} and \mathbf{B} , fourth-order tensors $\mathbf{\Gamma}$, where Einstein's summation convention is applied for tensor components with repeated indices taking 1, 2, 3. Further, $\mathbf{0}$ stands for the second-order zero tensor, \mathbf{I} for the second-order identity tensor possessing the Kronecker delta components δ_{ij} ($\delta_{ij} = 1$ for $i = j$; $\delta_{ij} = 0$ for $i \neq j$), $\text{tr} \mathbf{A} = A_{ij}\delta_{ij}$ for the trace, $\mathbf{A}' = \mathbf{A} - (\text{tr} \mathbf{A})\mathbf{I}/3$ for the deviatoric part, \mathbf{A}^{-1} for the inverse tensor satisfying $\mathbf{AA}^{-1} = \mathbf{I}$ and $\|\mathbf{A}\| = \sqrt{A_{ij}A_{ij}}$ for the magnitude, and symbol \mathbf{A}^T for the transposed tensor. The symmetric and the antisymmetric parts of \mathbf{A} are denoted by $\text{sym}[\mathbf{A}] \equiv (\mathbf{A} + \mathbf{A}^T)/2$ and $\text{ant}[\mathbf{A}] \equiv (\mathbf{A} - \mathbf{A}^T)/2$, respectively. (\cdot) and (\circ) stand for the material-time derivative and the proper objective time-derivative, respectively. The symbol $\langle \cdot \rangle$ signifies the Macaulay's bracket defined by $\langle s \rangle = (s + |s|)/2$, specifically $s < 0 : \langle s \rangle = 0$ and $s \geq 0 : \langle s \rangle = s$ for an arbitrary scalar variable s . $(\cdot)_0$ denotes initial values of variables.

2 Strain rate

Denoting the current position vector and the velocity vector of a material particle by \mathbf{x} and \mathbf{v} , respectively, the velocity gradient tensor is defined as $\mathbf{l} = \partial \mathbf{v} / \partial \mathbf{x}$. The strain rate tensor and the continuum spin tensor are defined as $\mathbf{d} \equiv \text{sym}[\mathbf{l}]$ and $\mathbf{w} \equiv \text{ant}[\mathbf{l}]$, respectively. Limiting the elastic deformation to be infinitesimal, the strain rate \mathbf{d} can be additively decomposed into an elastic strain rate \mathbf{d}^e and a plastic strain rate \mathbf{d}^p , i.e.

$$\mathbf{d} = \mathbf{d}^e + \mathbf{d}^p \quad (1)$$

as verified exactly based on the multiplicative decomposition of the deformation gradient tensor (Hashiguchi [32, 33]).

First, let \mathbf{d}^e be given by the hypoelastic relation (Truesdell [65]):

$$\mathbf{d}^e = \mathbf{E}^{-1} : \overset{\circ}{\boldsymbol{\sigma}} \quad (2)$$

where \mathbf{E} is the fourth-order tensor describing the elastic stiffness modulus and $\boldsymbol{\sigma}$ is the Cauchy stress, (\circ) denoting

the proper objective time-derivative [30, 42]. Let \mathbf{E} be given in the Hooke's form as

$$\begin{cases} E_{ijkl} = \left(K - \frac{2}{3}G\right)\delta_{ij}\delta_{kl} + G(\delta_{ik}\delta_{jl} + \delta_{il}\delta_{jk}) \\ E_{ijkl}^{-1} = -\frac{1}{2G}\delta_{ij}\delta_{kl} + \frac{3K - 2G}{36KG}(\delta_{ik}\delta_{jl} + \delta_{il}\delta_{jk}) \end{cases} \quad (3)$$

where K and G are the pressure-dependent elastic bulk modulus and shear modulus, respectively, leading to the nonlinear elasticity as will be formulated in Sect. 4.1.

3 Formulation of constitutive equation of granular materials

The elastoplastic constitutive equation of granular materials for describing cyclic loading behavior is formulated below by modifying and elaborating the past formulations based on the subloading surface model [19, 20, 26, 28, 30, 34]. The incorporation of the rigorous evolution rule of the elastic-core, in which the most elastic response appears since the subloading surface shrinks to a point in the stress space, is of crucial importance for the description of the cyclic loading behavior. In addition, the incorporation of the rigorous evolution rule of the rotational hardening is required for the description of the induced anisotropy. The basic constitutive equation of granular materials taking account of these phenomena will be formulated in this section.

3.1 Normal-yield and subloading surfaces

For granular materials, the conventional yield surface, renamed the *normal-yield surface*, is given by

$$f(\boldsymbol{\sigma}, \boldsymbol{\beta}) = F(H) \quad (4)$$

where F is the stress-valued *isotropic hardening function* of strain-valued isotropic hardening variable H . The rotational hardening variable of the yield surface incorporated first by Sekiguchi and Ohta [63] is denoted by $\boldsymbol{\beta}$, while the yield surface rotates around the origin of the stress space. Consequently, the normal-yield surface expands or contracts and rotates around the fixed reference point $\boldsymbol{\alpha}$ which is fixed to the origin of the stress space, i.e., $\boldsymbol{\alpha} = \mathbf{0}$ in geomaterials. f is assumed to be a function of $\boldsymbol{\sigma}$ in homogeneous degree-one fulfilling $f(|s|\boldsymbol{\sigma}, \boldsymbol{\beta}) = |s|f(\boldsymbol{\sigma}, \boldsymbol{\beta})$ for an arbitrary scalar variable s .

As described in Introduction, the mutual slips of solid particles in materials are not induced simultaneously but induced gradually exhibiting a smooth transition from the elastic to the plastic state, which leads to the continuous variation of the elastoplastic tangent stiffness modulus tensor. Then, it is postulated that the plastic strain rate

develops gradually as the stress approaches the yield surface. To describe the approaching degree of stress to the yield surface, the *subloading surface* which passes through the current stress and has a shape and a direction similar to the yield surface, renamed the *normal-yield surface*, is incorporated. Then, the approaching degree to the normal-yield surface is represented by a scalar variable which is defined by the ratio of the size of the subloading surface to that of the normal-yield surface, while the ratio is called the *normal-yield ratio* and designated by the symbol R ($0 \leq R \leq 1$). Further, to describe the cyclic loading behavior, the similarity center of the normal-yield and the subloading surfaces, denoted by the symbol \mathbf{c} , is assumed to translate with the plastic deformation. Here, it is called the *elastic-core* because the most elastic response is induced when the stress lies on it, i.e., $\boldsymbol{\sigma} = \mathbf{c}$ leading to $R = 0$.

Consequently, the subloading surface for the normal-yield surface in Eq. (4) is represented by (see Fig. 1)

$$f(\bar{\boldsymbol{\sigma}}, \boldsymbol{\beta}) = RF(H) \quad (5)$$

where the following relation holds by virtue of the similarity of the subloading surface to the normal-yield surface.

$$\bar{\boldsymbol{\sigma}} \equiv \boldsymbol{\sigma} - \bar{\boldsymbol{\alpha}} (= R\boldsymbol{\sigma}_y) = \hat{\boldsymbol{\sigma}} + R\mathbf{c} \quad (6)$$

where

$$\hat{\boldsymbol{\sigma}} \equiv \boldsymbol{\sigma} - \mathbf{c} \quad (7)$$

It follows from Eq. (6) that

$$\bar{\boldsymbol{\alpha}} \equiv (1 - R)\mathbf{c} \quad (\bar{\boldsymbol{\alpha}} - \mathbf{c} = R(\boldsymbol{\alpha} - \mathbf{c}); \boldsymbol{\alpha} = \mathbf{0}) \quad (8)$$

leading to

$$\dot{\bar{\boldsymbol{\alpha}}} = (1 - R)\dot{\mathbf{c}} - \dot{R}\mathbf{c} \quad (9)$$

$\bar{\boldsymbol{\alpha}}$ is the conjugate (similar) point in the subloading surface to the reference point $\boldsymbol{\alpha}$ ($= \mathbf{0}$) in the normal-yield surface as shown in the (p, q) plane in Fig. 1, where $p \equiv -(\text{tr } \boldsymbol{\sigma})/3$ and $q \equiv \sigma_l - \sigma_a$, σ_l and σ_a representing the lateral and the axial stresses, respectively. $\boldsymbol{\sigma}_y$ is the conjugate point on the normal-yield surface to the current stress point $\boldsymbol{\sigma}$ on the subloading surface.

The material-time derivative of Eq. (5) reads:

$$\frac{\partial f(\bar{\boldsymbol{\sigma}}, \boldsymbol{\beta})}{\partial \bar{\boldsymbol{\sigma}}} : \dot{\bar{\boldsymbol{\sigma}}} - \frac{\partial f(\bar{\boldsymbol{\sigma}}, \boldsymbol{\beta})}{\partial \bar{\boldsymbol{\sigma}}} : \dot{\bar{\boldsymbol{\alpha}}} + \frac{\partial f(\bar{\boldsymbol{\sigma}}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} = \dot{R}F + RF \quad (10)$$

noting Eq. (6), which can be rewritten as

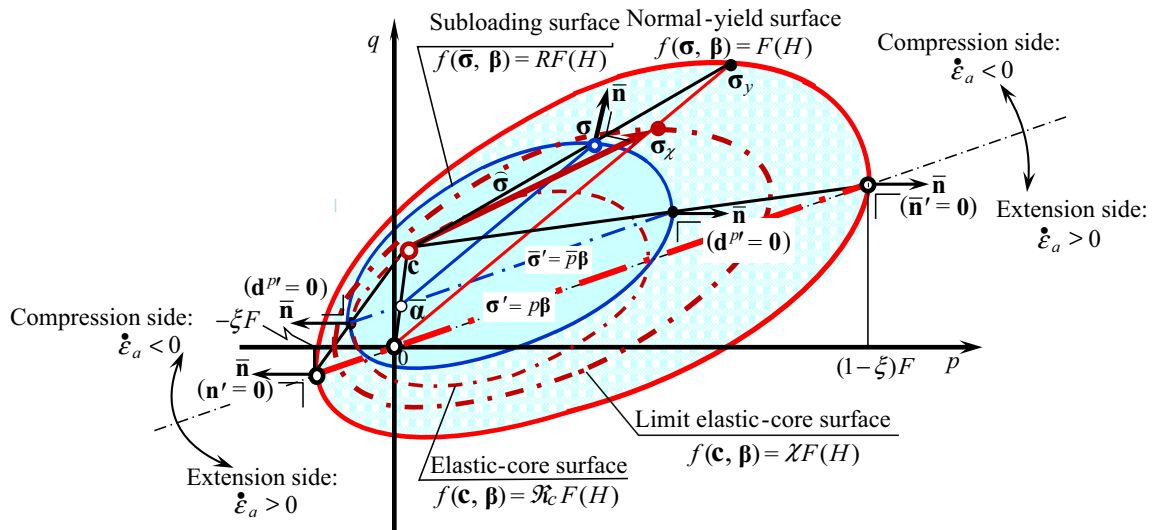


Fig. 1 Rotated normal-yield, subloading, elastic-core and limit elastic-core surfaces shown in the (p, q) plane

$$\bar{\mathbf{n}} : \overset{\circ}{\sigma} - \bar{\mathbf{n}} : \left[\overset{\circ}{\alpha} + \frac{\dot{F}}{F} \bar{\sigma} + \frac{\dot{R}}{R} \bar{\sigma} - \frac{1}{RF} \left(\frac{\partial f(\bar{\sigma}, \beta)}{\partial \beta} : \overset{\circ}{\beta} \right) \bar{\sigma} \right] = 0 \tag{11}$$

where

$$\bar{\mathbf{n}} \equiv \frac{\partial f(\bar{\sigma}, \beta)}{\partial \bar{\sigma}} / \left\| \frac{\partial f(\bar{\sigma}, \beta)}{\partial \bar{\sigma}} \right\| \quad (\|\bar{\mathbf{n}}\| = 1) \tag{12}$$

noting the following equation based on the Euler’s theorem for the function of $\bar{\sigma}$ in homogeneous degree-one.

$$\frac{1}{\frac{\partial f(\bar{\sigma}, \beta)}{\partial \bar{\sigma}}} = \frac{\frac{\partial f(\bar{\sigma}, \beta)}{\partial \bar{\sigma}} : \bar{\sigma}}{\bar{\mathbf{n}} : \bar{\sigma}} \bar{\mathbf{n}} = \frac{f(\bar{\sigma}, \beta)}{\bar{\mathbf{n}} : \bar{\sigma}} \bar{\mathbf{n}} = \frac{RF}{\bar{\mathbf{n}} : \bar{\sigma}} \bar{\mathbf{n}} \tag{13}$$

3.2 Evolution rule of normal-yield ratio

The rate of the normal-yield ratio R is given by the following equation based on the fundamental concept of the subloading surface described in Sect. 3.1.

$$\dot{R} = U(R) \|\mathbf{d}^p\| \text{ for } \mathbf{d}^p \neq \mathbf{0} \tag{14}$$

where U is a monotonically decreasing function of R , satisfying the following conditions (see Fig. 2).

$$U(R) \begin{cases} \rightarrow +\infty & \text{for } R = 0 : \text{ quasi-elastic state} \\ > 0 & \text{for } R < 1 : \text{ sub-yield state} \\ = 0 & \text{for } R = 1 : \text{ normal-yield state} \\ < 0 & \text{for } R > 1 : \text{ over normal-yield state} \end{cases} \tag{15}$$

Equation (14) with Eq. (15) is incorporated into a consistency condition of the subloading surface so that the stress is controlled to be attracted automatically to the normal-yield surface during the plastic loading process,

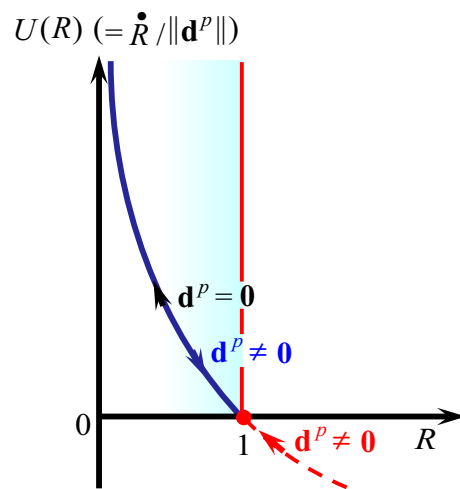


Fig. 2 Function $U(R)$ in the evolution rule of normal-yield ratio

even if it goes out from the normal-yield surface in numerical calculations, because of $\dot{R} < 0$ for $R > 1$ by Eq. (14) with Eq. (15) 4. Let function U satisfying Eq. (15) be given simply by

$$U(R) = u \cot\left(\frac{\pi}{2}R\right) \tag{16}$$

where u is a material function in general, and its specific form will be presented in Sects. 3.6 and 4.5. The smaller u is, the gentler is the transition from the elastic to the plastic state. Equation (14) with Eq. (16) can be analytically integrated as follows:

$$\begin{cases} R = \frac{2}{\pi} \cos^{-1} \left(\cos \left\{ \frac{2}{\pi} R_0 \exp \left[-\frac{2}{\pi} u (\varepsilon^p - \varepsilon_0^p) \right] \right\} \right) \\ \varepsilon^p - \varepsilon_0^p = \frac{2}{\pi} \frac{1}{u} \ln \frac{\cos \left(\frac{\pi}{2} R_0 \right)}{\cos \left(\frac{\pi}{2} R \right)} \end{cases} \quad (17)$$

for the initial condition $R = R_0$ for $\varepsilon^p = \varepsilon_0^p$, where $\varepsilon^p \equiv \int_0^t \|\mathbf{d}^p\| dt$ (time).

3.3 Evolution rule of elastic-core

Let the rigorous evolution rule of the elastic-core be formulated in this section based on the following facts.

- (1) In the physical view point, a smooth elastic–plastic transition is not described if the elastic-core lies on the normal-yield surface at which a remarkable plastic deformation is induced,
- (2) In the mathematical view point, the subloading surface is not determined uniquely if the elastic-core, i.e., the similarity center lies on the normal-yield surface, noting the fact: If the elastic-core lies on the normal-yield surface and the stress coincides with the elastic-core, R is indeterminate as known from the relation $\mathbf{0} = R\mathbf{0}$ which is induced by substituting $\boldsymbol{\sigma} = \mathbf{c} = \boldsymbol{\sigma}_y$ into $\boldsymbol{\sigma} - \mathbf{c} = R(\boldsymbol{\sigma}_y - \mathbf{c})$ based on the similarity of the subloading surface to the normal-yield surface.

Consequently, the elastic-core is not allowed to approach the normal-yield surface unlimitedly.

To avoid the unlimited approach of the elastic-core to the normal-yield surface, first let the following surface, called the *elastic-core surface*, be introduced as shown in Fig. 1, which passes through the elastic-core \mathbf{c} and possesses a similar shape and orientation to the normal-yield surface with respect to the null stress ($\boldsymbol{\sigma} = \boldsymbol{\alpha} = \mathbf{0}$).

$$f(\mathbf{c}, \boldsymbol{\beta}) = \mathfrak{R}_c F(H), \quad \mathfrak{R}_c = f(\mathbf{c}, \boldsymbol{\beta})/F(H) \quad (18)$$

where the variable \mathfrak{R}_c is the ratio of the size of the elastic-core surface to that of the normal-yield surface, called the *elastic-core yield ratio*. It plays the role of a measure for the approaching degree of the elastic-core to the normal-yield surface. Since the elastic-core must lie inside the normal-yield surface as described above, the elastic-core yield ratio has to be less than unity. Then, the inequality

$$f(\mathbf{c}, \boldsymbol{\beta}) \leq \chi F(H), \quad \text{i.e. } \mathfrak{R}_c \leq \chi \quad (19)$$

must hold, where $\chi (< 1)$ is a material constant exhibiting the maximum value of \mathfrak{R}_c . The time-differentiation of Eq. (19) at the limit state in which \mathbf{c} lies on the *limit elastic-core surface* $f(\mathbf{c}, \boldsymbol{\beta}) = \chi F(H)$ yields

$$\frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \mathbf{c}} : \overset{\circ}{\mathbf{c}} + \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \overset{\circ}{\boldsymbol{\beta}} - \chi \dot{F} \leq 0 \quad \text{for } \mathfrak{R}_c = \chi$$

which can be rewritten as

$$\begin{aligned} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \mathbf{c}} : \overset{\circ}{\mathbf{c}} + \left(\frac{1}{\chi F} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \mathbf{c}} : \mathbf{c} \right) \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \overset{\circ}{\boldsymbol{\beta}} \\ - \left(\frac{1}{\chi F} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \mathbf{c}} : \mathbf{c} \right) \chi \dot{F} \leq 0 \quad \text{for } \mathfrak{R}_c = \chi \end{aligned} \quad (20)$$

Making use of the relation $[\partial f(\mathbf{c}, \boldsymbol{\beta})/\partial \mathbf{c}] : \mathbf{c} (= f(\mathbf{c}, \boldsymbol{\beta})) = \chi F$ on account of Euler's homogeneous function $f(\mathbf{c}, \boldsymbol{\beta})$ of \mathbf{c} in degree-one, Eq. (20) is further rewritten as

$$\frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \mathbf{c}} : \left[\overset{\circ}{\mathbf{c}} + \left(\frac{1}{\chi F} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \overset{\circ}{\boldsymbol{\beta}} - \frac{\dot{F}}{F} \right) \mathbf{c} \right] \leq 0 \quad \text{for } \mathfrak{R}_c = \chi \quad (21)$$

Equations (19) and (21) (rate form) are called the *enclosing condition of elastic-core*. Let the following relation be adopted.

$$\overset{\circ}{\mathbf{c}} + \left(\frac{1}{\chi F} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \overset{\circ}{\boldsymbol{\beta}} - \frac{\dot{F}}{F} \right) \mathbf{c} = c_e \|\mathbf{d}^p\| (\boldsymbol{\sigma}_\chi - \mathbf{c}) \quad (22)$$

where c_e is the material constant controlling the translational rate of the elastic-core and $\boldsymbol{\sigma}_\chi$ is the stress on the limit elastic-core surface conjugate to the current stress $\boldsymbol{\sigma}$ on the subloading surface (see Fig. 1), i.e.

$$\boldsymbol{\sigma}_\chi = \frac{\chi}{R} \bar{\boldsymbol{\sigma}} \quad \left(\frac{\bar{\boldsymbol{\sigma}}}{R} = \frac{\boldsymbol{\sigma} - \bar{\boldsymbol{\alpha}}}{R} = \frac{\boldsymbol{\sigma}_\chi - \boldsymbol{\alpha}}{\chi} \right) \quad (23)$$

and $\boldsymbol{\sigma}_y$ is the conjugate point on the normal-yield surface to the current stress on the subloading surface (see Fig. 1). The inequality in Eq. (21) is fulfilled in Eq. (22) as verified by

$$\begin{aligned} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \mathbf{c}} : \left[c_e \|\mathbf{d}^p\| (\boldsymbol{\sigma}_\chi - \mathbf{c}) \right] &= c_e \|\mathbf{d}^p\| \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \mathbf{c}} \\ : (\boldsymbol{\sigma}_\chi - \mathbf{c}) &\leq 0 \quad \text{for } \mathfrak{R}_c = \chi \end{aligned} \quad (24)$$

noting that $\partial f(\mathbf{c}, \boldsymbol{\beta})/\partial \mathbf{c}$ is the outward-normal of the elastic-core surface at the current elastic-core \mathbf{c} and makes an obtuse angle with $\boldsymbol{\sigma}_\chi - \mathbf{c}$ when \mathbf{c} lies on the limit elastic-core surface $f(\mathbf{c}, \boldsymbol{\beta}) = \chi F(H)$ as far as it is the convex surface, while $\boldsymbol{\sigma}_\chi$ lies on the limit elastic-core surface. The irrational evolution rule adopting $\boldsymbol{\sigma}_y$ instead of $\boldsymbol{\sigma}_\chi$ by which the enclosing condition cannot be satisfied in general was proposed by Hashiguchi [26–28] and used by Hashiguchi et al. [41] and Fincato and Tsutsumi [12]. Further, the evolution rule

$$\begin{aligned} \dot{\mathbf{c}} + \left(\frac{1}{\chi F} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} - \frac{\dot{F}}{F} \right) \mathbf{c} &= c_e \left(\mathbf{d}^p - \frac{\mathfrak{R}_c}{\chi} \hat{\mathbf{n}}_c \|\mathbf{d}^p\| \right) \\ &= c_e \left(\bar{\mathbf{n}} - \frac{\mathfrak{R}_c}{\chi} \hat{\mathbf{n}}_c \right) \|\mathbf{d}^p\| \end{aligned} \quad (25)$$

was used by Hashiguchi [30], Hashiguchi and Ueno [39], Iguchi et al. [45–47], etc., where

$$\hat{\mathbf{n}}_c \equiv \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \mathbf{c}} \Big/ \left\| \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \mathbf{c}} \right\| \quad (26)$$

which is the normalized outward-normal of the elastic-core surface at the current elastic-core \mathbf{c} . It cannot be applicable to the generic deformation behavior, since it depends only on the unit outward-normal tensors $\bar{\mathbf{n}}$ and $\hat{\mathbf{n}}_c$ independent of the size and the shape of the normal-yield surface as seen in the right-hand side of Eq. (25).

Consequently, the following evolution rule of \mathbf{c} is given from Eq. (22) with Eq. (23) as follows:

$$\dot{\mathbf{c}} = c_e \|\mathbf{d}^p\| \left(\frac{\chi}{R} \bar{\boldsymbol{\sigma}} - \mathbf{c} \right) + \left(\frac{\dot{F}}{F} - \frac{1}{\chi F} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \mathbf{c} \quad (27)$$

The substitution of Eq. (27) in Eq. (9) yields:

$$\begin{aligned} \dot{\bar{\boldsymbol{\alpha}}} &= (1 - R) \left[c_e \|\mathbf{d}^p\| \left(\frac{\chi}{R} \bar{\boldsymbol{\sigma}} - \mathbf{c} \right) + \left(\frac{\dot{F}}{F} - \frac{1}{\chi F} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \mathbf{c} \right] \\ &\quad - \dot{R} \mathbf{c} \end{aligned} \quad (28)$$

3.4 Consistency condition

The substitution of Eq. (28) in Eq. (11) leads to the consistency condition:

$$\begin{aligned} \bar{\mathbf{n}} : \dot{\bar{\boldsymbol{\sigma}}} - \dot{\bar{\mathbf{n}}} : \left\{ (1 - R) \left[c_e \|\mathbf{d}^p\| \left(\frac{\chi}{R} \bar{\boldsymbol{\sigma}} - \mathbf{c} \right) \right. \right. \\ \left. \left. + \left(\frac{\dot{F}}{F} - \frac{1}{\chi F} \frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \mathbf{c} \right] - \dot{R} \mathbf{c} + \frac{\dot{F}}{F} \bar{\boldsymbol{\sigma}} + \frac{\dot{R}}{R} \bar{\boldsymbol{\sigma}} \right. \\ \left. - \frac{1}{RF} \left(\frac{\partial f(\bar{\boldsymbol{\sigma}}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \bar{\boldsymbol{\sigma}} \right\} = 0 \end{aligned}$$

i.e.

$$\begin{aligned} \bar{\mathbf{n}} : \dot{\bar{\boldsymbol{\sigma}}} - \dot{\bar{\mathbf{n}}} : \left[c_e (1 - R) \|\mathbf{d}^p\| \left(\frac{\chi}{R} \bar{\boldsymbol{\sigma}} - \mathbf{c} \right) + \frac{\dot{F}}{F} [\bar{\boldsymbol{\sigma}} + (1 - R) \mathbf{c}] \right. \\ \left. + \frac{\dot{R}}{R} (\bar{\boldsymbol{\sigma}} - R \mathbf{c}) - \frac{1}{RF} \left(\frac{\partial f(\bar{\boldsymbol{\sigma}}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \bar{\boldsymbol{\sigma}} \right. \\ \left. - \frac{1 - R}{\chi F} \left(\frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \mathbf{c} \right] = 0 \end{aligned} \quad (29)$$

By virtue of the relations

$$\begin{cases} \bar{\boldsymbol{\sigma}} + (1 - R) \mathbf{c} = \boldsymbol{\sigma} \\ \bar{\boldsymbol{\sigma}} - R \mathbf{c} = \hat{\boldsymbol{\sigma}} \end{cases}$$

deduced from Eq. (6), Eq. (29) is simplified to the equation:

$$\begin{aligned} \bar{\mathbf{n}} : \dot{\bar{\boldsymbol{\sigma}}} - \dot{\bar{\mathbf{n}}} : \left[c_e (1 - R) \|\mathbf{d}^p\| \left(\frac{\chi}{R} \bar{\boldsymbol{\sigma}} - \mathbf{c} \right) + \frac{\dot{F}}{F} \bar{\boldsymbol{\sigma}} + \frac{\dot{R}}{R} \hat{\boldsymbol{\sigma}} \right. \\ \left. - \frac{1}{RF} \left(\frac{\partial f(\bar{\boldsymbol{\sigma}}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \bar{\boldsymbol{\sigma}} - \frac{1 - R}{\chi F} \left(\frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \mathbf{c} \right] = 0 \end{aligned} \quad (30)$$

3.5 Plastic strain rate

The associated flow rule is adopted for the subloading surface, i.e.

$$\mathbf{d}^p = \dot{\lambda} \bar{\mathbf{n}} \quad (\dot{\lambda} > 0) \quad (31)$$

where $\dot{\lambda}$ is the plastic multiplier, i.e., positive proportionality factor. Here, note that the associated flow rule can be applied to the subloading surface even for soils as will be described in Sect. 3.7.

Substitution of Eqs. (14) and (31) in Eq. (30) yields:

$$\bar{\mathbf{n}} : \dot{\bar{\boldsymbol{\sigma}}} - \dot{\lambda} \dot{M}^p = 0 \quad (32)$$

where

$$\begin{aligned} \dot{M}^p \equiv \bar{\mathbf{n}} : \left[\frac{F'h}{F} \bar{\boldsymbol{\sigma}} + \frac{U}{R} \hat{\boldsymbol{\sigma}} + c(1 - R) \left(\frac{\bar{\boldsymbol{\sigma}}}{R} - \frac{\mathbf{c}}{\chi} \right) \right. \\ \left. - \frac{1}{RF} \left(\frac{\partial f(\bar{\boldsymbol{\sigma}}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \bar{\boldsymbol{\sigma}} - \frac{1 - R}{\chi F} \left(\frac{\partial f(\mathbf{c}, \boldsymbol{\beta})}{\partial \boldsymbol{\beta}} : \dot{\boldsymbol{\beta}} \right) \mathbf{c} \right] \end{aligned} \quad (33)$$

$$F' \equiv dF/dH, \quad h \equiv \dot{H}/\dot{\lambda}, \quad \mathbf{b} \equiv \dot{\boldsymbol{\beta}}/\dot{\lambda} \quad (34)$$

The explicit functions of F' , h , and \mathbf{b} will be formulated later in Eqs. (68), (77) and (81), respectively, in Sect. 4 for granular materials.

It follows from Eq. (32) that

$$\dot{\lambda} = \frac{\bar{\mathbf{n}} : \dot{\bar{\boldsymbol{\sigma}}}}{\dot{M}^p}, \quad \mathbf{d}^p = \frac{\bar{\mathbf{n}} : \dot{\bar{\boldsymbol{\sigma}}}}{\dot{M}^p} \bar{\mathbf{n}} \quad (35)$$

The strain rate is given from Eq. (35) along with Eqs. (1) and (2) as follows:

$$\mathbf{d} = \mathbf{E}^{-1} : \dot{\bar{\boldsymbol{\sigma}}} + \frac{\bar{\mathbf{n}} : \dot{\bar{\boldsymbol{\sigma}}}}{\dot{M}^p} \bar{\mathbf{n}} \quad (36)$$

from which the proportionality factor described in terms of the strain rate in the flow rule (31), denoted by $\dot{\Lambda}$ instead of $\dot{\lambda}$, is derived as follows:

$$\dot{\bar{\Lambda}} = \frac{\bar{\mathbf{n}} : \mathbf{E} : \mathbf{d}}{\bar{M}^p + \bar{\mathbf{n}} : \mathbf{E} : \bar{\mathbf{n}}} \quad (37)$$

Using Eq. (37), the stress rate is given from Eq. (36) as follows:

$$\begin{aligned} \overset{\circ}{\boldsymbol{\sigma}} &= \mathbf{E} : \mathbf{d} - \frac{\bar{\mathbf{n}} : \mathbf{E} : \mathbf{d}}{\bar{M}^p + \bar{\mathbf{n}} : \mathbf{E} : \bar{\mathbf{n}}} \mathbf{E} : \bar{\mathbf{n}} \\ &= \left(\mathbf{E} - \frac{(\mathbf{E} : \bar{\mathbf{n}}) \otimes (\bar{\mathbf{n}} : \mathbf{E})}{\bar{M}^p + \bar{\mathbf{n}} : \mathbf{E} : \bar{\mathbf{n}}} \right) : \mathbf{d} \end{aligned} \quad (38)$$

The loading criterion is given as follows (Hashiguchi [21, 22]):

$$\begin{cases} \mathbf{d}^p \neq \mathbf{0} & \text{for } \dot{\bar{\Lambda}} > 0, \\ \mathbf{d}^p = \mathbf{0} & \text{for other} \end{cases} \quad (39)$$

or

$$\begin{cases} \mathbf{d}^p \neq \mathbf{0} & \text{for } \mathbf{n} : \mathbf{E} : \mathbf{d} > 0, \\ \mathbf{d}^p = \mathbf{0} & \text{for other} \end{cases} \quad (40)$$

where the judgment of yielding whether the stress reaches the yield surface is not required because the plastic strain rate develops continuously as the stress approaches the yield surface.

3.6 Modification of unloading–reloading response

The variation of the accumulated plastic strain $\varepsilon_b^p - \varepsilon_a^p$ ($\varepsilon^p = \int_0^t \|\mathbf{d}^p\| dt$) for a certain variation $R_b - R_a$ of the normal–yield ratio induced during the plastic deformation process from the state a to the state b is identical regardless of loading processes, e.g. initial loading, reloading and inverse loading, proportional and non-proportional loadings as known from Eq. (17), if the parameter u in Eq. (16) is a constant. It leads to the impertinent description that the reloading stress–strain curve after a partial unloading returns to the preceding stress–strain curve too gently. Therefore, it leads to the inadequate prediction of cyclic loading behavior, resulting in an unrealistically large plastic strain accumulation during cyclic loading. The material parameter u is then extended to describe the generalized Masing effect [50] as follows:

$$u = \bar{u} \exp(u_c \mathfrak{R}_c C_n) \quad (\bar{u} \exp(-u_c \chi) \leq u \leq \bar{u} \exp(u_c \chi)) \quad (41)$$

$$\left(= \begin{cases} \bar{u} \exp(u_c \chi) & \text{(largest) for } \mathfrak{R}_c = \chi \text{ and } C_n = 1 \\ \bar{u} & \text{(average) for } \mathfrak{R}_c = 0 \text{ or } C_n = 0 \\ \bar{u} \exp(-u_c \chi) & \text{(smallest) for } \mathfrak{R}_c = \chi \text{ and } C_n = -1 \end{cases} \right)$$

where

$$C_n \equiv \hat{\mathbf{n}}_c : \bar{\mathbf{n}} \quad (-1 \leq C_n \leq 1) \quad (42)$$

\bar{u} (average value of u) and u_c are material constants and u is a continuous function of variables \mathfrak{R}_c and C_n . The forms of the function u for particular states are shown in the bracket. $C_n = 1, 0$ and -1 indicate states for which the plastic strain rate is directed outward-normal, tangential and inward-normal, respectively, to the elastic-core surface. By this modification, a realistic description is given for the phenomenon in which the reloading curve after a partial unloading returns rapidly to the preceding loading curve and instead the curvature of the inverse loading curve decreases leading to the generalized Masing effect.

3.7 Basic characteristics of subloading surface model

The subloading surface model possesses the following distinguished features.

- (1) The plastic deformation is not induced abruptly but develops gradually. In fact, mutual slips between material particles, e.g., crystal particles in metals and soil particles in sands and clays, are not induced simultaneously but induced gradually from parts in which mutual slips can be induced easily. Therefore, the plastic strain rate develops continuously as the stress approaches the yield surface. Then, the elastoplastic constitutive equation is required to satisfy the following smoothness condition (Hashiguchi [21, 22, 24]).

$$\lim_{\delta \boldsymbol{\sigma} \rightarrow 0} \overset{\circ}{\boldsymbol{\sigma}}(\boldsymbol{\sigma} + \delta \boldsymbol{\sigma}, \mathbf{H}, H; \mathbf{d}) \rightarrow \overset{\circ}{\boldsymbol{\sigma}}(\boldsymbol{\sigma}, \mathbf{H}, H; \mathbf{d}) \quad (43)$$

where \mathbf{H} and H are the second-order tensor-valued and the scalar-valued internal variables, respectively, and $\delta(\cdot)$ stands for an infinitesimal variation. The rate-linear constitutive equation is described as

$$\overset{\circ}{\boldsymbol{\sigma}} = \mathbb{M}^{\text{ep}}(\boldsymbol{\sigma}, \mathbf{H}, H) : \mathbf{d} \quad (44)$$

where the fourth-order tensor \mathbb{M}^{ep} is the tangent stiffness modulus tensor, which is the function of the stress and internal variables, and can be described generally by

$$\mathbb{M}^{\text{ep}} = \frac{\partial \boldsymbol{\sigma}}{\partial \boldsymbol{\varepsilon}} \quad (45)$$

where

$$\boldsymbol{\varepsilon} \equiv \int_0^t \mathbf{d} dt \quad (46)$$

Consequently, Eq. (44) can be rewritten as

$$\lim_{\delta \boldsymbol{\sigma} \rightarrow 0} \mathbb{M}^{\text{ep}}(\boldsymbol{\sigma} + \delta \boldsymbol{\sigma}, \mathbf{H}, H) \rightarrow \mathbb{M}^{\text{ep}}(\boldsymbol{\sigma}, \mathbf{H}, H) \quad (47)$$

The smoothness condition is satisfied in the

subloading surface model, while it is violated in the yield point by the other elastoplastic models because they assume the purely elastic domain.

- (2) The smooth transition from the elastic to the plastic state and the continuous variation of the tangent stiffness modulus tensor are described by the subloading surface model. However, it is violated at the yield point in any other elastoplasticity models because they assume a purely elastic domain inheriting the conventional elastoplasticity model.
- (3) The yield judgment whether the stress reaches the yield surface is unnecessary since the plastic strain rate develops continuously as the stress approaches the normal-yield surface. In contrast, a yield judgment is required in the other elastoplasticity models because they assume a surface enclosing a pure elastic domain, while the determination of the yield stress is accompanied with an arbitrariness because the tested stress vs. strain curve are usually smooth.
- (4) The tangent stiffness modulus changes always continuously, while it changes abruptly at the yield point in the other elastoplasticity model.
- (5) The plastic strain rate can be described for any small stress variation and for cyclic loading under any small stress amplitudes since a pure elastic domain is not assumed. However, it cannot be described during the stress variation inside the yield surface enclosing a purely elastic domain in the other elastoplasticity models.
- (6) The automatic stress-controlling function is furnished such that the stress is always attracted to the normal-yield surface. In particular, it is noticeable that the stress is automatically pulled back to the normal-yield surface when it goes over the surface in numerical calculation because of $\dot{R} < 0$ for $R > 1$

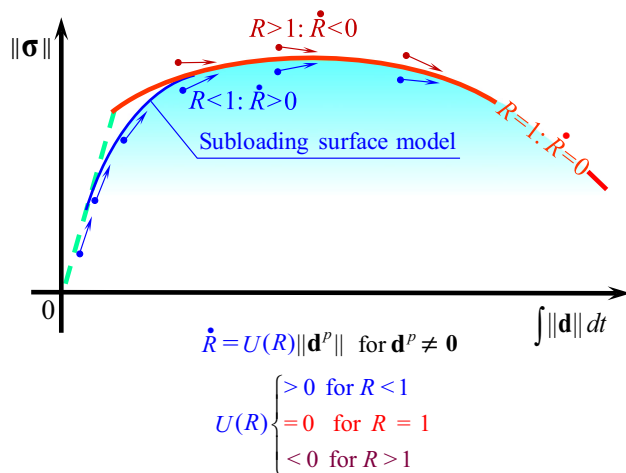


Fig. 3 Stress is automatically attracted to the normal-yield surface in subloading surface model

from Eq. (14) with Eq. (15)₄ (see Fig. 3). In contrast, the particular operation to pull back the stress is required in the other elastoplasticity models.

- (7) The outward-normal \mathbf{n} at the peak stress point in the subloading surface is approximately coincides with the outward-normal of the plastic potential surface adopted in the Drucker–Prager model [9] as shown in Fig. 4 in which \mathbf{n}^{DP} designates the outward-normal of the Drucker–Prager yield surface. Therefore, the associated flow rule can be adopted in the subloading surface model. On the other hand, the non-associated flow must be adopted in the Cap model composed of the Drucker–Prager model in the over-consolidated state and the Cam-clay model [61, 62] in the normal-consolidated state in order to suppress the excessive volumetric expansion in the over-consolidated state, which leads to the asymmetry of the elastoplastic tangent modulus tensor.
- (8) The exact finite strain elastoplastic constitutive equation, i.e., the multiplicative hyperelastic-based plastic constitutive equation can be formulated only by incorporating the subloading surface model (Hashiguchi [29, 32, 33], Hashiguchi and Yamakawa [42]).

4 Material functions for granular materials

The material functions included in the subloading surface model described above are now formulated for a wide range of granular materials involving clays and sands, generalizing the past formulations to achieve the accurate description of cyclic loading behavior up to the cyclic mobility.

4.1 Elastic moduli

The elastic bulk modulus K and the elastic shear modulus G are given by Hashiguchi [31], modifying the former equations to describe the pressure-dependency (Hashiguchi [23, 32], Hashiguchi and Chen [34]) as follows:

$$K = \frac{-\dot{p}}{d^e_v} = \frac{p + \vartheta F}{\tilde{\kappa}}, \quad G = \frac{\|\dot{\boldsymbol{\sigma}}'\|}{2\|\mathbf{d}^e\|} = G_0 \left(\frac{p + \vartheta F}{p_0 + \vartheta F_0} \right)^n \tag{48}$$

where $\tilde{\kappa}$ represents the slope of the swelling curve in the linear relation $\ln v - \ln p$ with both logarithms of volume v and pressure p in the isotropic consolidation. The term ϑF is incorporated so that K and G are applicable even for the negative pressure range. Here, ϑ is a material constant

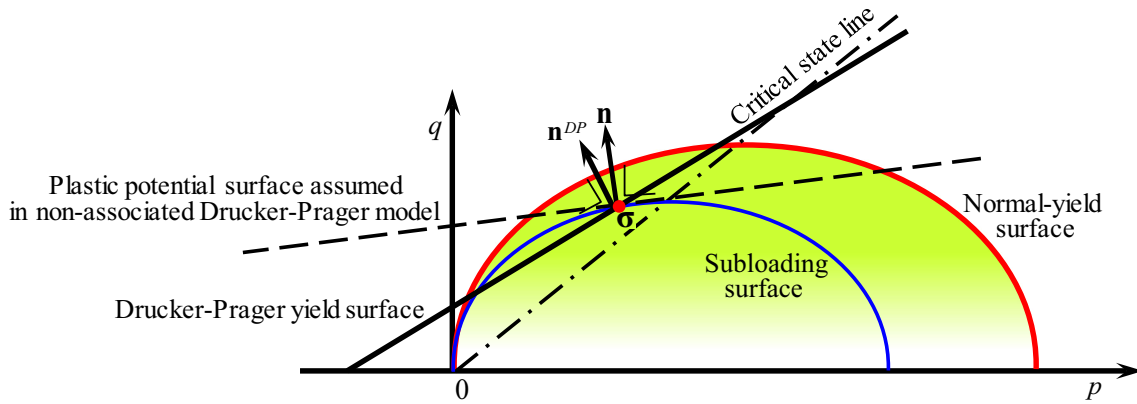


Fig. 4 Outward-normal of subloading surface, which coincides approximately with the plastic potential surface assumed in the Drucker–Prager model [9]

leading to $\varepsilon_v^e \rightarrow \infty$ (infinite volume expansion) for $p \rightarrow -\vartheta F$, while ϑ ($0 \leq \vartheta < 0.1$) for common clays and sands (Hashiguchi and Chen [34], Hashiguchi and Mase [35]). d_v^e is the elastic volumetric strain rate $d_v^e \equiv \text{tr } \mathbf{d}^e$. The power function of the pressure in Eq. (48) for the shear modulus is referred to Tatsuoka et al. [64]. n (≤ 1) is a material constant, while $n \cong 0.5$ is chosen for most sands. Consequently, the elastic tangent modulus depends naturally on pressure p and isotropic hardening function F . The elastic moduli in Eq. (48) hold for granular materials consistently in the frameworks of the infinitesimal elasticity, the hypoelasticity and multiplicative hyperelasticity as was verified by Hashiguchi [31].

4.2 Yield and subloading functions

Let the normal-yield surface be given by the modified Cam-clay model (Burland [7], Roscoe and Burland [61]):

$$p \left[1 + \left(\frac{\|\boldsymbol{\sigma}'\|/p}{M} \right)^2 \right] = F \tag{49}$$

i.e.

$$\left(\frac{p - (1/2)F}{F/2} \right)^2 + \left(\frac{\|\boldsymbol{\sigma}'\|}{MF/2} \right)^2 = 1 \tag{50}$$

where M is the stress ratio $\|\boldsymbol{\sigma}'\|/p$ in the critical state. To take account of the influence of the third deviatoric stress invariant, M is extended as follows (Hashiguchi [25]):

$$M = \frac{7}{8 + \cos 3\theta_\sigma} M_c = \frac{14\sqrt{6} \sin \phi_c}{(3 - \sin \phi_c)(8 + \cos 3\theta_\sigma)} \tag{51}$$

where θ_σ is the so-called Lode angle defined by

$$\cos 3\theta_\sigma \equiv \sqrt{6} \text{tr}(\boldsymbol{\rho}')^3, \quad \boldsymbol{\rho}' \equiv \frac{\boldsymbol{\sigma}'}{\|\boldsymbol{\sigma}'\|} \tag{52}$$

and

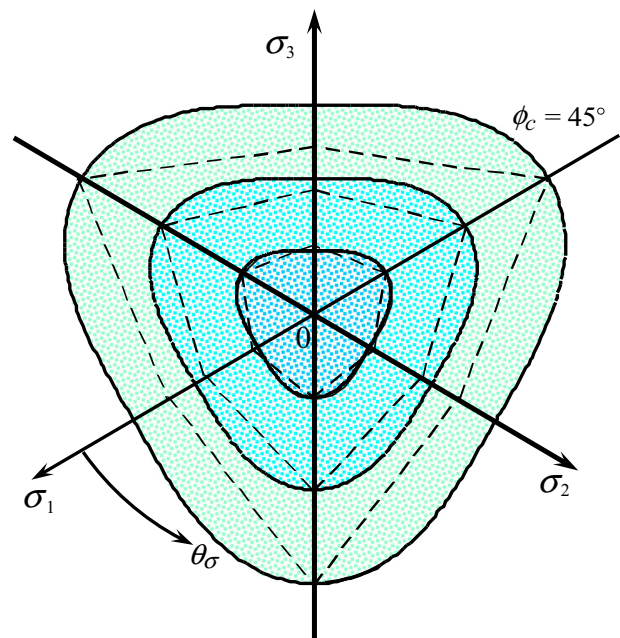


Fig. 5 Section of the critical state surface in π -plane. (Coulomb–Mohr criterion is shown by dotted lines for reference)

$$M_c \equiv \frac{2\sqrt{6} \sin \phi_c}{3 - \sin \phi_c} \tag{53}$$

ϕ_c being the angle of internal friction in the critical state for the axisymmetric, i.e., triaxial compression stress state ($\theta_\sigma = \pi/3 \pm n(2\pi/3)$) (n : integer). The section of the normal-yield surface at the critical state given by Eq. (49) with Eq. (51) always fulfills the convexity condition as shown in Fig. 5 for the section of the critical state surface in the deviatoric stress plane. Equation (51) would be the simplest equation taking account of the third deviatoric stress invariant and fulfilling the convexity among various conical surface equations in the principal stress space [25].

Sands possess quite small strength in the negative pressure range. However, it is of crucial importance to shift the yield surface to the negative pressure range in order to describe rigorously the cyclic mobility, in which the stress changes actively around the null stress state. Here, it should be noticed that the shift of the yield surface is of the importance also in the computational aspect.

To shift the yield surface to the negative pressure range, Eq. (50) is modified by $\xi F(p \rightarrow p + \xi F)$ as follows (Hashiguchi and Mase [35]).

$$\left(\frac{p - ((1/2) - \xi)F}{F/2}\right)^2 + \left(\frac{\|\sigma'\|}{MF/2}\right)^2 = 1 \tag{54}$$

where ξ is the material constant, while it must fulfill $\xi \leq 1/2$ since the tensile yield stress is smaller than the compression yield stress, and further the inequality $\xi < \vartheta$ is required since the volume does not become infinite by the elastic deformation inside the yield surface, satisfying $p \geq -\xi F > -\vartheta F$. The yield surface in Eq. (54) is shown in Fig. 6 in the pressure-deviatoric stress plane.

Equation (54) is extended by taking account of the rotation of the yield surface around the origin of the stress space into account and thus by replacing σ' with $\sigma' - p\beta$, where β is referred to as the *rotational hardening variable* represented by a deviatoric tensor, the evolution rule of which will be formulated in Sect. 4.4. This extension leads to

$$\left(\frac{p - ((1/2) - \xi)F}{F/2}\right)^2 + \left(\frac{\|\widehat{\sigma}'\|}{MF/2}\right)^2 = 1 \tag{55}$$

where

$$\widehat{\sigma}' \equiv \sigma' - p\beta \tag{56}$$

$$\widehat{M}(\cos 3\theta_{\widehat{\sigma}'}) = \frac{14\sqrt{6} \sin \phi_c}{(3 - \sin \phi_c)(8 + \cos 3\theta_{\widehat{\sigma}'})} M_c \tag{57}$$

$$\cos 3\theta_{\widehat{\sigma}'} \equiv \sqrt{6} \text{tr}(\mathbf{t}'_{\widehat{\sigma}'})^3, \quad \mathbf{t}'_{\widehat{\sigma}'} \equiv \frac{\widehat{\sigma}'}{\|\widehat{\sigma}'\|} \tag{58}$$

Equation (55) is rewritten as follows:

$$(1 - \xi)\xi F^2 + (1 - 2\xi)pF - (p^2 + \widehat{\rho}^2) = 0 \tag{59}$$

where

$$\widehat{\rho} \equiv \frac{\|\widehat{\sigma}'\|}{\widehat{M}} \tag{60}$$

Equation (59) is the quadratic equation of the hardening function F . By solving this equation, it can be described in the separated form to the function $f(p, \widehat{\rho})$ and the hardening function F as follows:

$$f(p, \widehat{\rho}) = F, \tag{61}$$

$$f(\sigma, \beta) = f(p, \widehat{\rho}) = \begin{cases} p[1 + (\widehat{\rho}/p)^2] & \text{for } \xi = 0 \\ \frac{1}{\xi}(\widehat{\rho} - \bar{\xi}p) & \text{for } \xi \neq 0 \end{cases}$$

where

$$\bar{\xi} \equiv 2(1 - \xi)\xi, \quad \bar{\xi} \equiv 1 - 2\xi, \quad \widehat{p}_\rho \equiv \sqrt{p^2 + 2\bar{\xi}\widehat{\rho}^2} \tag{62}$$

Note that an analytically separated form cannot be derived from the yield conditions other than the modified Cam-clay model, e.g. the original Cam-clay model (Schofield and Wroth [62]) which is shifted to the negative pressure range by $p \rightarrow p + \xi F$ leading to $p \exp[\|\sigma'\|/(pM)] = F \rightarrow (p + \xi F) \exp[\|\sigma'\|/\{(p + \xi F)M\}] = F$.

Further, let the subloading stress function $f(\bar{\sigma}, \beta)$ in Eq. (5) be given from Eq. (61) by replacing σ for the normal-yield surface to $\bar{\sigma}$ for the subloading surface, i.e., $\widehat{\sigma}'$ to $\widehat{\bar{\sigma}}'$ as follows:

$$f(\bar{\sigma}, \beta) = f(\bar{p}, \widehat{\bar{\rho}}) = \begin{cases} \bar{p}[1 + (\widehat{\bar{\rho}}/\bar{p})^2] & \text{for } \xi = 0 \\ \frac{1}{\xi}(\widehat{\bar{\rho}} - \bar{\xi}\bar{p}) & \text{for } \xi \neq 0 \end{cases} \tag{63}$$

where

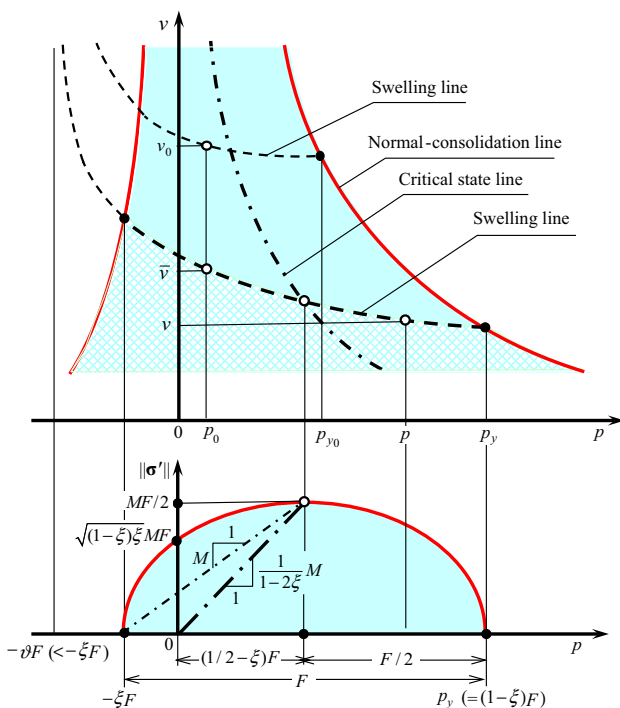


Fig. 6 Yield surface of granular materials with tensile strength

$$\widehat{p}_\rho \equiv \sqrt{\widehat{p}^2 + 2\widehat{\xi}\widehat{p}^2} \quad (\widehat{p} \equiv -(\text{tr}\widehat{\sigma})/3) \tag{64}$$

$$\widehat{p} \equiv \frac{\|\widehat{\sigma}'\|}{\widehat{M}}, \quad \widehat{\sigma}' \equiv \widehat{\sigma} - \widehat{p}\beta \tag{65}$$

$$\widehat{M}(\cos 3\theta_{\widehat{\sigma}}) = \frac{14\sqrt{6} \sin \phi_c}{(3 - \sin \phi_c)(8 + \cos 3\theta_{\widehat{\sigma}})} \tag{66}$$

$$(\cos 3\theta_{\widehat{\sigma}}) \equiv \sqrt{6} \text{tr}(\widehat{p}')^3, \quad \widehat{p}' \equiv \frac{\widehat{\sigma}'}{\|\widehat{\sigma}'\|} \tag{67}$$

4.3 Isotropic hardening by volumetric and deviatoric plastic strain rates

The hardening function in Eqs. (4) or (5) is given as follows (Hashiguchi [32]):

$$F(H) = F_0 \exp\left(\frac{H}{\widehat{\lambda} - \widehat{\kappa}}\right), \quad F' \equiv \frac{dF}{dH} = \frac{1}{\widehat{\lambda} - \widehat{\kappa}} F \tag{68}$$

where $\widehat{\lambda}$ and $\widehat{\kappa}$ stand for the slopes of the normal-consolidation curve and the swelling curve, respectively, in the $\ln v - \ln(p + \vartheta F)$ plane (Hashiguchi [23]), which is based on the following isotropic consolidation characteristics.

$$\begin{cases} \varepsilon_v^e = \ln \frac{v}{\bar{v}} = -\widehat{\kappa} \ln \frac{p + \vartheta p_y}{p_0 + \vartheta p_{y0}} \\ \varepsilon_v^p = \ln \frac{\bar{v}}{v_0} = -(\widehat{\lambda} - \widehat{\kappa}) \ln \frac{p_y}{p_{y0}} \end{cases} \tag{69}$$

where ε_v^e and ε_v^p are the elastic and the plastic logarithmic volumetric strains, respectively; v , v_0 and \bar{v} are the current volume, the initial volume and the volume unloaded state to the initial pressure, respectively; and p , p_0 , p_y and p_{y0} are the current, the initial and the current yield and the initial yield pressure, respectively, as shown in Fig. 6. Replacing the pressures p_y and p_{y0} to the isotropic hardening function F and its initial value F_0 , respectively, in Eq. (69), one has

$$\begin{cases} \varepsilon_v^e = -\widehat{\kappa} \ln \frac{p + \vartheta F}{p_0 + \vartheta F_0} \\ \varepsilon_v^p = -(\widehat{\lambda} - \widehat{\kappa}) \ln \frac{F}{F_0} \end{cases} \tag{70}$$

The isotropic hardening function is given from Eq. (70) by choosing the isotropic hardening variable H to be the plastic volumetric contraction, i.e., $H = -\varepsilon_v^p$ as follows:

$$F(H) = F_0 \exp\left(\frac{H}{\widehat{\lambda} - \widehat{\kappa}}\right) \tag{71}$$

The rate of isotropic hardening/softening variable H was extended by Nova [57] and Wilde [67] to incorporate the influence of the deviatoric plastic strain rate, called the *deviatoric (isotropic) hardening*, in addition to the

volumetric plastic strain rate described above. Here, aiming at establishing a quantitative description of cyclic mobility, let \dot{H} be extended to depend on the plastic volumetric strain rate d_v^p and the plastic shear rate d_s^p for a wide range of stress up to the negative pressure as follows (Fig. 6):

$$\dot{H} = -d_v^p + d_s^p = \dot{\lambda} h \tag{72}$$

$$d_v^p \equiv \text{tr} \mathbf{d}^p, \quad \mathbf{d}^p \equiv \mathbf{d}^p - (d_v^p/3) \mathbf{I} \tag{73}$$

$$d_s^p = \mu_d \|\mathbf{d}^p\| \frac{\chi_d^a - 1}{\chi_d^a - 1 + b} \begin{cases} > 0 & \text{(deviatoric hardening) for } \chi_d > 1 \\ = 0 & \text{(deviatoric non-hardening) for } \chi_d = 1 \\ < 0 & \text{(deviatoric softening) for } \chi_d < 1 \end{cases} \tag{74}$$

$$\chi_d \equiv \frac{\|\sigma'\|/(p + \vartheta F)}{M_d} \tag{75}$$

$$M_d(\cos 3\theta_{\sigma'}) \equiv \frac{14\sqrt{6} \sin \phi_d}{(3 - \sin \phi_d)(8 + \cos 3\theta_{\sigma'})} \tag{76}$$

$$h = -\text{tr} \bar{\mathbf{n}} + \mu_d \|\bar{\mathbf{n}}\| \frac{\chi_d^a - 1}{\chi_d^a - 1 + b} \tag{77}$$

where μ_d , $\phi_d (< \phi_c)$, $a (> 1)$ and $b (> 1)$ are material constants. Here, $\phi_d < \phi_c$ holds in general since the deviatoric stress can increase over the critical state ($\text{tr} \bar{\mathbf{n}} = 0 \rightsquigarrow \text{tr} \mathbf{d}^p = 0$) when the stress ratio increases at a low pressure under the undrained condition even in loose sands. M_d in Eq. (76) is formulated analogously to M in Eq. (51). Hardening and softening are induced outside and inside, respectively, the conical surface $\|\sigma'\| = M_d(p + \vartheta F)$ which is called the *deviatoric hardening boundary surface*. It is extended to accommodate the negative pressure p ($-\vartheta F < p \leq 0$) by replacing the normalized stress ratio $(\|\sigma'\|/p)/M_d$ to χ_d defined in Eq. (75). The deviatoric hardening rate depends nonlinearly on the modified stress ratio χ_d with the upper limit $\mu_d \|\mathbf{d}^p\|$ of the plastic shear rate d_s^p (Fig. 7).

4.4 Rotational hardening: Anisotropic hardening by rotation of yield surface

The rotational hardening rule is formulated by

- (1) postulating that the central axis $\sigma'/p = \beta$ of the normal-yield surface rotates towards the conjugate generating line $\sigma'/p = \widehat{M}_r \widehat{p}'$ on the rotational limit surface $\|\sigma'\|/p = \widehat{M}_r$,
- (2) noting that the anisotropic hardening is induced only by the deviatoric plastic strain rate as follows (Fig. 8):

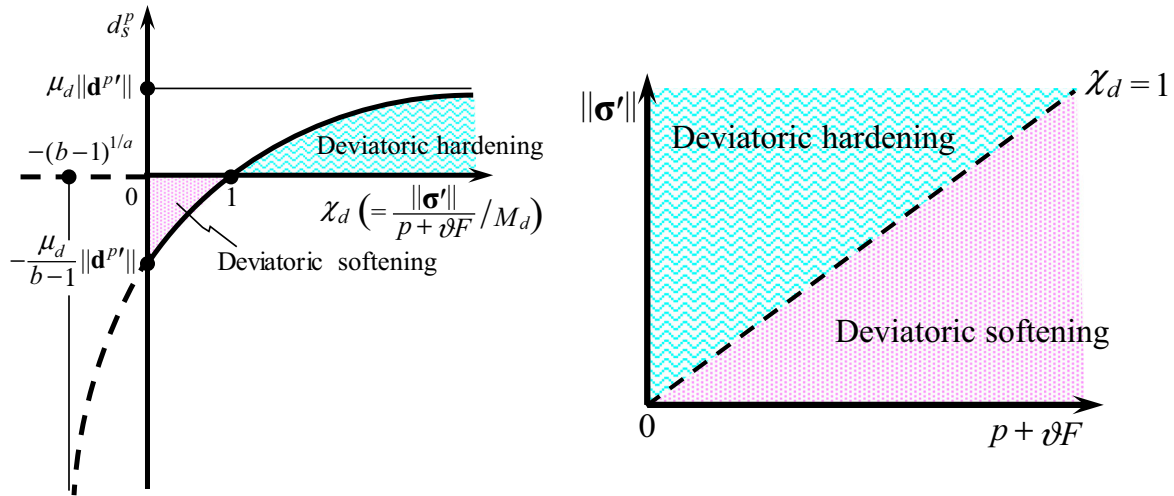


Fig. 7 Deviatoric hardening/softening

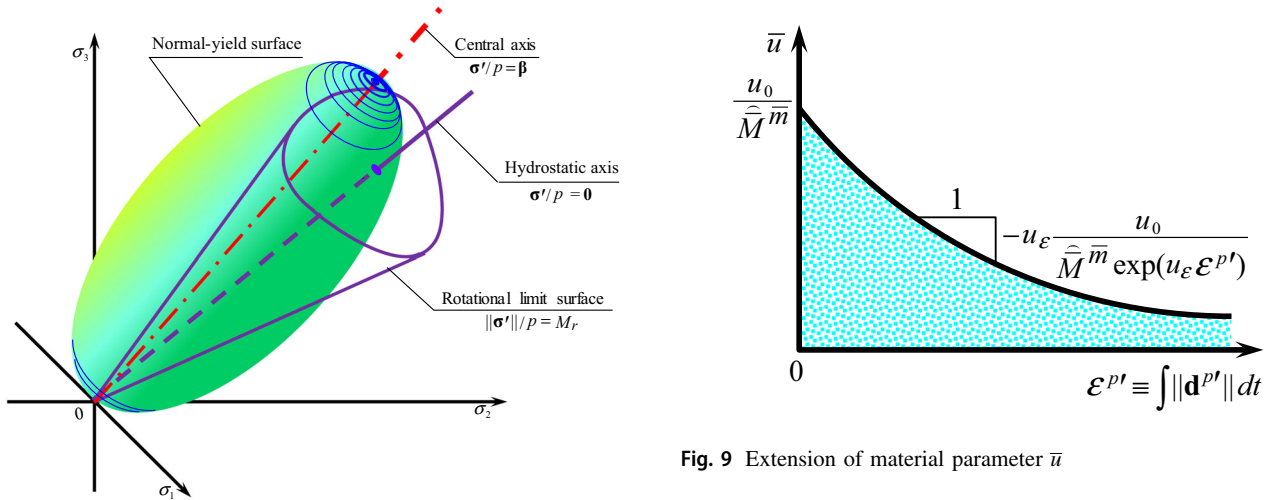


Fig. 8 Yield surface and rotational limit surfaces illustrated in the principal stress space

$$\mathring{\beta} = b_r \left(\mathbf{d}^{p'} - \frac{1}{\widehat{M}_r} \|\mathbf{d}^{p'}\| \beta \right) = \mathbf{b} \dot{\lambda} \tag{78}$$

where

$$\widehat{M}(\cos 3\theta_{\widehat{\sigma}}) = \frac{14\sqrt{6} \sin \phi_r}{(3 - \sin \phi_r)(8 + \cos 3\theta_{\widehat{\sigma}})} \tag{79}$$

$$\widehat{\sigma}' \equiv \widehat{\sigma}' - \bar{p}\beta, \quad \mathbf{t}'_{\widehat{\sigma}} \equiv \frac{\widehat{\sigma}'}{\|\widehat{\sigma}'\|}, \quad \cos 3\theta_{\widehat{\sigma}} \equiv \sqrt{6} \text{tr}(\mathbf{t}'_{\widehat{\sigma}})^3 \tag{80}$$

$$\mathbf{b} = b_r \left(\bar{\mathbf{n}}' - \frac{1}{\widehat{M}_r} \|\bar{\mathbf{n}}'\| \beta \right) \tag{81}$$

Fig. 9 Extension of material parameter \bar{u}

b_r is the material constant regulating the rate of the rotation. \widehat{M}_r is formulated analogously to M in Eqs. (51), (57) and (76), where ϕ_r is the material constant denoting the limit of the rotational angle of the normal-yield and subloading surfaces. Note here that Eq. (78) takes a form analogous to the evolution equation of back stress tensor in the nonlinear kinematic hardening rule [5]. For the subloading surface in Eq. (63) based on the modified Cam-clay model, the rotation does not occur, i.e., $\mathring{\beta} = \mathbf{0}$ when the stress lies on the central axis of the subloading surface, fulfilling $\widehat{\sigma}' = \bar{p}\beta$, for which $\bar{\mathbf{n}}' = \mathbf{0}$ leading to $\mathbf{d}^{p'} = \mathbf{0}$ holds as shown in Fig. 1.

4.5 Extension of material parameter for evolution of normal-yield ratio

The material parameter \bar{u} in the evolution equation (41) of the normal-yield ratio is extended as follows (Fig. 9):

$$\bar{u} = \frac{u_0}{\widehat{M}^{\bar{m}} \exp(u_e \varepsilon^{p'})} \tag{82}$$

where

$$\varepsilon^{p'} \equiv \int_0^t \|\mathbf{d}^{p'}\| dt \tag{83}$$

u_0 , \bar{m} and u_e are material constants. The material parameter \bar{u} becomes smaller inducing a larger plastic strain rate as the accumulation of the deviatoric plastic strain rate proceeds, while this trend is more notable for larger values of u_e . In addition, \bar{u} is inversely proportional to \widehat{M} and therefore a larger plastic strain rate is induced in the compression side than in the tension side, while this trend is more notable for larger value of \bar{m} .

5 Summary of material parameters and their physical meanings

Not a few material parameters, i.e., nineteen material constants and three initial values of internal variables are incorporated in the present elastoplastic constitutive equation in order to describe the cyclic mobility accurately. They are shown collectively in the following.

Material constants:

- Elastic moduli $\tilde{\kappa}$, G_0 , n
- Yield surface (ellipsoid) ϕ_c , ξ (< 0.5)
- Isotropic hardening/softening
 - { volumetric: $\tilde{\lambda}$, ϑ ($> \xi$)
 - { deviatoric: μ_d , ϕ_d ($< \phi_c$), a (≥ 1), b (≥ 1),
- Anisotropic (rotational) hardening: b_r , ϕ_r
- Normal - yield ratio u_c , \bar{u} , u_e , \bar{m}
- Similarity - center : c_e , χ (< 1)

Initial values of internal variables:

- Isotropic hardening function F_0 .
- Rotational hardening β_0 .
- Elastic-core c_0 .

F_0 , G_0 , ϑ , ϕ_c , ξ , \bar{u} , c_e and χ are larger but ϕ_d , $\tilde{\lambda}$ and $\tilde{\kappa}$ are smaller in denser granular materials with higher strength for same arrangement of particles.

The main influences of the material constants on the deformation behavior are described below.

1. The transition from the elastic to plastic state is gentler for smaller u_0 values.
2. Axial strain is induced more intensely in the triaxial compression side for larger \bar{m} values.
3. Strain rate increases more rapidly with the accumulated deviatoric plastic strain for larger u_e values.

4. The difference between the tangent stiffness moduli in reloading and reverse loading is larger and thus the difference between the curvatures of the stress–strain curves in them is larger for larger u_e values.
5. Plastic deformation begins sooner after unloading for larger c_e values for which the closed hysteresis loop is depicted so that the strain accumulation is suppressed. On the other hand, the open hysteresis loop is depicted for $c_e = 0$, returning to the initial subloading surface model.
6. The material constant ϕ_d , which regulates the boundary of the deviatoric hardening and softening, is larger for a looser material with a wider range of deviatoric softening.
7. The normal-yield and subloading surfaces rotates in a wider range for a larger ϕ_r value. They rotate more

Table 1 Physical properties of tested sands, test condition and material parameters in simulations

Figure	Figure 10	Figure 11	Figure 12	Figure 13
Material	Toyoura sand		Tone river sand	Edo river sand
Initial void ratio e_0	0.722	0.718	0.739	0.762
Relative density D_r (%)	76	77	84	88
Confining pressure p_c (kPa)	100		100	160
Stress ratio amplitude q/p_c	± 0.39	± 0.44	± 0.80	± 0.60
Cyclic number	9	7	87	20
$\tilde{\lambda}$	0.004		0.002	0.002
$\tilde{\kappa}$	0.0005		0.001	0.001
G_0 (MPa)	100		10	100
ϑ	0.1		0.1	0.04
ϕ_c ($^\circ$)	30		32	32
ξ	0.005		0.05	0.01
μ_d	5		3	3
ϕ_d ($^\circ$)	25		20	22
a	1		3	3
b	6		18	13
b_r	30		1	50
ϕ_r ($^\circ$)	28		10	29
u_c	3		1	2
\bar{u}	20		46.5	45
u_e	9		0.975	6
\bar{m}	12.5		3.4	3.8
c_e	20		20	40
F_0 (kPa)	350		400	460
c_0 (kPa)	-60 I		-60 I	-60 I

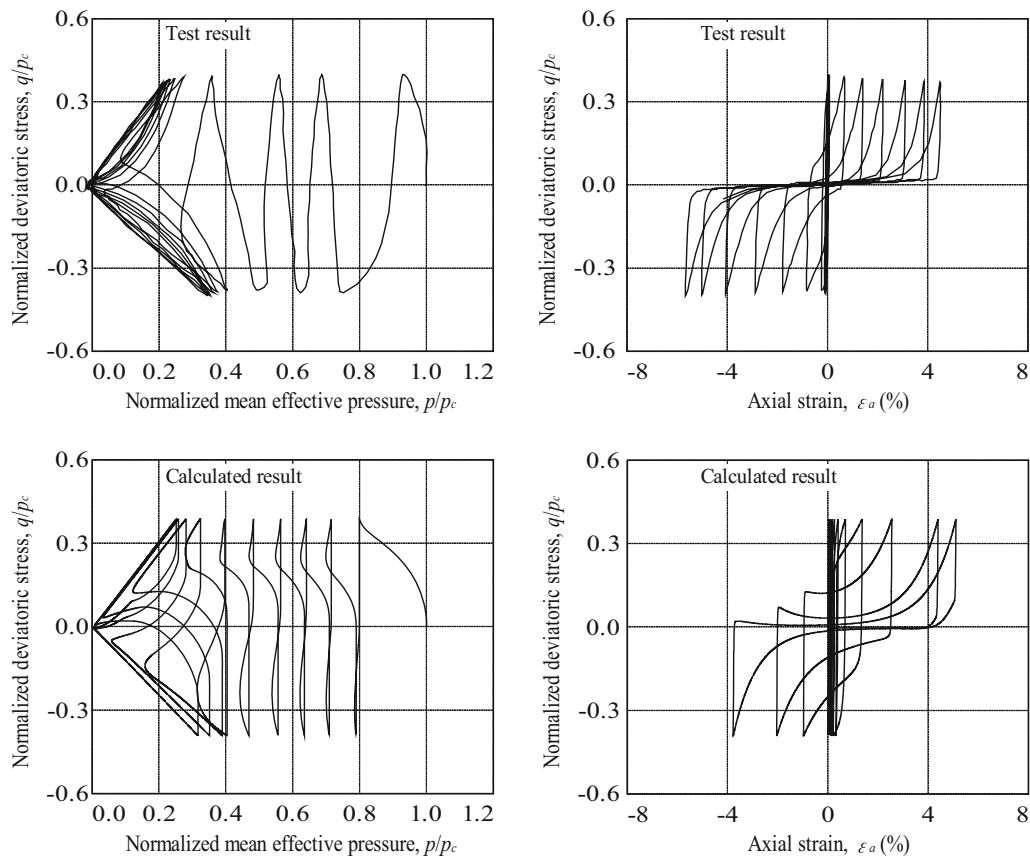


Fig. 10 Simulation of test data for Toyoura sand after Yamada and Noda [68] ($e_0 = 0.722$, $D_r = 76\%$, $p_c = 100$ kPa, stress ratio amplitude = ± 0.39 , cyclic number = 9)

rapidly with the deviatoric plastic strain for larger value of b_r .

The identifications of the material parameters are commented below.

$\tilde{\lambda}$ and $\tilde{\kappa}$ are determined by the curve-fitting to the test data of the isotropic consolidation. Also, they can be known easily by $\tilde{\lambda} = \lambda/(1 + e_0)$, $\tilde{\kappa} = \kappa/(1 + e_0)$, if there is past data of λ and κ in the $e - \log p$ linear relation.

G_0 and n are determined from the stress vs. strain curve under constant pressure inside the yield surface, while we may use $n = 0.5$ usually.

ϕ_c is determined by the inclination of the critical state line in the triaxial compression.

ξ is determined from the pressures in the isotropic compression and the extension test.

μ_d , ϕ_d , a , and b are determined from the undrained triaxial compression test data.

b_r and ϕ_r are determined from the loading and unloading process in the triaxial compression or extension test.

u_c , \bar{u} , u_e , and \bar{m} are inferred from the reloading curve in the drained test.

c_e and χ are determined from the loading–unloading curve, while we may put $\chi = 0.7$ usually.

F_0 can be determined from the initial normal-consolidation stress.

β_0 may be chosen to be $\beta_0 = \mathbf{0}$ in isotropic-consolidated sands but it can be calculated from K_0 -value in a K_0 -consolidation.

c_0 is determined from the stress at which the most elastic behavior is observed.

6 Simulations of test data for cyclic mobility

The constitutive equation of granular materials formulated in the previous sections is applied to the simulations of various test data on the cyclic mobility under the undrained condition with the constant deviatoric stress amplitudes.

All the test data adopted for the simulations were obtained by the cyclic triaxial compression/extension tests with symmetric constant deviatoric stress amplitudes from the isotropic stress state under constant total lateral confining pressures denoted by p_c . The initial isotropy, i.e., $\beta_0 = \mathbf{0}$ is assumed and the typical values $n = 0.5$ and $\chi =$

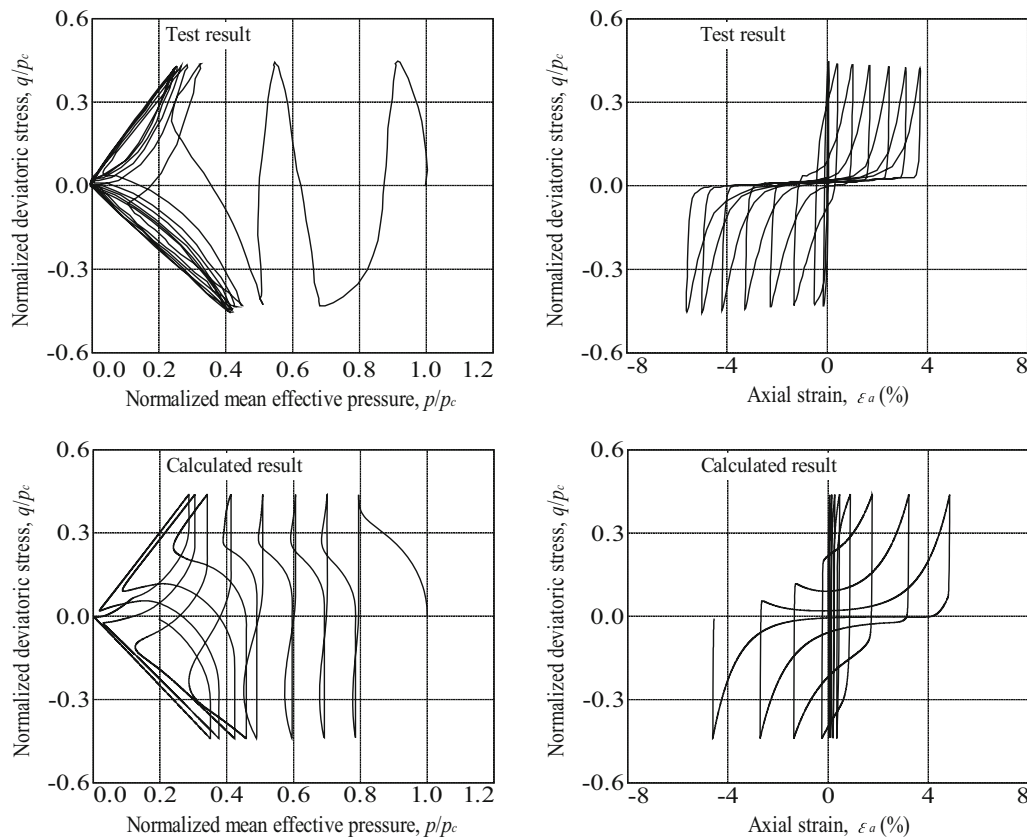


Fig. 11 Simulation of test data for Toyoura sand after Yamada and Noda [68] ($e_0 = 0.718$, $D_r = 77\%$, $p_c = 100$ kPa, stress ratio amplitude = ± 0.44 , cyclic number = 7)

0.7 are used for all test data. Further, the substructure spin is set to be zero because of the triaxial test in which the rotation of material is not induced so that the corotational-time derivative coincides with the material-time derivative. The physical properties of the tested sands, the test condition and the material parameters used in the simulations are listed in Table 1.

The simulation results of the Toyoura sand with the almost same initial void ratio ($e_0 = 0.722$ and 0.718) are shown for Figs. 10 and 11, while the same values of the material parameters are used in these simulations. Further, the simulations of the Tone river sand and the Edo river sand are shown for Figs. 12 and 13, respectively. The butterfly-shaped stress paths after the gradual decreases of the pressure and the deviatoric stress vs. axial strain curves with the increasing amplitudes of axial strain are simulated in these figures. The test results are simulated well in these calculations, where various stress ratios and numbers of loading cycles are applied. The maximum amplitudes of the axial strain are simulated well even for the high loading cycles up to the eighty-seven cycles. However, the decreases of the pressure are more gradual in the simulations than in the test results. The amplitudes of the axial strain in the simulations are smaller at the beginnings and later become

larger compared with the test results in Figs. 10 and 11 for the Toyoura sand. The stress paths in the simulations are warped inversely to the test results in Figs. 12 and 13 for the Tone river sand and the Edo river sand. Further improvements would be required for a more accurate prediction of cyclic mobility.

Since seismic waves cause the cyclic loading at high frequency, the ground deformation occurs under fully or nearly undrained condition during an earthquake. However, when the earthquake ceases, the vertical load due to the weight of the ground and the gravity of the building floating during the earthquake acts to the soil grounds pushing out the pore water, and thus the soil skeleton deforms under the drained conditions. Therefore, it is required to describe both the cyclic mobility under undrained condition and the monotonic loading behavior under drained condition by a unique elastoplastic constitutive equation with a unified set of material parameters. On the other hand, although there are few test results of cyclic mobility and drained tests on specimens of the same material and the same void ratio, the results of drained tests obtained using Toyoura used in the above-mentioned cyclic mobility are provided from the same experimenter to the authors [69]. The simulation result under the drained

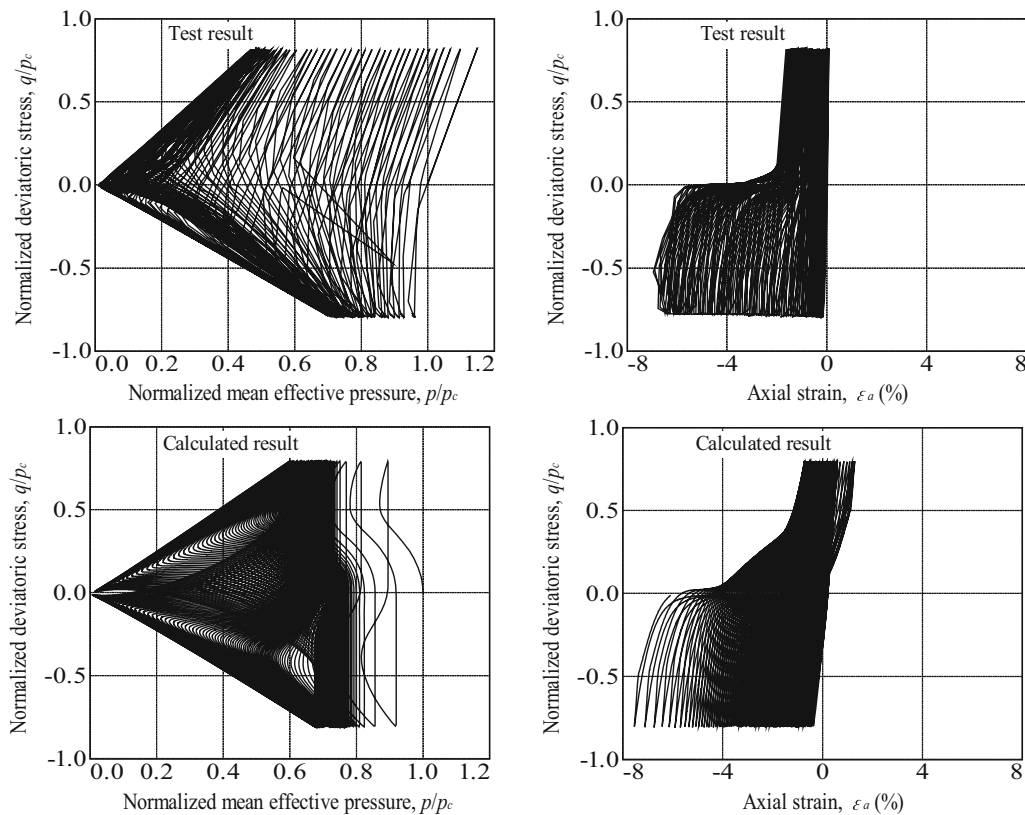


Fig. 12 Simulation of test data for Tone river sand after Kiyota et al [48, 49]. ($e_0 = 0.739$, $D_r = 84\%$, $p_c = 100$ kPa, stress ratio amplitude = ± 0.80 , cyclic number = 87)

condition is shown in Fig. 14, where the tested and the calculated stress and volume strain curves are shown. Therein, since the initial void ratio $e_0 = 0.748$ in the drained test is sufficiently close to $e_0 = 0.722$ and 0.718 in the above-mentioned cyclic mobility test, the material parameters shown in Table 1 (which are used in Figs. 10 and 11) are used. A fairly good simulation to the test result is seen in this figure.

Incidentally, for reference, the simulation to the drained test result with the initial void ratio $e_0 = 0.924$ (looser than in Fig. 14) of the Toyoura sand is shown in Fig. 15, choosing the material parameters as follows:

$$\tilde{\lambda} = 0.02, \tilde{\kappa} = 0.0007, G_0 = 100 \text{ MPa}, \vartheta = 0.1, \phi_c = 30^\circ,$$

$$\xi = 0.005, \mu_d = 1, \phi_d = 18^\circ, a = 1, b = 6, b_r = 1, \phi_r = 5^\circ,$$

$$u_c = 3, \bar{u} = 10, u_e = 9, \bar{m} = 10, c_e = 5,$$

$$F_0 = 170 \text{ kPa}, \mathbf{c}_0 = -60 \mathbf{I}$$

7 Conclusion

In this article, the subloading surface model is elaborated to describe the cyclic mobility observed prior to the liquefaction induced by earthquakes. The results obtained in this study are summarized as follows:

1. The elastoplastic constitutive equation of geomaterials is elaborated by the formulations of
2. the rigorous evolution rule of the elastic-core, which is of crucial importance for the description of cyclic loading behavior,
3. the evolution rule of the rotational hardening, which is of crucial importance for the description of the anisotropic hardening behavior of granular materials,
4. the material functions for the elastic moduli extended to the pressure dependence, the yield condition extended to the negative pressure range, the isotropic hardening rule based on not only the volumetric but also the deviatoric plastic deformations for the accurate description of the cyclic loading behavior up to the cyclic mobility.
5. Then, the validity of the proposed constitutive equation based on the subloading surface model for granular materials was verified by the comparisons with the test

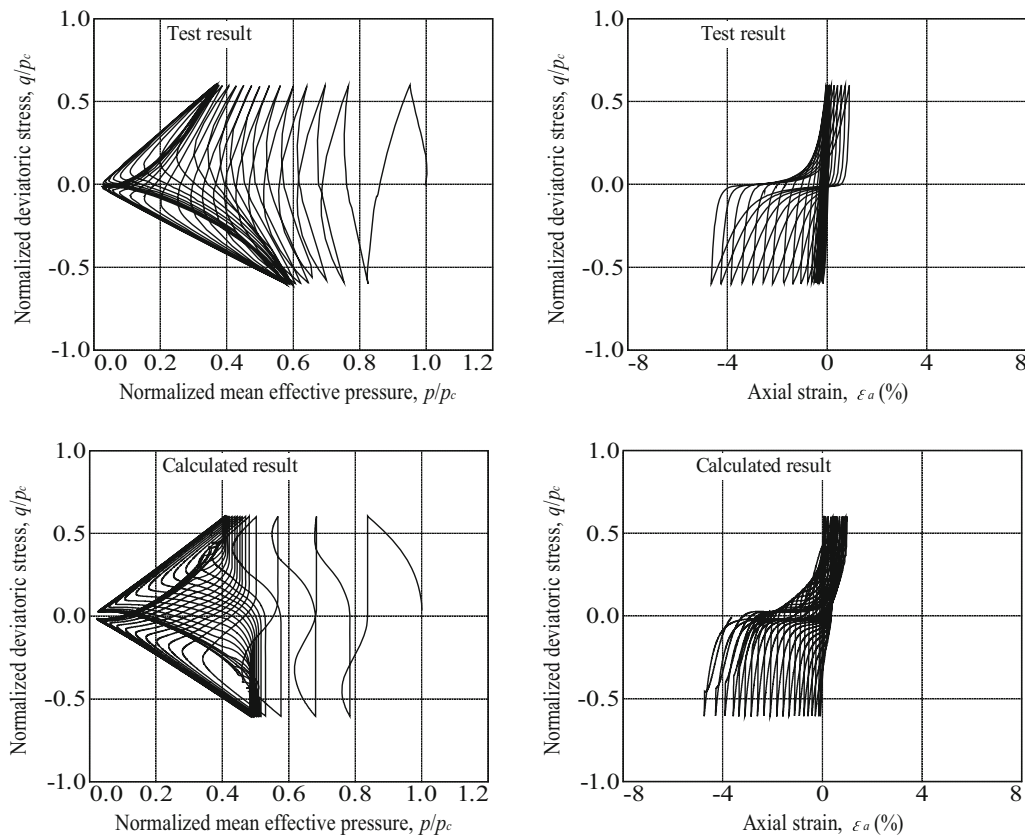


Fig. 13 Simulation of test data for Edo river sand after Kiyota et al [48, 49]. ($e_0 = 0.762$, $D_r = 88\%$, $p_c = 160$ kPa, stress ratio amplitude = ± 0.60 , cyclic number = 20)

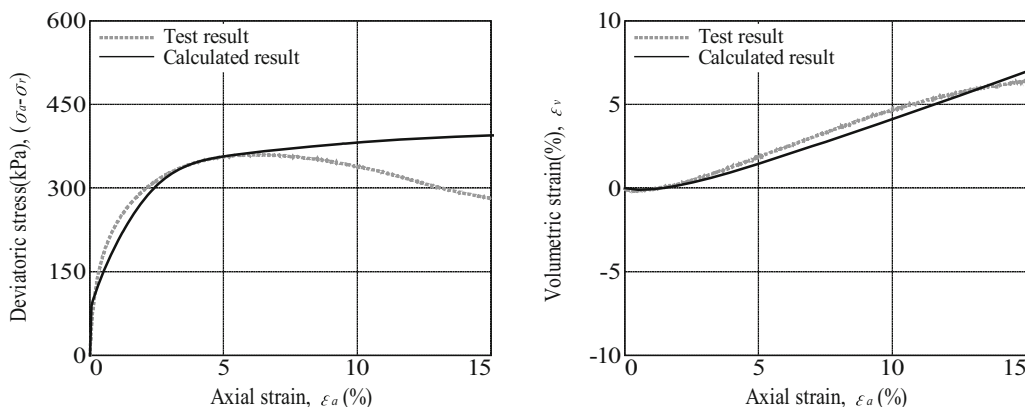


Fig. 14 Simulation of test data for Toyoura sand in drained condition after Yamada [69] ($e_0 = 0.748$, $D_r = 69\%$, $p_c = 98.1$ kPa)

data on the cyclic mobility in three kinds of sands for various numbers of cycles under various stress amplitudes. The butterfly-shaped stress path in the pressure vs. deviatoric stress plane and the S-shaped deviatoric stress vs. axial strain loops were reproduced accurately.

6. The present formulation would provide the substantial foundation for the elastoplastic description of cyclic mobility, although further elaboration is desirable for a highly accurate description of the cyclic mobility.

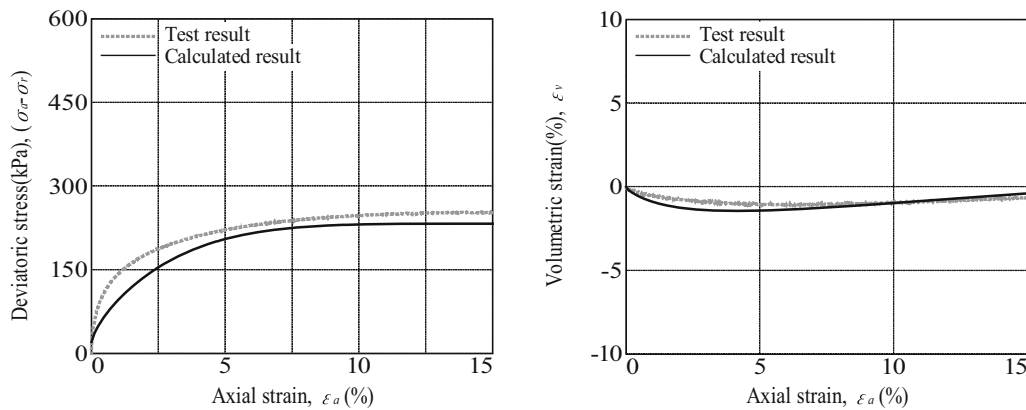


Fig. 15 Simulation of test data for Toyoura sand in drained condition after Yamada [69] ($e_0 = 0.924$, $D_r = 18\%$, $p_c = 98.1$ kPa)

7. The present constitutive model would be applicable to the predictions of the deformation behavior not only during the earthquake but also after it.

Not a few material constants are incorporated in order to describe the cyclic mobility accurately by the elastoplastic constitutive equation because it is to be one of the most complicated mechanical phenomena in the natural world. Further study is required to clarify their definite physical meanings for formulating a more sophisticated constitutive equation of geomaterials.

The formulation falls within framework of the hypoelastic-based plasticity which is limited to the description of the infinitesimal elastic deformation and requires the cumbersome time-integrations of corotational rates of stress and internal variables. Then, it should be extended to the multiplicative hyperelastic-based plasticity (Hashiguchi [29, 33], Hashiguchi and Yamakawa [42]) with a rigorous description of hyperelastic equation of granular materials (Hashiguchi [31]) for further accurate description of cyclic mobility up to the finite deformation.

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