

ording to [68Ray]. Although the crystal structure of Be_5Nb was reported to be CaCu_5 (D_{2d}) type [77Ray], the lattice parameters were unknown.

[92Bri1] and [92Bri2] examined decomposition processes of sputter-deposited Be-Nb alloys containing 3 to 14 at.% Nb at temperatures up to 450 °C by means of XRD, SEM, etc. No evidence of the Be_5Nb phase was found, and $\text{Be}_{17}\text{Nb}_2$ was stable to room temperature. Accordingly, [92Bri1] and [92Bri2] preferred the diagram of [Massalski1] in this composition range. A modified Be-Nb phase diagram is shown in Fig. 1. In addition, [92Bri1] observed a homogeneity range in Be_{12}Nb from -5.5 to 7.8, which is probably due to vacancies on the Nb sublattice.

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Cr-Nb (Chromium-Niobium)

H. Okamoto

The Cr-Nb phase diagram in [Massalski2] was redrawn from that of [86Ven], which was primarily based on the work of [61Gol]. The (Nb) liquidus was drawn with a steep slope ($>10\,000$ °C/100 at.%), which is very unlikely [91Oka].

Figure 1 shows the Cr-Nb phase diagram revised by [92Tho] by means of XRD and chemical analyses of equilibrated samples. Invariant reaction temperatures and solid solution ranges have been determined more quantitatively.

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Gd-Mg (Gadolinium-Magnesium)

H. Okamoto

The Mg-Gd phase diagram in [Massalski2] was redrawn from [88Nay]. In this diagram, the Mg solidus, $(\beta\text{Gd})/(\beta\text{Gd}) + \text{MgGd}$, and $(\beta\text{Gd}) + (\alpha\text{Gd})/(\alpha\text{Gd})$ boundaries were speculative. According to [91Oka], the liquidus curvatures of neighboring compounds cannot be very different. Hence, the liquidus of Mg_2Gd appeared to be too flat in the diagram of [88Nay].

These problems are eliminated in the Mg-Gd phase diagram calculated by [92Cac]. However, the calculated boundaries were not in good agreement with experimental data along the (Mg) solvus and $(\alpha\text{Gd})/(\alpha\text{Gd}) + \text{MgGd}$ boundaries. Therefore, the Mg-Gd phase diagram shown in Fig. 1 is drawn by combining appropriate portions of [88Nay] and [92Cac].

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