

## Improvement in the intense pulsed emission stability of grown CNT films via an electroless plated Ni layer

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Carbon nanotube (CNT) films were grown on silicon wafers with and without a nickel layer (Si-CNT and Ni-CNT) via the pyrolysis of iron phthalocyanine. The nickel layer was prepared using the electroless plating method. To study the emission stability of Si-CNT and Ni-CNT cathodes during intense pulsed emission, emission characteristics were measured repeatedly with a diode structure using a Marx generator as a voltage source. For the peak values of the pulsed voltage, which were in the range between 1.62–1.66 MV (corresponding to electric field intensities between 11.57–11.85 V/ $\mu\text{m}$ ), the first cycle emission current was 109.4 A for Si-CNT and 180.5 A for Ni-CNT. By comparing the normalized emission currents of the Si-CNT and Ni-CNT cathodes, the improvement in the emission stability can be easily quantified. The number of emission cycles necessary for the peak current to decay from 100% to 50% increased from  $\sim 3$  for Si-CNT to  $\sim 11$  for a Ni-CNT film.

**carbon nanotube, nickel layer, intense pulsed emission, improved emission stability, normalized current**

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Carbon nanotubes (CNTs) have attracted a great deal of attention because of their potential use in emission applications. This is because of their nanometer size scale, high aspect ratio, high mechanical strength, good conductance and high chemical stability [1–3]. These characteristics give CNTs application in many devices, such as field emission displays, high-resolution electron-beam instruments, lamps, X-ray sources, and microwave amplifier tubes [4–9]. Emission stability is a parameter that is vital to electron emission applications, especially in intense pulsed emission systems. Unfortunately, there is currently no reported research on intense pulsed emission stability.

In this paper, we study the emission stability of CNT films grown on silicon wafers during intense pulsed emission, and report on improvements in intense pulsed emis-

sion stability when a nickel (Ni) layer is used as the growth substrate.

## 1 Experimental

### 1.1 Sample preparation

CNT films were grown on single-sided polished silicon wafers (n-type, (100) oriented, resistivities between  $10^{-2}$ – $10^{-3}$   $\Omega\text{cm}^{-1}$ , 2-inch diameter) with and without a Ni layer (Si-CNT and Ni-CNT). The Ni layer that was used for CNT film growth was prepared on a Si wafer using an electroless plating process. The plating solution was composed of nickel sulfate,  $\text{NaH}_2\text{PO}_2$ , and trisodium citrate. The pH value of the solution was adjusted to be between 8–10 by adding ammonia to the solution. A palladium catalyst was used for the Ni plating. The CNT film was synthesized via the

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pyrolysis of iron phthalocyanine (FePc) [10].

## 1.2 Morphology of the samples

A scanning electron microscope (SEM, JEOL JSM-6700F, Tokyo, Japan) was used to study the morphologies of the Ni layer, Si-CNT and Ni-CNT films.

## 1.3 Intense pulsed emission characteristics

The intense pulsed emission characteristics were measured using a diode structure with a pulse generator under a vacuum of  $\sim 5 \times 10^{-4}$  Pa. The CNT cathodes were used in single pulse mode (i.e., only one voltage pulse was applied for each test procedure) during the intense pulsed emission measurement. To study the emission stability, we repeatedly measured the emission currents of the Si-CNT and Ni-CNT films under voltage supplied by a Marx generator. All the pulsed voltages have the same half-value width, which was 100 ns.

## 2 Results and discussion

### 2.1 Morphology of the samples

Figure 1 shows SEM images of the Ni substrate. It can be

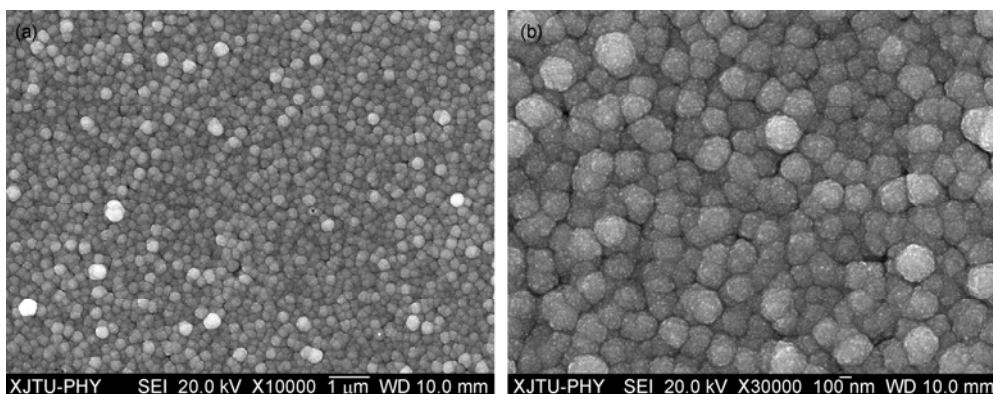
seen from Figure 1(a) that Ni the layer is homogeneous and compact. We estimated from Figure 1(b) that the size range of Ni particles was 150–300 nm.

SEM images of the Si-CNT and Ni-CNT films are shown in Figure 2. It can be seen that CNTs in both the Si-CNT and Ni-CNT films are aligned perpendicularly to the substrates and have a height of several microns.

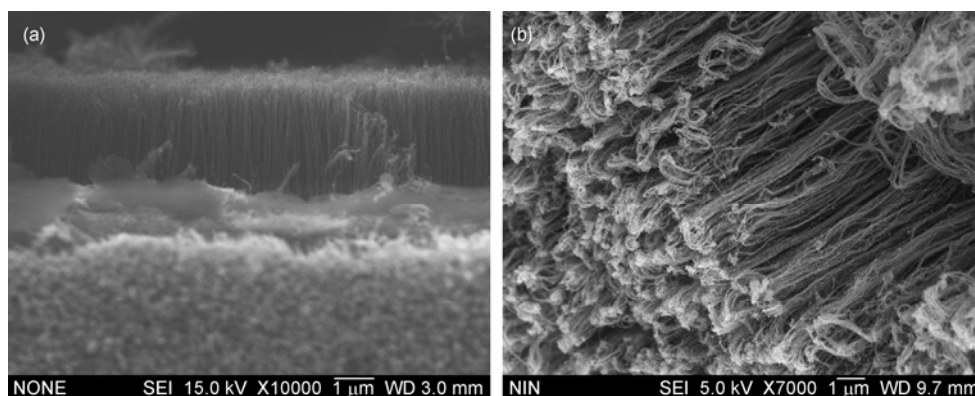
### 2.2 Stability characteristics of the intense pulsed emission current

Although all the measurements of emission current were performed using the same voltages from the Marx generator, the pulsed voltages between anode and cathode were different for each emission cycle. The range of peak values for the pulsed voltage and electric field intensity for all the emission stability measurements were 1.62–1.66 MV and 11.57–11.85 V/ $\mu\text{m}$ .

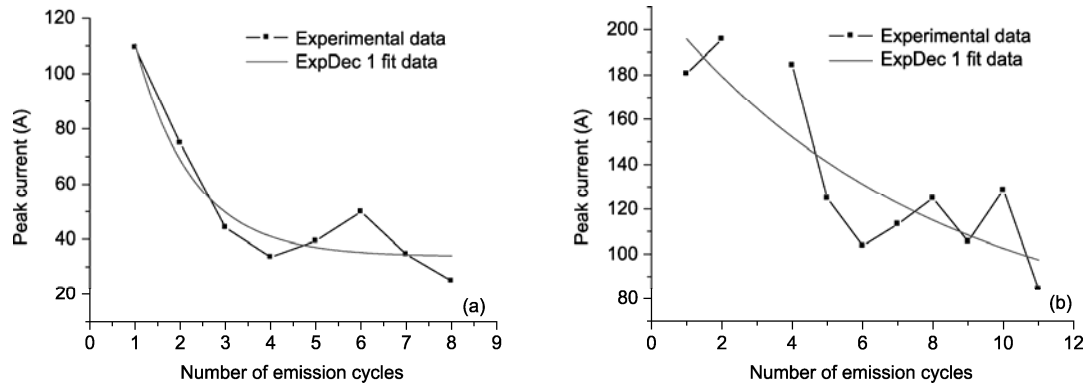
Figure 3 shows the results of repeated emission current measurements for Si-CNT and Ni-CNT cathodes. The absence of a data point for cycle number 3 in Figure 3 (b) is because the maximum current datum was over the upper limit of the recorder. It can be seen in Figures 3(a) and (b) that the peak current values of the first emission cycle of the Si-CNT and Ni-CNT films are 109.4 and 180.5 A. The peak values of the emission current generally decayed with



**Figure 1** SEM images of the Ni layer. (a) Low- and (b) high-magnification images.



**Figure 2** SEM images of the Si-CNT (a) and Ni-CNT (b) films.



**Figure 3** Current stability characteristics for the Si-CNT (a) and Ni-CNT (b) cathodes in an intense pulsed emission mode.

increasing measurement number (emission cycle number). There was some fluctuation in emission current in the measurement results. The fluctuation in the emission current can be attributed to fluctuations in the applied pulsed voltage. This affected the current measurement results, but it did not change the current decay trend.

Because a formula for emission current decay over time in an intense pulsed emission mode could not be found, we described the decay using an exponential decay model, which was used in previous work [11]. Assuming the emission current decayed according to a first order exponential decay model, we can get fitting curves using the first order “Fit Exponential Decay” function (ExpDec 1 model) in Origin 6.1 (Northampton, MA, USA).

The ExpDec 1 model can be described using

$$y = y_0 + A_1 e^{-x/t_1} \quad (1)$$

According to eq. (1) and the fitted results, the decay formulas for emission current of the Si-CNT and Ni-CNT cathodes can be written as

$$I_1 = I_{10} + A_{11} e^{-x/t_1} \\ = 33.75509 + 169.06883 e^{-x/1.27035} \quad (\text{A}), \quad (2)$$

$$I_2 = I_{20} + A_{21} e^{-x/t_2} \\ = 60.69897 + 153.88518 e^{-x/7.69049} \quad (\text{A}), \quad (3)$$

where  $I_1$  is the peak current of the Si-CNT cathode;  $I_2$  is the peak current of the Ni-CNT cathode; and  $x$  is the emission cycle number corresponding to  $I_1$  and  $I_2$ .

The smooth curves labeled “ExpDec 1 fit Data” in Figure 3(a) and (b) are the first order exponential decay model fits for the Si-CNT and Ni-CNT cathodes. We found from the experimental and fitted curves that the decay speed of the Ni-CNT film is lower than that of the Si-CNT film.

### 2.3 Comparison of stability characteristics

To compare the levels of emission current for the samples, we used the normalized emission current. The normalized emission current is defined as the ratio of the current values

of each point in fit curve to that of the first point. The normalized current is actually a dimensionless relative current, given by

$$I_n = \frac{I}{I_{\text{Max}}}, \quad (4)$$

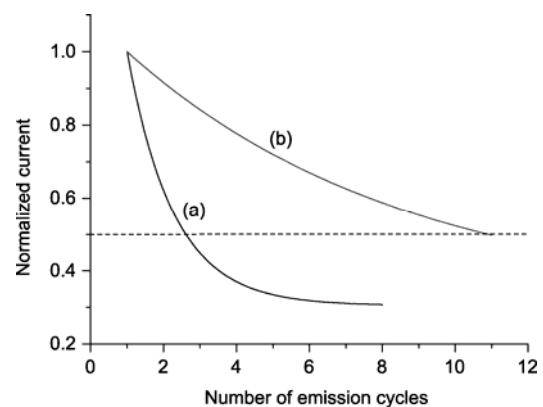
where  $I_n$  is the normalized current;  $I$  is the fitted current; and  $I_{\text{Max}}$  is the initial value (maximum value) for the fitted current.

To compare the decay trends directly, the normalized currents of Si-CNT and Ni-CNT cathodes were plotted together versus emission cycle number. The comparison curves are shown in Figure 4.

Because the normalized emission current denotes the percentage of the current value relative to the initial value, the emission cycle numbers corresponding to the same normalized current can be used for a comparison of the decay speed. Using 50% as a criterion, the emission cycle numbers that corresponded to a 50% decrease are ~3 for Si-CNT and ~11 for Ni-CNT.

### 2.4 Analysis of the mechanism for emission stability improvement

Generally, the factors that affect the conventional field



**Figure 4** Comparison of the emission current decay trends for Si-CNT (a) and Ni-CNT (b).

emission stability of DC modes include surface reactions between CNTs and gas molecules [12] and damage to the nanotubes [13] or the CNT-substrate interface [14]. Often, such damage can be caused by resistive heating from high-current field emission. The passivation of the CNT surface with amorphous carbon [15] can help shield the CNTs from reacting with ambient gas, and thus results in more stable field-emission behavior. However, the adhesion of CNTs to substrate is also critical for stability [16]. The strong electrostatic force exerted on the CNTs during emission can separate the CNTs from the substrate, causing current decay and arcing.

In intense pulsed emission modes, the CNT cathode emits a very high current. The damage caused resistive heating may be the primary factor in the decay of the emission current. Introducing a Ni layer has two positive effects resulting in improvement to the emission stability: (1) The resistance of the Ni layer is lower than that of the Si substrate. This can decrease the Joule heating during electron emission. (2) According to our experimental results, the bonding of CNTs to the Ni layer is better than that to the Si substrate. This can decrease the resistance of CNT-substrate interface. These two effects decrease the damage caused by resistive heating from the high-current field emission and improve the emission stability.

Moreover, when the density of the CNTs increases, the field emission intensity for the CNT cathode will be reduced through the field screening effect [17,18]. This results in a higher turn-on voltage and lower emission current. For extremely dense CNT films, only the few CNTs which extend out past the surface the film have strong emission capabilities. The probability these CNTs being damaged will be higher because they suffer higher emission currents, especially in an intense pulsed emission mode. Therefore, if the growth density of the CNTs can be restricted to the proper value, the emission characteristics and the emission stability can be further improved.

### 3 Conclusion

In summary, a Ni layer can be deposited on the surface of silicon wafer using the electroless plating method. CNT films can grow directly on silicon wafer with or without a Ni layer via the pyrolysis of FePc. The emission stability of CNT films grown on Si wafers during intense pulsed emission can be improved by introducing a Ni layer prior to CNT growth. The normalized emission current was used to contrast the differences in the decay trends in emission current of Si-CNT and Ni-CNT cathodes. The number of emis-

sion cycles corresponding to a normalized current decay from 100% to 50% increased from  $\sim 3$  for a Si-CNT film to  $\sim 11$  for a Ni-CNT film.

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