#### **ORIGINAL PAPER**



# Comparative investigation on the mechanical behavior of injection molded and 3D-printed thermoplastic polyurethane

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Received: 25 September 2020 / Accepted: 3 March 2021 / Published online: 26 March 2021 © The Author(s) 2021

### Abstract

3D printing opens up new possibilities for the production of polymeric structures that would not be possible with injection molding. However, it is known that the manufacturing method might have an impact on the mechanical properties of manufactured components. To this end, the mechanical behavior of test specimens made of thermoplastic polyurethane is compared for two different manufacturing methods. In particular, the SEAM technology (screw extrusion additive manufacturing) is compared to a conventional injection molding process. Uniaxial tension test specimens from both manufacturing methods are analyzed in two testing sequences (multi-hysteresis tests to analyze inelastic properties and uniaxial tension until rupture). To get as less perturbation as possible, the 3D-printed samples are printed with only one strand per layer. Moreover, a correction approach based on optical measurements is applied to determine the true cross-sectional area of the test specimens. The mechanical tests reveal that the inelastic material behavior is the same for both manufacturing methods. Instead, 3D-printed specimens show lower maximal stretch values at rupture and an increased variance in the results, which is related to the surface structure of 3D-printed specimens.

Keywords 3D printing  $\cdot$  TPU  $\cdot$  Build orientation  $\cdot$  Material testing  $\cdot$  SEAM  $\cdot$  MORPH

Dr Michael Johlitz, Guest Editor of this Special Issue [New Challenges and Methods in Experimental Investigations and Modelling of Elastomers] confirms that where he was co-author of a research article in this Special Issue, he was not involved in either the peer review or the decision-making process of that particular article.

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# Introduction

Injection molding is an established manufacturing procedure that has been continuously developed over many years of research. Nevertheless, there are limits to this process that are difficult or impossible to overcome. These are, for example, restrictions on the manufacturability of complex structures and the choice of the internal structure of a volumetric component. In contrast, 3D printing is a comparably slow process, but it can solve both of these problems better than injection molding. With this technology, the production of more complex geometries is feasible and the inner structure can easily be controlled by the help of infill strategies. However, the resulting material and component properties of 3D-printed and injection molded structures can differ, due to the different manufacturing methods. This results in a versatile field of research.

The printing process is controlled by numerous parameters, all of which have a significant impact on the resulting quality of printed parts, in terms of print quality and mechanical behavior. Some of these control parameters are: printing speed, printing temperature, bed adhesion and cooling. Investigations on the impact of printing parameters on the properties of manufactured parts are for example described in [2, 7, 8, 16].

In contrast to injection molding, where produced parts are always completely filled with material, 3D printing enables the use of infill strategies. This means that the interior of the parts does not necessarily have to be completely filled with material. Such a strategy saves material and contributes to the concept of lightweight construction. However, it also weakens the product against mechanical loading. The influences of infill strategies and percentages of infill volumes were for example investigated in [10] and [13]. But even a perfect 3D-printed part will always be different to an injection molded one because of the impact of build orientations resulting from layer-wise printing [1, 6, 14, 19].

There is a wide spectrum of printable materials, such as metals, ceramics and polymers, which can be printed by a variety of different additive manufacturing methods. This paper focuses on a polymeric material processed by an additive manufacturing process similar to FDM (fused deposition modeling). More specifically, a thermoplastic polyurethane (TPU) with a shore hardness of 86A is considered, which is a challenging material for 3D printing. Similar materials and their behavior have for example been investigated in [15, 17] and [18]. However, conventional 3D printers with a slicing software are used in those works. This results in samples that are thicker than one strand per layer. To reduce the directional dependencies of mechanical properties, a slicing software varies the building orientation from layer to layer. Thus, it is not possible to get unbiased information about directional influences of the material. To overcome this problem, 3D printing with the SEAM (screw extrusion additive manufacturing) technology is employed in this study. With this technology, it is possible to print big single-layer parts, from which samples can be cut out for mechanical tests in different directions with respect to the building direction. Thus, it is possible to obtain information about the material itself and the influence of build orientation resulting from the SEAM process.

Main objective of this study is the evaluation and comparison of mechanical properties of TPU samples produced by injection molding and by the SEAM technology. Here, special attention is paid to inelastic properties and directional dependencies due to the building process in additive manufacturing. To this end, "Manufacturing methods" gives an overview on the employed material and manufacturing methods. The "Experiments" then describes the experiments for mechanical testing. This includes uniaxial tension tests with two different loading sequences. First, a multihysteresis loading sequence up to moderate stretch levels is used to analyze the dependencies of the inelastic behavior on building direction. Second, monotonic uniaxial tension until rupture is used to analyse maximum strains at break as a first indication for adhesion properties between the layers. Beyond that, a correction procedure for determining the true cross-sectional area area is presented, which takes into account non-planar sample surfaces (resulting from the processes of printing and cutting). Finally, the results of multihysteresis and monotonic loading tests are summarized in the "Evaluation of the experiments".

## **Materials and methods**

#### Material

The investigated material is the TPU Desmopan<sup>®</sup> 487 from Covestro AG, Germany [9]. The material is delivered in form of pellets that can be processed within extrusion-based manufacturing methods like injection molding. The final material shows rubber-like material behavior, has a shore hardness of 86 A and a glass temperature of approximately  $-35 \,^{\circ}C$  [5]. Moreover, beneficial properties are grease and oil-resistance, low compression set and a good heat resistance [9].

#### Manufacturing methods

Two different extrusion-based manufacturing methods are applied to produce test specimens for mechanical testing. The next section gives a general overview of the SEAM process for 3D printing and explains the applied approach to produce test specimens. The second manufacturing method is conventional injection molding, which is regarded in the next following section.

#### SEAM process for 3D printing

The SEAM process is a very fast 3D-printing process with printing speeds up to 1 m/s (this exceeds most standard FDM printers up to 8 times) and a volume flow controlled output rate of up to 7 kg/h (see for example [4, 11]). The processing space allows to build pieces up to  $1100 \times 800 \times 600 \text{ mm}^3$ . It is, therefore, well suited for printing large-scale objects. This is only possible because of an extrusion based feeding system where pre-dried thermoplastic pellets are automatically fed into an extruder. Due to the applicable melting temperature of up to  $400 \,^\circ$ C, nearly all thermoplastics with and without fiber-reinforcement, high-performance thermoplastics and thermoplastic elastomers can be used. Due to the swivel angle range of  $\pm 45^\circ$  and torsion angle range of  $\pm 20^\circ$ , it is possible to print with real 5 axes. This allows the



Fig. 1 3D-printed specimen and definition of the cutting angles for the experiments

production of load path adapted component structures as well the integration of an endless fiber in the process.

In the current study, the part depicted in Fig. 1 was manufactured. The outer dimensions are  $262.50 \times 90 \text{ mm}^2$  and the outer radius is 22.50 mm. The part was printed with two different thicknesses (2 and 3 mm). Both versions can be printed with only one strand per layer. Thereby, printing layers were added in *z*-direction up to a total height of 150 mm.

During production, the building platform moved in three different spatial directions while the printing head remained fixed. The molten polymer comes out of the printing nozzle and is deposited on the platform continuously layer by layer. As cooling system, two nozzles with compressed air are installed. Thereby, cooling can be controlled by the air flow rate.

The investigated TPU was dried before printing, so that the material had residual moisture content of about 0.13 %. The printing temperature of the material was 220 °C and the melt pressure ranged between 50 and 60 bar. The screw speed was set to 20 rpm and the layer height was 0.7 mm. The table speed to achieve 2 mm and 3 mm thick walls was 112 and 70 mm/s, respectively. The air flow rate was set to 18 m/s.

The printed parts were then used to cut out S2 tension test specimens (total length 75 mm) by the help of a die cutter. Thereby, specimens were cut from the part in three different orientations  $(0^{\circ}, 45^{\circ} \text{ and } 90^{\circ})$  with respect to the building direction to investigate the impact of printing direction on the mechanical properties. Figure 1 depicts three different tension test specimens with varying cutting angles as well as the corresponding cutting orientation on the printed part. The cross-section of the specimens does not show a perfect rectangular shape. This is due to printing (grooves on the outer surfaces, see Fig. 1, right) and the cutting process. An approach to evaluate the true cross-sectional area is introduced in "Correction of cross-sectional area".

Table 1 Production parameters for injection molding

Plate thickness	2 mm	3 mm
Injection time	1.54 s	1.8 s
Holding pressure	450 bar	600 bai
Holding time	7 s	10 s
Plastication time	11 s	20 s
Residual cooling time	25 s	25 s

#### Injection molding

Injection molding was used to produce further uniaxial test specimens that can be compared to those resulting from the 3D-printing process presented in the previous section. To this end, plates with dimensions  $150 \times 100 \text{ mm}^2$  and thicknesses of 2 and 3 mm were produced on a *Battenfeld* injection molding machine (model HM 110/525). Before injection molding, the granulate was dried (residual moisture content 0.08 %). The screw speed was set to 100 rpm and the production was 25 mm<sup>3</sup>/s. Further production parameters are listed in Table 1.

In accordance with the procedure described in the previous section, S2 tension test specimens were cut from the injection molded plates by die cutting. Due to the cutting process, the cross-sectional shape of the specimens was not perfectly rectangular, what is regarded in "Correction of cross-sectional area".

#### Experiments

From the experimental point of view, the main goal of this study is the investigation of the impact of different manufacturing processes on the mechanical behavior. In addition, differences that might arise from different orientations of 3D printing and loading are investigated. To this end, 3D-printed specimens with three different cutting angles with respect to the building direction  $(0^\circ, 45^\circ, 90^\circ)$  are regarded and compared with injection molded S2 samples. This procedure is applied to specimens with two different thicknesses (2 and 3 mm). Thus, altogether eight different specimen groups are analyzed. Mechanical tests include two different testing sequences in uniaxial tension. The first sequence is a multi-hysteresis test with stretch amplitudes up to 2. Second, uniaxial tension until rupture of the samples is regarded. In the following, first the experimental setup is introduced in the next section. Next, the mentioned issue of non-perfect rectangular cross-sectional shapes is regarded in the next following section. The results of the test are provided in the fourth section.

#### **Experimental setup**

Uniaxial tension tests were carried out at room temperature (23°C) on a uniaxial testing machine from Zwick/Roell with a maximal load of 20 kN. A 1 kN load cell was installed to measure the comparably small force resulting from the soft TPU. The applied methods for strain measurement depend on the specific testing sequence.

The testing machine has a built-in clamping device (*Mul-tiXtens*) for strain measurement, which is used for the multihysteresis tests. At the start of the test, two clamps are placed on the sample at a pre-defined distance from each other. The strain is calculated from the change of the distance divided by the original distance. The benefit of this method is the possible actuation of the strain, which is used to load the specimens to specific strain levels. To avoid buckling of the specimens, unloading stops at zero force, which is not at zero strain because of permanent set.

The strain measurement device *MultiXtens* cannot be used for the tension tests until rupture. This is due to a limited measurement range and to secure the device, since a snap back due to rupture could possibly damage it. Therefore, a contact free method is used. Thereby, white marker points on the specimens are tracked by the help of a camera. Subsequently, the distances of the points are evaluated by image processing in MATLAB. The method to calculate strains from the distances remains the same as for the *MultiXtens*.

#### **Correction of cross-sectional area**

Engineering stress in uniaxial tension is not measured directly, but calculated by the quotient from tensile force and the initial cross-sectional area of the specimen in the measuring zone. Measuring the exact cross-section turns out to be difficult due to two reasons: first, the layer-wise 3D-printing process generates corrugated surfaces, i.e., surfaces with semicircles representing the different layers (see



**Fig. 2** Cross-section of a 3D-printed specimen with cutting angle  $0^{\circ}$  (left) and an injection molded specimen (right). In addition, length measures for the calculation of the cross-sectional area are provided

Fig. 2, left). And second, a trapezium shape on the left and right edges results from the cutting process. This appears in samples of both manufacturing methods (see Fig. 2).

The most common way to measure the initial cross-sectional area of S2 samples is to use a caliper gauge, which can only measure parallel edges. With this method, width b and height h can be determined as given in Fig. 2 and the cross-sectional area can be calculated by

$$A_c = h \cdot b \ . \tag{1}$$

However, it can be seen that this measuring method would not be sufficient to gather the true cross-sectional area. To regard both mentioned reasons for deviations from a perfect rectangular shape, two different correction factors  $\lambda_t$  and  $\lambda_s$ are introduced and determined by the help of microscope images of the cross-sections as given in Fig. 2. Note that this precise evaluation of the cross-section area would not be feasible for the specimens to be tested in experiments, since a destruction of them would be needed. A measurement after the tests would also be insufficient, because of inelastic deformation and the corresponding permanent set.

The first correction factor  $\lambda_t$  only regards the trapezium shape. Based on the dimensions depicted in Fig. 2, it is calculated by

$$\lambda_{t} = \frac{A_{t}}{A_{c}} \quad \text{with } A_{t} = \frac{h}{2}(b_{t} + b) .$$
<sup>(2)</sup>

Therein,  $A_c$  is the area measured by the caliper gauge according to 1 and  $A_t$  is the area of the trapezium. The width  $b_t$  is evaluated for the different specimen groups (i.e., different print orientations and thicknesses) by the help of microscope images. The second correction factor  $\lambda_s$  regards deviations

 Table 2
 Comparison of the different cross-sectional area correction factors for different specimen groups

	$\lambda_{\mathrm{t}}$	$\lambda_{\rm s}$
3 mm		
0°, 45°, 90°	0.92	0.93
Inj. mold.	0.92	1
2 mm		
0°, 45°, 90°	0.95	0.90
Inj. mold.	0.95	1

resulting from the semicircles in the 3D-printed specimens. It is calculated by

$$\lambda_{\rm s} = \frac{A_{\rm r}}{A_{\rm t}} \,, \tag{3}$$

with  $A_r$  being the true surface area that is measured by microscope images.

The results of the cross-section evaluation are summarized in Table 2. It can be seen that neither the trapezium nor the semicircle correction factor is constant for both thicknesses. This can be explained with the different height to width proportions. Smaller specimens are less influenced by the cutting process, but higher affected from the 3D-printed semicircles, which have nearly the same radius for 2 mm and 3 mm thick samples. A correction due to semicircles is not necessary for injection molded samples. Thus,  $\lambda_s = 1$  for the corresponding groups.

Based on the evaluated correction factors, the procedure to calculate the cross-sectional area of specimens to be mechanically tested is as follows. First, the outer dimensions are determined by a caliper gauge. Next, the virtual rectangular surface is calculated by Eq. (1) and the correction factors of Table 2 are applied to calculate the true surface area  $A_0$  via

$$A_0 = A_c \cdot \lambda_t \cdot \lambda_s . \tag{4}$$

#### **Evaluation of the experiments**

#### **Multi-hysteresis tests**

Multi-hysteresis tests were conducted on multiple samples of each specimen group (eight samples for each  $0^{\circ}$ ,  $45^{\circ}$  and  $90^{\circ}$  group; three samples for each injection molded group). The test sequence includes loading up to pre-defined stretch levels (1.2, 1.4, 1.6, 1.8, 2.0) with intermediate unloading to zero force. All amplitude levels were approached three times in a row before the next strain amplitude was set. The tests were carried out at a constant absolute value for the crosshead speed of 50 mm/min.



Fig. 3 Stress–stretch curves of all injection molded and printed  $90^{\circ}$  samples with a thickness of 2 mm

One example for the resulting engineering stress vs. stretch curves can be seen in Fig. 3. The picture shows injection molded samples (black lines) and 3D-printed samples with 90° cutting angle (gray lines) with a thickness of 2 mm. The stress–stretch curves clearly show all common rubber-like properties, such as Mullins effect, hysteresis and permanent set. Moreover, it can be seen that when using the correction factors from "Correction of cross-sectional area", there is no significant difference between 3D-printed and injection molded specimens. The same holds for the 3 mm specimens in Fig. 4.



Fig. 4 Stress–stretch curves of all injection molded and printed 90° samples with a thickness of 3 mm



Fig. 5 Maximum stress during multi-hysteresis tests for samples with a thickness of 2 mm



Fig. 6 Maximum stress during multi-hysteresis tests for samples with a thickness of 3 mm

All other results are combined in Figs. 5 and 6. The graphs show the mean values of the maximum stresses of all samples in each specimen group and the standard deviation (square root of the variance) of each group as error bars. The maximum stress is obtained when the test sequence reaches the maximum stretch ( $\lambda = 2$ ) for the first time.

Moreover, Figs. 5 and 6 also include a comparison of maximum stress values for the cases of uncorrected and corrected cross-sectional areas of the specimens. The correspondence of the mean values for the maximum stresses between 3D-printed and injection molded specimens gets clearly better with the use of the correction factors. The lowest average maximum stress for 2 and 3 mm can be



Fig. 7 Stretch curves for 3 mm thick samples for monotonic loading tests until rupture



Fig. 8 Maximum stretch values at rupture for 3 mm thick samples given as mean values (dots) and standard deviation (error bars)

seen at  $45^{\circ}$  cutting angle, the highest at  $90^{\circ}$  for 2 mm and  $0^{\circ}$  for 3 mm. Therefore, no connection can be made between the cutting angle and the maximum stress. The same applies for the standard deviation of the tests. For 2 mm thick specimens, the standard deviation is the lowest at  $0^{\circ}$  and the highest at  $45^{\circ}$ , but for 3 mm specimens, the lowest standard deviation is at  $90^{\circ}$  and the highest at  $0^{\circ}$ .

In summary, it can be stated that the production process of 3D printing does not adversely affect the material properties of the TPU. Only the variance of the specimens increases due to the manufacturing process, which might be attributed to properties of layer adhesion or to notch effects due to corrugated surfaces.

#### Uniaxial tension until rupture

Tension tests until rupture were carried out with three samples of each specimen group. The test includes monotonic loading with a constant cross-head speed of 50 mm/min until rupture of the sample. Examples of each group with 3 mm thickness can be seen in the force-stretch-diagram in Fig. 7. The remaining test results are combined in Fig. 8, which shows the average maximum stretch values for all cutting angles with the standard deviation (square root of the variance) as error bars.

In contrast to the multi-hysteresis test, the loading until rupture shows a clear dependence between cutting angle and stretch at rupture. The 0 degree samples have nearly the same maximum stretch as the injection molded ones. With higher cutting angle, the maximal stretch decreases obviously. Since all 3D-printed samples were manufactured in the same way, the differences in maximum stretch can be explained by the notch effect from the semicircles that comes from the manufacturing process during 3D printing. Nevertheless, in all cases nearly a maximum stretch of nearly 5 could be achieved, even for 3D-printed samples. This highlights the very good connection between the different layers in the 3D-printed samples.

## Conclusions

In this study, differences in the mechanical behavior between injection molded and 3D-printed TPU specimens were investigated. To get unbiased results, comparably big parts were printed by the SEAM technology with one strand per layer. From these parts, S2 tension test specimens were cut out in different directions with respect to the building direction. The mechanical behavior was tested and compared with corresponding specimens manufactured by injection molding.

To regard different cross-sectional shapes, two correction factors have been introduced, which can be used to get the correct cross-sectional area for a specimen measured with a caliper gauge.

Mechanical tests included uniaxial tension with different loading sequences. For the multi-hysteresis tests it can be seen that the 3D-printed specimens show the same inelastic material behavior as the injection molded ones. Only the variance between multiple samples is higher for the additive manufacturing method. In contrast, uniaxial tension until rupture shows an influence of the cutting angles. In particular, the maximal stretch decreases with higher cutting angle due to the notch effect of semicircles resulting from layer-wise printing.

It can be summarized that the inelastic material behavior of the printed material is the same for injection molding and the SEAM process (for the case of the presented processing parameters). However, a broader experimental basis (e.g., strain rate variations, temperature variations, loading cases other than uniaxial tension, fatigue tests) should be targeted in future work. Beyond that, it is also planned to investigate the local strain heterogeneities of the 45° and 90° specimens by the help of full field measurements (e.g., by digital image correlation) and finite-element simulations. For the latter, a choice of a suitable material model and a corresponding parameter identification have to be conducted. For the case of monotonic loading, standard hyperelastic material models might be sufficient. However, for the case of cyclic loading, an appropriate material model to capture the rubber-like phenomena must be applied. Here, for example, the MORPH model seems to be a promising approach (see e.g., [3, 12]).

Funding Open Access funding enabled and organized by Projekt DEAL.

#### Declarations

**Conflict of interest** The authors declare that they have no conflict of interest.

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