

Etching mechanism of barium strontium titanate (BST) thin films in CHF₃/Ar plasma

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Received February 22, 2011; accepted April 8, 2011

Reactive ion etching was used to etch barium strontium titanate thin films in a CHF₃/Ar plasma. BST surfaces before and after etching were analyzed by X-ray photoelectron spectroscopy to investigate the reaction ion etching mechanism, and chemical reactions had occurred between the F plasma and the Ba, Sr and Ti metal species. Fluorides of these metals were formed and remained on the surface during the etching process. Ti was almost completely removed because the TiF₄ by-product is volatile. Minor quantities of Ti–F could still be detected by narrow scan X-ray photoelectron spectra, and Ti–F was thought to be present in the form of a metal-oxy-fluoride. These species were investigated from O1s spectra, and a fluoride-rich surface was formed during etching. BaF₂ and SrF₂ residues were difficult to remove because of their high boiling point. The etching rate was limited to 12.86 nm/min. C–F polymers were not found on the surface, indicating that the removal of BaF₂ and SrF₂ was important for further etching. A 1-min Ar/15 plasma physical sputtering was carried out for every 4 min of surface etching, which effectively removed remaining surface residue. Sequential chemical reaction and sputtered etching is an effective etching method for barium strontium titanate films.

barium strontium titanate, reaction ion etching, X-ray photoelectron spectroscopy, plasma

Citation: Dai L P, Wang S Y, Shu P, et al. Etching mechanism of barium strontium titanate (BST) thin films in CHF₃/Ar plasma. Chinese Sci Bull, 2011, 56: 2267–2271, doi: 10.1007/s11434-011-4561-5

With the rapid development of electronics and information technology, there is a great demand for high K dielectric materials. An example is ferroelectric high-K materials [1–4] required for many microwave applications [5]. Ferroelectric thin films possessing high permittivity are of great interest for application to high-K dynamic random access memory (DRAM) cells. Barium strontium titanate (BST) thin film ferroelectric materials are promising as capacitor dielectric components in future high density DRAM devices, because of their high dielectric constant, low leakage current, low temperature coefficient in electrical processes, small dielectric loss and lack of fatigue/aging [6–8]. BST can potentially provide significant improvements in device performance; however, several problems need resolving to realize

highly integrated DRAMs involving BST thin films. To efficiently integrate BST thin films into Si-based circuitry, an etching process for micro-patterning technology must be developed. The etching mechanism includes two major processes: ion-assisted etching reaction; and sputtering process to remove the reaction by-products. Thus, identifying by-product compositions is important for optimizing etching processes. There are several reported studies on the etching of BST thin films using chlorine and fluorine-based plasma chemistry [9–17]. There are also several reports investigating the properties of CF₄/Ar plasmas both experimentally and theoretically. Unfortunately, the etching mechanism of BST films is still not clear, perhaps because BST thin films are a relatively new material to the microelectronics technology. Thus, most work has focused on technological aspects, with etching mechanisms having received little attention. Investigating the BST thin film dry-etching mechanism

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is important for improving etching performance.

In the current study, reactive ion etching (RIE) was carried out on BST thin films using CHF_3/Ar plasma. The etched surface was analyzed by X-ray photoelectron spectroscopy (XPS) to determine the composition and chemical states of all elements present. The relative atomic percentage of compositions was determined by narrow scan XPS spectra and sensitivity factors. We have further clarified the chemical reaction of BST films with F plasma, and measured the etching rate. From the etching mechanism we offer details of an optimized etching process.

1 Materials and methods

BST thin films were deposited on Si substrates by radio frequency magnetron sputtering. The target was a BST ceramic, radio frequency power was 100 W, substrate temperature was 500°C , and the chamber O_2 and Ar pressure during deposition was 4 Pa. Deposition was carried out for 3 h, followed by annealing at 600°C for 30 min. Two BST films were etched by RIE in a CHF_3/Ar plasma for 4 min. The first film was bare and the second carried a photoresist mask to allow the etching rate to be ascertained. The RIE power was 200 W, CHF_3/Ar flow rate was 50/10 sccm, and the pressure inside the reaction chamber held constant at 20×10^{-3} Torr. The Ar plasma post bombardment treatment RIE power was 200 W.

The etch rate of BST films was measured using a surface profiler (DEKTAK150). The surface chemical compositions of as-deposited and etched BST films were analyzed using a XSAM800 instrument under ultrahigh vacuum (2×10^{-7} Pa) conditions using XPS (VG ESCALAB220I-XL), with a monochromated $\text{MgK}\alpha$ source ($h\nu=1253.6$ eV). Scans for the elements Ba, Sr, Ti, O and F used an X-ray beam with a ~ 1 mm diameter. All samples were scanned after 30 s of sputtering to remove contamination effects from the film surface before XPS analysis.

2 Results and discussion

Figure 1 shows a typical XPS surface survey scan of a BST thin film after RIE in CHF_3/Ar plasma, at 20×10^{-3} Torr chamber pressure. Ba, Sr, Ti, O and F were detected in the film surface after etching, and elemental C was also present. The C_{1s} peak centered at ~ 284.8 eV was attributed to C–C/C–H bonding. No high energy shoulder peaks of C_{1s} –F were observed which is supported by F_{1s} spectra presented later in this report. C–C/C–H was thought to originate from instrumental contamination. During the etching process, C plasma may have reacted with O to produce CO_2 which desorbed from the BST surface. Compared with that of BST thin films specimens before etching, an increase of the F_{1s} peak was observed in the wide scan spectra of the etched

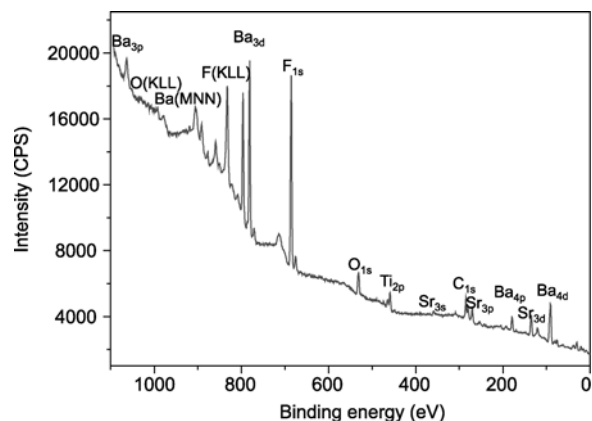


Figure 1 XPS wide scan spectra of a BST film after etching.

film. This indicated that fluorine-containing products accumulated on the etched surface.

XPS narrow scan spectra were recorded to determine the surface composition and chemical states of BST films before and after etching, and spectra for the Ba_{3d} , Sr_{3d} , Ti_{2p} , O_{1s} and F_{1s} spectral regions are shown in Figure 2. From these spectra, the chemical reactions occurring between the CHF_3 plasma and Ba, Sr, Ti and O elements were investigated. The Ar plasma acted solely as physical bombardment. The variations in Ba_{3d} , Sr_{3d} , Ti_{2p} , O_{1s} and F_{1s} XPS peaks are apparent in Figure 2. In Figure 2(a), the $\text{Ba}_{3d_{3/2}}$ peaks at 796.14 and 794.34 eV binding energy can be resolved into Ba–O and Ba–F components, respectively. $\text{Ba}_{3d_{5/2}}$ peaks at 781.07 and 778.88 eV binding energy could also be resolved into Ba–F and Ba–O components, respectively. The predominant Ba_{3d} peaks of etched BST shifted towards higher binding energy by about 1.80 and 2.19 eV, upon Ba–O bond being broken and reacted with the F plasma to form Ba–F. Figure 2(b) shows that the Sr_{3d} spectral region could be resolved into Sr–F and Sr–O components. Peaks at 135.96 and 134.41 eV binding energy corresponded to $\text{Sr}_{3d_{3/2}}$ and $\text{Sr}_{3d_{5/2}}$ of Sr–F, respectively, and peaks of $\text{Sr}_{3d_{3/2}}$ –O and $\text{Sr}_{3d_{5/2}}$ –O were observed at 134.62 and 133.09 eV, respectively. Compared with the spectral line of as-deposited BST, that of the etched BST showed a high energy shift of about 1.52 eV upon forming Sr–F bonds. As shown in Figure 2(c), Ti_{2p} spectra could also be resolved into Ti–F and Ti–O components. Peaks at 464.60 and 458.87 eV binding energy corresponded to $\text{Ti}_{2p_{1/2}}$ and $\text{Ti}_{2p_{3/2}}$ of Ti–F, respectively, and peaks at 463.57 and 457.84 eV binding energy corresponded to $\text{Ti}_{2p_{1/2}}$ –O and $\text{Ti}_{2p_{3/2}}$ –O, respectively. This indicated that the fluoride of Ti species was still present as a surface residue despite its high volatility. The Ti–F peak intensity became very weak and reflected a sharply decreasing Ti content. Figure 2(d) shows the O_{1s} spectral region, which shows the superposition of peaks from oxides of Ba, Sr and Ti. The O_{1s} peak at ~ 529.45 eV before etching shifted to ~ 531.73 eV following etching,

representing a binding energy shift about ~ 2.28 eV. Such chemical shift behavior may be influenced by two factors: the variation in oxide (Ba–O, Sr–O and Ti–O) relative percentages through the chemical reaction; and M–O–F being formed. Upon the incorporation of fluorine and its replacement of oxygen, M–O–F was formed and oxygen electrons became more tightly bound, thus having a higher binding energy than those of metal oxides. It is not surprising that Ti–F was still detected in the film despite its high volatility,

because it existed in the form of Ti–F–O. The intensity of O_{1s} peak decreased significantly and reflected the decreasing O element percentage during the etching process. Figure 2(e) shows the F_{1s} spectral region, and the peak at 694.62 eV binding energy corresponded to F_{1s} –C species of CHF_3 . The peak at 685.56 eV binding energy represented the superposition of those from F_{1s} –Ba, F_{1s} –Sr and F_{1s} –Ti species. No high energy shoulder peaks at 686.21–689.08 eV were apparent, indicating that no polymer containing C–F remained

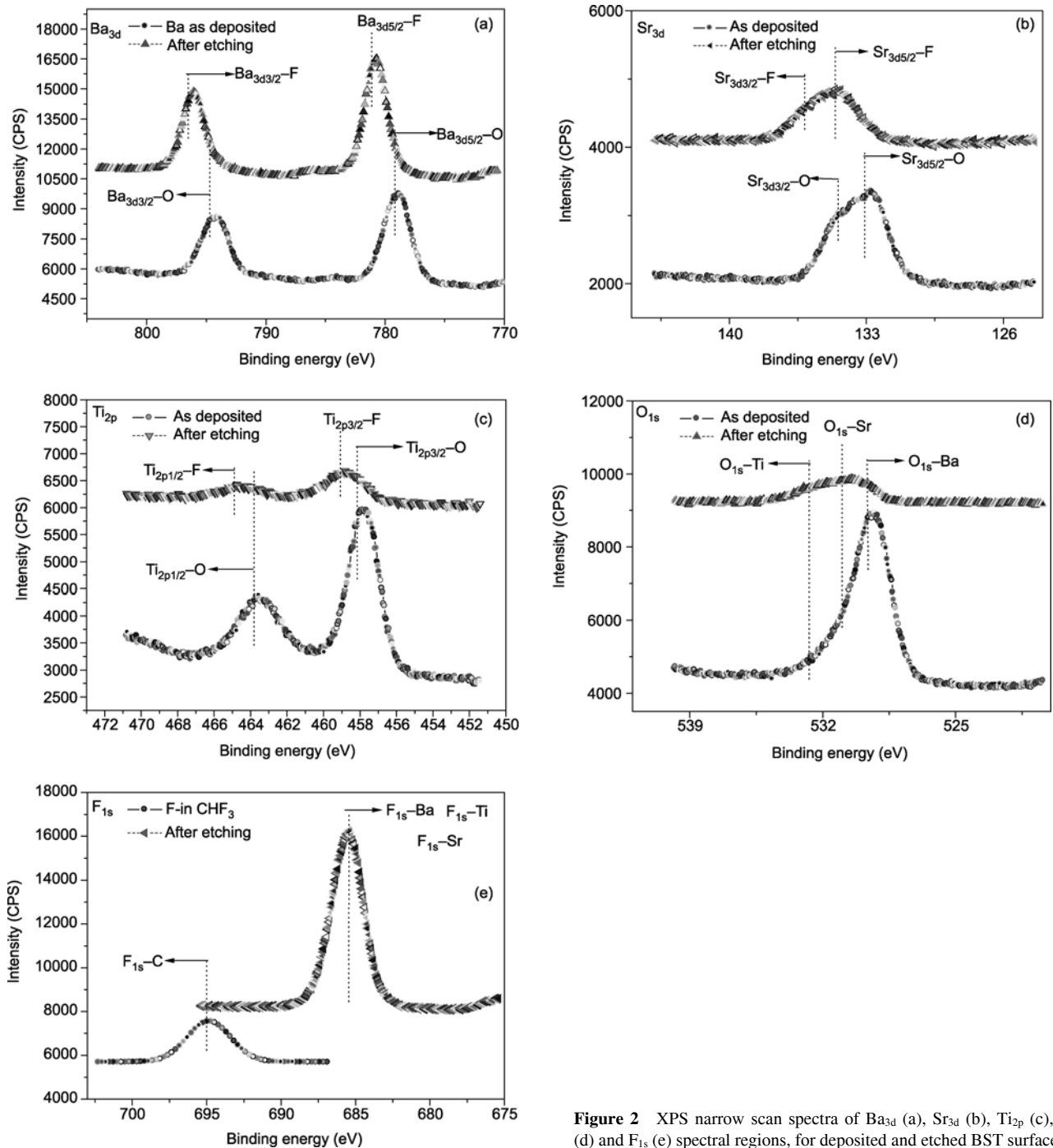


Figure 2 XPS narrow scan spectra of Ba_{3d} (a), Sr_{3d} (b), Ti_{2p} (c), O_{1s} (d) and F_{1s} (e) spectral regions, for deposited and etched BST surfaces.

on the BST film surface. From the above XPS analysis, the chemical reactions during etching can be summarized as follows: Ba–O, Sr–O and Ti–O bonds were broken during CHF₃/Ar plasma bombardment, and Ba, Sr and Ti atoms reacted with the F plasma to form fluorine-containing products, some of which were removed from the surface while other less volatile species remained. The C–F bond was also broken and C plasma then reacted with O to produce CO₂, which was volatile and desorbed from the surface. The C–H bonds are thought to have also been broken, with the H plasma then reacting with O or F to produce volatile H₂O or HF.

There were three components (BaO, SrO, TiO₂) in the BST film suspension, thus it is interesting to compare the relative compositions of as-deposited and etched BST surfaces. To investigate the role of CHF₃/Ar plasma and to further study the etching mechanism, the relative atomic percentages of the as-deposited and etched surfaces were calculated from integrated XPS narrow scan peak intensities, and are shown in Table 1. While the XPS compositions measured in this work do not necessarily represent the true compositions, the relative changes induced by ion bombardment should be accurate and provide clues about relevant surface reactions. The relative atomic percentage of F in the BST after etching was high (71.34%) which indicated that a fluoride-rich surface was formed during etching. Thus, chemical reactions occurred between the metal oxides and F plasma. Ba–O, Sr–O and Ti–O reacted to form Ba–F, Sr–F and Ti–F, respectively, and the relative O atomic percentage significantly decreased. The Ti atomic percentage significantly decreased because volatile TiF₄ (bp 284°C) was formed and could be easily removed from the BST surface. While some Ti–F was still detected in the narrow scan spectra, it was likely to exist in the form of Ti–O–F. The atomic percentages of Ba and Sr showed minor decreases upon etching, because it was difficult to remove the BaF₂ and SrF₂ by-products with high boiling points of 2260°C and 2489°C, respectively. The etch rate was calculated to be 12.86 nm/min, thus removal of BaF₂ and SrF₂ is important for further etching. Based on the above analysis, the RIE mechanism of BST in CHF₃/Ar plasma has been elucidated. However, the need for approaches to remove BaF₂ and SrF₂ still remain. Physical sputtering is likely to be effective for their removal, given their very high boiling points. A 1-min

Ar/15 plasma post bombardment treatment was carried out after every 4 min of surface etching. The relative atomic percentages of the post-treated surface are also shown in Table 1, and are similar to that of the as-deposited surface. This demonstrates that BaF₂ and SrF₂ remnants had been etched from the surface, and that Ar plasma post bombardment treatment is an effective supplement for the RIE of BST films.

3 Conclusions

BST thin films were etched in CHF₃/Ar plasma using RIE. The elemental composition and chemical states of as-deposited and etched BST films were investigated by XPS to determine the etching mechanism. Etching led to the accumulation of fluorine-containing by-products on the BST surface, with Ti fluorides being easily removed. Fluorides of Ba and Sr remained on the surface as these residues were hard to remove. A 1-min Ar/15 plasma post bombardment treatment was carried out on the etched surface, and was shown to be effective for removing these residues. Sequential chemical reaction and sputter etching is an effective etching procedure for BST films.

This work was supported by the University of Electronic Science and Technology of China (Y02002010301045) and the Foundation of State Key Laboratory of Electronic Thin Films and Integrated Devices (KFJJ200909).

Table 1 BST surface elemental compositions for as-deposited and etched surfaces

Condition	Ba (%)	Sr (%)	Ti (%)	O (%)	F (%)
As deposited	8.44	9.16	24.99	57.41	0
After etching	6.02	5.72	4.19	12.72	71.34
Ar post sputtering	8.69	9.23	24.50	50.34	7.24

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