



# Equations of state of $\alpha$ -SiC (6H) and $\beta'$ -Mg<sub>2</sub>Si<sub>1,1</sub> from single-crystal X-ray diffraction data and novel high-pressure magnesium silicide Mg<sub>2</sub>Si<sub>7</sub>

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## Abstract

SiC and (Fe, Mg)-silicide are candidate phases forming under reducing conditions in the Earth and planetary interiors. However, structural studies of SiC and Mg<sub>2</sub>Si at high pressure and their thermal stability are presently lacking. In this work, we applied single-crystal X-ray diffraction in a diamond anvil cell at high pressure and determined the equations of state of  $\alpha$ -SiC (6H) and  $\beta'$ -Mg<sub>2</sub>Si<sub>1,1</sub> up to 60 and 40 GPa, respectively, yielding bulk moduli of 226.0(4) and 56(1) GPa. We also report the formation of a novel orthorhombic Mg<sub>2</sub>Si<sub>7</sub> phase upon laser heating  $\beta'$ -Mg<sub>2</sub>Si<sub>1,1</sub> at ~45 GPa and 2000 °C [*Pbam*, *a* = 7.16(1) Å, *b* = 12.490(3) Å, *c* = 2.6545(3) Å, *V* = 237.5(3) Å<sup>3</sup>]. The structure of this compound contains layers formed by irregular 12-member silicon rings, which are arranged in channels filled with both Mg and Si atoms. No signs of the Mg<sub>2</sub>Si<sub>7</sub> phase were detected upon releasing the pressure in the DAC, which suggests that this phase is unstable under ambient conditions.

**Keywords** Single-crystal X-ray diffraction · Diamond anvil cell · Magnesium silicide · Silicon carbide

## Introduction

Silicon carbide (SiC) is a natural material that has attracted great industrial interest owing to its hardness, refractory, and semi-conductor properties. SiC is commonly found in the Solar System in the form of stardust and chondrite meteorites (e.g., Alexander 1993 and references therein), and has been proposed to potentially constitute a significant volume of planetary interiors in carbon-rich star systems (Madhusudhan et al. 2012). SiC is known as the rare mineral moissanite, which has been reported in bulk

kimberlitic chemistry and as inclusions in kimberlitic diamonds (Leung et al. 1990; Shiryayev et al. 2011). Several moissanite grains recovered from ophiolites and kimberlites have been reported to contain inclusions of metallic Si and Fe-silicides (Di Pierro et al. 2003; Trumbull et al. 2009; Shiryayev et al. 2011), and silicides were also found in meteorites (Ross et al. 2019) Because magnesium and silicon are both cosmochemically abundant elements, they are expected to be major components of rocky planets. SiC and (Fe, Mg)-silicide are, therefore, candidates to form Si-bearing minerals under reducing conditions in the Earth and planetary interiors. Additionally, magnesium silicide is considered a promising narrow bandgap semiconductor and has thus been investigated at high pressure in recent years.

SiC<sub>4</sub> tetrahedrons are the main building blocks of SiC crystal structures at ambient pressure. The tetrahedrons are connected at the corners to form layers, and different stacking forms can generate a variety of different SiC polytypes. The most well-studied phases are the zinc-blende type structure (also known as the 3C polytype or  $\beta$ -SiC) and the hexagonal wurtzite-type structure (6H polytype). The 6H polytype together with other hexagonal and rhombohedral structures are also referred to as  $\alpha$ -SiC (Shaffer 1969). Previous high-pressure experimental studies have extensively investigated 3C and 6H polytypes using Brillouin spectroscopy

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and powder X-ray diffraction (XRD), with reported bulk moduli values of approximately 220–260 GPa (e.g., Daviau and Lee 2018 and references therein). Although some of the reported SiC compressibility data are in reasonable agreement, there remain significant differences in the range of reported observations, as summarized in Table 1. The transition of the 6H-SiC polytype to the rock salt B1-type structure is expected to occur at ~100 GPa based on calculations (Eker and Durandurdu 2008). This phase transition is predicted to occur in two stages: 6H-SiC is first compressed along the *c*-direction and then undergoes shear deformation on the *a*-*b* plane. This phase transition was experimentally confirmed during shock compression at ~105 GPa (Sekine and Kobayashi 1997). The phase transition of 3C polytype (a cubic structure with three tetrahedral layers) to the same rock salt structure was experimentally observed using powder XRD in a DAC at ambient temperature and ~100 GPa (Yoshida and Onodera 1993), and at ~60–70 GPa in later works (Miozzi et al. 2018 and references therein). However, studies on crystal structures of SiC and their behaviour at high pressure are presently lacking.

In this work, we studied Mg-bearing silicide endmember. First-principles calculations suggest that cubic Mg<sub>2</sub>Si ( $\beta$ -Mg<sub>2</sub>Si, anti-fluorite type structure) undergoes a reversible phase transition to an orthorhombic phase at ~7 GPa and to a hexagonal polymorph at ~20 GPa (Yu et al. 2010). This prediction has been partially confirmed in several experimental studies involving powder diffraction in a DAC using ethanol-methanol (Strössner et al. 1987; Yoshida and Onodera 1993) or NaCl pressure medium (Wang et al. 2016). But the experimental data are largely inconsistent. Cubic Mg<sub>2</sub>Si has also been shown to decompose to hexagonal Mg<sub>9</sub>Si<sub>5</sub> and Mg at ~800 K and 5 GPa (Gaida et al. 2021). Magnesium silicide is usually assumed to have the composition Mg<sub>2</sub>Si; however, phase compositions can vary depending on the degree of Mg and Si ordering, which may lead to the formation of different crystal structures.

Here we report the first compressibility study on  $\alpha$ -SiC (6H) and  $\beta'$ -Mg<sub>2</sub>Si<sub>1.1</sub> using single-crystal XRD in a diamond anvil cell (DAC) at room temperature up to 60 and 40 GPa,

respectively. We also report decomposition of  $\beta'$ -Mg<sub>2</sub>Si<sub>1.1</sub> at high pressure and high temperature. These materials are candidates to form Si-bearing minerals under reducing conditions in the Earth and planetary interiors.

## Methods

### Sample synthesis

The experimental samples were synthesized using a multi-anvil apparatus. SiC was synthesized using the Iris 1500 ton multi-anvil press with an Osugi-type guide block system at the Bayerisches Geoinstitut (BGI), University of Bayreuth, Germany (Ishii et al. 2016; Ishii et al. 2019). This phase appeared as a side product of phase egg synthesis (AlSiO<sub>3</sub>OH). The starting material consisted of finely ground Al(OH)<sub>3</sub>:SiO<sub>2</sub> = 1:1 placed into a welded Pt capsule. The capsule was inserted in a standard BGI 7/3 assembly, compressed to ~27 GPa, and then heated to 1300 °C and maintained at this temperature for 3 h. The pressure calibration was conducted in separate runs (Liu et al. 2017). Possible scenarios of carbon contamination in the starting material are as follows. (1) CO<sub>2</sub> may be absorbed from the air during sample storage or preparation. (2) The agate mortar used for sample grinding might be contaminated if previously used with carbonates, even after a thorough cleaning with SiO<sub>2</sub>. (3) Carbon may be absorbed from ethanol that did not completely evaporate after sample grinding.

Mg-Si crystals were synthesized from commercial Mg<sub>2</sub>Si powder (99.99% metal basis Alfa Aesar) using the Sumitomo 1200 ton multi-anvil press with a split-sphere guide block system at BGI. The pressure calibration was determined in separate runs (Keppler and Frost 2005). The sample was placed into a welded Pt capsule in a standard BGI 10/4 assembly, compressed to ~20 GPa, and then heated to ~1200 °C and maintained at this temperature for 1 h. The sample was quenched by shutting off the power and then decompressed to ambient pressure.

**Table 1** Experimental equations of state for SiC polytypes

Polytype	Maximum pressure (GPa)	EOS	K <sub>0</sub> (GPa)	K <sub>0</sub> '	Reference
3C, rock-salt	205	BM EOS	224 (5)	4.1(0.3)	Miozzi et al. (2018)
3C	8.1	BM EOS	237 (2)	4 (fixed)	Wang et al. (2016)
3C, 6H	95	BM EOS	260 (9)	2.9(0.3)	Yoshida and Onodera (1993)
6H	68.4	BM EOS	230 (4)	4 (fixed)	Bassett et al. (1993)
6H	27	BM EOS	218 (5)	4.19 (fixed)	Amulele et al. (2004)
6H	60	BM EOS	226.0 (0.4)	4 (fixed)	This study

All previous studies were conducted using powder diffraction in a diamond anvil cell  
*BM EOS* Burch-Murnaghan equation of state

## Sample selection

The recovered crystals were first checked inside a BETSA-type membrane-driven DAC without a pressure medium to ensure that they were pure phases of sufficiently high quality for further experiments. Several crystals were placed in a row inside the DAC and still images were collected by rotating the DAC from  $-5^\circ$  to  $+5^\circ$ . The phases were confirmed by comparing the obtained powder patterns with the patterns generated from CIF files in Dioptas (Prescher and Prakapenka 2015). The crystals with the best reflection quality were chosen for the high-pressure experiments.

## High-pressure DAC experiments

We used a BETSA-type membrane-driven DAC with tungsten carbide seats and Boehler-Almax diamonds with 250 or 300  $\mu\text{m}$  culets. Rhenium gaskets were pre-indented to  $\sim 30$   $\mu\text{m}$  thickness, and a hole of approximately 110  $\mu\text{m}$  in diameter was drilled using a SPI RedPower R4 modulated fiber laser ( $\lambda \sim 1070$  nm) to form a sample chamber. The pre-selected crystals were then placed in the sample chamber, together with a ruby for pressure determination. The cell was loaded with Ne at the European Synchrotron Radiation Facility (ESRF), Grenoble, France to achieve quasi-hydrostatic conditions during compression. Each DAC was equipped with a steel membrane connected to the helium gas bottle via a pump with a pipe, which allowed the pressure to be remotely controlled within the pressure chamber to achieve desirable pressure steps. During compression, the pressure was determined using the shift of the ruby fluorescence lines (Mao et al. 1986).

## Single-crystal X-ray diffraction

Single-crystal XRD data were collected at the ID15B beamline at ESRF. Prior to XRD measurement, the experimental geometry was precisely defined based on a freshly collected diffraction profile from a vanadinite calibrant. The beamline was equipped with a MAR555 flat panel detector for the  $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$  experiment, which was upgraded to an Eiger2 X 9 M CdTe detector with dimensions of  $340 \times 370$  mm for the SiC experiment. The operating conditions included an X-ray energy of 30 keV, the wavelength of 0.4133  $\text{\AA}$ , and beam size of  $\sim 15$   $\mu\text{m}$  at FWHM. Data were collected at each pressure point in the omega range of  $\pm 32^\circ$ , which was defined by opening the membrane cells (Supplementary Fig. 1, Kantor et al. 2013), using a  $0.5^\circ$  step per frame and exposure time of 1 s per step. The integration of the reflection intensities and absorption corrections were performed using CrysAlis<sup>Pro</sup> software (Agilent 2014). The structure solution and refinement were carried out using the isotropic approximation in

Jana2006 (Petříček et al. 2014) with Superflip (Palatinus and Chapis 2007) and SHELXT (Sheldrick 2015).

After compressing  $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$  to the maximum pressure, the samples were heated to 2000 K using a double-sided laser heating system at the ID18 beamline at ESRF (Aprilis et al. 2017). Particular care was taken when analyzing the samples after laser heating because the products often became polycrystalline. We first collected a map of still images around the sample by moving the sample stage, and then selected the best spot for data collection where single-crystal reflections (domains) were observed. After processing, reflections of the phase of interest were manually found using the Ewald explorer in CrysAlis<sup>Pro</sup>; otherwise the data processing routine was similar to that of the unheated samples. Additional experimental detail could be found in Supplementary Table 1.

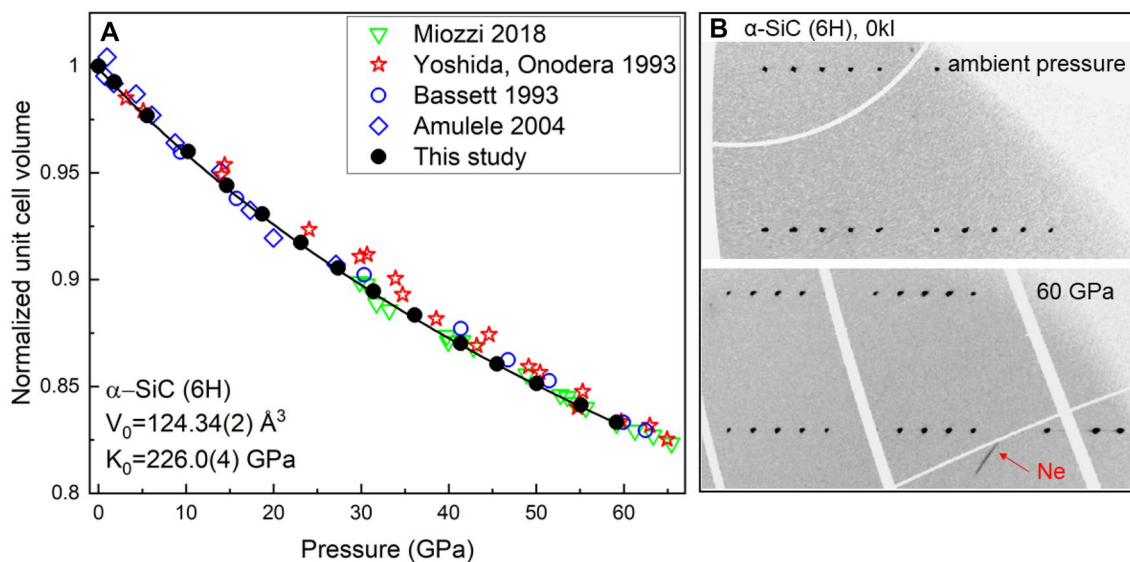
## Results and discussion

### Compressibility of $\alpha$ -SiC (6H)

The structure of the sample studied here consisted of six tetrahedral layers (Fig. 2A, inset) characterized by space group  $P63mc$  (186) with lattice parameters of  $a = 3.0814(8)$   $\text{\AA}$ ,  $c = 15.121(2)$   $\text{\AA}$ , and  $V = 124.34(2)$   $\text{\AA}^3$ , which is indicative of the 6H polytype within the broad class of  $\alpha$ -SiC phases (Shaffer 1969).

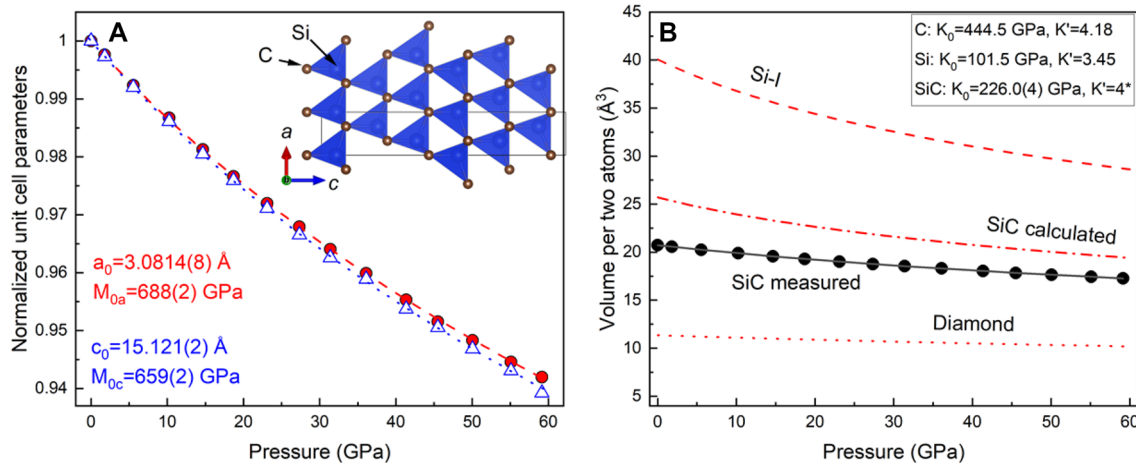
No phase transitions were observed during the compression of  $\alpha$ -SiC (6H) to 60 GPa (Fig. 1A). Reciprocal-space analysis of the SiC crystal at the highest obtained pressure (Fig. 1B) did not reveal any extra reflections or splittings, which confirms that our results are in accordance with the literature data.

Figure 1A shows the unit cell volume as a function of pressure at ambient temperature. Fitting of the pressure–volume data with a second-order Birch–Murnaghan equation of state yields  $K_0 = 226.0(4)$  GPa and  $V_0 = 124.34(2)$   $\text{\AA}^3$ , the latter of which was fixed from the experimental data at ambient conditions. A comparison of our result with previous studies (Table 1 after Daviau and Lee (2018) and references therein) demonstrates good agreement with the pioneer high-pressure XRD diffraction study on the 6H polytype by Bassett et al. (1993) and a more recent study by Miozzi et al. (2018). The single-crystal diffraction data collected here in a quasi-hydrostatic pressure medium provide increased accuracy of the bulk modulus determination compared with previous work, and the new results extend the pressure ranges from 8.1 GPa (Wang et al. 2016) and 27 GPa (Amulele et al. 2004) in previous studies to 60 GPa. Our data do not agree with the compressibility reported by Yoshida and Onodera (1993) who studied a mixture of cubic (3C) and hexagonal



**Fig. 1** **A** Equation of state of  $\alpha$ -SiC (6H). Black solid curve shows the fit of the experimental data (black circles) with a second-order Birch-Murnaghan equation of state with  $V_0 = 124.34(2) \text{ \AA}^3$  and  $K_0 = 226.0(4) \text{ GPa}$ .  $V_0$  was fixed based on the data at ambient conditions. For comparison, we also plot some experimental data for different SiC polytypes (Table 1). See text for discussion. The error bars

are within data points **B** Reflections of hexagonal P63mc  $\alpha$ -SiC (6H) in the 0kl plane of a precession-like image in reciprocal space reconstructed using the UNWARP procedure within CrysAlis<sup>Pro</sup> software. As could be observed from the comparison of diffraction image collected at ambient pressure and at  $\sim 60 \text{ GPa}$ , there is no sign of a phase transformation



**Fig. 2** **A** Normalized unit cell parameters of  $\alpha$ -SiC (6H) fitted with a linear second-order Birch-Murnaghan equation of state as a function of pressure. The compressibility is similar along the  $a$ - and  $c$ -axes with  $M_{0a} = 688(2) \text{ GPa}$  and  $M_{0c} = 659(2) \text{ GPa}$ . **B** Volume of the SiC formula unit in  $\alpha$ -SiC (6H) during compression (black circles, black solid curve). The equations of state for 2C in diamond (red dotted

curve) and 2Si in Si-I with the diamond structure (red dashed curve) are plotted for comparison. The red dash-dot curve represents the SiC volume calculated from Si-I and diamond volumes at a given pressure. Note that Si-I is stable only up to  $\sim 13 \text{ GPa}$ . The error bars are within data points

(6H) phases. The long-range order of different polytypes is not expected to cause such a significant difference ( $\sim 40 \text{ GPa}$ ) of the bulk modulus, thus the discrepancy is likely artificial and may be explained by the different experimental setup, non-hydrostatic conditions, or different pressure calibrants used in that study.

The  $c$ -axis is found to be slightly more compressible ( $M_{0c} = 659(2) \text{ GPa}$ ) than the  $a$ -axis ( $M_{0a} = 688(2) \text{ GPa}$ ) (Figs. 1A, 2). The tetrahedron volume decreased during compression from  $\sim 3.5 \text{ \AA}^3$  at ambient conditions to  $\sim 3 \text{ \AA}^3$  at 50 GPa. The bulk modulus obtained by fitting the SiC<sub>4</sub> tetrahedron volume with a second-order Birch-Murnaghan

equation of state yields  $K_0 = 225.7(4)$  GPa, which is indistinguishable from that of the unit cell volume within the experimental uncertainties. The compressibility of bulk 6H-SiC is, therefore, defined by the compressibility of the Si–C bonds. A comparison of the compressibility in diamond (Dewaele et al. 2008) and Si-I with the diamond structure (Anzellini et al. 2019) shows that SiC has a smaller volume than the calculated average Si and C volumes per atom in the unit cells of diamond and Si-I (Fig. 2B). This may indicate that SiC is thermodynamically favorable with respect to a mixture of Si and C over a wide pressure range.

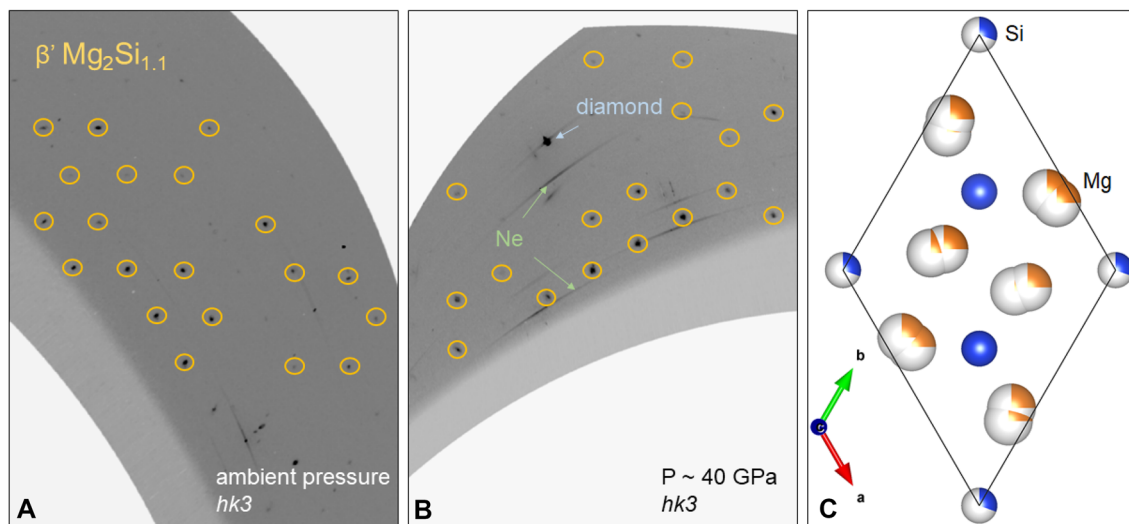
### Compressibility of $\beta'$ -Mg<sub>2</sub>Si<sub>1.1</sub>

Computational studies have predicted the phase transitions of Mg<sub>2</sub>Si from an anti-fluorite type structure (cubic, *Fm-3 m*) to anti-cotunnite (orthorhombic, *Pmna*) at ~6–8 GPa and to a Ni<sub>2</sub>In-type structure (hexagonal P63/mmc) at ~20–24 GPa (Yu et al. 2010; Huan et al. 2016). In contrast, experimental studies have not been entirely consistent and reports have included the coexistence of anti-cotunnite and Ni<sub>2</sub>In-type structured phases (Hao et al. 2009), the presence of hexagonal structures (Cannon and Conlin 1964; Peun et al. 1995), and an unknown monoclinic phase stable above 11.1 GPa (Zhu et al. 2012). This can be explained by the variable chemistry of Mg-Si compounds. Indeed, Huan et al. (2016) computed that Mg<sub>2</sub>Si is stable at low and high pressures and is nearly energetically equivalent to hexagonal Mg<sub>9</sub>Si<sub>5</sub> at 6–24 GPa.

Several studies have explored the stability of Mg-Si hexagonal phases at high pressure. For example, Ji et al. (2013) synthesized hexagonal Mg<sub>9</sub>Si<sub>5</sub> at 5 GPa and 1000 °C [ $a = 12.411(1)$  Å,  $c = 12.345(1)$  Å, space group *P6<sub>3</sub>*], and Gaida et al. (2021) synthesized the same phase at pressures up to 18 GPa and 1200 °C, which is close to our synthesis conditions. Both studies used a mixture of Mg and Si as the starting material.

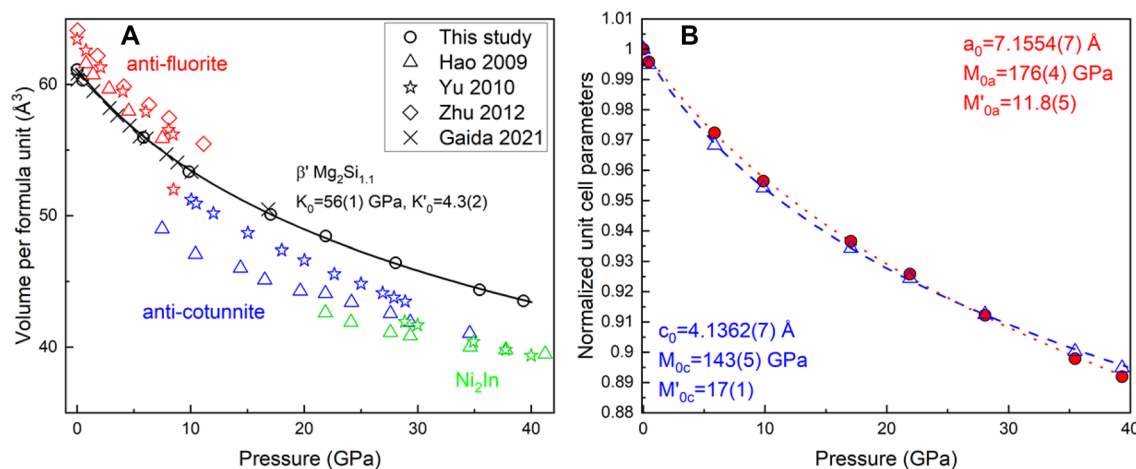
In this study, we synthesized another Mg-Si hexagonal phase that was recovered after the multi-anvil experiment. The sample is hexagonal with lattice parameters of  $a = 7.1554(7)$  Å and  $c = 4.1362(7)$  Å, which correspond to previously reported  $\beta'$ -Mg<sub>2</sub>Si<sub>1.1</sub> that has been observed during the precipitation hardening of Al–Mg–Si alloys (Andersen et al. 1998; Vissers et al. 2007). In this structure, Si atoms partly occupy edge positions, two Si atoms are located inside the unit cell at (2/3, 1/3, 1/4), and Mg atoms are placed inside the unit cell and can be described either as one Mg in a fully occupied site or two Mg atoms located close to each other with partial occupation (Fig. 3C). Previous reports on this phase described samples with a needle-like shape and size of 2–200 nm. In contrast, the high-pressure synthesis in a multi-anvil press at 20 GPa and 1200 °C applied in this study facilitated crystal growth up to ~10–20 μm in lateral dimensions.

The compression of  $\beta'$ -Mg<sub>2</sub>Si<sub>1.1</sub> up to 40 GPa in a quasi-hydrostatic Ne pressure medium did not result in any phase transformations (Fig. 3). The lattice parameters were accurately determined based on at least 170



**Fig. 3** **A** Reflections of hexagonal  $\beta'$ -Mg<sub>2</sub>Si<sub>1.1</sub> (in yellow circles) in the *hk3* plane of a precession-like image of reciprocal space, reconstructed using the UNWARP procedure of CrysAlis<sup>Pro</sup> software at ambient pressure inside a DAC prior to Ne gas loading, and **B** at ~40 GPa. Patterns show no sign of a phase transition, as no new single-crystal reflections are observed. **C** Structure of hexagonal

$\beta'$ -Mg<sub>2</sub>Si<sub>1.1</sub> at ambient conditions.  $R_{\text{int}} = 5.5\%$ ,  $R_1 = 7.8\%$ . Si atoms are marked as blue, Mg atoms are yellow. Atomic positions ( $x, y, z$ , occupancy): Si1 (1/3, 2/3, 1/4, 1), Si2 [0, 0, 0.3087(10), 0.306(8)], Mg1 [0.6829(9), 0.5970(12), 0.743(7), 0.355(14)], Mg2 (0.730(2), 0.664(3), 0.738(10), 0.145(13)]



**Fig. 4** **A** Compressibility of  $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$ . Fitting of experimental data (black circles) with a third-order Birch-Murnaghan equation of state (black solid curve) yielding  $K_0=56(1)$  and  $K'_0=4.3(2)$ .  $V_0$  was fixed based on the data at ambient conditions. For comparison, we show the volume per formula unit for anti-fluorite  $\text{Mg}_2\text{Si}$  (red), anti-cotunnite (blue), and  $\text{Ni}_2\text{In}$ -type  $\text{Mg}_2\text{Si}$  (green), in addition to hexagonal  $\text{Mg}_9\text{Si}_5$  (black crosses). Data from Yu et al. (2010) are based on calculations. **B** Normalized unit cell parameters of  $\beta'$ - $\text{Mg-Si}$  as a function of pressure. The error bars are within data points

single-crystal reflections, showing a monotonic and anomaly-free decrease with pressure (Fig. 4A). However, a refinement of the atomic coordinates was not feasible owing to the structure complexity and limited diffraction data quality. These observations deviate from the previously reported high-pressure behavior of stoichiometric  $\text{Mg}_2\text{Si}$ , which undergoes multiple phase transitions in the pressure range up to 40 GPa (Fig. 4A).

Fitting the pressure–volume data for  $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$  with a third-order Birch-Murnaghan equation of state yields  $K_0=56(1)$  GPa,  $K'=4.3(2)$ , and  $V_0=183.40(4)$  Å<sup>3</sup> (Supplementary Fig. 2). The compressibility of  $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$  is therefore similar to that of cubic anti-fluorite  $\text{Mg}_2\text{Si}$  ( $K_0=57.03(2)$  GPa,  $K'=4$  (fixed),  $V_0=62.34(1)$  Å<sup>3</sup>; Hao et al. 2009). Our data are also in good agreement with the compressibility of hexagonal  $\text{Mg}_9\text{Si}_5$  recently reported by Gaida et al. (2021) [ $K_0=58(3)$  GPa,  $K'=4.1(7)$ ,  $V_0=1643(2)$  Å<sup>3</sup>].

The reported bulk moduli of  $\text{Mg}_2\text{Si}$  high-pressure phases are significantly higher than that determined here for  $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$ . For example, anti-cotunnite  $\text{Mg}_2\text{Si}$  has  $K_0=102.65(5)$  and  $\text{Ni}_2\text{In}$ -type  $\text{Mg}_2\text{Si}$  has  $K_0=163.83(10)$  (Hao et al. 2009). We demonstrate that  $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$  does not undergo phase transitions as does cubic  $\text{Mg}_2\text{Si}$ , and is considerably more compressible than the high-pressure forms of the latter phase. There are several possible reasons for this divergent behavior: (1) different transformational pathways for different starting materials, which is a well-known phenomenon in silicon dioxide (Bykova

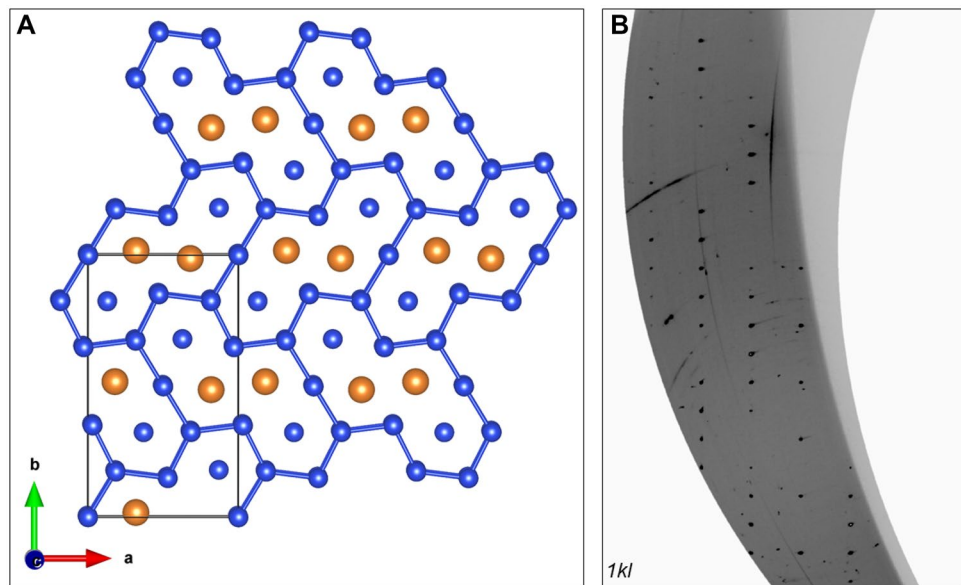
et al. 2018); (2) differing chemical compositions of the materials (i.e.  $\text{Mg/Si}$  ratio); and (3) different experimental conditions, particularly the hydrostaticity of the pressure medium. Our results thus highlight the need for further investigation of  $\text{Mg}_2\text{Si}$  polymorphs at high pressure using modern single-crystal X-ray diffraction methods.

### Decomposition of $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$ at high temperature and novel $\text{Mg}_2\text{Si}_7$

After laser heating to 2000 K at  $\sim 45$  GPa,  $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$  decomposed and formed a novel  $\text{Mg}_2\text{Si}_7$  phase (Figs. 4, 5), which is orthorhombic ( $Pbam$ , 55) with  $a=7.16(1)$  Å,  $b=12.490(3)$  Å,  $c=2.6545(3)$  Å, and  $V=237.5(3)$  Å<sup>3</sup>. The structure of this phase was solved and refined to  $R_1=0.08$  based on 96 reflections (see supplementary CIF file). Even if the X-ray scattering factors of Mg and Si are similar, the quality of the diffraction data allows an unambiguous assignment of the structural positions to determine the composition of the phase as  $\text{Mg}_2\text{Si}_7$ .

The remarkable features of this phase include its layered structure, which consists of layers formed by irregular 12-member silicon rings. The rings are arranged in channels filled with both Mg and Si atoms (Fig. 5). The bond length of the Si–Si forming channels is  $\sim 2.55$  Å. The distance from a Si atom inside the channel to the closest Si atom is  $\sim 2.61$  Å. A comparison of the known bond lengths at ambient conditions show that this bond length is substantially larger than the double Si–Si bond length

**Fig. 5** **A** Novel high-pressure  $\text{Mg}_2\text{Si}_7$  phase. A Z-shaped framework with Si atoms creates infinite channels filled with Mg and Si atoms. Space group  $P6_3/m$ , (55) with  $a = 7.16(1)$  Å,  $b = 12.490(3)$  Å,  $c = 2.6545(3)$  Å, and  $V = 237.5(3)$  Å<sup>3</sup>. VESTA software was used to generate the structure image. **B** Reflections of the novel  $\text{Mg}_2\text{Si}_7$  phase synthesized after laser heating to  $\sim 2000$  K in the  $1kl$  plane of a precession-like image of reciprocal space. From the reflections arrangement in comparison with Fig. 3A and B one could note, that hexagonal  $\text{Mg}_2\text{Si}_{1.1}$  transformed to the phase with orthorhombic symmetry



in silenes ( $\sim 2.14$  Å) and normal Si–Si bonds ( $\sim 2.3$  Å), but also shorter than the longest known Si–Si  $\sigma$  bond ( $\sim 2.7$  Å) and recently reported single  $\pi$  bond ( $\sim 2.8$  Å) (Kyushin et al. 2020). The obtained distance is similar to that between atoms in neighboring hexagonal layers of hcp Si in high-pressure Si polymorphs (2.5344 Å at 42.5 GPa; Hanfland et al. 1999) and in hexagonal  $\text{Mg}_9\text{Si}_5$  ( $\sim 2.5$  Å; Gaida et al. 2021). The Mg–Mg distance in the novel  $\text{Mg}_2\text{Si}_7$  structure is  $\sim 2.61$ – $2.65$  Å, which is significantly shorter than that in  $\text{Mg}_9\text{Si}_5$  (3.1 Å). The shortest Mg–Si distance in the novel  $\text{Mg}_2\text{Si}_7$  phase is  $\sim 2.6$  Å, which is longer than that in  $\text{Mg}_9\text{Si}_5$  (2.42 Å). The Si atoms located inside the channels are 12-fold coordinated with Si chain-forming atoms and an average bond length  $\sim 2.7$  Å.

No signs of the  $\text{Mg}_2\text{Si}_7$  phase were detected upon releasing the pressure in the DAC, which suggests that this phase is unstable at ambient conditions.

As silicides are known in meteorites, and magnesium and silicon are cosmochemically abundant elements, one can hypothesize magnesium silicide(s) (or solid solutions based on Mg–Si phases) may be present in the interior of reduced rocky planets. Our results suggest that the high-pressure behavior of Mg–Si is complex and warrant further investigations.

## Conclusions

We report the equations of state for  $\alpha$ -SiC (6H) up to 60 GPa based on high-precision single-crystal XRD in a DAC. We also studied the compressibility of  $\beta'$ - $\text{Mg}_2\text{Si}_{1.1}$  up to 40 GPa and report a novel  $\text{Mg}_2\text{Si}_7$  structure consisting of layers formed by irregular 12-member silicon rings that contain channels filled with both Mg and Si atoms.

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## Declarations

**Conflict of interest** The authors declare that they have no conflict of interest. I.K and L.D. were funded by DFG 939/9-2 and DFG 939/13-2.

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