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Preparation of superhydrophobic silver nano coatings with feather-like structures by electroless galvanic deposition

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Superhydrophobic silver nanocoatings with feather-like morphology are fabricated on copper substrates by electroless galvanic deposition. The coating exhibit superhydrophobicity with a contact angle of 156.4° and glide angle of 4° without any further surface modification. Scanning electron microscope (SEM), X-ray diffraction (XRD) and contact angle measurements are used to investigate the morphology, crystal structure and superhydrophobicity, respectively, of the coatings. The coatings exhibit high thermal stability; their water contact angle did not change when the coatings were heated to 200°C for 2 h. The mechanism of superhydrophobicity of the silver coating is discussed based on the work of Amirfazli, Wenzel and Cassie.

silver nano coating, superhydrophobicity, contact angle, composite interface

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The wettability of a solid surface is a very important interfacial behavior. Recently, superhydrophobic surfaces have attracted great interest because of their potential for application in the fields of water repellency, antisticking, self-cleaning and antifouling [1–8]. Wettability is normally characterized by measurement of the contact angle (CA) of a surface. Generally, surfaces with water contact angles larger than 150° and slide angles lower than 10° are called superhydrophobic [9,10].

In nature, many plants and insects have evolved to exhibit perfect superhydrophobicity; for example, the lotus leaf [7,11], water-strider legs [12], the hind wings of the waterman [13], cicadae [14], termite wings and backs of beetles [15,16]. Research has revealed that the dual-scale hierarchical structures of natural surfaces can not only ensure practical superhydrophobic performance but also provide mechanical durability [17]. Inspired by the fascinating functions of such structures in nature [18,19], two kinds of approaches have been proposed to artificially fabricate superhydrophobic surfaces. One is to create micro-/nanostructures on hydrophobic surfaces. Many techniques have been developed to generate rough solid surfaces, including crystallization control [20], phase separation [21,22], template synthesis [23,24], electrochemical deposition [25], and chemical vapor deposition [26]. The other approach is to chemically modify micro-/nanostructured surfaces with materials with low surface free energy [7]. Such materials include fluoroalkylsilane [27], fluoropolymers [28], organic polymers [29], and alkylketene dimers [30]. However, most of the techniques described above require multiple steps, specific substrates, special equipment or harsh chemical treatment, which seriously hinder the practical application of superhydrophobic materials. In this paper, we present a simple method to fabricate large-area superhydrophobic silver coatings on unmodified copper substrate. The structure and superhydrophobicity of the silver coating can be controlled simply by adjusting the initial concentration of silver nitrate and reaction time.

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1 Experimental

1.1 Fabrication of silver coatings on copper substrate

Copper substrates with a size of 2.0 cm×2.0 cm were ultrasonically cleaned in ethanol for 1 h. The substrates were rinsed with distilled water and air-dried at room temperature. The copper substrates were then vertically dipped into silver nitrate solution, held for a Specified period, removed and then heated at 70°C for 30 min. Substrates were cooled befor subsequent measurement and characterization.

1.2 Characterization

Surface morphologies were observed by a scanning electron field-emission microscope (FE-SEM, Hitachi 4800, Japan). The crystal structure of silver powder pealed from the copper substrate were characterized using an X-ray diffractometer (XRD, D/MAX-2400, Rigaku, Japan) with Cu K α radiation. CAs were measured using a CA meter (DSA100, Krüss, Germany) at room temperature. Water droplets with a volume of 5 μ L were placed at five different positions for each sample. Images of water droplets were obtained using a digital camera.

2 Results and discussion

Silver coatings were deposited on copper substrate through the following galvanic exchange reaction:

$$Cu + 2Ag^+ = 2Ag + Cu^{2+}$$

Surface morphologies of the as-prepared silver coatings were characterized by FE-SEM. Under the conditions of a low concentration of silver nitric (0.00625 mol/L) and reaction time of 90 s, silver coatings with various fractal-like structures formed on the copper substrate. The fractal-like structures were surrounded by tiny, almost indistinguishable voids, shown as Figure 1(a). The CA of water on this sample was ~117.4° (inset of Figure 1(a)). When the concentration of Ag⁺ was increased to 0.025 mol/L, silver nanostructures with leaf-like or dendritic morphologies surrounded by coral-like clusters begun to form on the copper substrate. The CA of water on the silver nanoleaves increased to 154.4° (shown in Figure 1(b) and its inset). The magnified image in Figure 1(c) shows the leaves contain vein-like structures. The length of the middle-sized veins in the silver nanoleaves are about 10 µm, while those of the lateral and minor veins are approximately 3-3.5 and 1 µm, respectively. The leaf-like veins are composed of spherical particles with diameters of 200-300 nm that were stringed together by a thread. The thickness of the silver nanocoating fabricated with an Ag⁺ concentration of 0.025 mol/L and reaction time of 90 s is about 35 µm (Figure 1(d)). The composite leaflike structures with micro- and nanoscale components play an important role in the introduction of superhydrophobicity because of their effect on surface roughness. When the reaction time is increased from 90 to 150 s at a fixed silver concentration of 0.025 mol/L, the stability of the coating decreased, and its morphology varied from leaf- to featherlike (Figure 1(e)). The CA of water on the feather-like surface was about 154.5°. A digital photograph of a water droplet on the silver coating shown in Figure 1(e) is presented in Figure 1(f).

The above analysis reveals that both initial Ag^+ concentration and reaction time play important roles in the generation of superhydrophobic silver coating, because Ag^+ concentration and reaction time both affects the morphology and surface roughness of the coating. The silver nanocoatings formed with an Ag^+ concentration of 0.025 mol/L and reaction time of 90–120 s possess suitable surface roughness to exhibit optimal superhydrophobicity, with a water CA of 154.4°–156.4° and slide angle of about 4°. The correlations between water CA and initial Ag^+ concentration and reaction time are shown in Figure 2(a) and (b).

Silver coatings with leaf-like structure were placed in oven for 2 h at 200°C, and then cooled naturally. No apparent changes were observed in the CA of water, revealing that the silver coating with a leaf-like structure has high thermal stability. To test the dynamic behavior of water droplets on the optimal superhydrophobic silver coating, water droplets with a volum of 8 μ L were added dropwise onto the surface using a syringe. Interestingly, water droplets bounced up and down on the coating and then rolled off easily.

The surface morphology of the silver coating is related to both Ag^+ concentration and reaction time. According to ref. [31], the formation of silver nanocoatings follows a process of initial reduction, nucleation, adsorption, growth, braching, and further growth. As a result, the leaf-like nanostructures form through the attachment of silver particles and are controlled by the strength of reducing agent(s) in the system [32].

The leaf-like silver coating was scraped from the copper substrate and analyzed by XRD to determine its crystal structure. The CRD pattern in Figure 3 contains four peaks in the region of 35° -80° that can be indexed as the (111), (200), (220), and (311) planes of face-centered cubic Ag, consistent with JCPDS No. 4-783.

It is known that both surface morphology and surface free energy play important roles in the generation of superhydrophobicity. The mechanism of the transition from hydrophobicity to superhydrophobicity can be described based on the reports of Amirfazli groups [33,34], Wenzel [35] and Cassie et al. [36]. Water droplets on a silver nanocoating with low roughness form a noncomposite state (Wenzel state) where the droplets fill the asperities in the surface through the 3D capillary effect. CA and surface roughness increased as reaction time and Ag^+ concentration increased, causing the state to change from noncomposite (Wenzel state) to



Figure 1 (Color online) SEM images of Ag coatings deposited on copper substrates by galvanic exchange reaction. ((a), (b)) Images of samples fabricated from aqueous solutions with initial Ag⁺ concentration of 0.00625, 0.025 mol/L and reaction time of 90 s, respectively; (c) magnification of Figure 1(b); (d) cross-sectional image of the sample fabricated with an initial Ag⁺ concentration of 0.025 mol/L and reaction time of 90 s; (e) sample fabricated with an initial Ag⁺ concentration of 0.025 mol/L and reaction time of 90 s; (e) sample fabricated with an initial Ag⁺ concentration of 0.025 mol/L and reaction time of 90 s; (e) sample fabricated with an initial Ag⁺ concentration of 0.025 mol/L and reaction time of 90 s; (e) sample fabricated with an initial Ag⁺ concentration of 0.025 mol/L and reaction time of 90 s; (e) sample fabricated with an initial Ag⁺ concentration of 0.025 mol/L and reaction time of 90 s; (e) sample fabricated with an initial Ag⁺ concentration of 0.025 mol/L and reaction time of 90 s; (e) sample fabricated with an initial Ag⁺ concentration of 0.025 mol/L and reaction time of 90 s; (e) sample fabricated with an initial Ag⁺ concentration of 0.025 mol/L and reaction time of 150 s; (f) digital photograph of water on the Ag coating shown in Figure 1(e). Insets are the optical images of water droplets on the surfaces of corresponding silver coatings.



Figure 2 Variation of the CA of water with the concentration of silver nitrate at fixed reaction time of 90 s (a), and reaction time at fixed initial Ag^+ concentration of 0.025 mol/L (b).



Figure 3 XRD pattern of the silver coating scraped from copper substrate.



Figure 4 Transition from Wenzel state (wet contact between the liquid and rough substrate) to the Cassie state (non-wet contact between the liquid and rough substrate).

composite (Cassie state). In the composite state, CA and glide angles do not depend on surface roughness but are dominated by heterogeneity in the solid surface. As a result, water droplets do not fill the grooves, and remain stable because of the trapped air beneath them. Water droplets then began to roll off when the surface was tilted slightly, indicating the coating was superhydrophobic. The transition from the Wenzel to Cassie state is shown in Figure 4.

3 Conclusions

A simple technique to fabricate superhydrophobic silver coatings on copper substrate has been demonstrated. The CA of water on the silver coating could be increased to 156.4° using 0.025 mol/L silver nitrate and reaction time of 120 s without any further surface modification. Water droplets did not readily stick to the coating, and begun to roll off when the surface was tilted slightly. Superhydrophobicity is induced because of the hierarchical structures of these silver nanocoatings.

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- Mumm F, Van Helvoort A T J, Sikorski P. An easy route to superhydrophobic copper based droplet microfluidic systems. ACS Nano, 2009, 3: 2647–2652
- 2 Parkin I P, Palgrave R G. Self-cleaning coatings. J Mater Chem, 2005, 15: 1689–1695
- 3 Sun T L, Feng L, Gao X F, et al. Bioinspired surfaces with special wettability. Acc Chem Res, 2005, 38: 644–652
- 4 Xu Q F, Wang J N. A superhydrophobic coating on aluminium foil with an anti-corrosive property. New J Chem, 2009, 33: 734–738
- 5 Tuteja A, Choi W, Ma M L, et al. Designing superoleophobic surfaces. Science, 2007, 318: 1618–1622
- 6 Erbil H Y, Demirel A L, Avci Y, et al. Transformation of a simple plastic into a super-hydrophobic surface. Science, 2003, 299: 1377– 1380
- 7 Feng L, Li S, Li Y, et al. Super-hydrophobic surfaces: From natural to artificial. Adv Mater, 2002, 14: 1857–1860
- 8 Nakajima A, Hashimoto K, Watanabe T. Recent studies on super-hydrophobic films. Mon Chem, 2001, 132: 31–41
- 9 Oner D, McCarthy T J. Ultrahydrophobic surfaces. Effect of topography length scales on wettability. Langmuir, 2000, 16: 7777–7782
- 10 Cao L, Price T P, Weiss M, et al. Super water- and oil-repellent surfaces on intrinsically hydrophilic and oleophilic porous silicon films. Langmuir, 2008, 24: 1640–1643
- 11 Barthlott W, Neinhuis C. Purity of the sacred lotus, or escape from contamination in biological surfaces. Planta, 1997, 202: 1–8
- 12 Gao X F, Jiang L. Water-repellent legs of water striders. Nature, 2004, 432: 36
- 13 Wang Y S, Shi Y L, Feng X J, et al. Investigation of superhydrophobicity on water boatman's hind wings (in Chinese). Chin Sci Bull (Chin Ver), 2012, 57: 1227–1230
- 14 Lee W, Jin M K, Yoo W C, et al. Nanostructuring of a polymeric substrate with well-defined nanometer-scale topography and tailored surface wettability. Langmuir, 2004, 20: 7665–7669
- 15 Andrew R P, Chris R L. Water capture by a desert beetle. Nature, 2001, 414: 33–34
- 16 Watson G S, Cribb B W, Watson J A. How micro/nanoarchitecture facilitates anti-Wetting: An elegant hierarchical design on the termite wing. ACS Nano, 2010, 4: 129–136
- 17 Li W, Amirfazhi A. Hierarchical structures for natural superhydrophobic surfaces. Soft Matter, 2008, 4: 462–466
- 18 Liu K S, Yao X, Jiang L. Recent developments in bio-inspired special wettability. Chem Soc Rev, 2010, 39: 3240–3255
- 19 Liu K S, Jiang L. Bio-inspired self-cleaning surfaces. Ann Rev Mater Res, 2012, 42: 231–263
- 20 Shibuichi S, Onda T, Satoh N, et al. Super-water-repellent fractal surfaces. Langmuir, 1996, 12: 2125–2127
- 21 Xie Q D, Fan G Q, Zhao N, et al. Facile creation of a bionic superhydrophobic block copolymer surface. Adv Mater, 2004, 16: 1830– 1833
- 22 Han J T, Lee D H, Ryu C Y, et al. Fabrication of superhydrophobic surface from a supramolecular. Organosilane with quadruple hydrogen bonding. J Am Chem Soc, 2004, 126: 4796–4797
- 23 Sun M H, Luo C X, Xu L P, et al. Artificial lotus leaf by nanocasting. Langmuir, 2005, 21: 8978–8981
- 24 Feng L, Song Y L, Zhai J, et al. Creation of super-hydrophobic surface from amphiphilic polymer. Angew Chem Int Ed, 2003, 42: 800–802
- 25 Shi F, Wang Z Q, Zhang X, et al. Combining layer-by-layer assembly with electrodeposition of silver aggregates for fabricating superhydrophobic surfaces. Adv Mater, 2005, 17: 1005–1009
- 26 Lau K K S, Bico J, Teo K B K, et al. Superhydrophobic carbon nanotube forests. Nano Lett, 2003, 3: 1701–1705
- 27 Nakajima A, Fujishima A, Hashimoto K, et al. Preparation of transparent superhydrophobic boehmite and silica films by sublimation of aluminum acetylacetonate. Adv Mater, 1999, 11: 1365–1368
- 28 Morra M, Occhiello E, Garbassi F. Contact angle hysteresis in oxygen plasma treated poly(tetrafluoroethylene). Langmuir, 1989, 5: 872– 876
- 29 Youngblood J P, McCarthy T J. Ultrahydrophobic polymer surfaces

prepared by simultaneous ablation of polypropylene and sputtering of poly(tetrafluoroethylene) using radio frequency plasma. Macromolecules, 1999, 32: 6800–6806

- 30 Shibuichi S, Onda T, Satoh N, et al. Super water-repellent surfaces resulting from fractal surfaces. J Phys Chem, 1996, 100: 19512– 19517
- 31 Xu X H, Zhang Z Z, Yang J. Fabrication of biomimetic superhydrophobic surface on engineering materials by a simple electroless galvanic deposition method. Langmuir, 2010, 26: 3654–3658
- 32 Mdluli P S, Revaprasadu N. Time dependant evolution of silver

nanodendrites. Mater Lett, 2009, 63: 447-450

- 33 Li W, Amirfazli A. Microtextured superhydrophobic surfaces: A thermodynamic analysis. Adv Colloid Interface Sci, 2007, 132: 51–68
- 34 Li W, Amirfazli A. A thermodynamic approach for determining the contact angle hysteresis for superhydrophobic surfaces. J Colloid Interface Sci, 2005, 292: 195–201
- 35 Wenzel R N. Resistance of solid surfaces to wetting by water. Ind Eng Chem, 1936, 28: 988–994
- 36 Cassie A B D, Trans S B. Wettability of porous surfaces. Faraday Soc, 1944, 40: 546–551
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