



# Heat capacity, thermal expansion, and thermal diffusivity of NaUO<sub>2</sub>BO<sub>3</sub>

Karin Popa<sup>1</sup> · Ondrej Beneš<sup>1</sup> · Dragoş Staicu<sup>1</sup> · Jean-Christophe Griveau<sup>1</sup> · Eric Colineau<sup>1</sup> · Alice Seibert<sup>1</sup> · Jean-Yves Colle<sup>1</sup> · Sarah Stohr<sup>1</sup> · Philippe E. Raison<sup>1</sup> · Joseph Somers<sup>1</sup> · Rudy J. M. Konings<sup>1</sup>

Received: 21 July 2017 / Accepted: 11 December 2017 / Published online: 20 December 2017  
© The Author(s) 2017. This article is an open access publication

## Abstract

In the present studies, the thermal behaviour of NaUO<sub>2</sub>BO<sub>3</sub> has been investigated. This compound is a potential product of interaction between the coolant (Na), control rods (B<sub>4</sub>C), and the oxide fuel, which could form under accidental conditions in sodium-cooled fast reactors. The thermal expansion, the heat capacity, and thermal diffusivity of NaUO<sub>2</sub>BO<sub>3</sub> have been measured. The thermal conductivity of the material is derived from these results and presented here for the first time.

**Keywords** Sodium uranyl borate · Solid state reactions · Thermal expansion · Thermodynamic properties · Calorimetry · Thermal analysis

## Introduction

Sodium uranyl borate NaUO<sub>2</sub>BO<sub>3</sub> (or NaUBO<sub>5</sub>) is the only anhydrous borate that forms in the Na–U–B–O phase diagram [1–3]. This compound has been reported for the first time by Hoekstra [4], and its crystal structure was determined by Gasparin [5] from X-ray diffraction data measurements on single crystal. NaUO<sub>2</sub>BO<sub>3</sub> crystallises in an orthorhombic unit cell in the space group *P*cam (*N*<sup>o</sup>57). In this structure, uranium atoms are sevenfold coordinated, the boron atoms have three coplanar surrounding oxygen and sodium atoms are sixfold coordinated and linked together by three oxygen atoms along the *c* axis as shown in Fig. 1.

An extensive thermodynamic characterisation has been performed by Chernurukov et al. [6, 7]. The latter authors measured the enthalpy of formation and the low-temperature heat capacity, from which the Gibbs energy was derived as  $\Delta_f G^\circ(298.15\text{ K}) = -(2122.5 \pm 5.0)\text{ kJ mol}^{-1}$ . This results in a reaction energy from the oxides of

$\Delta_r G^\circ(298.15\text{ K}) = -(190.5 \pm 5.1)\text{ kJ mol}^{-1}$ , indicating that the compound can form under oxidising conditions when the oxides of sodium, uranium, and boron are mixed together. This could occur for instance in nuclear borosilicate waste glasses (containing about 10 wt% sodium oxide) or in the debris of damaged nuclear reactor cores containing oxidised fuel and B<sub>4</sub>C control rod material, such as in Fukushima case (reaction with the sodium ions from the cooling sea water).

In the present study, we have extended the thermodynamic characterisation of this compound to higher temperatures and determined the thermal expansion, the enthalpy increment, and the thermal diffusivity, from which the heat capacity and thermal conductivity are derived. In addition, we re-measured the low-temperature heat capacity in order to verify the change in the slope observed by Karyakin et al. [6] in their measurements.

## Materials and methods

### Synthesis and material processing

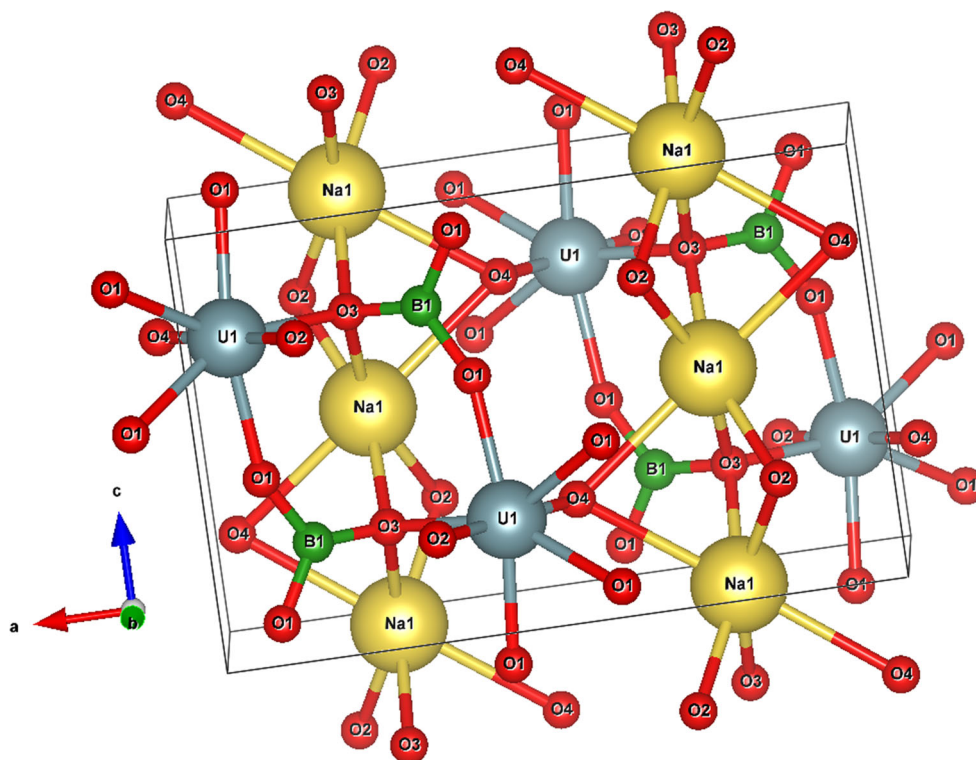
NaUO<sub>2</sub>BO<sub>3</sub> was produced using a simplified solid state reaction procedure. Thus, stoichiometric amounts of commercial NaHCO<sub>3</sub> (Alfa Aesar, 99.7%), H<sub>3</sub>BO<sub>3</sub> (Sigma-Aldrich, 99.999%), and UO<sub>2.10</sub> (COGEMA powder, stoichiometry calculated from the value of the lattice

**Electronic supplementary material** The online version of this article (<https://doi.org/10.1007/s10973-017-6923-y>) contains supplementary material, which is available to authorized users.

✉ Karin Popa  
karin.popa@ec.europa.eu

<sup>1</sup> European Commission, Joint Research Centre,  
P.O. Box 2340, 76125 Karlsruhe, Germany

**Fig. 1** Sketch of the crystal structure of  $\text{NaUO}_2\text{BO}_3$  (atomic positions from Gasperin [5])



parameter) were reacted for 5 h at 1173 K under an air atmosphere. Pellets of 3.5 mm diameter and 1–2 mm in height were produced by pressing and re-heated at 1173 K under air for another 5 h.

### XRD analysis

$\text{NaUO}_2\text{BO}_3$  was characterised at room temperature by X-ray powder diffraction (XRD) using a Bruker D8 diffractometer mounted in a Bragg–Brentano configuration with a curved Ge (1, 1, 1) monochromator, a ceramic copper tube (40 kV, 40 mA) and equipped with a LynxEye position sensitive detector. The data were collected by step scanning in the angle range  $10^\circ \leq 2\theta \leq 120^\circ$  at a  $2\theta$  step size of  $0.0092^\circ$ . For the measurements, the powder was deposited on a silicon wafer to minimise the background and dispersed on the surface with several drops of isopropanol. The structural refinement was performed using the Fullprof2k suite [8]. The shape of the peaks was described by a Pseudo-Voigt function, and the background was fitted by a linear interpolation between a set of about 50 background points.

The thermal expansion of  $\text{NaUO}_2\text{BO}_3$  was measured by high-temperature X-ray diffraction. The data were collected on a second Bruker D8 X-ray diffractometer mounted with a curved Ge (1, 1, 1) monochromator, a copper ceramic X-ray tube (40 kV, 40 mA), a LynxEye position sensitive detector and equipped with an Anton

Paar HTK 2000 chamber. Measurements were conducted up to 1073 K under helium, in the angle range  $16^\circ \leq 2\theta \leq 90^\circ$  with a  $2\theta$  step size of  $0.017^\circ$ .

### Raman and IR spectroscopy

$\text{NaUO}_2\text{BO}_3$  was characterised in the temperature range from 6 to 290 K by Raman spectroscopy using a Jobin–Yvon® T 64000 equipped with a macro-inlet. To obtain the Raman spectra, a laser excitation source was used to illuminate the pressed powder sample fixed with indium inside the cold finger of a closed cycle He cryostat. The scattered radiation from the surface was detected with a low-noise  $\text{LN}_2$ -cooled CCD detector after separation of the Raman signal from the Rayleigh component. The excitation source was a  $\text{Kr}^+$  Coherent® cw laser radiating at 647 nm with a power of 50 mW. Spectra were obtained with an acquisition time of 10 s.

FT-IR spectra were measured using Bruker Alpha spectrometer employing the attenuated total reflectance (ATR) technique with a resolution of  $0.01 \text{ cm}^{-1}$  in a range  $375\text{--}4000 \text{ cm}^{-1}$ . No special preparation for powder samples was necessary for this technique.

### DTA/TG measurements

The thermal behaviour was investigated using a Netzsch STA 449C DTA/TG, using an alumina crucible and

different atmospheres (Ar and air). The temperature was controlled by a Pt-PtRh (10%) thermocouple. The applied heating and cooling rates were 10 K min<sup>-1</sup>.

### Heat capacity measurements

The low-temperature heat capacity of NaUO<sub>2</sub>BO<sub>3</sub> was measured on a pellet of 51.9 mg from 293.4 to 1.82 K at 0 T with a PPMS-9T instrument (quantum design) using the hybrid adiabatic relaxation method, in the absence and the presence of the magnetic field. The sample was fixed on the sapphire platform by a small amount of thermal conductive grease (Apiezon-N). The heat capacity contributions of the puck and of the layer of grease were determined separately through the addenda protocol determination. A complete description of the technique and the method has been reported by Lashley et al. [9], whereas details of the instrument used in this work can be found in Javorský et al. [10]. Based on comparison with standard materials and experience for other compounds, we estimate the uncertainty of the measurements to be better than 3%, depending on the temperature range.

Solid pieces of 53.7–62.9 mg were further used to measure the enthalpy increments using a Setaram multi-detector high-temperature calorimeter (MDHTC-96) using a drop detector. For more details about the technique, we refer to our previous studies [11, 12]. The measurements were carried out under an argon atmosphere. The temperature range of the experiment was from 482.5 to 1190.5 K. Each isothermal run consisted of three drops of analysed material. Before and after each sample, a reference material (platinum ingots of 99.95 at.% purity) was dropped to determine the sensitivity (signal vs. heat ratio) of the detector. The drops were separated by time intervals of 20 min, long enough to re-stabilise the monitored heat flow signal. All evaluations of the background subtraction and peak integration were done using commercially available software for data processing. The reported temperatures were corrected (ITS-90) in accordance with the calibration curve obtained prior to measurement using several high-purity standard metals (Sn, Pb, Zn, Al, Ag, Ni) with various melting temperatures in order to cover the whole temperature range of the measurement.

### Thermal diffusivity measurements

The thermal diffusivity measurements were carried out using a laser flash device, designed and constructed in house [13]. The curves were measured with pyrometers. The samples were heated at the measurement temperatures in a high-frequency furnace, in the temperature range from 500 to about 1130 K. The samples were disc fragments

with a thickness of about 1 mm. The faces were checked to ensure that they were plane and parallel, without defects.

## Results and discussions

### Structural characterisation and thermal expansion

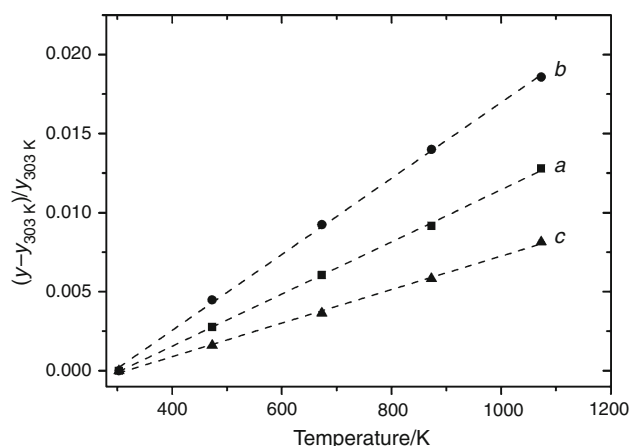
The synthesis resulted in pure phase NaUO<sub>2</sub>BO<sub>3</sub> (space group *P*cam, *Z* = 4), as confirmed by room temperature XRD and vibrational spectroscopy. The results of our refinement are in good agreement with the previous literature reports. Table 1 summarises the main experimental crystallographic data of the present studies.

The DTA/TG and measurements performed under air and argon indicated that NaUO<sub>2</sub>BO<sub>3</sub> is stable up to 1470 K (supplementary material S1). However, a slight change in colour may be observed at about 1200 K, indicating that a transformation is occurring already at this temperature. Moreover, the HTXRD measurements performed under inert atmosphere indicated the appearance of U<sub>3</sub>O<sub>8</sub> as a secondary phase at 1273 K. This is the reason of limiting our heat capacity and thermal expansion measurement around 1200 K.

The relative variation of the cell parameters of NaUO<sub>2</sub>BO<sub>3</sub> with temperature is presented in Fig. 2. The

**Table 1** Unit cell parameters of NaUO<sub>2</sub>BO<sub>3</sub>

	<i>a</i> /Å	<i>b</i> /Å	<i>c</i> /Å	<i>V</i> /Å <sup>3</sup>
Present work	10.725(2)	5.784(2)	6.863(2)	425.74
Gasperin [5]	10.712(3)	5.780(1)	6.862(2)	424.86
Karyakin et al. [6]	10.705(2)	5.788(1)	6.862(2)	424.17



**Fig. 2** Relative variation of the cell parameters of NaUO<sub>2</sub>BO<sub>3</sub> with temperature

**Table 2** Experimental low-temperature heat capacity data for NaUO<sub>2</sub>BO<sub>3</sub> in the absence of the magnetic field

<i>T</i> /K	<i>C<sub>p</sub></i> /J K <sup>-1</sup> mol <sup>-1</sup>	<i>T</i> /K	<i>C<sub>p</sub></i> /J K <sup>-1</sup> mol <sup>-1</sup>	<i>T</i> /K	<i>C<sub>p</sub></i> /J K <sup>-1</sup> mol <sup>-1</sup>	<i>T</i> /K	<i>C<sub>p</sub></i> /J K <sup>-1</sup> mol <sup>-1</sup>
293.40	142.83	186.75	115.77	71.886	52.331	26.788	11.935
293.42	142.80	184.81	115.38	69.908	50.642	26.404	11.616
293.42	142.79	182.80	115.01	67.933	48.920	26.000	11.282
291.00	142.33	180.89	114.51	65.957	47.190	25.641	10.952
291.41	142.43	178.87	114.11	63.982	45.423	25.224	10.596
291.41	142.45	176.90	113.68	62.006	43.660	24.775	10.244
289.45	142.19	174.96	113.27	60.030	41.883	24.443	9.9551
287.07	141.60	172.95	112.77	58.057	40.116	23.976	9.5584
285.15	140.98	170.97	112.12	56.083	38.323	23.626	9.2884
283.19	140.40	168.99	111.49	54.111	36.511	23.162	8.9033
281.25	139.65	167.03	110.82	52.136	34.694	22.790	8.6148
279.29	138.94	165.04	110.06	50.160	32.867	22.375	8.2808
277.33	138.26	163.06	109.26	48.190	31.018	21.987	7.9789
275.38	137.72	161.09	108.46	46.196	29.166	21.574	7.6633
273.41	137.28	159.11	107.63	44.230	27.350	21.178	7.3681
271.46	136.94	157.13	106.72	44.403	27.549	20.781	7.0732
269.50	136.57	155.15	105.86	43.798	26.947	20.372	6.7791
267.54	136.17	153.17	104.91	43.395	26.596	19.983	6.5022
265.58	135.68	151.19	104.00	42.992	26.223	19.575	6.2072
263.62	135.25	149.21	103.02	42.590	25.860	19.192	5.9356
261.66	134.78	147.23	102.15	42.188	25.490	18.776	5.6487
259.70	134.30	145.25	101.19	41.784	25.122	18.405	5.3954
257.74	133.88	143.28	100.18	41.380	24.758	17.972	5.1091
255.78	133.46	141.30	99.136	40.976	24.388	17.621	4.8814
253.82	133.01	139.32	98.089	40.573	24.029	17.171	4.5883
251.85	132.59	137.33	97.004	40.052	23.583	16.831	4.3725
249.88	132.13	135.35	95.964	39.631	23.210	16.378	4.0918
247.93	131.64	133.37	94.897	39.245	22.853	16.031	3.8800
245.96	131.23	131.38	93.836	38.839	22.498	15.596	3.6215
243.99	130.69	129.40	92.722	38.431	22.144	15.229	3.4080
242.03	130.24	127.42	91.599	38.026	21.794	14.813	3.1730
240.06	129.71	125.43	90.471	37.625	21.433	14.436	2.9632
238.09	129.17	123.45	89.359	37.238	21.089	14.456	2.9747
236.12	128.62	121.47	88.171	36.841	20.741	14.229	2.8515
234.14	128.08	119.48	86.982	36.446	20.391	14.026	2.7437
232.17	127.51	117.50	85.808	36.044	20.039	13.828	2.6424
230.20	126.98	115.51	84.615	35.642	19.695	13.636	2.5445
228.22	126.55	113.53	83.355	35.245	19.351	13.447	2.4510
226.25	126.07	111.54	82.181	34.850	19.009	13.258	2.3567
224.29	125.61	109.56	80.910	34.447	18.684	13.065	2.2628
222.32	125.07	107.57	79.606	34.038	18.373	12.880	2.1739
220.35	124.55	105.58	78.306	33.636	18.024	12.698	2.0887
218.38	123.97	103.59	76.946	33.232	17.662	12.517	2.0065
216.40	123.39	101.61	75.588	32.836	17.287	12.342	1.9290
214.43	122.83	99.627	74.154	32.442	16.917	12.170	1.8539
212.45	122.30	97.644	72.711	32.037	16.545	12.001	1.7817
210.48	121.81	95.667	71.269	31.620	16.168	11.834	1.7104
208.51	121.37	93.684	69.784	31.192	15.795	11.668	1.6423

**Table 2** (continued)

<i>T</i> /K	<i>C<sub>p</sub></i> /J K <sup>-1</sup> mol <sup>-1</sup>	<i>T</i> /K	<i>C<sub>p</sub></i> /J K <sup>-1</sup> mol <sup>-1</sup>	<i>T</i> /K	<i>C<sub>p</sub></i> /J K <sup>-1</sup> mol <sup>-1</sup>	<i>T</i> /K	<i>C<sub>p</sub></i> /J K <sup>-1</sup> mol <sup>-1</sup>
206.53	120.83	91.700	68.308	30.781	15.441	11.504	1.5755
204.55	120.28	89.713	66.763	30.380	15.101	11.344	1.5116
202.60	119.78	87.735	65.225	29.975	14.777	11.177	1.4469
200.60	119.20	85.764	63.641	29.575	14.426	11.019	1.3876
198.62	118.72	83.783	62.035	29.179	14.080	10.863	1.3291
196.68	118.20	81.801	60.415	28.799	13.734	10.710	1.2734
194.67	117.67	79.832	58.791	28.387	13.293	10.560	1.2201
192.69	117.18	77.837	57.294	28.024	13.033	10.414	1.1691
190.71	116.69	75.846	55.607	27.564	12.610	10.271	1.1214
188.73	116.26	73.866	53.991	27.207	12.290	10.126	1.0735
9.985	1.0292	6.727	0.29107	4.483	0.08256	3.040	0.02563
9.846	0.98623	6.614	0.27586	4.420	0.07926	3.000	0.02463
9.707	0.94372	6.471	0.25766	4.359	0.07599	2.960	0.02366
9.572	0.90350	6.374	0.24607	4.299	0.07286	2.919	0.02273
9.438	0.86478	6.285	0.23471	4.239	0.06985	2.881	0.02191
9.307	0.82720	6.196	0.22433	4.181	0.06687	2.841	0.02116
9.176	0.79137	6.111	0.21453	4.122	0.06410	2.804	0.02043
9.047	0.75719	6.025	0.20500	4.059	0.06086	2.765	0.01995
8.920	0.72341	5.941	0.19625	4.002	0.05847	2.701	0.01854
8.796	0.69209	5.857	0.18742	3.946	0.05607	2.645	0.01758
8.672	0.66133	5.775	0.17940	3.892	0.05376	2.589	0.01665
8.549	0.63162	5.695	0.17168	3.838	0.05146	2.532	0.01577
8.424	0.60224	5.615	0.16432	3.785	0.04930	2.478	0.01498
8.305	0.57545	5.535	0.15731	3.734	0.04733	2.425	0.01421
8.188	0.54973	5.458	0.15063	3.682	0.04541	2.365	0.01331
8.074	0.52584	5.382	0.14456	3.631	0.04359	2.314	0.01261
7.961	0.50263	5.306	0.13821	3.582	0.04184	2.266	0.01193
7.849	0.48066	5.232	0.13240	3.533	0.04027	2.217	0.01123
7.740	0.45896	5.158	0.12650	3.485	0.03864	2.168	0.01056
7.632	0.43879	5.088	0.12118	3.437	0.03720	2.124	0.00997
7.526	0.41935	5.017	0.11619	3.391	0.03572	2.076	0.00939
7.421	0.40081	4.942	0.11110	3.344	0.03437	2.031	0.00890
7.317	0.38266	4.873	0.10638	3.299	0.03304	1.988	0.00844
7.216	0.36586	4.807	0.10218	3.254	0.03165	1.943	0.00799
7.116	0.34957	4.739	0.09795	3.210	0.03026	1.897	0.00754
7.015	0.33380	4.676	0.09394	3.167	0.02901	1.854	0.00713
6.918	0.31887	4.608	0.08983	3.125	0.02783	1.822	0.00685
6.821	0.30430	4.545	0.08600	3.083	0.02673		

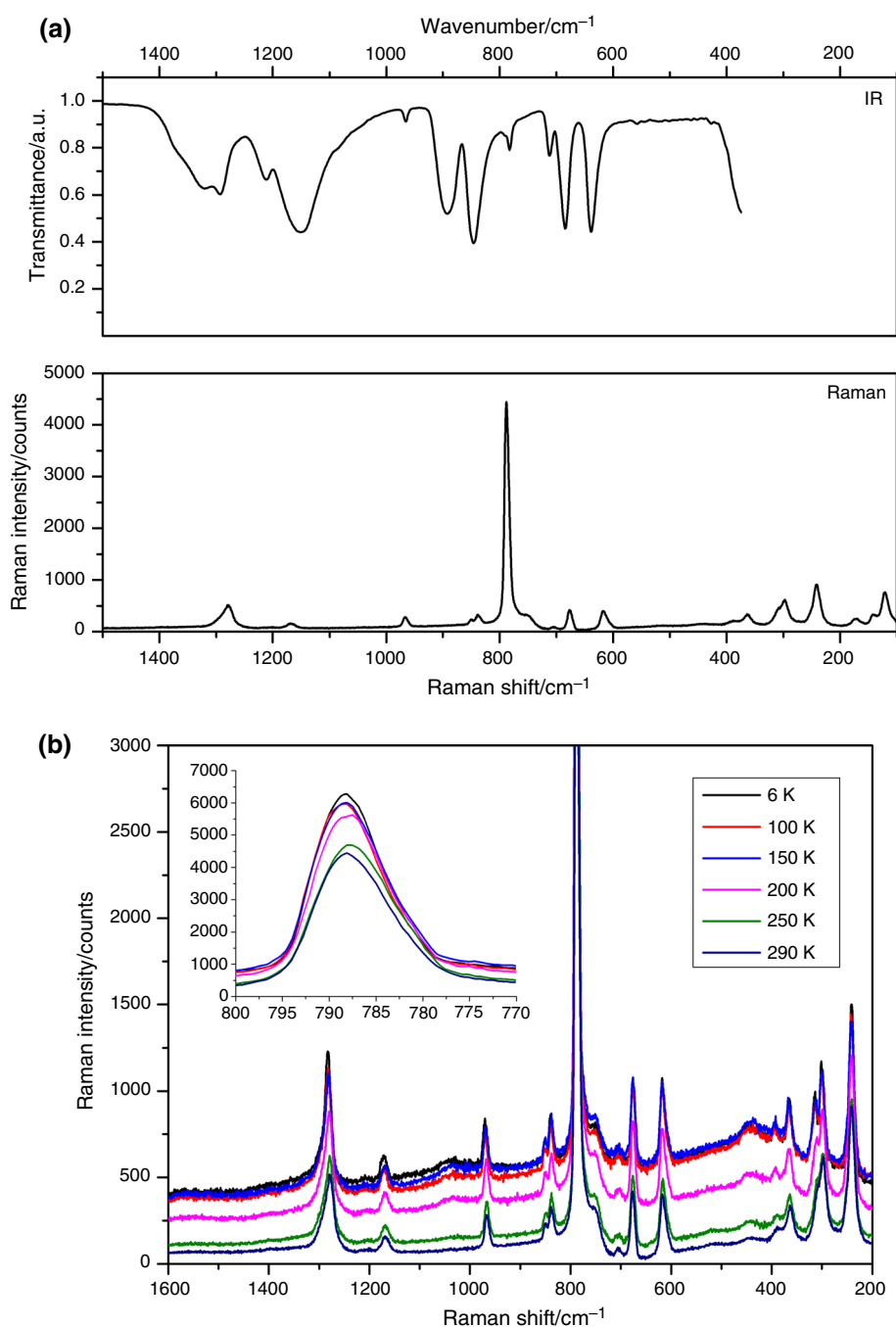
linear thermal expansion is anisotropic and varies more along the *c* axis. This can be explained by the fact that NaO6 units are made of weak Na–O bonds and are linearly connected along the *c* axis as shown in Fig. 1. As often observed, thermal expansion varies inversely with the bond strength.

## Heat capacity

The low-temperature heat capacity measurements data on NaUO<sub>2</sub>BO<sub>3</sub> are presented in Table 2.

The heat capacity data have been fitted to a series of overlapping polynomials, and we derive for the  $C_p(298.15\text{ K}) = (143.2 \pm 2.9)\text{ J K}^{-1}\text{ mol}^{-1}$ ,  $S^\circ(298.15\text{ K}) = (174.9 \pm 3.5)\text{ J}\cdot\text{K}^{-1}\cdot\text{mol}^{-1}$ , and  $\{H^\circ(298.15\text{ K}) - H^\circ(0\text{ K})\} = (26.37 \pm 0.53)\text{ kJ mol}^{-1}$ . The errors were

**Fig. 3** **a** IR and Raman spectra of  $\text{NaUO}_2\text{BO}_3$  at room temperature; **b** series of Raman spectra as a function of temperature in the range 6 to 290 K. The inset shows the full sizes of the most intense peak at  $788\text{ cm}^{-1}$ . No significant change in position is observed. The slight asymmetry at lower wavenumbers is increasing with increasing temperature



estimated from the accuracy of the equipment, which is in the order of 2%.

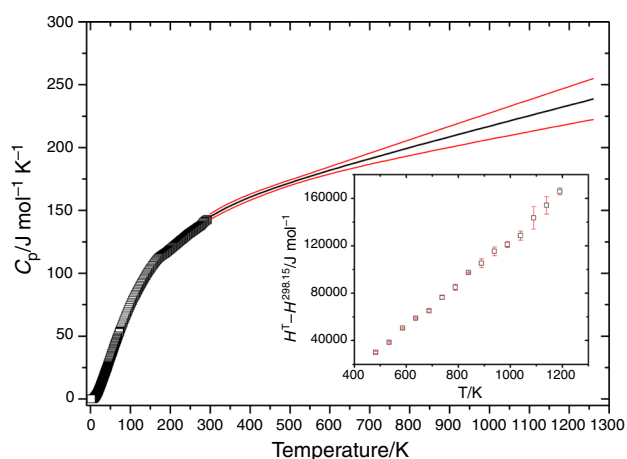
The low-temperature heat capacity measurements indicate an anomalous change in the slope of the curve at about 175 K. Our results agree satisfactory with the measurements reported by Karyakin et al. [6] (supplementary material S2), including the anomaly that was not identified as such by them. It should be noted that our measurements were performed from room temperature downwards, whereas Karyakin et al. measured from low temperature

upwards. This phenomenon was further analysed by performing X-ray diffraction and IR, respectively, and Raman spectroscopic measurements in the low-temperature range. Figure 3a, b summarises the IR and Raman spectra at room temperature and the temperature-dependent Raman spectra in the range from 6 to 290 K. The IR spectrum corresponded to earlier investigations by Hoekstra [4]. To our knowledge, there is no Raman spectrum of sodium uranyl borate reported in the literature. Anyhow, the temperature-dependent series showed no significant differences in the

**Table 3** Enthalpy increment  $H^\circ(T_m) - H^\circ(T_a)$  of NaUO<sub>2</sub>BO<sub>3</sub> ( $T_m$ —measurement temperature,  $T_a$ —ambient temperature)

$T_m/K$	$T_a/K$	$H^\circ(T_m) - H^\circ(T_a)/\text{kJ mol}^{-1}$
482.5	294.2	29.94 ± 0.88
533.9	294.2	38.34 ± 0.49
585.8	294.2	50.38 ± 0.43
636.5	294.2	58.76 ± 0.63
687.5	294.2	65.11 ± 1.00
738.0	294.2	76.30 ± 1.34
788.5	294.2	84.90 ± 2.57
838.7	294.2	97.35 ± 0.52
889.0	294.2	105.22 ± 3.71
938.7	294.2	115.29 ± 3.75
989.0	294.2	121.12 ± 2.59
1039.3	294.2	128.59 ± 3.89
1089.6	294.2	143.51 ± 9.45
1139.7	294.2	154.08 ± 7.32
1190.5	294.2	165.87 ± 2.95

The error indicated is the standard deviation of the average value from several sample drops



**Fig. 4** Heat capacity ( $\text{J K}^{-1} \text{mol}^{-1}$ ) of NaUO<sub>2</sub>BO<sub>3</sub> as a function of temperature, (square) low-temperature data, (solid black line) the heat capacity obtained from the drop calorimeter, (solid red line) confidence band. Inset: the measured enthalpy increments with their standard deviations

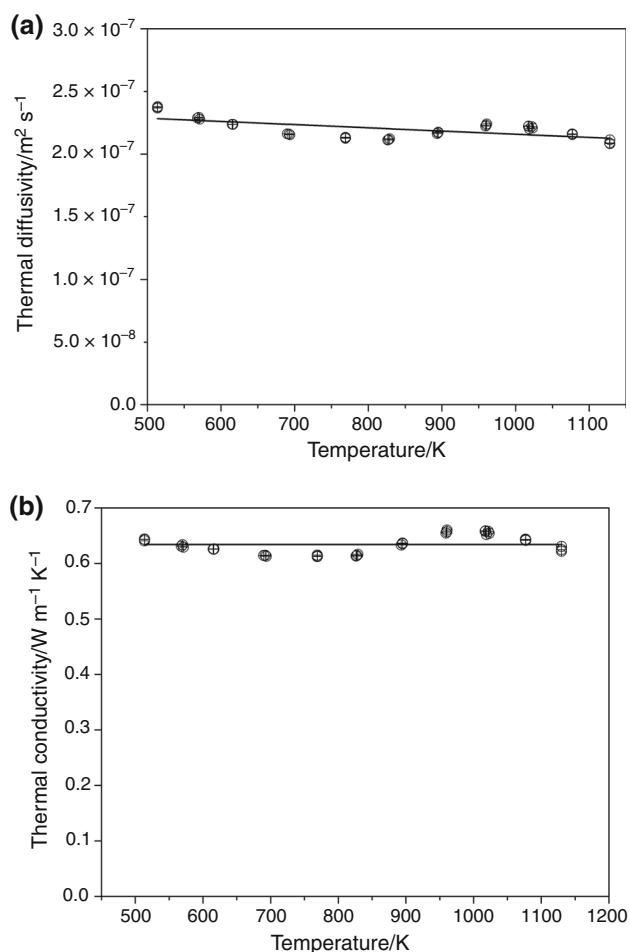
spectra with temperature variation. This is in agreement with the X-ray diffraction data; neither of them showed any clear indication of a structural transformation. We therefore conclude that the origin of the anomaly in the heat capacity must be of a diffuse nature, which could be related to a slow and slight structural reordering of oxyanionic units, which cannot be detected by X-ray diffraction or Raman spectroscopy.

The enthalpy increment data derived for NaUO<sub>2</sub>BO<sub>3</sub> are shown in Table 3 and in Fig. 4. All measured enthalpy

**Table 4** Thermal diffusivity and thermal conductivity of NaUO<sub>2</sub>BO<sub>3</sub>

$T/K$	Thermal diffusivity $10^{-7} \text{ m}^2 \text{ s}^{-1}$	Thermal conductivity $\text{W m}^{-1} \text{ K}^{-1}$
513.6	2.376	0.641
513.6	2.385	0.643
513.8	2.387	0.644
513.9	2.384	0.643
568.1	2.297	0.631
569.9	2.304	0.633
571.1	2.289	0.630
615.3	2.250	0.626
616.1	2.248	0.626
616.3	2.250	0.626
690.0	2.176	0.614
692.2	2.175	0.615
693.6	2.169	0.613
769.0	2.143	0.612
769.3	2.146	0.613
769.1	2.152	0.615
826.5	2.131	0.613
827.8	2.135	0.615
829.0	2.142	0.616
893.5	2.186	0.633
894.9	2.194	0.636
895.1	2.198	0.637
959.3	2.253	0.655
960.4	2.261	0.657
960.9	2.271	0.660
1018.9	2.234	0.652
1022.0	2.252	0.658
1023.1	2.241	0.654
1017.4	2.254	0.658
1017.8	2.257	0.659
1077.2	2.190	0.642
1077.3	2.194	0.643
1076.9	2.196	0.643
1130.0	2.127	0.625
1130.0	2.116	0.622
1130.0	2.147	0.631

increment data were fitted using a combined linear regression together with the room temperature heat capacity data obtained from the PPMS measurements performed as part of this study. A Maier–Kelly-type second-order polynomial equation was used, similar to our previous studies for alkaline and alkaline-earths uranates [14, 15]. The following heat capacity equations were obtained:



**Fig. 5** Thermal diffusivity (a) and conductivity (b) of  $\text{NaUO}_2\text{BO}_3$

$$C_{p,m}^o / \text{J K}^{-1} \text{mol}^{-1} = (188.3 \pm 27.8) + (11.20 \times 10^{-3} \pm 3.97 \times 10^{-3})(T/K) - (4.308 \times 10^6 \pm 1.431 \times 10^6)(T/K)^{-2}. \quad (1)$$

### Thermal diffusivity and conductivity

The results of the thermal diffusivity measurements are listed in Table 4, and are shown in Fig. 5a. The relative uncertainty on the thermal diffusivity measurements is of 5%, mainly due to sample thickness variations. The sample thickness change due to thermal dilatation was corrected using the thermal expansion data obtained in this work. The results show very moderate temperature dependence and can be described by the following equation:

$$\alpha / (\text{m}^2 \text{s}^{-1}) = 2.398 \times 10^{-7} - 2.093 \times 10^{-11}(T/K). \quad (2)$$

The thermal conductivity of the compound was calculated from the measured thermal diffusivity, heat capacity, and density using the formula:

$$\lambda = \rho \times \alpha \times C_p. \quad (3)$$

The density of the samples, measured by the Archimedes technique at ambient temperature, was of  $5.37 \text{ g cm}^{-3}$  (98% of the theoretical density, with  $\text{TD} = 5.50 \text{ g cm}^{-3}$ ). The temperature dependence of the density was calculated using the thermal dilatation data obtained in this work.

The obtained sample thermal conductivity (Fig. 5b) showed no significant temperature dependence and was therefore considered to be temperature independent, with a value of  $\lambda / (\text{W m}^{-1} \text{K}^{-1}) = 0.634$ , which is very low for a ceramic material [16, 17].

The relative uncertainty on the calculated thermal conductivity is estimated to be of 10%, resulting from the sum of the uncertainties on the thermal diffusivity (5%), specific heat (3%), and density (2%).

### Conclusions

- The linear thermal expansion of  $\text{NaUO}_2\text{BO}_3$  is very anisotropic along all the crystallographic axes.
- The anomalous change in the curve of the low-temperature heat capacity at about 175 K found by Karyakin et al. [6] was reproduced, indicating that it is a genuine effect.
- The thermal conductivity of  $\text{NaUO}_2\text{BO}_3$  was found to be very low for a ceramic material.

**Acknowledgements** The authors acknowledge Daniel Bouexière, Co Boshoven, Herwin Hein, Michael Holzhäuser, Davide Robba, and Marc Sierig, for technical support.

**Open Access** This article is distributed under the terms of the Creative Commons Attribution 4.0 International License (<http://creativecommons.org/licenses/by/4.0/>), which permits unrestricted use, distribution, and reproduction in any medium, provided you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made.

### References

1. Wang S, Alekseev EV, Ling J, Liu G, Depmeyer W, Albrecht-Schmitt TE. Polarity and chirality in uranyl borates: insights into understanding the vitrification of nuclear waste and the development of nonlinear optical materials. *Chem Mater.* 2010;22:2155–63.
2. Wang S, Alekseev EV, Diwu J, Miller HM, Oliver AG, Liu G, Depmeyer W, Albrecht-Schmitt TE. Functionalization of borate networks by the incorporation of fluoride: syntheses, crystal structures, and nonlinear optical properties of novel actinide fluoroborates. *Chem Mater.* 2011;23:2931–9.
3. Silver MA, Albrecht-Schmitt TE. Evaluation of f-elements borate chemistry. *Coord Chem Rev.* 2016;323:36–51.



4. Hoekstra HR. Uranyl metaborate and sodium uranyl borate. In: Fields P, et al., editors. Lanthanide and actinide chemistry, advances in chemistry. Washington, DC: ACS; 1967. p. 320–30.
5. Gasperin M. Synthèse et structure du borouranate de sodium,  $\text{NaBUO}_5$ . *Acta Cryst.* 1988;C44:415.
6. Karyakin NV, Chernorukov NG, Knyazev AV, Feoktistova OV, Alimzhanov MI, Feoktistova MA. The thermodynamic properties of sodium uranoborate. *Russ J Phys Chem.* 2002;76:347–50.
7. Chernorukov NG, Knyazev AV, Kortikova OV, Sergacheva IV. Thermochemistry of alkali metal uranoborates and their crystal hydrates. *Radiochemistry.* 2003;45:122–6.
8. Rodriguez-Carjaval J. Recent advances in magnetic structure determination by neutron powder diffraction. *Phys B.* 1993;192:55–69.
9. Lashley JC, Hundley MF, Migliori A, Sarrao JL, Pagliuso PG, Tarling TW, Jaime M, Cooley JC, Hulst WL, Morales L, Thoma DJ, Smith JL, Boerio-Goates J, Woodfield BF, Stewart GR, Fisher RA, Phillips NE. Critical examination of heat capacity measurements made on quantum design physical property measurement system. *Cryogenics.* 2003;43:369–78.
10. Javorský P, Wastin F, Colineau E, Rebizant J, Boulet P, Stewart G. Low-temperature heat capacity measurements on encapsulated transuranium samples. *J Nucl Mater.* 2005;344:50–5.
11. Popa K, Shvareva T, Mazeina L, Colineau E, Wastin F, Konings RJM, Navrotsky A. Thermodynamic properties of  $\text{CaTh}(\text{PO}_4)_2$  synthetic cheralite. *Am Mineral.* 2008;93:1356–62.
12. Beneš O, Popa K, Reuscher V, Zappia A, Staicu D, Konings RJM. High-temperature heat capacity of  $\text{PuPO}_4$  monazite-analogue. *J Nucl Mater.* 2011;418:182–5.
13. Sheindlin M, Halton D, Musella M, Ronchi C. Advances in the use of laser-flash techniques for thermal diffusivity measurements. *Rev Scient Instr.* 1998;69:1426–36.
14. Popa K, Wastin F, Colineau E, Konings RJM. The heat capacity of  $\text{BaUO}_4$ . *J Chem Thermodynamics.* 2007;39:104–7.
15. Konings RJM, Popa K, Colineau E, Wastin F. The low-temperature heat capacity of  $\text{CaUO}_4$  and  $\text{SrUO}_4$ . *J Chem Thermodynamics.* 2008;40:220–4.
16. Pujula M, Sánchez-Rodríguez D, Lopez-Olmedo JP, Farjas J, Poura P. Measuring thermal conductivity of powders with differential scanning calorimetry. A simplified method. *J Therm Anal Calorim.* 2016;125:571–7.
17. Almeida TF, Leite FHG, Faira RT Jr, Holanda JNF. Thermal study of calcium silicate material synthesized with solid waste. *J Therm Anal Calorim.* 2017;128:1265–72.