

Layered Fe(III) doped TiO₂ thin-film electrodes for the photoelectrocatalytic oxidation of glucose and potassium hydrogen phthalate

YU Hua¹, WANG Jing^{2,3}, ZHANG ShanQing^{1*}, LI XinJun² & ZHAO HuiJun¹

¹ Environmental Futures Centre and Griffith School of Environment, Gold Coast Campus, Griffith University, QLD 4222, Australia;

² Guangzhou Institute of Energy Conversion, Chinese Academy of Sciences, Guangzhou 510640, China;

³ Graduate University of the Chinese Academy of Sciences, Beijing 100049, China

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Four types of TiO₂ thin-film electrodes were fabricated from TiO₂ and Fe(III) doped TiO₂ sols using a layer-by-layer dip-coating technique. Electrodes fabricated were TF (pure TiO₂ surface, Fe(III)-TiO₂ bottom layer), FT (Fe(III)-TiO₂ surface, pure TiO₂ bottom layer), TT (both layers pure TiO₂) and FF (both layers Fe(III)-TiO₂). The photoelectrochemical behavior of these electrodes was characterized using linear sweep voltammetry (LSV), electrochemical impedance spectroscopy (EIS) and steady-state photocurrent measurements in aqueous 0.1 mol L⁻¹ NaNO₃ containing varying concentrations of glucose or potassium hydrogen phthalate (KHP). EIS and LSV results revealed that exciton separation efficiency followed the sequence of TF > TT > FT > FF. Under a constant potential of +0.3 V, steady-state photocurrent profiles were recorded with varying organic compound concentrations. The TF electrode possessed the greatest photocatalytic capacity for oxidizing glucose and KHP, and possessed a KHP anti-poisoning effect. Enhanced photoelectrochemical performance of the TF electrode was attributed to effective exciton separation because of the layered TF structure.

Fe(III)-doped TiO₂, charge separation efficiency, photoelectrocatalysis

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Photoelectrocatalysis offers a powerful method for the degradation and monitoring of organic compounds in wastewater [1–7]. Semiconductor photoanodes play an important role in these processes [7–9], and among them TiO₂ is widely used because of its excellent chemical and photochemical stability, and outstanding photoelectrochemical efficiency [10–13]. To further its use in applications, the photoelectrocatalytic efficiency of TiO₂ photoanodes needs to be improved, and many studies have focused on this goal. The most effective and practical approach to achieve this is to incorporate features to facilitate exciton separation, such as band-gap engineering. For example, mixed phase TiO₂ electrodes (i.e. anatase/rutile) may possess higher activity in

the photoelectrocatalytic degradation of organic compounds than pure anatase electrodes because the electron-transfer pathway from anatase phase to rutile phase facilitates the separation of photoelectron and photohole, extending the lifetime of the photoelectron and photohole [14]. Other TiO₂ modifications such as Pt¹⁵ and Ag¹⁶ doping have also been shown to improve organic degradation efficiency in photoelectrocatalytic processes, perhaps due to the Pt or Ag nanoparticles acting as electron traps and preventing exciton recombination.

Metal ion doping [17,18] has also attracted significant attention for modifying TiO₂ thin-film electrodes because of its narrowing of the band gap, suppressing of exciton recombination, and improving of photoelectrocatalytic performance [19]. The doping effects of different metals on

*Corresponding author (email: s.zhang@griffith.edu.au)

TiO₂ film electrodes have been investigated extensively [20]. Our studies [16–21] indicated that both metal ion type and the doped composite hierarchical structure affected photoelectrocatalytic activity. The layered structure that favored the formation of the electron flow gradient was beneficial for the improvement of exciton separation and, therefore, photoelectrocatalytic activity [16–21].

In the current work, layered Fe(III) doped TiO₂ electrodes were fabricated from TiO₂ and Fe(III) doped sols using a layer-by-layer dip-coating technique. The effect of Fe(III) doping on TiO₂ electrode photoelectrocatalytic behavior was systematically investigated. The layered Fe(III) doped structure is expected to be beneficial for enhancing exciton separation and thus photoelectrocatalytic activity of TiO₂ thin-film photoanodes.

1 Experimental

1.1 Preparation of TiO₂ and Fe(III)-TiO₂ sols

The TiO₂ sol was prepared by the hydrolysis of tetrabutyl titanate as described in our previous report [16]. Briefly, 68 mL of tetrabutyl titanate and 16.5 mL of diethanolamine were dissolved in 210 mL of absolute ethanol, with the mixture stirred vigorously for 1 h at room temperature (solution A). A mixture of 3.6 mL of H₂O and 100 mL of ethanol (solution B) was added dropwise to Solution A under continuous stirring. After stirring for 0.5 h, polyethylene glycol (PEG, *M_w* 2000 Daltons, 30% w/w of the solid TiO₂ sol) was added, and the resulting alkoxide solution was aged for 24 h at room temperature to subsequently form the TiO₂ sol.

The preparation of the Fe(III)-TiO₂ sol was the same as described above except that aqueous Fe(NO₃)₃ was added as solution B, rather than H₂O. The molar ratio of Fe/Ti was 0.5% for the Fe(III)-TiO₂ sol.

1.2 Preparation of thin-film electrodes

Thin-film electrodes were fabricated using a layer-by-layer dip-coating technique, from the TiO₂ and Fe(III)-TiO₂ sols. Fluorine-doped tin oxide (FTO) conductive glass was used as the substrate, and was pretreated sequentially with detergent, water and ethanol, and dried at 80°C. For the dip-coating process, FTO glass was dipped into the TiO₂ or Fe(III)-TiO₂ sol and withdrawn at a speed of 2 mm s⁻¹. Different types of TiO₂ thin-film electrodes were fabricated following the procedures in Table 1. Once the desired coating had been obtained, TiO₂ electrodes were dried in air at room temperature, and subsequently annealed in air at 500°C for 1 h. TiO₂ and Fe(III)-TiO₂ sol coatings were denoted as T and F, respectively, and four types of the TiO₂ film electrodes were obtained. These were TF (pure TiO₂ surface layer, Fe(III)-TiO₂ bottom layer), FT (Fe(III)-TiO₂ surface layer, pure TiO₂ bottom layer), TT (both layers pure TiO₂) and FF (both layers Fe(III)-TiO₂), as shown in Table 1.

Table 1 Coating sequences for the TiO₂ thin-film electrodes

Coating mode	Thin-film composition	
	Bottom	Surface
TT	10 coats of TiO ₂	10 coats of TiO ₂
TF	10 coats of Fe(III)-TiO ₂	10 coats of TiO ₂
FF	10 coats of Fe(III)-TiO ₂	10 coats of Fe(III)-TiO ₂
FT	10 coats of TiO ₂	10 coats of Fe(III)-TiO ₂

1.3 Photoelectrochemical measurements

Photoelectrochemical experiments were carried out at room temperature in a three-electrode cell with a quartz window for UV illumination, as shown in Figure 1. The TiO₂ film electrode, a saturated Ag/AgCl electrode and a platinum mesh were used as the working, reference and counter electrodes, respectively. Aqueous 0.1 M (1 M=1 mol L⁻¹) NaNO₃ was employed as the supporting electrolyte. The TiO₂ electrode was inserted into an electrode holder, and illumination was achieved with a 300W Xenon arc lamp (Trustech, Beijing, China) with an UV band pass filter (UG-5, Schott). Electrochemical characterization by linear sweep voltammetry (LSV), electrochemical impedance spectroscopy (EIS) and steady-state photocurrent (*i_{ph}*) were measured by a CHI660A electrochemical workstation (CH Instruments, USA) under constant UV illumination (light intensity of 6.2 mW/cm² at 365 nm).

2 Results and discussion

2.1 Photoelectrochemical characterization

Electrochemistry is a powerful tool for studying charge transport and recombination processes at TiO₂/electrolyte interfaces [22]. In this work, TiO₂ film electrodes were initially characterized with EIS and LSV techniques. EIS Nyquist plots are commonly used to investigate charge transport resistance and exciton separation efficiency [23]. The size of the arc radius on the EIS Nyquist plot indicates the rate of photoelectrochemical reaction at the electrode [24],

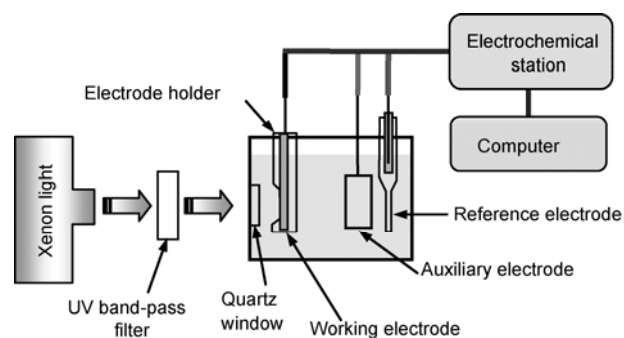


Figure 1 Schematic of the three electrode cell used for photoelectrochemical measurements.

and is a useful parameter for characterizing electron transport resistance [16]. Figure 2 shows impedance spectra of different TiO₂ film electrodes under UV illumination with light intensity of 6.2 mW cm⁻² in aqueous 0.1 mol L⁻¹ NaNO₃. Frequencies for EIS measurement were scanned from 10⁵ to 0.1 Hz. The circular radii varied with doping, and were in the sequence TF < TT < FT < FF, as shown in Figure 2. This suggested that electron transport resistance of the electrodes was also in the sequence TF < TT < FT < FF.

The photoelectrochemical behavior of the TiO₂ electrodes was further characterized with LSV measurements in aqueous 0.1 mol/L NaNO₃ under the constant UV illumination [14]. The photocurrent originates from the photooxidation of water and organic compounds within the electrolyte, and photocurrent density corresponds to the amount of free carriers under the same potential bias, which is related to the amount of photo-generated holes within the TiO₂ electrode [16]. LSV curves of the TiO₂ electrodes are shown in Figure 3. Photocurrents of all samples increased with potential from -0.1 to 0.5V, and photocurrent values of the thin-film TiO₂ electrodes followed the sequence TF > TT > FT > FF. Under the same potential, stronger current indicates more electrons are collected from the photoelectrochemical reaction, and suggests better separation efficiency. The TF electrode showed the strongest photocurrent, and therefore exhibited the best exciton separation efficiency. The LSV result is coincident with that of the EIS measurement, because better exciton separation efficiency provides a higher free charge carrier concentration, leading to a lower electron transport resistance. Therefore the photocurrent sequence TF > TT > FT > FF is in agreement with the electron transport resistance order TF < TT < FT < FF.

An advantage of metal ion doping in TiO₂ is the temporary trapping of photo-generated charge carriers by the dopant, and the inhibition of their recombination during their migration from the materials core to surface [25]. For the electrodes fabricated in this study, Fe(III) ions present in

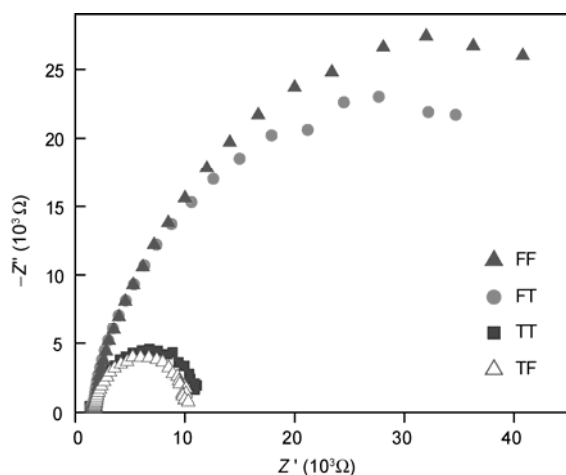


Figure 2 EIS Nyquist plots of the TiO₂ thin-film electrodes, under constant UV illumination.

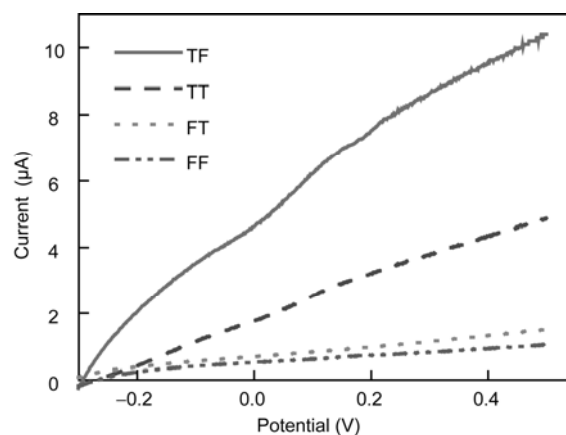


Figure 3 LSV plots for TiO₂ thin-film electrodes in aqueous 0.1 M NaNO₃ under constant UV illumination (scan rate was 5 mV s⁻¹).

the doped TiO₂ electrodes were expected to act as electron traps [26]. For the TF electrode, photo-generated electrons were attracted by the Fe(III) layer at the bottom of the electrode, and this attraction provided a further driving force for exciton separation at the electrode surface. Separation efficiency of the TF electrode was enhanced compared with that of the TT electrode, as evidenced by the lower electron transport resistance from EIS measurements and the higher photocurrent under the same potential from LSV measurements.

In contrast, the FT and FF electrodes had a much greater electron transport resistance and lower exciton separation efficiency than the TT electrode, for two reasons. First, part of the TiO₂ surface exposed to the UV light was covered by the Fe(III) species, so less charge carriers were generated under the same illumination conditions. Second, the additional attractive force on photo-generated electrons was towards the Fe(III) species at the FT electrodes top layer, and in opposing direction to the conductive substrate. In the photoelectrochemical process, electrons are removed from the conducting substrate at the bottom. Temporary trapping of electrons by Fe(III) on the surface hinders their subsequent removal. For the FF electrode, Fe(III) species were evenly distributed throughout, and such electron attraction processes affected the whole electrode. Because of these two effects, the FF and FT electrodes showed much larger electron transport resistance and lower photocurrent than the TT electrode.

2.2 Photoelectrochemical oxidation of organic compounds

TiO₂ based photoelectrocatalytic degradation is widely used in the treatment of wastewater, because it can oxidize a wide spectrum of organic compounds in aqueous solution [14]. The general reaction for the complete mineralization of an organic compound, i.e. C_yH_mO_jN_kX_q, at the TiO₂ electrode surface can be represented by eq. (1) [13]:

Figure 4 shows a typical photocurrent response for a common TiO₂ electrode under a constant potential (+0.3 V vs AgCl/Ag) and constant UV illumination in aqueous solution. Upon illumination, a sharp current spike was observed, which was because of the oxidation of adsorbed water molecules and organic species. The current subsequently decreased gradually and obtained a steady-state. The overall steady-state photocurrent (i_{total}) is generally composed of two parts: one part (i_{net}) arises from oxidation of organic species (e.g. glucose or KHP), and the other (i_{water}) is due to water oxidation [14]. i_{net} can be calculated by subtracting i_{water} in the electrolyte without organic compounds, from i_{total} in electrolyte with organic compounds, as indicated in eq. (2). i_{net} represents the oxidation rate of the organic species¹⁴, and is indicative of the organic compound concentration in solution [27].

$$i_{\text{net}} = i_{\text{total}} - i_{\text{water}} \quad (2)$$

(i) Photoelectrochemical oxidation of glucose. Figure 5 shows i_{net} profiles of different TiO₂ film electrodes in aqueous 0.1 M NaNO₃ with increasing glucose concentration, under a potential bias of +0.3 V and UV illumination. TF electrodes showed the highest i_{net} for each concentration, and i_{net} values were found to follow the sequence TF > TT > FT > FF, consistent with the aforementioned LSV and EIS results. For all electrode types, i_{net} increased with glucose concentration and two distinct stages were observed. At low concentration, i_{net} increased linearly with glucose concentration. This was predominantly because of the mass transport limiting of organic compounds to the electrode surface. At higher concentrations, i_{net} deviated from linearity, and this was likely because of the saturated adsorption of glucose, with the photoelectrocatalytic reaction subsequently becoming the rate-determining step. The TF electrode possessed the highest i_{net} across both concentration ranges, which suggested that TF photoelectrocatalytic activity was greater than that of the other electrodes. The greatest photocatalytic activity was directly related to the largest

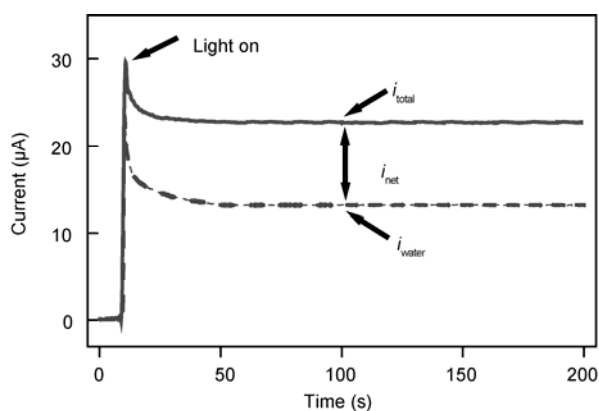


Figure 4 Typical photocurrent response of TiO₂ thin-film electrodes in aqueous 0.1 M NaNO₃ in the presence (solid line) and absence (dashed line) of organic compounds.

number of photoholes on the TF surface, because of efficient exciton separation. The results also showed that i_{net} of FT and FF electrodes were smaller than that of the TT electrode. This suggested the FT and FF modification did not lead to positive improvement on the photocatalytic performance of the TiO₂ electrodes. This can also be rationalized with the aforementioned EIS and LSV measurements, because FT and FF modification did not lead to the net electron attraction force, only the loss of active sites due to Fe(III) coverage.

Overall, TF electrodes showed the highest photocurrent and photoelectrocatalytic oxidation efficiency, and glucose was easily oxidized because of the layered bottom-doping structure of the electrode.

(ii) Photoelectrochemical oxidation of KHP. Potassium hydrogen phthalate (KHP) is an aromatic organic compound, and because of the stability of the benzene ring, KHP is difficult to oxidize and can poison TiO₂ electrodes. Thus, it is frequently used as a standard degradation substrate for estimating photocatalytic efficiency of TiO₂ electrodes [28]. In this study, the photoelectrocatalytic degradation of KHP was carried out under the same experimental conditions as those for glucose above. Figure 6 shows i_{net} profiles for the oxidation of different KHP concentrations at the various electrodes. i_{net} values of all electrodes increased linearly with KHP concentrations in the low concentration range, similar to those observed for the oxidation of glucose because these i_{net} values were obtained under mass transport limitation. As shown in Figure 6, the linear ranges of KHP concentrations ended at 0.6, 0.5, 0.3 and 0.2 mM/L, for TF, TT, FT and FF electrodes, respectively. The TF electrode had the widest linear range, and it is well-established that electrodes with higher photoelectrocatalytic activity have a faster reaction rate and therefore a wider linear range [27]. The large amount of photo-holes provides a faster reaction rate and efficient oxidation of KHP. Thus, we concluded that TF had a higher photoelectrocatalytic oxidation capability

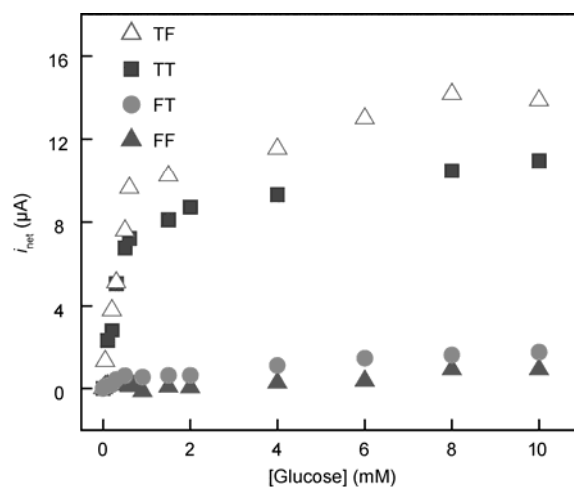


Figure 5 The relationship between glucose concentration and net photocurrent, i_{net} , for the four different electrode types.

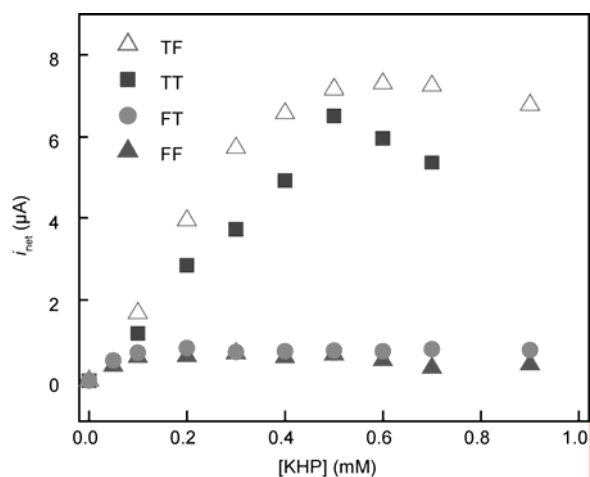


Figure 6 The relationship between KHP concentration and net photocurrent, i_{net} , for the four different electrode types.

than the other electrodes, attributed to the layered Fe(III) doped structure. This resulted in the best exciton separation efficiency among the four kinds of electrodes tested.

In contrast to the i_{net} profiles in Figure 5, i_{net} behaved differently after reaching its maximum in Figure 6. In Figure 6, with increasing KHP concentration, i_{net} for TT, FT and FF decreased, probably due to poisoning of the TiO_2 photocatalyst by the high KHP concentration [14–29]. An i_{net} decrease for TF was not observed, and i_{net} further increased with increasing KHP concentration. This suggested TF possessed an anti-poisoning effect, most likely because of the enhanced photocatalytic activity of the TF layered structure.

3 Conclusions

Layered Fe(III) doped TiO_2 electrodes were prepared by a dip-coating technique, and characterized using electrochemical methods. Electrode photocatalytic activity was evaluated by linear sweep voltammetry (LSV), and charge transport resistance was investigated by electrochemical impedance spectroscopy (EIS). EIS results demonstrated that TF thin-film electrodes possessed the greatest exciton separation efficiency. Greater separation efficiency results in more effective photoholes on the electrode surface for the photooxidation of water and organic compounds. This was confirmed by LSV measurements in aqueous solution with and without organic compounds. The highest photocurrent was obtained for the TF electrode, for both oxidizing water and the photoelectrocatalytic degradation of glucose and KHP. In addition, the TF electrode showed a promising KHP anti-poisoning effect.

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