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**SPECIAL TOPIC: Advanced Photocatalytic Materials** 

# Facile preparation of self-assembled MXene@Au@CdS nanocomposite with enhanced photocatalytic hydrogen production activity

Juanjuan Yin<sup>1,2</sup>, Fangke Zhan<sup>2</sup>, Tifeng Jiao<sup>1,2\*</sup>, Wenhan Wang<sup>3</sup>, Guangcong Zhang<sup>2</sup>, Jinghua Jiao<sup>2</sup>, Guiyuan Jiang<sup>3\*</sup>, Qingrui Zhang<sup>2</sup>, Jianmin Gu<sup>2</sup> and Qiuming Peng<sup>1</sup>

ABSTRACT Photocatalytic hydrogen production is considered a promising approach to generating clean sustainable energy. However, the conventional co-catalyst (e.g., Pt) used in photocatalytic hydrogen production is high-cost and difficult to obtain. Here, we designed and prepared a ternary nanocomposite MXene@Au@CdS, which can be used in the field of efficient and excellent photocatalytic hydrogen production. The MXene@Au@CdS has a hydrogen production rate of 17,070.43  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> (tested for 2 h), which is 1.85 times that of pure CdS nanomaterials. The improved hydrogen production performance of the MXene@Au@CdS is attributed to: (i) MXene provides more active adsorption sites and reaction centers for Au and CdS nanoparticles; (ii) the synergistic effect of Au's strong surface plasmon resonance expands the optical response range of CdS. Therefore, this work solves the problem of the solid connection between the surface functional groups of photocatalyst, and achieves rapid interface charge transfer and long-term stability during the hydrogen production.

Keywords: nanocomposite, MXene, photocatalytic hydrogen production

## INTRODUCTION

As technology advances, the environmental crisis and growing energy problems require clean and eco-friendly renewable energy. Photocatalytic hydrogen production is considered to be a promising approach to clean environments and sustainable energy, and has attracted widespread attention in the industry [1-3]. In addition,

the use of solar energy to generate hydrogen from water is expected to be a strategy to solve global energy problems [4]. In particular, photocatalytic water decomposition using semiconductor photocatalysts has shown great potential. In recent years, cocatalysts have been promising in enhancing the activity and stability of photocatalysts [5– 8]. However, the cocatalyst (e.g., Pt) used for hydrogen evolution generally is faced with high cost and difficulty in production, and the commercializing of the photocatalyst is a challenge. Therefore, it is crucial to find an inexpensive and highly active cocatalyst for the process of utilizing solar energy to produce hydrogen.

CdS has been extensively studied due to its visible light response and suitable energy band alignment for water decomposition [9,10]. However, pure CdS particles tend to aggregate and show high photo-generated charge recombination rate, resulting in shrinking surface area and inferior photocatalytic performance. Therefore, in order to address these issues, some studies have achieved efficient photocatalytic performance by adding various cocatalysts [11]. Among them, precious metal nanoparticles (such as Au, Ag) are effective co-catalysts to enhance the photocatalytic activity of CdS due to their local surface plasmon resonance [12]. The excited electrons on the Au nanoparticles could enter the conduction band of CdS by exciting local surface plasmon resonance, thereby improving the photocatalytic activity. In addition, the position of the noble metal also greatly affects the photocatalytic hydrogen production performance [13]. Metal/semiconductor heterostructures possess prominent

<sup>&</sup>lt;sup>1</sup> State Key Laboratory of Metastable Materials Science and Technology, Yanshan University, Qinhuangdao 066004, China

<sup>&</sup>lt;sup>2</sup> Hebei Key Laboratory of Applied Chemistry, School of Environmental and Chemical Engineering, Yanshan University, Qinhuangdao 066004, China

<sup>&</sup>lt;sup>3</sup> State Key Laboratory of Heavy Oil Processing, China University of Petroleum Beijing, Beijing 102249, China

<sup>\*</sup> Corresponding authors (emails: tfjiao@ysu.edu.cn (Jiao T); jianggy@cup.edu.cn (Jiang G))

role in regulating the charge transfer in photocatalysis. Xiao's group [14,15] have done a series of studies on photooxidation catalysis. For example, Zeng *et al.* [14,15] prepared a cascaded charge transfer channel in a multilayer photoanode by *in-situ* layer-by-layer construction to achieve plasma-induced photoelectrochemical water oxidation. Li *et al.* [16,17] investigated the general self-assembly of metal/metal chalcogenide heterostructures induced by surface linkers regulating multipliable photoreduction catalysis with adjustable charge flow.

Naguib et al. [18] reported a two-dimensional (2D) transition metal carbide/carbonitride (MXene) and added a new member to the 2D material family. MXene possesses a large number of hydrophilic functional groups (-OH and -O) on its surface which promote its strong interaction with water molecules and make it easy to establish strong connections with various semiconductors. The excellent metal conductivity of MXene ensures its effective carrier transfer, and the exposed metal sites (such as Ti, Nb, or V) on MXene are more likely to undergo redox reactions than carbon materials [19]. In addition, MXene could stably function in agueous solution. The above excellent properties render MXene a promising photocatalytic material. MXene materials demonstrate excellent physical and chemical properties such as abundant surface groups, strong electrical conductivity, excellent ductility and flexibility, which are applied in the fields of energy storage and conversion, air pollution and sewage treatment, electromagnetic shielding [20], etc. Yang et al. [21] reported that the velocity of holes on MXene were much higher than that of electrons according to the calculation results from density functional theory. Ran et al. [22] synthesized a CdS/Ti<sub>3</sub>C<sub>2</sub> composite and applied it to photolysis of hydrogen under visible light. The experimental results show that Ti<sub>3</sub>C<sub>2</sub> is an excellent cocatalyst, which greatly enhances the photohydrolysis effect of CdS.

Here, we designed a new and efficient three-component  $Ti_3C_2$  MXene@Au@CdS heterostructure for efficient photocatalytic hydrogen production under visible light. The Au nanoparticles have strong surface plasmon resonance, and hence they can be used as sensitizers to improve the photocatalytic efficiency and extend the optical response spectrum of CdS.  $Ti_3C_2$  MXene provides a stable matrix for the dispersion of the nanoparticles with high-efficiency charge separation. The synergistic effect from CdS to  $Ti_3C_2$  MXene migration could accelerate the photocatalytic process. Moreover the –O terminating groups on  $Ti_3C_2$  MXene can facilitate the release of H<sub>2</sub>. Therefore, this study purposed a strategy to obtain

rich active surfaces for strong bonds with the photocatalyst, rapid transfer of interface charges and long-term stability. On the other hand, it provides MXene with a new application. Besides, this stable and cost-effective semiconductor photocatalyst could effectively realize solar water decomposition.

## **EXPERIMENTAL SECTION**

#### Materials

Ethanol ( $C_2H_5OH$ ) and chloroauric acid tetrahydrate (HAuCl<sub>4</sub>·4H<sub>2</sub>O) were obtained from Sinopharm Holding Chemical Reagent Co., Ltd. (Beijing, China). Dimethyl sulfoxide (DMSO) and hydrofluoric acid were purchased from Tianjin Chemical Industry. Ultrapure water was prepared by using a Milli-Q Millipore filtration system (Millipore Co., Bedford, Massachusetts, USA). All chemicals were used as received without further purification.

#### Preparation of MXene

The MXene was prepared according to previous report [18]. Briefly, the MXene was obtained by etching away the Al layer of  $Ti_3AlC_2$  at 40°C with HF. After HF treatment, the sample was washed with demonized water until the pH value of the solution was around 6, and then dried under vacuum at 80°C for 24 h to obtain  $Ti_3C_2$  MXene.

#### Synthesis of Ti<sub>3</sub>C<sub>2</sub> MXene@Au composite

The Ti<sub>3</sub>C<sub>2</sub> MXene@Au composite was prepared by selfreduction method. Briefly, 100 mg of Ti<sub>3</sub>C<sub>2</sub> MXene was added to 100 mL of ultrapure water and stirred. Next, 3 mL of HAuCl<sub>4</sub> (0.1 mol L<sup>-1</sup>) was slowly added to the above suspension with constant stirring to trigger the selfreduction. After about half an hour, the mixture was centrifuged and washed several times with ultrapure water. Finally, lyophilization was carried out at  $-60^{\circ}$ C for 48 h to obtain the Ti<sub>3</sub>C<sub>2</sub> MXene@Au composite.

#### Preparation of MXene@Au@CdS composite

First, 0.04 g of  $Ti_3C_2$  MXene@Au solid powder was added to 35 mL of DMSO and stirred for about 20 min, and then 0.1 g of cadmium acetate was put into the solution and continuously stirred for about 2 h. Next, the mixed solution was transferred to a 50-mL autoclave and maintained at 180°C for 12 h. After cooling naturally, the mixed solution was taken out, and the supernatant was removed, leaving a precipitate, washed several times with ethanol, and collected by centrifugation. Then, 40 mL of ethanol was added to the precipitate, and the mixture was transferred to an autoclave, and maintained at 100°C for

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20 h. Finally, the sample was washed several times with ultrapure water to finally obtain  $Ti_3C_2$  MXene@Au@CdS composite.  $Ti_3C_2$  MXene@CdS was prepared in the same manner with  $Ti_3C_2$  MXene as the reactant.

#### Photocatalytic hydrogen production test

The hydrogen production performance of the prepared samples were tested under visible light (> 420 nm) from a 300W Xe arc lamp (Beijing Perfect Light Co., Ltd.) with a 420 nm cut filter. Generally, 4 mg of the prepared photocatalyst was dispersed in 80 mL of a mixed aqueous solution of 0.35 mol L<sup>-1</sup> Na<sub>2</sub>S and 0.25 mol L<sup>-1</sup> Na<sub>2</sub>SO<sub>3</sub>. In addition, a solution with 1 wt% of Pt promoter was added to the photocatalyst. Finally, the amount of H<sub>2</sub> produced was measured by using a gas chromatograph (GC) equipped with a thermal conductivity detector (TCD).

#### Photoelectric property test

Photoelectrochemical measurements were performed by using a CHI660 electrochemical workstation (Solartron) with a standard three-electrode configuration, and the prepared composite nanomaterials were used as photo-anodes with Pt foil and Ag/AgCl electrodes as the counter and reference electrodes, respectively, and Na<sub>2</sub>SO<sub>4</sub> (1 mol L<sup>-1</sup>) as the electrolyte. The simulated solar light source was a combination of a 300 W xenon lamp and an

AM 1.5G filter (Newport). Electrochemical impedance spectroscopy (EIS) was tested in a 1 mol  $L^{-1}$  Na<sub>2</sub>SO<sub>4</sub> solution with a direct current bias of 0.6 V against Ag/AgCl in the frequency range of  $10^{-1}$  to  $10^{5}$  Hz, where the tested alternative-current voltage was 10 mV.

#### Characterization

The CdS, Ti<sub>3</sub>C<sub>2</sub> MXene and Ti<sub>3</sub>C<sub>2</sub> MXene-based nanocomposite were characterized with scanning electron microscope (SEM), transmission electron microscope (TEM, HT7700, High-Technologies Corp., Japan) and high-resolution TEM (HRTEM, JEM-2100F, JEOL). Xray diffraction (XRD) was obtained on an X-ray diffract meter equipped with a CuKa X-ray radiation source and a Bragg diffraction device (SMART LAB, Rigaku, Japan). A NETZSCH STA 409 PC Luxxsi simultaneous thermal analyzer (Netzsch Instruments Manufacturing Co., Ltd., Germany) was used for thermo gravimetric (TG) analysis of the samples. UV-visible diffuse reflectance spectra (DRS) were obtained on a UV-visible spectrophotometer (Hitachi U3010) with BaSO4 as the substrate. X-ray photoelectron spectroscopy (XPS) data was obtained with Thermo Scientific ESCALab 250Xi under 200 W monochromatic AlKa radiations. The photocatalytic activity test was performed on a Labsolar II photocatalytic water analysis device produced by Beijing Perfila Co., Ltd.



Figure 1 SEM images of (a) CdS, (b) MXene, (c) MXene@Au, (d) MXene@CdS and (e) MXene@Au@CdS; (f-i) the elemental mappings of MXene@Au@CdS.

#### **RESULTS AND DISCUSSION**

#### Structural characterization

As shown in Fig. 1a, the size of CdS nanoparticles is around 150 nm. Fig. 1b shows the uniform layer structure of MXene, similar to the accordion [23]. When the chloroauric acid solution was added to a Ti<sub>3</sub>C<sub>2</sub> MXene solution of a certain concentration, the Au nanoparticles were obtained by the self-reduction and uniformly dispersed on the Ti<sub>3</sub>C<sub>2</sub> MXene, effectively preventing the aggregation of Au nanoparticles during the reduction process. The size of the Au nanoparticles is about 80-100 nm, as shown in Fig. 1c. CdS nanoparticles in-situ grew by adding cadmium acetate to the solution of  $Ti_3C_2$ MXene@Au. As shown in Fig. 1d, the sheet of MXene is not very obvious and the surface becomes rough after the process of generating CdS. Fig. 1e shows that the prepared Au nanoparticles and CdS nanoparticles uniformly disperse on the MXene layer with no large-scale aggregation. The MXene not only provides sites for the growth of Au and CdS nanoparticles, but also reduces the aggregation of particles during the preparation process. As shown in Fig. 1f-i, the mapping images of MXene @Au@CdS show the presence of Ti, Au, Cd, and S, and most of Au and CdS nanoparticles are loaded on the surface of MXene or between the MXene layers.

As shown in Fig. 2a, the CdS nanoparticles exhibit a uniform spherical structure with an average diameter of about 150 nm. MXene shows a typical layered structure with a uniform thickness (Fig. 2b). As shown in Fig. 2c, d, Au and CdS particles are reduced and uniformly attached to the surface of the MXene and each layer. Fig. 2e is a topographical view of the MXene@Au@CdS showing no significant aggregation, indicating that MXene is a good

substrate for dispersion growth. Fig. 2f is an HRTEM image of the heterostructure MXene@Au@CdS. The lattice spacing of the (111) crystal plane of CdS and the lattice spacing of the (111) crystal plane of Au in the HRTEM image are 0.336 and 0.238 nm, respectively, which are consistent with the data corresponding to the peak positions of CdS and Au in XRD. These results indicate that tight connections are formed between MXene, Au and CdS, which contribute to charge separation and transfer [24]. The energy dispersive X-ray spectroscopy (EDS) test of MXene@Au@CdS was performed, as shown in Fig. 3 [25,26], further confirming the heterostructure of MXene@Au@CdS and consistent with the TEM results.

XRD was used to characterize the prepared MXene, MXene@Au, MXene@CdS, CdS, and MXene@Au@CdS, as shown in Fig. 4a. For pure MXene, the diffraction peaks well match to the MXene reported [27]. In the curve of the MXene@Au@CdS, the peaks at 26.64°, 44.44° and 51.86° can be attributed to the (111), (220) and (311) of CdS. Furthermore, the peaks at 38.20°, 64.61° and 77.80° are assigned to the corresponding (111), (220) and (331) Au crystal faces. Therefore, it confirms that the nanocomposite MXene@Au@CdS has been successfully prepared [28].

The thermal stability of the obtained CdS, MXene@CdS and MXene@Au@CdS nanocomposites can be analyzed by TG under nitrogen [29–33], as shown in Fig. 4b. As the temperature gradually increases, the weight loss of the materials also gradually increases [34–36]. When the temperature rose to 400°C, the weight loss of the material reached the largest, with about 4.5%, 4% and 1.5% loss for CdS, MXene@CdS and MXene@Au@CdS, respectively. For CdS, the sequential addition of MXene and Au en-



Figure 2 TEM images of (a) CdS, (b) MXene, (c) MXene@Au, (d) MXene@CdS, (e) MXene@Au@CdS and HRTEM image of (f) MXene@Au@CdS.



Figure 3 TEM images and elemental mapping images of MXene@Au@CdS nanocomposite.



Figure 4 XRD patterns (a) and TG curves (b) of prepared MXene composites.



Figure 5 XPS profiles of the MXene@Au@CdS (a); C/Ti/Au/Cd/S bonding states in the MXene@Au@CdS (b-f).

hances the thermal stability of the material.

The chemical interface binding state of MXene @Au@CdS was demonstrated by XPS, as shown in Fig. 5a–f. The C 1s spectrum shows two distinct peaks at 284.84 and 288.67 eV, respectively, due to the C–OR and C–F bonds [37]. The peaks at 458.44 and 464.01 eV on the Ti 2p spectrum are attributed to the  $2p_{3/2}$  and  $2p_{1/2}$  peaks of Ti<sub>3</sub>C<sub>2</sub> [38]. For the Au 4f spectrum, two characteristic peaks of Au0 in the Au 4f<sub>7/2</sub> and Au 4f<sub>5/2</sub> states can be observed at 83.49 and 87.15 eV [39]. In addition, the binding energies of Cd 3d (Fig. 5e) and S 2p (Fig. 5f)

are consistent with the data of CdS [40].

A nitrogen adsorption-desorption measurement was performed on the obtained samples to observe the specific surface area and porosity, as shown in Fig. 6. Pure MXene exhibits a typical H3 isotherm, indicating its mesoporous properties. Table 1 lists the Brunauer-Emmett-Teller (BET) surface areas, pore volumes and pore sizes of the prepared samples. The BET surface area, pore volume and pore size of MXene (MXene@Au@CdS) are 22.2672 (34.5170) m<sup>2</sup> g<sup>-1</sup>, 0.0563 (0.0622) cm<sup>3</sup> g<sup>-1</sup> and 9.7948 (7.3110) nm, respectively, indicating that loading



Figure 6 N2 adsorption-desorption curves (a) and the pore size distributions (b) of MXene, MXene@Au, MXene@Au@CdS and CdS.

Table 1 The BET data of as-prepared materials

Catalysts	Surface area $(m^2 g^{-1})$	Pore volume $(\text{cm}^3 \text{g}^{-1})$	Pore size (nm )
MXene	22.2672	0.0563	9.7948
MXene@Au	12.2502	0.0316	9.4052
MXene@CdS	17.8133	0.0421	9.0111
MXene@Au@CdS	34.5170	0.0622	7.3110
CdS	45.4100	0.1096	9.1249

with Au@CdS nanoparticles, the MXene keeps a porous structure and possesses a relatively large specific surface area, which can provide some active sites for hydrogen production reactions.

As shown in Fig. 7, the MXene, MXene@Au, MXene @CdS, MXene@Au@CdS and CdS were investigated by UV-visible DRS spectra. As the wavelength increases, the absorption intensity of MXene gradually decreases, which can be mainly attributed to the unique absorption of carbonaceous materials [41]. In the DRS spectrum, the absorbance intensity of MXene is relatively low due to the small amount of MXene added. In addition, the CdS sample shows a steep absorption peak at 475 nm as expected, as shown in Fig. 7a. The absorption band of MXene@CdS composite is concentrated at about 475 nm, showing the characteristic of CdS. The curve of the obtained MXene@Au@CdS shows spectral characteristics of CdS and MXene. Moreover, since the amount of MXene added in the process of self-reduction reaction is relatively excessive, there is a shielding effect on the absorption peak of the formed Au nanoparticles. Thus, there is no obvious surface plasmon resonance (SPR) peak for Au nanoparticles in the DRS results, as shown in Fig. 7b. The above results indicate that the prepared ternary composite MXene@Au@CdS effectively enhances the absorption of CdS in visible light range, beneficial for



Figure 7 UV-vis DRS spectra of MXene, MXene@Au, MXene@CdS, and MXene@Au@CdS (a); enlarged curves of MXene@Au, MXene@Au@CdS and AuNPs solution (b).

the subsequent photocatalytic hydrogen production reaction [42,43].

#### Photocatalytic performance of the MXene@Au@CdS

According to literature report, under visible light irradiation ( $\lambda$ >420 nm), the activity of photocatalytic hydrogen production has been greatly improved [44-46]. Thus, the present prepared samples were evaluated for photocatalytic H<sub>2</sub> release in aqueous Na<sub>2</sub>S and Na<sub>2</sub>SO<sub>3</sub> solutions under visible light irradiation, with the results of hydrogen evolution tests shown in Fig. 8a. The amount of released hydrogen was not detected in the absence of photocatalyst or light irradiation. In addition, the evolution of hydrogen from MXene@Au is inconspicuous, indicating that the nanocomposite MXene@Au has weak photocatalytic hydrogen evolution activity. Both of CdS composite materials show higher photocatalytic hydrogen production activity than pure CdS. The amount of hydrogen generated gradually increases from CdS, MXene @CdS, to MXene@Au@CdS ternary nanocomposite,. The composite MXene@CdS has a hydrogen production rate of 12,340.50  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, while the MXene@ Au@CdS production shows a hydrogen rate of 17,070.43  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, which corresponds to 1.336 and 1.85 times that of CdS (9234.46  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>), respectively. This is because CdS forms a stronger bond with MXene and MXene@Au during hydrothermal reactions, which greatly promotes the rapid transfer of interface charge, highlighting the great potential of MXene. Moreover the powerful SPR with the sensitizer Au can greatly expand the optical response range of CdS. In addition, MXene is inactive against H<sub>2</sub> production under visible light irradiation, further verifying its role as a cocatalyst rather than a photocatalyst. In order to investigate the stability of MXene@Au@CdS, cyclic hydrogen evolution experiments were performed, as shown in Fig. 8b. The MXene@Au@CdS shows relatively high stability and durability after four consecutive cycle tests.

In order to study the overall charge transfer efficiency and surface charge transfer efficiency of the obtained



Figure 8 Hydrogen evolution profiles of CdS and MXene-based composites under visible-light illumination (a); 4 mg of the photocatalyst was dispersed in a mixed aqueous solution of 80 mL of 0.35 mol  $L^{-1}$  Na<sub>2</sub>S and 0.25 mol  $L^{-1}$  Na<sub>2</sub>SO<sub>3</sub>; cycle runs for the photocatalytic H<sub>2</sub> production over MXene@Au@CdS nanocomposite (b).



Figure 9 (a) *i-t* curves, (b) EIS of MXene, MXene@Au, MXene@CdS, MXene@Au@CdS and CdS.



Figure 10 Schematic of the MXene@Au@CdS preparation flow chart and energy band structure diagram.

nanocomposites, the transient photocurrent (TPC) response and EIS tests were performed, as shown in Fig. 9. The anode photocurrent increases during illumination and disappears without illumination (Fig. 9a). The MXene@Au@CdS as the photoanode produces higher photocurrent than other samples, demonstrating that MXene@Au@CdS has higher charge separation efficiency. At the same time, the photocurrent is very stable in continuous on/off cycles. Under visible light irradiation, since the semicircular diameter of MXene@Au@CdS is smaller than those of other three materials, showing that the interface charge transfer resistance of MXene@Au@CdS is smaller, as shown in Fig. 9b. Therefore, it can be concluded that MXene@Au@CdS has better interface charge transfer capability on the surface than other samples.

The illustration of preparation and hydrogen production for the MXene@Au@CdS is shown in Fig. 10. First, a certain amount of concentrated HF solution was added to  $Ti_3AlC_2$  suspension at room temperature, and Al layer in the  $Ti_3AlC_2$  was etched to obtain  $Ti_3C_2$  MXene. Next, a certain amount of HAuCl<sub>4</sub> solution was added to the  $Ti_3C_2$  MXene suspension under stirring to trigger the selfreduction. And then MXene@Au composite was obtained after a suitable reaction time. Subsequently MXene@Au powder was added to DMSO with stirring for about 20 min. Then, appropriate amount of  $Cd(CH_3CO_2)_2$  solid was added to the above suspension, and continuously stirred for 2 h. After that, the mixed solution was poured into an autoclave, which was heated up to 180°C and kept for 12 h. Finally, ternary MXene@Au@CdS was obtained. A reasonable mechanism of electron-hole separation for the MXene@Au@CdS is proposed, as shown in Fig. 10. CdS is excited by visible light to generate electrons in the conduction band (CB) and holes in the valence band (VB) [46–48]. Since the original  $E_{\rm F}$  of n-type CdS is much more negative than the original  $E_{\rm F}$  of MXene, the close contact between CdS and MXene will lead the electrons to transfer from CdS to MXene. At the same time, a fixed positive charge remains near the interface of CdS and MXene to form holes. Therefore, a Schottky junction is formed between MXene and CdS. Finally, the terminal photoelectrons of functional group -O in MXene can effectively reduce protons in aqueous solution by releasing H<sub>2</sub> gas. The electrons transfer by two paths. One is that the Schottky junction effectively captures the lightsensing electrons, which exist as an electron trap. The other is accelerated by the strong SPR absorption of Au which could extend the response spectrum of CdS [49]. This study demonstrates a new clue for the design of catalytic materials [50-58] and self-assembled functionalized nanocomposites [59-65].

#### **CONCLUSIONS**

In summary, a new ternary nanocomposite MXene @Au@CdS was designed and applied in the field of photocatalytic hydrogen production, showing high catalytic activity. Under visible light irradiation, the hydrogen production rate of the MXene@Au@CdS is 17,070.43  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, which is 1.85 times that of CdS. This nanocomposite not only avoids the problem of lack of rich surface functions to establish a strong connection

with photocatalyst, but also achieves rapid transfer of interface charges and long-term stability. The present study provides a strategy for opening up a new application field for MXene-based catalytic composite material. The as-prepared stable and cost-effective semiconductor photocatalyst is expected to effectively realize solar water decomposition.

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**Author contributions** Yin J, Jiao T, and Jiang G designed the project and performed the experiments. Zhan F, Wang W, Zhang G, Jiao J, Zhang Q, Gu J and Peng Q characterized the materials and discussed the results of the experiments. All the authors contributed to the general discussion.

Conflict of interest The authors declare no conflict of interest.



Juanjuan Yin is a PhD candidate supervised by Prof. Tifeng Jiao at the School of Environmental and Chemical Engineering of Yanshan University. Her current research interest focuses on MXene-based nanocomposites for photo- and electrocatalytic hydrogen production.



**Guiyuan Jiang** received his BE and Master degrees from China University of Petroleum, Beijing and PhD degree from the Institute of Chemistry, CAS in 2000, 2003 and 2006, respectively. He joined China University of Petroleum, Beijing in 2006, and became a full professor in 2012. He was a visiting postdoctor at the University of California, Riverside in 2010, and visiting scholar in Tsinghua University in 2013–2014. His research interest mainly focuses on the energy catalysis, including catalytic conversion of light hydrocarbons and artificial photosynthesis.

# 自组装纳米复合材料MXene@Au@CdS的制备及 其光催化制氢活性

 尹娟娟<sup>1,2</sup>, 展方可<sup>2</sup>, 焦体峰<sup>1,2\*</sup>, 王文涵<sup>3</sup>, 张广聪<sup>2</sup>, 焦菁华<sup>2</sup>, 姜桂元<sup>3\*</sup>, 张庆瑞<sup>2</sup>, 谷建民<sup>2</sup>, 彭秋明<sup>1</sup>



Tifeng Jiao received his PhD in physical chemistry from the Institute of Chemistry, Chinese Academy of Sciences (CAS). He was a postdoctoral fellow of CNRS (Centre National de la Recherche Scientifique) with A.P. Girard-Egrot (Université Claude Bernard Lyon 1, France). Currently, he is a full professor and vice director of the School of Environmental and Chemical Engineering, Yanshan University. His current research interests include the synthesis of new self-assembled nanostructured materials and nanocomposites, and their related properties.

**摘要** 光催化制氢被认为是一种有效获得清洁可持续能源的方法. 常规光催化制氢使用的助催化剂(如Pt)具有成本高和难以获得的 缺点.本文设计并制备了三元纳米复合材料MXene@Au@CdS,可用 于高效光催化制氢. MXene@Au@CdS的氢气产生率为17070.43 μmol g<sup>-1</sup> h<sup>-1</sup> (测试时间2 h),是纯CdS纳米材料的1.85倍. MXene @Au@CdS优异的制氢性能归因于: (i) MXene为Au和CdS纳米颗粒 提供了更多的活性吸附位点和反应中心. (ii) 金的强表面等离子体 共振协同效应使得CdS的光学响应范围增大.本工作解决了光催化 剂表面官能团之间的固态连接问题,并在制氢过程中实现了快速 界面电荷转移和长期稳定性.