#### **ORIGINAL PAPER**



# Electrochemical deposition of nickel from aqueous electrolytic baths prepared by dissolution of metallic powder

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#### **Abstract**

A new method of preparation of aqueous electrolyte baths for electrochemical deposition of nickel targets for medical accelerators is presented. It starts with fast dissolution of metallic Ni powder in a  $HNO_3$ -free solvent. Such obtained raw solution does not require additional treatment aimed to removal nitrates, such as the acid evaporation and Ni salt precipitation-dissolution. It is used directly for preparation of the nickel plating baths after dilution with water, setting up pH value and after possible addition of  $H_3BO_3$ . The pH of the baths ranges from alkaline to acidic. Deposition of 95% of ca. 50 mg of Ni dissolved in the bath takes ca. 3.5 h for the alkaline electrolyte while for the acidic solution it requires ca. 7 h. The Ni deposits obtained from the acidic bath are physically and chemically more stable and possess smoother and crack-free surfaces as compared to the coatings deposited from the alkaline bath. A method of estimation of concentration of  $H_2O_2$  in the electrolytic bath is also proposed.

**Keywords** Medical accelerator targets  $\cdot$  Nickel electrodeposition  $\cdot$  Nickel dissolution;  $\cdot$  H<sub>2</sub>O<sub>2</sub> determination

#### Introduction

Solid nickel targets are frequently used for production of medical radioisotopes by means of nuclear reactions with charged particles from accelerators. Several Cu, Co and Ni radioisotopes of medical importance are produced in reactions of nickel nuclei with high energy protons, deuterons or  $\alpha$  particles [1–8]. As an example, <sup>64</sup>Cu, a very important radioisotope frequently used in positron emission tomography [1, 3, 9–14] and considered for use in radiotherapy [10, 13, 14], is produced in medical accelerators using <sup>64</sup>Ni(p,n)<sup>64</sup>Cu (mainly), <sup>64</sup>Ni(d,2n)<sup>64</sup>Cu and <sup>61</sup>Ni( $\alpha$ ,p)<sup>64</sup>Cu reactions [1–3, 11–13, 15–24]. The solid Ni targets used in these processes are typically obtained through electrodeposition on a substrate with a good electronic and heat conductivity, e.g. gold [2, 16–23, 25–27], platinum [15, 28, 29]

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or copper covered with a protective layer of Au or Ag [24, 30]. Unfortunately, some of natural nickel isotopes, which act as targets in production of the medical radioisotopes, are of low natural abundance (from 0.9 to 3.6% for <sup>61</sup>Ni, <sup>62</sup>Ni and <sup>64</sup>Ni [31]). Thus, high yield of the nuclear reaction in question is achieved only for a target containing nickel which is highly enriched with the respective isotope. Isotopically enriched nickel, however, is usually very expensive, and, typically, only a very small amount of such material is available for preparation of electrolyte baths. Such bath usually contains from few tens to ca. 120 mg of Ni [15, 18, 20, 22, 23, 26, 28, 32]. Work with such small amount of the electroactive species leads to serious consequences for the electroplating procedure. Firstly, it is not possible to prepare a bath containing high concentration of the nickel. Reported concentration of isotopically enriched nickel in the baths used for the target deposition typically varies from few tens of millimolar to ca. 0.4 M [33]. This is much less than for typical Watts or sulphamate baths used for industrial nickel plating [34, 35]. Secondly, the amount of Ni dissolved in the bath is calculated in such a way that deposition of all the metal will produce the target layer with a required thickness. Thus, the whole or almost whole (e.g. at least 95%) amount of the nickel is expected to be deposited. It is a common approach to continue the target electrodeposition until the



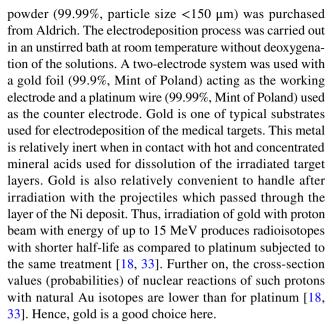
bath becomes colourless (naked eye judgement) [15, 19, 20, 26, 28, 29, 36, 37]. This is in contrast to the industrial plating procedures which do not require complete metal deposition and which left the bath usually only weakly depleted with the metal after completing the deposition.

Although numerous works report electrochemical plating of Ni targets for accelerators [1, 2, 6, 15–28, 30, 36–38], there are some aspects of the electrodeposition process which still need to be improved. Thus, <sup>64</sup>Ni and <sup>62</sup>Ni are most often available as a metallic powder and there is a need to develop a proper method of its dissolution. Apparently, a concentrated HNO<sub>3</sub> is a good medium to dissolve the metallic Ni. Excess of the acid, however, should be removed from the bath after completing the dissolution. Further on, the presence of the nitrates in the electrolyte bath may lead to unwanted precipitation of Ni(OH)<sub>2</sub> when NO<sub>3</sub> ions undergo cathodic reduction [39–41]. Finally, too high concentration of the nitrates reduces faradaic efficiency of the electrodeposition [42, 43]. This problem is overcome by evaporation of HNO<sub>3</sub> from the liquid containing dissolved nickel and by application of several subsequent steps of dissolution, evaporation and final filtration of such obtained deposit [2, 16, 17, 23, 25, 27]. As an alternative, HNO<sub>3</sub> can be neutralised with an excess of a hydroxide [20]. This leads to precipitation of Ni(OH)<sub>2</sub> which is subjected to repeated procedures of centrifugation and separation from the supernatant [20]. Both these approaches, however, are time consuming and include additional laboratory procedures. Metallic Ni can be also dissolved in concentrated HCl, but again, this step must be followed by additional procedures, such as evaporation and dissolution, aimed to remove the acid [5, 21, 32, 37]. The other problem related to the electrochemical preparation of the targets lies in the fact that the complete deposition of the whole amount of Ni dissolved in the bath usually requires a very long time, up to 12 h [18, 23, 25], 24 h [1, 2, 15–17, 20, 30, 44] or even 48 h or more [2, 22, 38]. Finally, poor quality Ni deposits, e.g. with cracks or poorly adhered to the substrate, are unacceptable [19, 21, 24, 30, 45].

This manuscript presents preliminary results of works aimed to solve the above-mentioned problems. We report a method of preparation of an aqueous nickel containing electrolyte bath which can be used as a starting point in development of a fast and simple procedure of production of Ni targets for accelerators. Besides, the method can be used also for other applications, where the plating baths prepared by dissolution of metallic Ni powder are of interest.

## **Experimental**

All the solutions were prepared using high purity chemicals (Avantor (POCh), Riedel de Haën, Chempur and B&K) and  $18.2~M\Omega$  cm water purified in a Millipore system. Ni



The electrodeposition was performed in a galvanostatic mode using a Rohde&Schwarz HMP2020 programmable power supply or a CHInstruments CHI660D electrochemical analyser. Currents of 50–60 mA cm<sup>-2</sup> (in respect to the geometric area) were applied; an analysis of the current density influence on the deposition process will be reported in a separate communication. UV-Vis spectra of the solutions were recorded with a Shimadzu UV-1800 spectrophotometer while energy-dispersive X-ray fluorescence (ED-XRF) analysis of the deposits was performed using a Shimadzu 800HS2 system with 50 kV exciting X-ray beam. pH was measured using a Schott Lab870 pH-meter and a SI Analytics BlueLine electrode. Morphology of the deposits was analysed using a LEO 435VP scanning electron microscope (SEM).

## **Results and discussion**

#### **Nickel powder dissolution**

Dissolution of Ni powder (Ni(0)) was carried out in absence of HNO<sub>3</sub>. Mass of a single batch of the Ni powder used for the bath preparation was equal to 50–54 mg. Complete deposition of such amount of the metal leads to formation of a layer with a thickness of 112–121 µm for the surface area of 0.5 cm<sup>2</sup>. In order to obtain a reasonably high concentration of Ni ions in the electrolyte bath, the volume of the solution used for the metal dissolution as well as the volume of the final plating bath should be as low as possible. The powder was dissolved in a solution containing 0.1 ml of 30% H<sub>2</sub>O<sub>2</sub> and 0.5 ml of 3M H<sub>2</sub>SO<sub>4</sub>. The dissolution is a fast and vigorous process which takes less than 5 min at room temperature. It can be made even faster by heating the mixture up



to 70–80 °C. When the dissolution does not proceed with a sufficient rate, an additional amount of  $H_2O_2$ , up to 0.1 ml, can be added. Such prepared liquid is referred further in the text as raw solution and constitutes the basis for preparation of the final electrolyte baths.

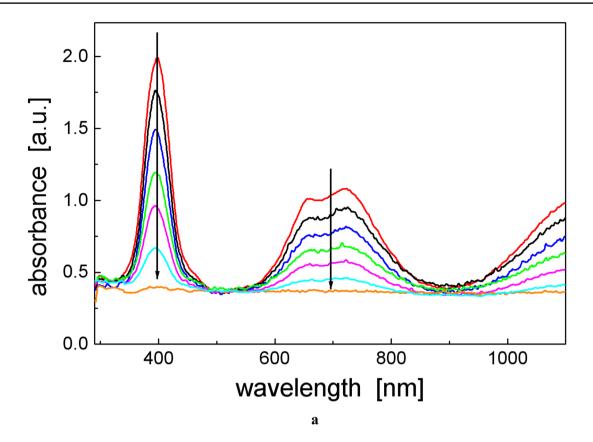
Application of a mixture containing  $H_2O_2$  and a mineral acid (H<sub>2</sub>SO<sub>4</sub> [44, 46–48] or HCl [5]) to dissolve metallic Ni has been reported in the earlier literature. Such approach was used for dissolution of powders [5, 44, 47, 48] and crushed electrical circuit boards [46]. This is the first step in preparation of electrolytic baths used for deposition of accelerator targets described in [5, 44, 48]. The procedure presented by us differs in several aspects from those reported by the above-mentioned papers. Thus, in contrast to [5], we use H<sub>2</sub>SO<sub>4</sub> instead of HCl. The presence of the latter in the electrolytic bath may enhance anodic dissolution of Pt counter electrode [49, 50] and may result in extensive Cl<sub>2</sub> evolution [51] when the anode is subjected to a strong anodic polarisation. In contrast to [44], we add only a single portion of H<sub>2</sub>O<sub>2</sub> instead of gradual addition of several batches of the peroxide during the dissolution process. We also use a higher concentration of both H<sub>2</sub>SO<sub>4</sub> (2.50 M) and H<sub>2</sub>O<sub>2</sub> (1.63 M) as compared to [44, 46, 47], where the respective concentrations varies from 0.1 to 2 M for the acid and from 0.06 to 0.22 M for the peroxide. Peroxide-to-nickel molar ratio used in the present work is ca. 1.1 which is similar to 1.2–1.3 reported in [44, 46] but less than 3.5 described in [47]. Concentration of the nickel cations in the raw solution prepared by us varies from 1.4 to 1.6 M and is comparable to 1.73 M from [5] but significantly higher than 0.017-0.17 M obtained in [44, 46, 47]. It is important to note that we did not observe significant differences between electroplating from baths prepared from raw solutions when concentrations of Ni<sup>2+</sup>, H<sub>2</sub>SO<sub>4</sub> and H<sub>2</sub>O<sub>2</sub> were varied within the limits reported above. As compared to [44, 47], the procedure described in this paper can be successfully applied at room temperature and dissolves nickel faster. It also allows dissolving higher amount of Ni powder in respect to the volume of the liquid medium as compared to [47], but the presented approach seems to be economically beneficent only when applied to small amounts of the metal. It should be noted that an enhancement in the Ni powder dissolution can be obtained by adding other than H<sub>2</sub>O<sub>2</sub> oxidising agent, e.g. Fe<sup>3+</sup> [47]. This approach, however, may result in unwanted iron co-deposition when the latter is not removed from the electrolyte bath prior to the electrodeposition.

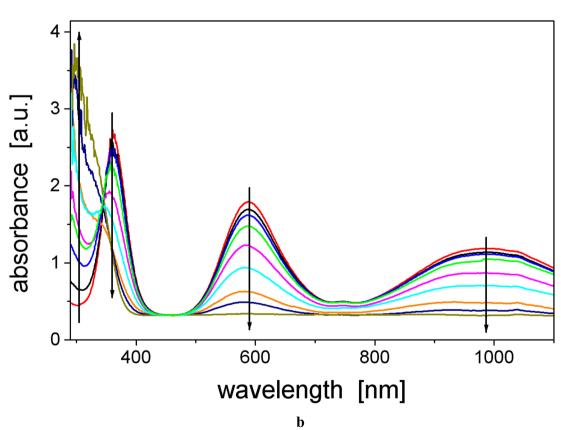
In the next step, the raw solution is converted into the electrolytic bath. Two of the most often applied Ni electrodeposition baths are the alkaline one containing ammonia Ni<sup>2+</sup> complexes (pH of 9–11) [2, 15, 16, 18–20, 22, 24, 26–28, 38] and the acidic one, usually containing sulphates (pH of 3–5) [19, 21, 23, 30, 34]. pH of our raw solution

is lower than 0 being too low as for the intended applications. The most obvious way of neutralisation of such type of the solution is by means of addition of a proper amount of a hydroxide. However, gradual neutralisation of the raw solution with aqueous ammonia or alkali metal hydroxide solutions leads to slow precipitation of a green deposit and this process occurs already at pH lower than ca. 0.3. The precipitate is dissolved when the pH is lowered back to the more acidic values. It is also completely dissolved when the pH is raised above 7 by adding  $NH_{3aq}$ . The latter solution has a blue colour due to presence of ammonia Ni<sup>2+</sup> complexes and can be used for preparation of the alkaline electrolyte bath. When the pH is decreased back below 7, the precipitate reappears and can be dissolved only after shifting pH below ca. 0.3. The precipitate in question probably contains nickel hydroxide sulphates, i.e. double salts (or mixed complexes) containing Ni(OH)<sub>2</sub> and NiSO<sub>4</sub>. Generation of these species was reported to take place at pH of 2 during neutralisation of NiSO<sub>4</sub> with NaOH [52].

Formation of the above-mentioned precipitate has a strong impact on the further treatment of the raw solution. Thus, preparation of the alkaline bath with pH of 9.2–9.9 is relatively simple and relies on neutralisation of the raw solution with a proper amount of 25% ammonia solution until a clear liquid free from the precipitate and with required pH is obtained. Subsequently, distilled water is added to the total volume of 2.0-2.1 ml. More complex is preparation of the acidic bath with pH of 3-5. Such bath cannot be obtained by neutralisation of the raw solution with NH<sub>3aq</sub> due to the fact that the above-mentioned Ni-containing precipitate is stable in this pH range. Therefore, an alternative method of the raw solution neutralisation had to be developed. We found that this task can be successfully completed by addition of sodium or ammonia acetate instead of the hydroxides or NH<sub>3aq</sub>. Solubility of these acetates in water is very high, and they dissolve very fast at room temperature. Addition of the acetates allows to increase pH up to the values higher than 5. As an example, setting up pH in the range of 4.1–4.5 requires the addition of 0.128–0.167 g ml<sup>-1</sup> of CH<sub>3</sub>COONa·3H<sub>2</sub>O. It is worth to note that the presence of the acetates in the electrolytic bath may also facilitate formation of good quality Ni deposits [53]. Addition of the acetates is preceded by dilution of the raw solution with distilled water up to the volume of 2.0-2.1 ml. Boric acid was added to the both the alkaline and the acidic electrolyte giving the final concentration of 0.31–0.34 M. H<sub>3</sub>BO<sub>3</sub> is known to improve quality of Ni deposits although the exact mechanism of the acid influence on the nickel electrodeposition is still discussed [33, 54–57]. Thus, H<sub>2</sub>BO<sub>2</sub> is considered as a hydrogen evolution reaction inhibitor, surface pH regulator and a Ni<sup>2+</sup> complexing agent [34, 53-59]. Concentration of the nickel cations in both the alkaline and the acidic bath varies from 0.37 to 0.45 M. Apparently, variation of the









◄Fig. 1 UV-Vis spectra recorded during Ni electrodeposition from the acidic (a) and the alkaline (b) bath. Current density of 60 mA cm<sup>-2</sup>; the arrows indicate directions of the spectra changes with time

nickel and H<sub>3</sub>BO<sub>3</sub> concentrations within the limits quoted above has no influence on the deposition process.

## **Nickel electrodeposition**

### In situ UV-Vis analysis of the plating process

The electrodeposition was carried out for two types of the baths: the alkaline one (initial pH of 9.2–9.9) and the acidic one (initial pH of 4.1-4.5). The plating was performed under controlled current conditions in a two-electrode system which is a set-up frequently used for the Ni targets preparation [2, 5, 16, 18, 20-23, 26, 28, 33]. In situ UV-Vis measurements with the electrolytic cell placed inside a spectrometer were used to track progress in the plating process. The amount of the electrodeposited nickel equals the amount of the metal removed from the bath. The UV-Vis spectra recorded during the nickel electrodeposition from the alkaline and from the acidic baths are shown in Fig. 1. The spectra reveal shapes typical for aqueous solutions of NiSO<sub>4</sub> [60–62]; noises in the curves are produced by gases evolving on the cathode (hydrogen) and on the anode (oxygen). It follows that the time evolution of the spectra depends on the bath pH. Thus, in the acidic bath only, a gradual decrease of Ni peaks at 395 and 659–721 nm is observed without development of new signals (Fig. 1a). The most intensive Ni(II) peak at 395 nm was selected to analyse nickel removal from the bath. Evolution of the spectra recorded in the alkaline bath is somewhat more complex because the decrease in the Ni signals at ca. 361 and 583 nm is accompanied by a simultaneous development of absorption bands below ca. 340 nm (Fig. 1b). Formation of the latter signals is independent on the H<sub>2</sub>O<sub>2</sub> presence, and these bands can be attributed to nitrates or other nickel-oxygen compounds [63, 64] which are formed during NH<sub>3</sub> oxidation at the platinum anode [65, 66]. These signals eventually overlap the main Ni peak at ca. 362 nm. Hence, the progress in the Ni electrodeposition from the alkaline bath was analysed using the less intense Ni signal at ca. 588 nm.

It is worth to mention that the system studied was set up in such a way that the spectrophotometer beam was directed into a space between the working and the counter electrodes. This was done in order to reduce the optical path disruption by solid bodies. It follows then that the UV-Vis measurements reflect changes in the nickel cations concentration in the centre of the cuvette which, in turn, may differ from Ni<sup>2+</sup> content variation near the electrode surface. Therefore, the results presented in Figs. 1, 2 and 3 provide reliable

information about the rate of removal of nickel cations from the bulk of the solution but an analysis of reported UV-Vis data in terms of kinetics of the nickel reduction should be treated with caution since the kinetics is determined by the Ni<sup>2+</sup> content next to the electrode surface.

Figure 2 shows rate of removal of  $Ni^{2+}$  ions from the bath due to their reduction and incorporation into the deposited metal layer. The relative fraction of the nickel cations remaining in the bath at time *t* counted from the start of the deposition,  $Ni_{bath}$ , is calculated using Eq. (1):

$$Ni_{bath}(t) = \frac{A(t)}{A_0} \cdot 100\% \tag{1}$$

where  $A_0$  and A(t) denote background corrected absorbance before (t = 0) and during the deposition at time t, respectively. An analysis of the figure shows that the rate of Ni electrodeposition from the analysed baths strongly depends on pH. Thus, the deposition rate increases with pH, and, for 50–60 mA cm<sup>-2</sup>, the plating of more than 99.5% of Ni takes ca. 5 h for pH of 9.3 (Fig. 2) while for pH of 4.3, ca. 95% of Ni is deposited within ca. 7.5 h (Fig. 2). At the end of the deposition process, the pH of the acidic bath decreases to the level of 1.5–2.3, in a rough agreement with other reports [19, 21], while in the alkaline bath, the final pH reaches the values of 7.1–8.4. It should be stressed, however, that these value were measured after the experiment and they represent pH averaged for the whole volume of the bath. Thus, conclusions about pH changes near the electrode surface during the deposition should be drawn with care [59]. In order to simplify the deposition process as much as possible, we did not apply any procedures, such as gradual addition of respective chemicals [37], aimed to maintain the pH during the electrodeposition.

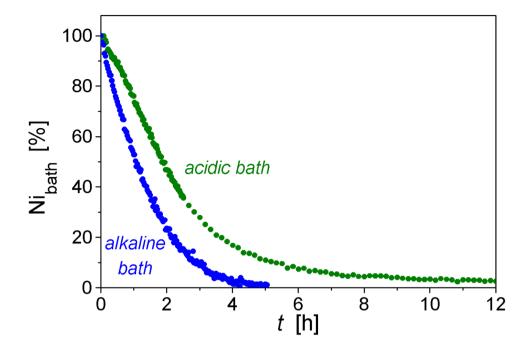
Complete or almost complete ( $\geq$ 95%) deposition of the nickel dissolved in the bath within a reasonable time, such as several hours, requires application of a relatively high current density [33]. This, in turn, leads to a strong polarisation of the cathode which establishes conditions of vigorous hydrogen evolution. This effect significantly reduces faradaic efficiency of the process [19, 26, 67]. The UV-Vis spectra shown in Fig. 2 not only allow tracking progress in the deposition process but also provide data for analysis of faradaic efficiency of the system ( $\varepsilon$ ). For a given time window  $\Delta t = t_1 - t_2$ , the latter is expressed by the following Eq. (2):

$$\varepsilon(t_2) \frac{\left(\frac{A(t_1) - A(t_2)}{A_0}\right) \cdot N_0 \cdot n \cdot F}{Q_{\rm m}} \cdot 100\% \tag{2}$$

where  $N_0$  is the initial number of moles of nickel present in the bath (t = 0), n is the number of electrons exchanged during the Ni<sup>2+</sup> reduction (n = 2), F represents the Faraday constant,  $Q_m$  stands for measured charge passed in the system



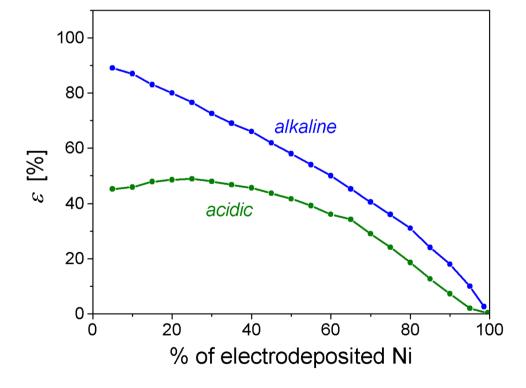
Fig. 2 Amount of Ni remaining in the electrolyte bath, Ni<sub>bath</sub>, calculated from the UV-Vis spectra as a function of the deposition time from an alkaline and an acidic bath. Current density of 60 mA cm<sup>-2</sup>



during  $\Delta t$  while  $A(t_1)$  and  $A(t_2)$  indicate absorbance at the beginning and at the end of the time window, respectively. The efficiency is calculated as a temporary value for  $\Delta t = 100$  s and  $t_1$  and  $t_2$  values continuously growing with the deposition progress. Such calculated values are shown in Fig. 3 as a function of the amount of Ni electrodeposited from the start of the plating process to the time of  $t_2$ . This plot shows temporary efficiency during 100 s of the plating at the time when

the amount of the deposited metal reaches the value given on the abscissa axis. The progress in the nickel deposition was determined on the basis of plots like the ones shown in Fig. 2 (after applying mathematical smoothing procedure). It follows that the faradaic efficiency decreases with the deposition time, and, for the alkaline bath, its maximum value is significantly higher as compared to the acidic electrolyte (ca. 91% vs. ca. 53% for deposition of first few per cent of the metal).

**Fig. 3** Temporary Faradaic efficiency of the Ni electrodeposition,  $\varepsilon$ , as a function of the amount of electrodeposited metal. Data for the alkaline (initial pH of 9.3) and the acidic bath (initial pH of 4.4), current of 60 mA cm<sup>-2</sup>,  $\Delta t = 100$  s (details in the text)





#### Physical and chemical characterisation of the deposits

Morphology of the deposit was analysed using scanning electron microscopy. The SEM images shown in Fig. 4 reveal that the quality of the deposit surface is generally better for lower pH of the bath. The layers deposited from the alkaline bath have a very high surface roughness and contain numerous spherical, cauliflower-like structures and "wells". The latter are typical for the deposition taking place around a high diameter hydrogen bubble attached to the surface [68]. Electrodeposition from the acidic bath, on the other hand, generates significantly smoother surfaces, free of cracks and "wells" and with only a small amount of the spherical features (Fig. 4b). An analysis of SEM images suggests that the deposits obtained from the acidic bath are composed with crystallites of a smaller size as compared to those plated from the alkaline solution.

The deposits plated from the acidic bath are more physically and chemically stable as compared to those plated from the alkaline electrolyte. The latter are brittle and fine fragments of the deposit come off from the substrate easily. Chemical stability of the deposits was evaluated on the basis of a dissolution test. The deposits were placed in concentrated HNO<sub>3</sub> (65%) at room temperature, and UV-Vis spectra were recorded as a function of the time of the sample immersion in the acid. Figure 5 reveals that the deposit from the alkaline bath dissolves significantly faster than the one deposited from the acidic bath. Eventually, significant fractions of the deposit plated from the alkaline layer peel off from the substrate at the advanced stages of the dissolution process. These fragments fall on the bottom of the cuvette and Ni cations originating from their dissolution slowly diffuse towards the optical path of the spectrometer. This effect leads to a very slow increase in the Ni absorbance seen in Fig. 5 at the final stages of the dissolution of the deposit plated from the alkaline bath.

It is likely to assume that the bath composition influence on the morphology of the deposit surface (Fig. 5) is mirrored by different chemical stability of the layers plated from different baths. Thus, the surface roughness of the deposits plated from the alkaline bath is significantly higher as compared to those deposited from the electrolyte with the lower pH values (Fig. 4). This indicates that the deposits from the alkaline bath possess a higher contact area between dissolving medium and the nickel surface and contain a higher concentration of low coordination number surface atoms prone to easy dissolution as compared to the layers obtained from the acidic electrolyte. Finally, the deposits obtained from the alkaline bath could have a low adhesion to the substrate.

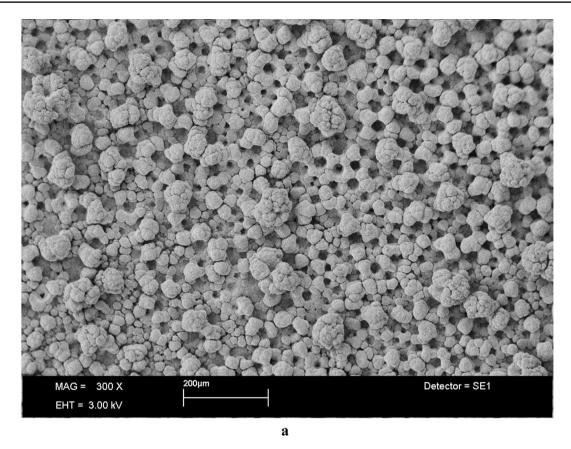
Figures 1, 2, 3, 4 and 5 show that the deposition process strongly depends on the bath composition. It is likely that this effect is caused not only by the pH variation but is related also to differences in the chemical composition of

the acidic and the alkaline bath. Some of the chemicals used for the electrolyte preparation are present only in one type of the bath while chemical forms of those common for both baths may differ due to pH dependent dissociation. Thus, in the alkaline bath, the Ni<sup>2+</sup> exists as an ammonia complex, while in the acidic bath, it forms complexes with water molecules (Ni<sup>2+</sup><sub>aq</sub>), sulphates and/or acetates, depending on the concentration of the complex components [19, 62, 69–72]. A comparison of the spectra recorded in solutions containing sulphates and acetates does not reveal significant differences which point out to similar wavelengths of absorption bands of Ni<sup>2+</sup> complexes formed by these two ligands [62, 72, 73]. Further on, it was reported that UV-Vis signals of Ni<sup>2+</sup> complexes with acetates overlap (Ni(H<sub>2</sub>O)<sub>6</sub>)<sup>2+</sup> bands [72]. Therefore, the identity of Ni<sup>2+</sup> complexes present in the bath cannot be unambiguously resolved on the basis of UV-Vis spectra only.

The complexed cations undergo charge transfer reaction leading to formation of Ni<sup>+</sup> intermediates [57, 69, 74, 75]. The latter are also expected to form complexes with respective agents present in the bath [57, 69, 74, 75]. This leads to a conclusion that the electron transfer during reduction of both Ni<sup>2+</sup> and Ni<sup>+</sup> and accompanying processes, such as adsorption of Ni containing species at the electrode surface, may be strongly affected by the bath composition. The bath chemistry may also influence transport of the electroactive species in the solution since various Ni<sup>2+</sup> complexes may differ in respect to their diffusion coefficients [76]. Finally, different species are expected to be adsorbed at the cathode surface depending on the bath composition. This, in turn, may affect morphology of the deposit [77, 78].

The pH and the bath composition have an impact also on the hydrogen evolution reaction (HER) which takes place simultaneously to the metal deposition. They affect the electrode activity towards HER [79–81] while pH variation also results in a shift of the reversible potential of H<sup>+</sup>/H<sub>2</sub>(H<sub>2</sub>O/ H<sub>2</sub>) red-ox couple in respect to the Ni/Ni<sup>2+</sup> system [77, 82]. These factors not only may influence the faradaic efficiency but also may have an impact on conditions of H<sub>2</sub> bubbles formation. The presence of "wells" seen on SEM image taken for the deposit plated from the alkaline bath (Fig. 4) suggests a strong influence of the H<sub>2</sub> bubbles existence on the deposit morphology. The bubble formation depends on the electrode surface properties, which can be modified by the adsorption phenomena, and on the hydrogen solubility in the electrolyte, which depends on the composition of the latter [83–86]. A complex relation between the hydrogen bubbles adhesion to the electrode and morphology (roughness) and wettability of the surface of the latter is reported in the literature [87]. Finally, bubbles of gases released from both the cathode and the anode may affect transport of the electroactive species in the electrolyte (blocking and/or convection) [88]. It follows then that the thorough analysis of



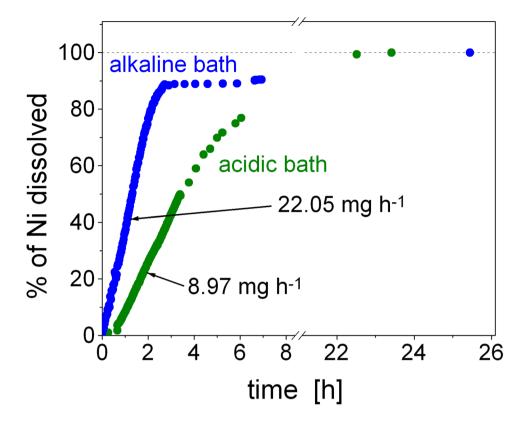


MAG = 300 X EHT = 15.00 kV

Fig. 4 SEM images of the Ni deposits obtained at  $60~\text{mA cm}^{-2}$  and from **a** alkaline and **b** acidic bath. Deposition of more than 98% of Ni, magnification of  $300\times$ 



**Fig. 5** Dissolution of Ni deposits in concentrated HNO<sub>3</sub> at room temperature. The deposits were plated from the alkaline and from the acidic baths at current densities of 60 and 50 mA cm<sup>-2</sup>, respectively



the factors governing the deposit morphology is complex and must consider variation of numerous parameters. At this stage of the studies, as the results are available for two bath compositions only, it is premature to conclude which of the above-mentioned factors is the most important one affecting the deposit morphology.

Figure 6 presents an ED-XRF spectrum recorded for nickel deposited from the acidic and the alkaline baths. It shows that the deposits contain pure nickel. In addition to the above described influence of the bath composition on the deposit properties, it is also worth mentioning that the electrodeposition from the alkaline bath leads to formation of a black deposit on the Pt anode. The same effect is observed also in the absence of  $H_2O_2$ when the deposition is carried out from a bath prepared by dissolution of NiSO<sub>4</sub> instead of the metallic Ni. The deposit dissolves immediately in concentrated HNO<sub>3</sub>, and, as a separate ED-XRF analysis reveals, it contains nickel (Fig. 7). The most likely explanation is formation of NiOOH in an anodic reaction proposed in [89]. An analysis shows that concentration of Ni ions originating from dissolution of this anodic deposit is below UV-Vis detection limit (the same analysis conditions as for the electrodeposition process). It follows then that the amount of Ni deposited on the anode constitutes a meaningless fraction of the overall mass of the metal dissolved in the bath.

## Estimation of H<sub>2</sub>O<sub>2</sub> content in the electrolyte bath

The presence of the hydrogen peroxide in the galvanic bath may affect nickel deposition process. Thus, H<sub>2</sub>O<sub>2</sub> can be electrochemically reduced using solid electrodes, such as Ni [90, 91]. When this reaction takes place parallel to the metal plating, it may reduce the Faradaic efficiency of the electrodeposition process [43]. On the other hand, it was reported that the presence of some amounts of  $H_2O_2$  in the nickel plating bath can be beneficial for elimination of pits in the deposit [92] and may reduce hydrogen uptake by the deposit [67]. As regards the anodic processes, Ni dissolution prevails over passivation in the presence of the peroxide in the electrolyte [93]. This may take place when the Ni electrode is accidentally left in the bath under open circuit conditions. It is then interesting to determine the amount of the H<sub>2</sub>O<sub>2</sub> remaining in the bath after completing the nickel dissolution. Hydrogen peroxide can be catalytically decomposed by Ni<sup>2+</sup> cations [94–97], and the rate of this reaction increases when pH is changed from the acidic to the alkaline one [95]. Because of this reason, as well as due to more promising results of the plating from the acidic bath, we focus our analysis on determination of H<sub>2</sub>O<sub>2</sub> in the latter electrolyte only.

The method applied by us is based on detection of Fe(III) species generated by Fe(II) oxidation with the hydrogen peroxide [98–101]. The detailed scheme of the reactions



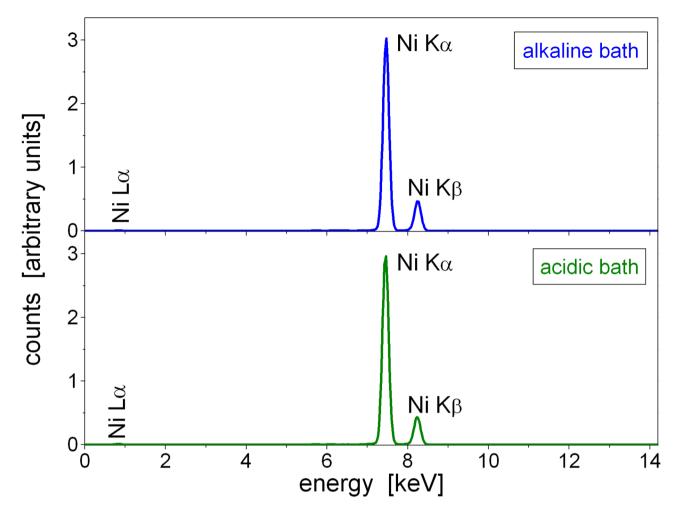


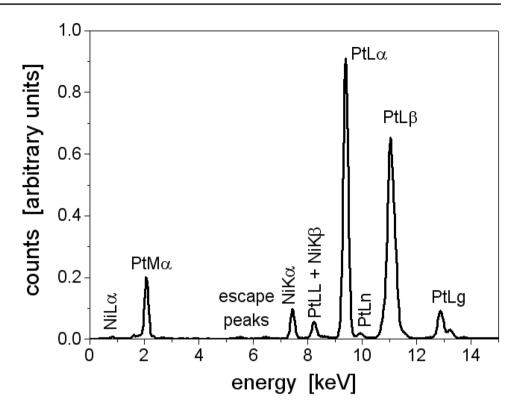
Fig. 6 ED-XRF spectrum of Ni deposited from the alkaline (initial pH = 9.3, top panel) and from the acidic bath (initial pH = 4.3, bottom panel). Current density of  $60 \text{ mA cm}^{-2}$ 

taking place in this process is complex [102–104], especially in the presence of the acetates which may react with generated radicals [105], and its discussion is out of scope of this paper. Due to a complex composition of the bath, we did not add any Fe(III) complexing agents which may improve attenuation coefficient of iron containing species. Test solutions free of H<sub>2</sub>O<sub>2</sub> were prepared by dissolving NiSO<sub>4</sub> instead of the Ni powder. The amount of H<sub>2</sub>SO<sub>4</sub>, H<sub>2</sub>BO<sub>2</sub> and CH<sub>3</sub>COONa added to the test solutions as were as Ni<sup>2+</sup> concentration in the latter were the same as for the acidic electrolyte bath. Such obtained solution was diluted to 6 ml. In the next step,  $0.403 \pm 0.006$  g of FeSO<sub>4</sub>·7H<sub>2</sub>O was added. Fe(II) was in a molar excess in respect to the maximum possible concentration of H<sub>2</sub>O<sub>2</sub> (0.39-0.93 M in respect to the undiluted bath). Further on, addition of such high amount of iron improves ratio of iron to nickel signals in the spectrum. This makes iron signals less affected by the nickel absorption bands. Hydrous FeSO<sub>4</sub> dissolves relatively quickly in the test solution and such prepared mixture remains stable over a long period of time. This is in contrast to other methods of  $\rm H_2O_2$  determination, such as the one based on a reaction with KMnO<sub>4</sub> [106, 107]. We found that the addition of the permanganates to the test solutions leads to immediate precipitation of a deposit even in the absence of the hydrogen peroxide. Therefore, we do not investigate further this approach.

Various amounts of  $\rm H_2O_2$  were added to the test solutions, and respective UV-Vis spectra show how the amount of Fe(III) and Fe(II) varies as a function of the peroxide concentration (Fig. 8). The solutions remain stable for at least several tens of minutes after adding a single portion of  $\rm H_2O_2$ , as follows from the spectra analysis. The spectra reveal out of scale signals due to Fe(II) and Fe(III) below ca. 482 nm [108, 109], strongly overlapping signals at ca. 448 and ca. 510 nm and a wide peak at 970–1005 nm. Nickel signals at 659–721 nm are poorly visible due to dilution of the bath while the peak at ca. 395 nm overlap Fe signals located in this wavelength range, including the



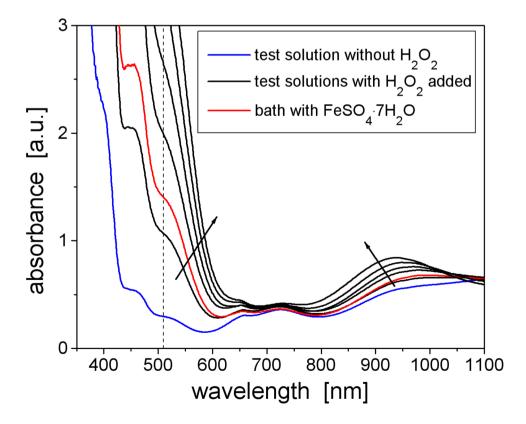
Fig. 7 ED-XRF spectrum of Pt anode after completing Ni deposition from the alkaline bath (initial pH of 9.3) with the current of 60 mA cm<sup>-2</sup>



ones attributed to acetate complexes of Fe(III) and located below ca. 480 nm [110, 111]. Signals at ca. 510 nm and ca. 970–1005 nm are assigned to acetate complexes of Fe(III) [111–113]. Their intensity increases with the  $\rm H_2O_2$ 

content due to generation of Fe(III) in a reaction with the latter. The maximum of the peak at 1005-970 nm shifts towards lower wavelengths with the addition of  $H_2O_2$ , i.e. with the increasing concentration of generated Fe(III). This

Fig. 8 UV-Vis spectra of tests solutions containing 0.403 g of FeSO<sub>4</sub>·7H<sub>2</sub>O and various amounts of  $\rm H_2O_2$  added (from  $7.8\cdot 10^{-5}$  to  $4.7\cdot 10^{-4}$  M per 6 ml of the solution). The arrows indicate evolution of the spectra with increasing amount of added  $\rm H_2O_2$ ; horizontal dotted line indicates location of the signal at 510 nm. A spectrum recorded for diluted acidic bath containing 0.403 g of FeSO<sub>4</sub>·7H<sub>2</sub>O is also shown





effect can be attributed to variation in stoichiometry of the Fe(III)-acetate complexes due to changes in the Fe(III) concentration. For a sufficiently high amount of H<sub>2</sub>O<sub>2</sub> added the signal at 510 nm eventually overlaps other bands which onset shifts to higher wavelength as the amount the peroxide added increases. Thus, due to a complex composition of the bath, as well as complicated shape of the UV-Vis spectra, we apply an analysis aimed to a rough estimation of the peroxide content and we do not made attempts to determine it with high accuracy. We found that only analysis of signals at 510 nm (corrected for baseline value at 610-630 nm) and ca. 970-1005 nm (corrected for arbitrarily selected baseline value at 1100 nm) delivers reasonable results. Respective calibration curves constructed on the basis of UV-Vis spectra recorded for the test solutions are shown in Fig. 9. Regardless on relatively high scatter of the experimental data points, the plots reveal a reasonably good linearity.

The spectra recorded for the test solutions were compared with the electrolyte bath prepared by dissolution of 50-53 g of Ni in a mixture containing 0.1 ml H<sub>2</sub>O<sub>2</sub> and 0.5 ml 3M H<sub>2</sub>SO<sub>4</sub>. The solution was then diluted with water to the volume of 2.1 ml. The added chemicals include CH<sub>3</sub>COONa (setting up pH of 4.3-4.4) and H<sub>3</sub>BO<sub>3</sub> (final concentration of 0.32 M). This was the standard composition of the acidic bath used in the experiments described in the previous chapters. Such prepared bath was subjected to dilution to 6 ml and  $0.403 \pm 0.006$  g of FeSO<sub>4</sub>·7H<sub>2</sub>O was added (red curve in Fig. 8). The bath treatment was simplified as much as possible in order to avoid accidental H<sub>2</sub>O<sub>2</sub> decomposition. Therefore, the electrolyte was subjected only to a slight dilution which does not affect pH significantly. A comparison of the results of analysis of the signal at 510 nm and the band at ca. 970–1005 nm shows that the amount of the peroxide determined using these two approaches differs by less than 15%. Thus, the reported below values are the averages of the results obtained for both these signals. The analysis shows that the estimated H<sub>2</sub>O<sub>2</sub> concentration in the bath varies from 0.057 to 0.070 M for 53 and 50 mg of dissolved Ni, respectively (concentrations given in respect to the undiluted bath). Although  $H_2O_2$  can be consumed in a reaction with the acetates, this process is rather slow in the absence of respective catalysts [114]. This, in connection with the fact that the peroxide is detected in the electrolytic bath, suggests that possible decomposition of H<sub>2</sub>O<sub>2</sub> via processes different than the nickel oxidation/ dissolution is of a minor importance. Thus, ca. 8.6·10<sup>-4</sup> and ca.  $8.3 \cdot 10^{-4}$  moles underwent the reaction with 53 and 50 mg of the metallic nickel powder, respectively. It follows then that the molar ratio of consumed H<sub>2</sub>O<sub>2</sub> to Ni is equal to 0.95. Taking into account that the reaction of H<sub>2</sub>O<sub>2</sub> with Ni(0) may proceed according two pathways with stoichiometries equal to 1:1 (formation of Ni(II)) or 3:2 (formation of Ni(III)) [115], one may conclude that most of the metal is dissolved in a reaction with the peroxide while dissolution through a direct reaction with H<sub>2</sub>SO<sub>4</sub> [116, 117] plays a minor role. Nickel is slowly dissolved in acids, including H<sub>2</sub>SO<sub>4</sub> [116, 118], but this process can be facilitated by presence of atmospheric oxygen dissolved in the solution [117].

Knowing the amount of H<sub>2</sub>O<sub>2</sub> present in the bath, one may estimate influence of the charge of the possible peroxide reduction on the overall faradaic efficiency of the process. It is assumed here that the electroreduction of H<sub>2</sub>O<sub>2</sub> is a two electron process [90, 119-122]). A comparison of the charge required for the complete deposition of Ni with the one needed for reduction of the peroxide remaining in the bath shows that the latter should not exceed 16% of the former. It is also important to note that H<sub>2</sub>O<sub>2</sub> reduction can be significantly retarded at potentials where the electrode surface becomes covered with adsorbed hydrogen [121]. This point, however, cannot be further evaluated because coverage of Ni electrode surface with adsorbed hydrogen at potentials of HER is a matter of debate [123]. Clearly, the peroxide reduction is only slightly responsible for the low Faradaic efficiency depicted in Fig. 3. It is worth to note that  $H_2O_2$  is expected to be decomposed also via catalytic oxidation on the platinum anode surface [119, 121, 124]. This reduces the amount of the peroxide which interacts chemically and electrochemically with the cathode and, consequently, diminishes the H<sub>2</sub>O<sub>2</sub> influence on the deposition process. Thus, the main factors responsible for differences between the faradaic efficiency calculated for the acidic and the alkaline baths are those related to the pH and the bath composition influence on the nickel electrodeposition and on the hydrogen evolution reaction (reversible potential and kinetic parameters). More data are required to conclude which of these two factors governs the faradaic efficiency.

An attempt was made to neutralise  $H_2O_2$  in the acidic bath using its reduction with sulphites [107, 125, 126].  $Na_2SO_3$  was added in a twofold excess over the  $H_2O_2$  concentration. Deposits obtained from this bath are of very poor quality, i.e. brittle, with very rough surface and, as follows from an ED-XRF analysis (not shown), they contain significant amounts of sulphur. Therefore, this approach was no longer considered. Another possibility of the peroxide removal is a catalytic decomposition using metallic Pt [44, 127]. Indeed, such reaction is observed in the system after immersing massive platinum, e.g. wire. However, we did not make further attempts to remove  $H_2O_2$  because quality of the deposits obtained from the acidic baths in the presence of the peroxide was found to be acceptable.



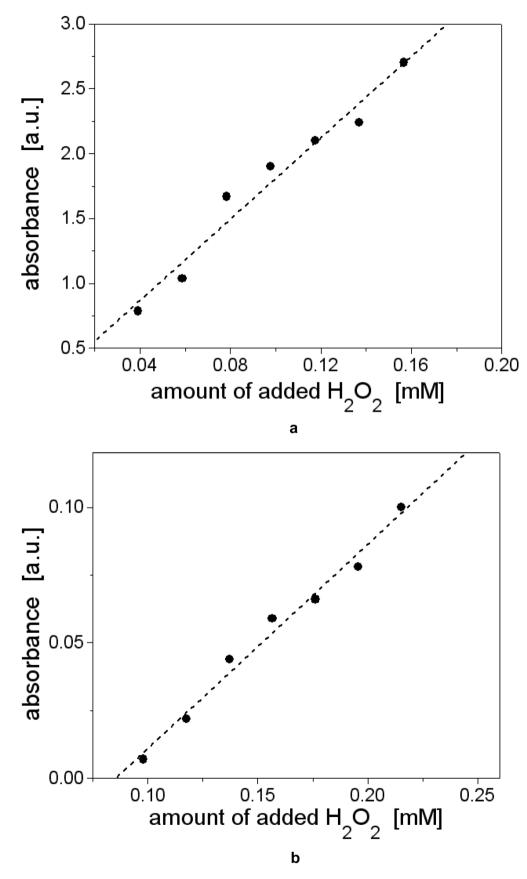


Fig. 9 UV-Vis spectroscopy calibration curves constructed on the basis of spectra recorded for test solutions. a Signal at 510 nm; b signal at 970–1005 nm

#### **Conclusions**

This paper deals with a new method of preparation of the electrolyte baths for nickel electrodeposition with metallic Ni powder used as a starting material. The powder can be quickly dissolved in a HNO<sub>3</sub>-free solution. Such obtained mixture can be used for preparation of nitrates-free plating baths without additional procedures aimed to remove the nitrates. Neutralisation of the solution with NH<sub>4</sub>OH or acetates allows obtaining electrolyte baths with pH ranging from the alkaline to the acidic one. Electrodeposition of more than 99% of ca. 50 mg of Ni dissolved in the bath takes ca. 5 h for the alkaline electrolyte while for the acidic solution, ca. 95% of the metal is deposited within ca. 7 h. The deposits obtained from the acidic baths are more physically and chemically stable than those deposited from the alkaline solution. The surface of the former is crackfree and is smoother as compared to the deposits obtained at higher pH. The described method can be applied for preparation of Ni targets used for accelerator production of radioisotopes for medical applications. The amount of  $H_2O_2$  remaining in the bath can be roughly estimated using a reaction of Fe(II) oxidation and by determination of such formed Fe(III) by means of UV-Vis spectroscopy.

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