

# Characterization and Mössbauer spectroscopy of steel slag generated in the ladle furnance in SIDERPERU steel plant

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Accepted: 1 February 2022 / Published online: 9 February 2022 © The Author(s) 2022

#### Abstract

The steel industry produces large amounts of slag coming from different stages during the steelmaking process every year. Currently, there are numerous attempts to recycle it or to use it in some other industry sectors and to preserve the environment. The characteristics of the slag depends on the steelmaking process and it is crucial to have it before any attempt of recycling. In this work, slag sample produced in the ladle furnace from SIDERPERU steel plant were collected and analyzed by using energy dispersion X-ray (EDX), X-ray diffraction (XRD), X-ray fluorescence spectroscopy (XRF), SQUID magnetometer and Mössbauer spectroscopy. The chemical analysis obtained by EDX and XRF indicate that the main elemental composition of the material is Fe, Ca, Si and Cr. XRD identifies that these elements are in the phases: FeO,  $Fe_3O_4$ ,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, Ca<sub>2</sub>SiO<sub>4</sub>, and Ca2.32Mn0.68SiO7. Magnetometry measurements suggest the Verwey transition for magnetite and the Morin transition for hematite are screened by the presence of superparamagnetic phases. The Mössbauer spectrum shows two doublets related to Fe<sup>2+</sup> and Fe<sup>3+</sup> ions with hyperfine parameters belonging to that of non-stoichiometric wustite. Also, the presence of hyperfine fields characteristic of the  $Fe_3O_4$  and  $Fe_2O_3$  phase identified at room temperature verifies the magnetometry analysis. The analysis of the sample used in this work reveals details connected with the steel fabrication processes and are helpful for posterior recycling attempts.

Keywords Ladle Furnace  $\cdot$  Steel slag  $\cdot$  Steelmaking process  $\cdot$  Mössbauer Spectroscopy  $\cdot$  Iron oxides

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This article is part of the Topical Collection on *Proceedings of the International Conference on the Applications of the Mössbauer Effect (ICAME 2021), 5-10 September 2021, Brasov, Romania* Edited by Victor Kuncser

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## **1** Introduction

The metallurgical slag is a by-product material generated during the steelmaking process. Large amounts of slags are produced in the steel industry and increase every year generating pollution and affecting the environment [1]. Some attempts in reusing steel slags include raw material for road construction, pavements, bricks and concrete [2–4]. The partial replacement of cement by steel slags makes a significant contribution to the steel industry and helps to reduce  $CO_2$  emissions [2]. Thus, finding an appropriate use of steel slags is very important for both, to reduce the potential impact on the environment and to maintain the sustainability of natural resources [5, 6].

Steel slags are formed during the melting of iron ore or iron scrap and subsequently discharged as waste [7]. Depending on the type of steel, the slags can be classified as carbon steel slag or stainless-steel slag [8, 9]. Slags can also be classified depending on the step during the steelmaking process they were generated i.e., basic oxygen furnace (BOF) slag, electrical arc furnace (EAF) slag, ladle furnace (LF) slag and continuous casting residue. Many research have been focused on the characterization of steel slags produced in an EAF. For example, Yildirim and Prezzi, reported that the main contents in the slags from EAFs are typically FeO, CaO and SiO<sub>2</sub> and similar works that report the common compounds [7, 8]. However, the characterization and recycling of LF slags (LFS) are less widespread. Nevertheless, there are some attempts to recycle LFS as Portland cement, soil stabilization and asphalt mixes due to high calcium content [10–13]. Since the composition of the LFS varies according to the scrap used, it is necessary to know the characteristics and chemical and physical properties of these slags before recycling and to decide the best strategy to recover the metals.

In this work, we present the characterization of the slags produced during the second step process of steel in a ladle furnace from SIDERPERU in order to facilitate the recycling of the steel in Peru.

## 2 Experimental

The samples were collected from SIDERPERU steel plant, Chimbote, Peru. The steelmaking process in this steel plant consists of three different stages: A melting process in an Electric Arc Furnace (EAF), the forming of chemical equilibrium between Fe, Ca and other components in a Ladle Furnace (LF), and the solidification through continuous casting. During the second stage, some melted steel remains stuck at the internal walls of the LF, which are discarded after the melting steel reaches its chemical equilibrium and dried to room temperature. Tons of this slag are produced every week and they are crushed for recycling. Several kilograms of this material were collected during this work and named as 'remanent' sample, as listed in Table 1. Whereas, before recycling, magnetic drums are

Sample	Source	Factory treatment
Remanent	Melted steel stuck in the internal walls of the ladle furnace	Dried in normal conditions
Demetallized	Crushed remanent sample	Magnetically separated

 Table 1
 Difference between slag samples collected after treatment in the ladle furnace at SIDERPERU

approached to the crushed material with the aim to extract the remaining Fe. The left-over material from this process is then thrown away by SIDERPERU and collected in this work and called as 'demetallized' slag (see Table 1).

The collected samples were powdered and sieved down to 72  $\mu$ m for better performance analysis in order to increase the accuracy of measurements. The obtained powders were stored in two Eppendorf (each containing one sample), from which they supplied for each characterization technique. The morphology of the samples was inspected through a scanning electron microscope (SEM, Philipps XL30) with a resolution of 10 nm at 30 kV, voltage range: 0.5 V to 30 kV and magnification of  $300,000\times$  adapted with an energy dispersive X-ray (EDX) spectrometer OXFORD Xplorer. An XRF-720 Shimadzu equipment was used to register the elemental composition of the sample, which counts with an advantage algorithm that allows analyzing a variety of types of matrices. The X-ray diffraction (XRD) measurements were performed in a Rigaku brand diffractometer using a cobalt target ( $\lambda_{K\alpha} = 1.79026$  Å) with  $10^{\circ} \le 2\theta \le 90^{\circ}$ , step width 0.02° and speed of 4 s/step. The Co-target was used for XRD since, in contrast to other targets, it increases the diffraction signal of iron-containing minerals. The magnetic measurements were carried out using a DC-MPMS-SQUID magnetometer (Quantum Design Inc.) in zero field cooling (ZFC) and field cooling (FC) modes. The magnetization response as a function of temperature was collected from 5 to 400 K and applying external magnetic fields of 500 Oe and 10 kOe. The applied field response of the magnetization was also measured in the range  $\pm 10$  kOe at 5, 300 and 400 K. The Mössbauer spectroscopy of the sieved samples were performed at room temperature (RT) using a Mössbauer spectrometer in transmission geometry with a radioactive source of <sup>57</sup>Co (Rh) with 50 mCi current activity with a constant acceleration function. The data were registered in a multichannel analyzer with counting data stored in 1024 channels platform. For the calibration measurement, a thin  $\alpha$ -Fe foil was used. The fitting process was carried out considering Lorentzian profiles using the WinNormos software.

#### **3** Results and Discussions

Figure 1a shows a dried piece of the remanent slag stucked in the ladle furnace from SID-ERPERU steel plant and in Fig. 1b a piece of the demetallized slag after crushing the remanent slag is also shown. Theses slags are hard sponge stones, with irregular shape and have many pores probably caused by release of gases during melting and quenching. Fig. 1c shows a SEM micrograph of remanent sample revealing a rough surface with many pores suggesting that it boiled at high temperatures during many hours. The SEM micrograph for the demetallized slag sample shown in Fig. 1d presents a quite spheroidal grain with no rough surface and diameter size around to 100  $\mu$ m revealing also the presence of nanoparticles. According to the SEM images, Ca and Fe are the predominate elements spread over the whole samples as indicated by the mapping colors. Figs. 1e,f show the respective EDX sum-spectra. Whereas, the respective XRF spectra are shown in Figs. 1g,h. The element quantification provided by both techniques are listed in Table 2. According to it, Fe, Ca and Si are the main component elements. However, the EDX analysis also detects Mn, Cr and Zn as secondary elements. Moreover, XRF also detects Cr, Ti, K, Sr, Nb, Zr, Zn for both samples. This difference in secondary elemental detection is due the accuracy



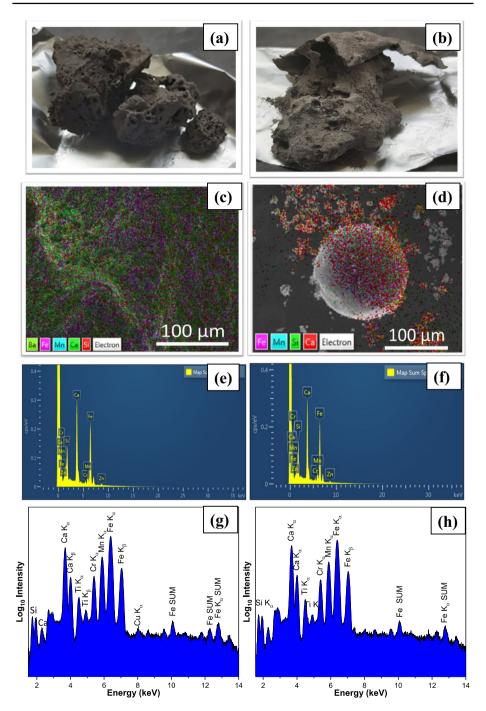


Fig. 1 a and b Ladle furnace slag remanent and demetallized, c and d SEM micrography and color mapping for remanent and demetallized slag, e and f EDX sum spectrum and g and h X ray fluorescence of remanent and demetallized ladle furnace slag.

Sample	EDX									
Remanent	Element	Fe	Ca	Si	Zn	Cr				
	wt (%)	46.98	30.22	12.2	2.2	1.49				
	XRF									
	Element	Fe	Ca	Si	Cr	Ti	Κ	Sr	Nb	Zr
	Mass %	47.2	29.3	17.6	3.49	1.217	0.359	0.311	0.122	0.064
	Element	Zn	Rb							
	Mass %	0.051	0.017							
Demetallized	EDX									
	Element	Fe	Ca	Si	Mn					
	wt (%)	26.8	53.45	9.12	10.13					
	XRF									
	Element	Fe	Ca	Si	Mn	Cr	Ti	Sr	Nb	Zr
	Mass %	35.54	32.89	9.62	8.047	2.616	0.359	0.46	0.043	0.076
	Element	Cu								
	Mass %	0.065								

 Table 2
 Elemental Composition of remanent and demetallized slag generated in the ladle furnace from
 SIDERPERU steel plant

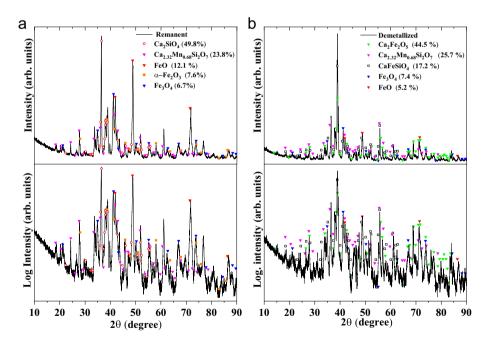


Fig. 2 X-ray diffractogram of metallurgical slag produced in a ladle furnace during steelmaking process in SIDERPERU plants. a) remanent slag, b) demetallized slag

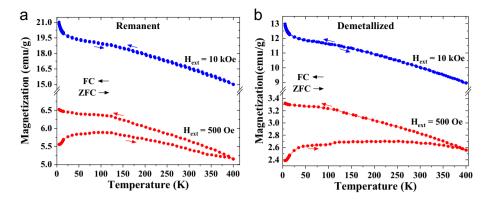


Fig. 3 Magnetization response as a function of the temperature for the remanent (a) and demetallized (b) slags collected from SIDERPERU steel plant

of both techniques. Compared to EDX, XRF has a larger penetration depth and is able to detect more elements.

Figure 2 shows the X-ray diffractograms of the slags. The abundance of the different phases identified by XRD are given in the legends in the plots. According to Fig. 2(a), the main crystalline phases identified by the XRD for the remanent slag are larnite ( $Ca_2SiO_4$ , PDF card N0. 00-029-0369), ankerite (Ca2.32Mn0.68Si2O7 PDF card N0. 00-040-0054), wustite (FeO, PDF card N0. 01-073-2143), hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, PDF card N0. 00-086-2368) and magnetite ( $Fe_3O_4$ , PDF card N0. 01-086-1355). Larnite could be formed by the reaction of CaO and SiO<sub>2</sub> due to reactants added during the melting process of the initial scraps in the electrical furnace [14], whereas ankerite Ca<sub>2 32</sub>Mn<sub>0.68</sub>Si<sub>2</sub>O<sub>7</sub> is a non-stoichiometric mineral that can occur due to reaction of MnO, CaO and SiO<sub>2</sub>. The X-ray diffractogram for the demetallized slag (see Fig. 2(b)) reveals a mineralogical phase of the brownmillerite subgroup known as srebrodolskite (Ca2Fe2O5, PDF card N0. 00-047-1744), ankerite (Ca<sub>2 32</sub>Mn<sub>0.68</sub>Si<sub>2</sub>O<sub>7</sub>, PDF card N0. 00-040-0054), kirschteinite (CaFeSiO<sub>4</sub>, PDF card N0. 00-011-0477), magnetite and wustite. The XRD for both samples are also shown in logarithmical scale in order to observe the smaller peaks. Srebrodolskite should occurr at temperatures above 750 °C by the reaction of the Fe components in the slags with CaCO<sub>3</sub>, which is used for lowering the melting point of the steel. α-Fe<sub>2</sub>O<sub>3</sub> was formed by the oxidation of wustite with the oxygen gas injected during melting of steel scrap in the initial electrical arc furnace. Moreover, kirschteinite could be formed by reaction of CaO, FeO and SiO<sub>2</sub> at 1460 °C during the reduction process of scrap melted in the ladle furnace. Furthermore, the non-stoichiometric ankerite might have formed during the reaction of MnO, CaO and SiO<sub>2</sub>. These mineralogical phases are commonly found in steel slag reported in literature [7, 8, 15, 16]. The results obtained here are in agreement with the EDX and XRF analysis above, which registered Fe and Ca as the major components.

Figures 3(a) and (b) show the magnetization response as a function of the temperature (M(T)) curves taken in ZFC and FC modes under 500 Oe and 10 kOe external field of the remanent and demetallized samples, respectively. Note that at any temperature and external field, the magnetization values for the remanent sample are higher than those for the demetallized one. This is because the remanent sample contains higher amount of iron than the demetallized sample as discussed in the chemical composition analysis above. Besides, it is not clear the cause of the relative high values of the magnetization in both cases. To name,

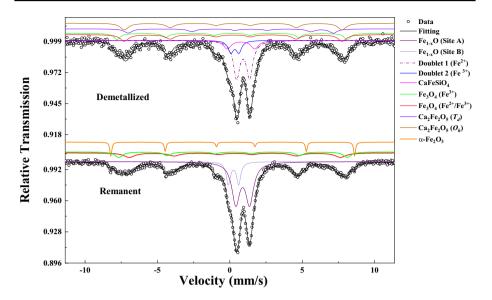


Fig. 4 Mössbauer spectra at RT of a slag produced in a ladle furnace during steelmaking process in SIDER-PERU plant

the magnetization values provided by the paramagnetic phases, such as wustite and iron silicates, are lower than 1 emu/g [17-19] in contrast to Fig. 3(a) and (b). Moreover, it is uncertain that the high magnetization values be caused by the ferrimagnetic magnetite nor antiferromagnetic hematite present in the samples, as detected by the XRD analysis above. Furthermore, the Verwey [20] and Morin [21] transitions for magnetite and hematite are not detected in the M(T) curves. Instead, an apparent blocking state is noted at low temperatures in the ZFC curves taken under 500 Oe applied field in both samples. Thus, the relative high values of the magnetization and this blocking state in the M(T) curves suggest that both samples contain certain amount of magnetic nanoparticles. Nanoparticles are not indicated by XRD due to the overlapping of multiple mineralogical phases, whose superparamagnetic signals in Fig. 3 might screen the Verwey and Morin signals of the magnetite and hematite, respectively. In addition, the irreversibility over the whole temperature range of the ZFC and FC curves obtained under 500 Oe for both samples, become reversible under a higher applied field, such as 10 kOe. In this way, increasing the applied field tends to align all magnetic domains present in the sample to the direction of this external field resulting in gaining of magnetization. As a complementary information, the magnetization as a function of the external field is given in the Supplemental Material S1.

The Mössbauer spectra for both samples at room temperature are shown in Fig. 4 and the obtained hyperfine parameters are listed in Table 3. The spectra for both samples are quite similar as shown in Fig. 4. The fitting process was performed using sextets and doublets subspectra. The two-broad lines absorption in the central parts of the spectra might come from the iron silicates and/or calcium iron compounds detected by the XRD analysis above. Moreover, according to the literature [22, 23], the electronic exchange between Fe<sup>2+</sup> in octahedral site and Fe<sup>3+</sup> in tetrahedral site and the electronic exchange of tetrahedral Fe<sup>3+</sup> and Fe<sup>2+</sup>/Fe<sup>3+</sup> octahedral sites produce two paramagnetic doublets that can be related to non-stoichiometric wustite. For the remanent slag sample, the hyperfine parameters associated to wustite are IS = 0.99 mm/s

Sample	Site	IS (mm/s)	QS (mm/s)	Bhf (T)	W (mm/s)	Area (%)
Remanent	Fe <sub>1-x</sub> O (Fe <sup>2+</sup> )	0.99(2)	1.01(1)	_	0.64(3)	53.3
	$Fe_{1-x}O(Fe^{3+})$	0.36(2)	0.71(1)	_	0.34(3)	8.0
	Magnetite (Fe <sup>2+</sup> /Fe <sup>3+</sup> )	0.47(3)	-0.02(2)	45.2(1)	0.85(5)	22.9
	Magnetite (Fe <sup>3+</sup> )	0.33(2)	0.02	48.8(4)	0.6(2)	13.9
	Hematite (Fe <sup>3+</sup> )	0.41(2)	-0.21(2)	52.2(0)	0.14(3)	1.9
Demetallized	Doublet 1 (Fe <sup>2+</sup> )	0.90(1)	1.00(2)	-	0.50(3)	33.5
	Doublet 2 (Fe <sup>3+</sup> )	0.46(1)	0.52(1)	-	0.42(3)	7.3
	CaFeSiO <sub>4</sub> (Fe <sup>2+</sup> )	0.87(1)	1.89(1)	-	0.63(1)	11.1
	Magnetite (Fe <sup>3+</sup> )	0.26(5)	-	49.2(5)	0.51(1)	10.6
	Magnetite (Fe <sup>2+</sup> /Fe <sup>3+</sup> )	0.34(1)	_	46.6(1)	0.70(2)	21.5
	$Ca_2Fe_2O_5(T_d)$	0.70(2)	-1.4(1)	43.9(2)	0.63(1)	11.5
	$Ca_2Fe_2O_5(O_h)$	0.77(3)	1.36(1)	50.85(2)	0.55(2)	4.5

 Table 3
 Hyperfine
 Mössbauer
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and QS = 1.01 mm/s, and IS = 0.36 mm/s and QS = 0.71 mm/s. Similarly, the hyperfine parameters IS = 0.90 mm/s and QS = 1.0 mm/s and a tiny doublet with IS = 0.38 mm/s and QS = 0.52 mm/s in the oxidation state of Fe<sup>3+</sup> were found for the demetallized slag. Note that, in contrast to the XRD of the demetallized sample above in which wustite represents the smaller component, the corresponding MS presents this oxide having the highest relative absorption area (%). This might be caused by the presence of superparamagnetic Fe-oxides (such as wustite nanoparticles), as suggested by the respective SEM micrograph (Fig. 1(d)) and the magnetometry analysis (Fig. 3(b)) above. In this way, the purple doublet marked with dashed dots in Fig. 4 (Doublet 1 (Fe<sup>2+</sup>)) is assumed to correspond to superparamagnetic wustite. Two sextets with hyperfine magnetic fields  $B_{hf}$ =45.2 T and  $B_{hf}$ =48.8 T, typical for magnetite in octa-hedral (Fe<sup>3+</sup>/Fe<sup>2+</sup>) and tetrahedral Fe<sup>3+</sup> sites [24–26], were found in the remanent slag. Additionally, another two sextets with hyperfine fields  $B_{hf} = 49.2$  T and 46.6 T that also could be associated to magnetite were obtained in demetallized slag which agrees to XRD results above. The rate areas between the octahedral and tetrahedral sites in the magnetite for both samples is around 1.8 (pure magnetite) [27], this suggests a nonstoichiometric magnetite due to partial substitutions by the other chemical elements registered by EDX and XRF above. A small sextet related to hematite hyperfine parameters [28, 29] is also observed for the remanent sample. The other two remaining sextets in demetallized slag belong to srebrodolskite (Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub>), which was also detected by XRD, in octahedral and tetrahedral sites with Fe<sup>2+</sup> and Fe<sup>3+</sup> oxidation states with  $B_{hf}$  = 43.9 T and  $B_{hf}$  = 50.85 T [30], respectively.  $Ca_2Fe_2SiO_4$  detected by XRD in the demetallized slag was associated with a doublet in Fe<sup>2+</sup> oxidation state with hyperfine parameters of IS = 0.87 mm/s and QS = 1.89 mm/s [31].

The information provided in the present work would be very useful for posterior attempts to recycle these types of steel slags. As a matter of fact, recently our research group has reported the reduction of the slags produced in SIDERPERU plant [5] and

also our initial achievements in recycling them into Q235 steel pieces [6], for which the information provided in the present work were very useful.

# 4 Conclusions

Remanent and demetallized slag produced in the ladle furnace from SIDERPERU steel plant were characterized by different techniques. The samples are hard sponge stones and dark grey in color. Fe, Ca, Si and Mn are the main elements detected by both EDX and XRF analysis. The common iron oxide phases detected by XRD were magnetite and wustite. Larnite, ankerite manganese and hematite were also found in remanent slag. Kirschteinite, srebrodolskite and ankerite manganese are the mineralogical phases found in the demetallized slag. The magnetic measurements suggest the presence of a superparamagnetic phase screening the Verwey transition signal of magnetite and the Morin transition of hematite in both samples. The Mössbauer spectroscopy analysis reveals that magnetic and wustite are not in stoichiometric state and that the wustite contributes with paramagnetic doublets in the spectra. The remanent and demetallized slag produced in the ladle furnace in SIDERPERU steel plant are promising materials for recycling in the steelmaking industry due to their high iron content minerals.

Supplementary Information The online version contains supplementary material available at https://doi.org/10.1007/s10751-022-01799-x.

Acknowledgements This work was supported by the CONCYTEC – World Bank – FONDECYT program "*Incorporación de Investigadores*", Contract No. 12 -2019 – FONDECYT – BM – INC. INV. The authors are indebted to Mr. E. Hurtado Aguilar and Mr. R.H Cervantes Altuna for allowing us access to SIDER-PERU to collect the samples.

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