



# Synthesis and photoelectrical performance of nanoscale PbS and Bi<sub>2</sub>S<sub>3</sub> co-sensitized on TiO<sub>2</sub> nanotube arrays

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**Abstract** TiO<sub>2</sub> films have been widely applied in photovoltaic conversion techniques. TiO<sub>2</sub> nanotube arrays (TiO<sub>2</sub> NAs) can be grown directly on the surface of metal Ti by the anodic oxidation method. Bi<sub>2</sub>S<sub>3</sub> and PbS nanoparticles (NPs) were firstly co-sensitized on TiO<sub>2</sub> NAs (denoted as PbS/Bi<sub>2</sub>S<sub>3</sub>(*n*)/TiO<sub>2</sub> NAs) by a two-step process containing hydrothermal and sonication-assisted SILAR method. When the concentration of Bi<sup>3+</sup> is 5 mmol/L, the best photoelectrical performance was obtained under simulated solar irradiation. The short-circuit photocurrent ( $J_{sc}$ ) and photoconversion efficiency ( $\eta$ ) of PbS/Bi<sub>2</sub>S<sub>3</sub>(5)/TiO<sub>2</sub> NAs electrode were 4.70 mA/cm and 1.13 %, respectively.

**Keywords** Solar cells · TiO<sub>2</sub> nanotube arrays · Bi<sub>2</sub>S<sub>3</sub> · PbS · Surface photovoltage

## 1 Introduction

Industrial development and population growth have led to a surge in the global energy demands. Solar energy is an important source of renewable energy and has been widely

applied in various fields including transport. Solar-powered cars [1] and aircrafts [2] depend on solar cells to convert sunlight into electricity to drive electric motors. In the future, they are expected to play a key role in reducing consumption of burning fossil fuels. Currently, crystalline silicon is the most common material used for solar cells. However, a large number of toxic substances are generated during the production of crystalline silicon.

Dye-sensitized solar cells (DSSCs) [3] have been well developed over the past two decades. With the advantages of facile preparation and low cost, nanoscale inorganic semiconductors are considered as ideal substitutes for organic dyes. Moreover, their band gaps can be conveniently tailored by controlling the size of nanoparticles [4]. PbS [5], Bi<sub>2</sub>S<sub>3</sub> [6], and other inorganic semiconductors have been used to sensitize *n*-type wide-band gap semiconductor, such as TiO<sub>2</sub>. TiO<sub>2</sub> nanotube arrays (TiO<sub>2</sub> NAs) are attracting considerable interest, because they can provide direct and efficient transport channels for photogenerated electrons, and promote the separation of photogenerated electrons and holes [7]. TiO<sub>2</sub> NAs can be grown directly on the surface of metal Ti by the anodic oxidation method [8]. Ti alloy with lightweight and high-strength is considered an ideal material for solar-powered cars and aircrafts. Therefore, we intend to sensitize TiO<sub>2</sub> NAs with inorganic semiconductors to construct solar cells, and study their photoelectric properties.

Previously, we had demonstrated that both Bi<sub>2</sub>S<sub>3</sub> and PbS nanoparticles are efficient sensitizers for TiO<sub>2</sub> NAs [6, 9]. Nevertheless, the strategy of TiO<sub>2</sub> sensitized by the single inorganic semiconductor limited the further improvement of the photoelectric performance and light absorption property. As a result, co-sensitized strategy has been adopted by many researchers. On the other hand, it has been reported that the bulk nano-heterojunction

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structure formed by Bi<sub>2</sub>S<sub>3</sub> and PbS quantum dots can enhance the carrier lifetime resulting from the separation of nanoscale phase, and then the photoelectric properties can be improved [10]. Thus, in this work, we first prepared PbS and Bi<sub>2</sub>S<sub>3</sub> nanoparticles to co-sensitize TiO<sub>2</sub> NAs, and studied their photoelectric properties. The preparation process and photoelectric properties of PbS and Bi<sub>2</sub>S<sub>3</sub> nanoparticles co-sensitized on TiO<sub>2</sub> NAs were discussed.

## 2 Methods

### 2.1 Preparation of TiO<sub>2</sub> NAs

An anodic oxidation method was used to grow TiO<sub>2</sub> NAs on the surface of Ti foil. In brief, Ti foil (3 cm × 1 cm × 0.25 mm) was anodized in ethylene glycol containing 0.25 wt% NH<sub>4</sub>F at a constant voltage of 60 V for 6 h, while a larger Pt foil (3 cm × 4 cm) was used as the counter electrode. All experiments were carried out at room temperature. The as-prepared samples were annealed at 450 °C for 3.5 h.

### 2.2 Preparation of Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> NAs

Bi<sub>2</sub>S<sub>3</sub> was deposited into anatase TiO<sub>2</sub> NAs by a hydrothermal method. At first, Bi(NO<sub>3</sub>)<sub>3</sub> (100 mL) and Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub> (100 mL) aqueous solutions with certain concentrations were prepared, respectively. The molar ratio of Bi<sup>3+</sup> to S<sub>2</sub>O<sub>3</sub><sup>2-</sup> was fixed at 2:3. In this work, three concentrations of Bi<sup>3+</sup> were chosen, 1, 5, and 10 mmol L<sup>-1</sup>. And then, both of Bi(NO<sub>3</sub>)<sub>3</sub> and Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub> aqueous solutions were mixed and stirred thoroughly to obtain precursor solution. The aforementioned precursor solution was transferred to a 25-mL Teflon-lined stainless steel autoclave containing anatase TiO<sub>2</sub> NAs sample. The autoclave was sealed and maintained at 100 °C for 24 h. The obtained sample is designated as Bi<sub>2</sub>S<sub>3</sub>(*n*)/TiO<sub>2</sub> NAs, where *n* represents the concentration of Bi<sup>3+</sup>.

### 2.3 Preparation of PbS/Bi<sub>2</sub>S<sub>3</sub>(*n*)/TiO<sub>2</sub> NAs

PbS nanoparticles (PbS NPs) were attached to Bi<sub>2</sub>S<sub>3</sub>(*n*)/TiO<sub>2</sub> NAs using a sonication-assisted successive ionic layer adsorption and reaction (SILAR) method. In other words, the as-prepared Bi<sub>2</sub>S<sub>3</sub>(*n*)/TiO<sub>2</sub> NAs electrode was successively immersed into 5 mmol/L Pb(NO<sub>3</sub>)<sub>2</sub> aqueous solution, D.I. water, 5 mmol L<sup>-1</sup> Na<sub>2</sub>S aqueous solution, and D.I. water again for 20 s each. According to our previous study [9], the SILAR cycles were carried out five times under ultrasonic waves of 20 kHz and 100 W to form PbS NPs of lower 4 nm. The sample was denoted as PbS/Bi<sub>2</sub>S<sub>3</sub>(*n*)/TiO<sub>2</sub> NAs.

## 2.4 Characterization

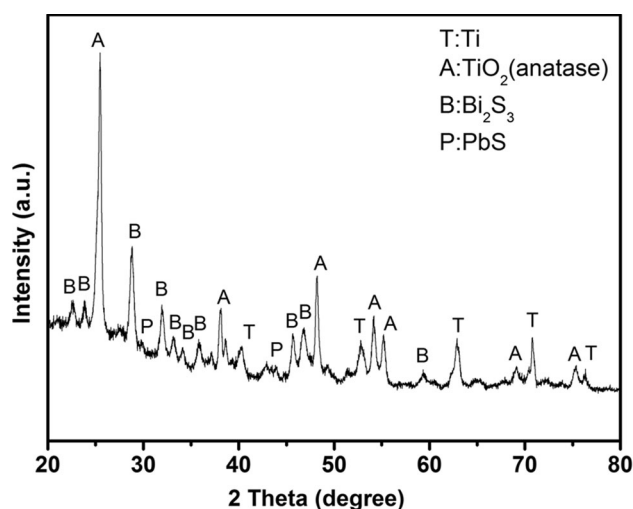
The morphologies and structure of all samples were examined using a field emission scanning electron microscope (FESEM, JSM 7001F, JEOL, Japan) and powder X-ray diffraction (XRD, PANalytical diffractometer), respectively. Energy dispersive X-ray analysis (EDX) attached to FESEM was used to qualitatively and quantitatively analyze the chemical composition of samples.

A surface photovoltage (SPV) measurement system was used to characterize the transport characteristics of photo-generated electron–holes pairs. For more details, please refer to Refs [6, 9, 11].

Photoelectric property of electrodes was studied using electrochemical work-station (Lanlike 2006A, China) in 0.5 mol/L Na<sub>2</sub>S electrolyte. PbS/Bi<sub>2</sub>S<sub>3</sub>(*n*)/TiO<sub>2</sub> NA (on Ti foil) was used as the working electrode, while Pt foil and SCE were used as counter and reference electrode, respectively. The electrodes were tested under simulated sunlight at AM 1.5 (100 mW/cm<sup>2</sup>) from a 500 W xenon lamp and an AM 1.5 filter. The effective surface area of the electrode was 1.0 × 1.5 cm for illumination.

## 3 Results and discussion

Previously, Bi<sub>2</sub>S<sub>3</sub> NPs had successfully been deposited into TiO<sub>2</sub> NAs by hydrothermal method at 100 °C [6]. And PbS NPs also can be attached to TiO<sub>2</sub> NAs by a sonication-assisted SILAR method [9]. So, we believed that nanoscale Bi<sub>2</sub>S<sub>3</sub> and PbS can be successively deposited on TiO<sub>2</sub> NAs using two methods mentioned above. Figure 1 shows the XRD pattern of PbS/Bi<sub>2</sub>S<sub>3</sub>(1)/TiO<sub>2</sub> NAs. Besides Ti and

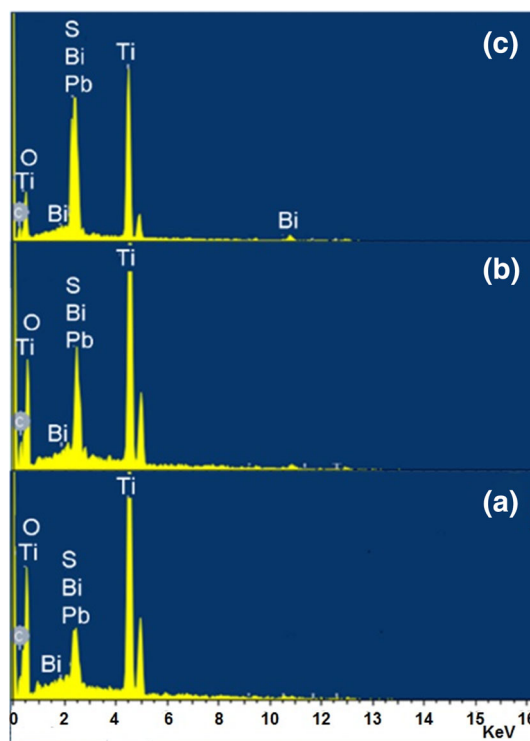


**Fig. 1** XRD pattern of PbS/Bi<sub>2</sub>S<sub>3</sub>(1)/TiO<sub>2</sub> NAs. Symbols T, A, B, and P represent the peaks of metal Ti, anatase TiO<sub>2</sub>, orthorhombic Bi<sub>2</sub>S<sub>3</sub>, and cubic PbS, respectively

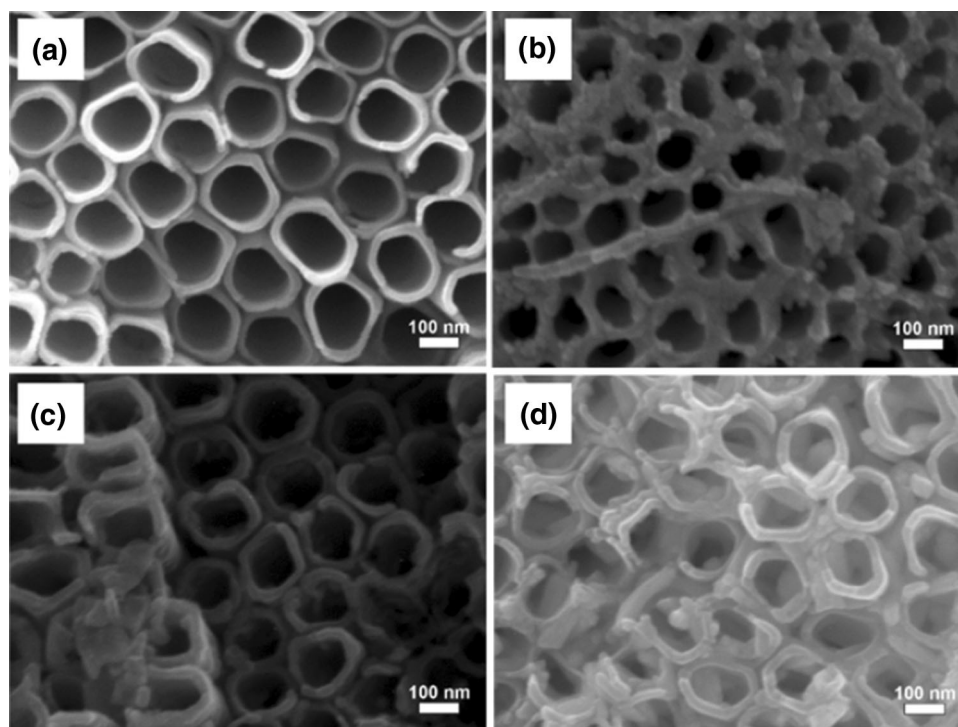
anatase  $\text{TiO}_2$ , X-ray diffraction peaks of orthorhombic  $\text{Bi}_2\text{S}_3$  and cubic  $\text{PbS}$  can also be found. Symbols T, A, B, and P in Fig. 1 represent the peaks of Ti,  $\text{TiO}_2$ ,  $\text{Bi}_2\text{S}_3$ , and  $\text{PbS}$ , respectively. The XRD result confirms that  $\text{PbS}/\text{Bi}_2\text{S}_3(n)/\text{TiO}_2$  NAs composite can be formed.

Figure 2 shows the typical top-view FESEM images of pure  $\text{TiO}_2$  NAs and  $\text{PbS}/\text{Bi}_2\text{S}_3(n)/\text{TiO}_2$  NAs. The regularly arranged  $\text{TiO}_2$  NAs fabricated by anodic oxidation are shown in Fig. 2a; its diameter and wall thickness are about 140 and 20 nm, respectively. The views of  $\text{PbS}/\text{Bi}_2\text{S}_3(n)/\text{TiO}_2$  NAs (Fig. 2b, c, and d) illustrate that some nanoparticles are decorated into  $\text{TiO}_2$  nanotubes, including interior of nanotubes and outside of nanotube walls. According to the XRD result, we confirm that these nanomaterials are  $\text{Bi}_2\text{S}_3$  and  $\text{PbS}$ . The filling degrees of  $\text{TiO}_2$  NAs increased along with the concentration of  $\text{Bi}^{3+}$ . We assume that the content of  $\text{PbS}$  in all samples remains unchanged, because  $\text{PbS}$  NPs were synthesized using the same conditions. The content of  $\text{Bi}_2\text{S}_3$  in  $\text{TiO}_2$  NAs was determined by EDX experiments, as shown in Fig. 3. The results of EDX analysis show that the mass fractions of the Bi element are 4.18 %, 14.24 %, and 30.36 % with respect to that of  $\text{PbS}/\text{Bi}_2\text{S}_3(n)/\text{TiO}_2$  NAs, respectively for  $n = 1, 5, \text{ and } 10$ .

$\text{TiO}_2$  is an important  $n$ -type semiconductor with a wide-band gap ( $E_g = 3.2 \text{ eV}$ ), while  $n$ -type  $\text{Bi}_2\text{S}_3$  has a narrow



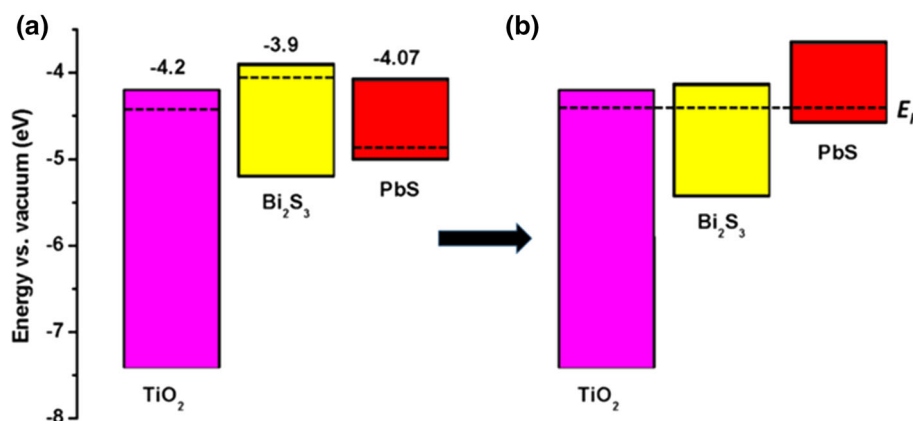
**Fig. 3** EDX spectra of  $\text{PbS}/\text{Bi}_2\text{S}_3(n)/\text{TiO}_2$  NAs. a, b, and c for  $n = 1, 5, \text{ and } 10$ , respectively



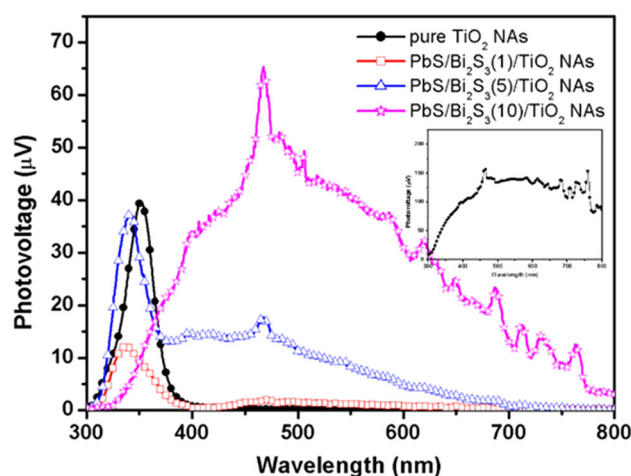
**Fig. 2** FESEM images: top-view of a pure  $\text{TiO}_2$  NAs and b, c, and d  $\text{PbS}/\text{Bi}_2\text{S}_3(n)/\text{TiO}_2$  NAs composites ( $n = 1, 5, \text{ and } 10$ , respectively)

$E_g$  of 1.3 eV. Photogenerated electrons can easily transfer from the Bi<sub>2</sub>S<sub>3</sub> surface to TiO<sub>2</sub>, because Bi<sub>2</sub>S<sub>3</sub> has a more negative conduction band (CB) [12]. The  $E_g$  of bulk PbS is only 0.41 eV, but is directly dependent on the size of nanoparticles. According to Refs. [13] and [14], PbS NPs of 4 nm or lower has a more negative CB than that of TiO<sub>2</sub>. Previously, we found that PbS NPs (<4 nm) can be fabricated using a sonication-assisted SILAR method when cycle number ( $n$ ) is 5 [9]. Figure 4a schematically illustrates energy band diagram of PbS, Bi<sub>2</sub>S<sub>3</sub>, and TiO<sub>2</sub>. Thus, we hope to construct PbS/Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> NAs heterojunction to promote the separation and transport of photogenerated electron–hole pairs.

SPV system is an effective tool to investigate the behavior of photogenerated charge carriers. SPV spectra of PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NAs are shown in Fig. 5. Simultaneously, corresponding SPV spectrum of pure Bi<sub>2</sub>S<sub>3</sub> is shown in the inset of Fig. 5. The signal of SPV is due to the difference of surface potential barriers before and after light illumination [15]. Noticeable SPV response ranging from 300 to 400 nm is found for TiO<sub>2</sub> NAs attributed to its wide  $E_g$ . For pure Bi<sub>2</sub>S<sub>3</sub>, it has a strong response in the whole-tested wave spectrum, especially 400–800 nm, because Bi<sub>2</sub>S<sub>3</sub> has a narrow  $E_g$  (1.3 eV). The SPV response of PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NAs increases gradually with the amount of Bi<sub>2</sub>S<sub>3</sub> in the range over 400 nm. This is because SPV response is directly related to the amount of semiconductor. However, in the range of 300–400 nm, PbS/Bi<sub>2</sub>S<sub>3</sub>(5)/TiO<sub>2</sub> NAs has strongest SPV response intensity in all PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NAs. This result suggests that there is one or more heterojunction among PbS, Bi<sub>2</sub>S<sub>3</sub>, and TiO<sub>2</sub> NAs, which aids the separation of photogenerated electrons and holes. Both Bi<sub>2</sub>S<sub>3</sub> and TiO<sub>2</sub> are  $n$ -type semiconductors, the direction of an internal electric field in Bi<sub>2</sub>S<sub>3</sub> ( $n$ )/TiO<sub>2</sub> heterojunction is opposite to that of built-in electric field in  $p$ -type PbS. Thus, when PbS NPs are coupled with Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NAs, the SPV response in 300–400 nm spectral range appears to decline.



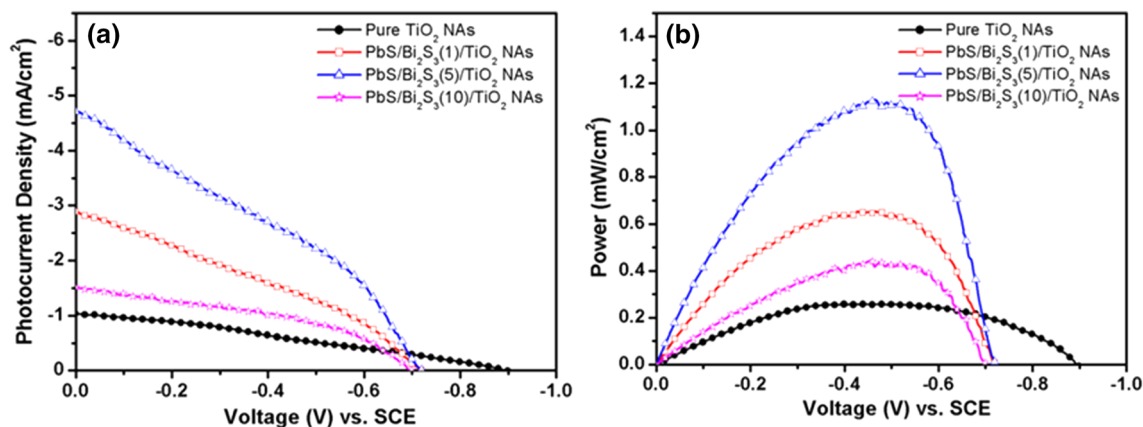
**Fig. 4** Energy band diagram of PbS, Bi<sub>2</sub>S<sub>3</sub>, and TiO<sub>2</sub> before **a** and after **b** forming heterojunction. The dotted lines indicate the position of Fermi level



**Fig. 5** Surface photovoltage (SPV) spectra of pure TiO<sub>2</sub> NAs and PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NAs ( $n = 1, 5,$  and  $10,$  respectively), and pure Bi<sub>2</sub>S<sub>3</sub> (*inset*)

The  $J$ - $V$  and  $P$ - $V$  characteristics of pure TiO<sub>2</sub> NAs and PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NAs electrodes in 0.5 mol L<sup>-1</sup> Na<sub>2</sub>S electrolyte are shown in Fig. 6a, b, respectively. And then, the corresponding parameters of photoelectrical performances are summarized in Table 1. The plain TiO<sub>2</sub> NAs electrode exhibits a negligible photoelectrical property. The short-circuit photocurrent ( $J_{sc}$ ) and photo conversion efficiency ( $\eta$ ) of plain TiO<sub>2</sub> NAs electrode are 1.01 mA/cm<sup>2</sup> and 0.26 %, respectively. For PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NA electrodes, the  $J_{sc}$  firstly increases and then decreases with the concentration of Bi<sup>3+</sup> ( $n, n = 1, 5,$  and  $10$ ), reaching 2.88, 4.70, and 1.51 mA/cm<sup>2</sup>, respectively. Accordingly, the highest  $\eta$  of 1.13 % is obtained from PbS/Bi<sub>2</sub>S<sub>3</sub>(5)/TiO<sub>2</sub> NAs electrode, which is around four times higher than that of plain TiO<sub>2</sub> NAs electrode. This result indicates that PbS and Bi<sub>2</sub>S<sub>3</sub> nanoparticles co-sensitized on TiO<sub>2</sub> NAs can remarkably improve their photoelectric property.





**Fig. 6**  $J$ - $V$  (a) and  $P$ - $V$  (b) characteristics of pure TiO<sub>2</sub> NA and PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NA electrodes ( $n = 1, 5$ , and  $10$ , respectively)

**Table 1** Parameters of photoelectric property of TiO<sub>2</sub> NA and PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NA electrodes

Sample	$J_{sc}$ (mA/cm <sup>2</sup> )	$V_{oc}$ (V)	$FF$	$\eta$ (%)
Pure TiO <sub>2</sub> NAs	1.04	0.90	0.28	0.26
PbS/Bi <sub>2</sub> S <sub>3</sub> (1)/TiO <sub>2</sub> NAs	2.88	0.72	0.31	0.65
PbS/Bi <sub>2</sub> S <sub>3</sub> (5)/TiO <sub>2</sub> NAs	4.70	0.72	0.33	1.13
PbS/Bi <sub>2</sub> S <sub>3</sub> (10)/TiO <sub>2</sub> NAs	1.52	0.70	0.42	0.45

$J_{sc}$ ,  $V_{oc}$ ,  $FF$ , and  $\eta$  are the short-circuit current density, open-circuit voltage, fill factor, and overall power conversion efficiency, respectively

The improved photoelectrical property of PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NAs electrodes may be attributed to several reasons. Firstly, the light response of TiO<sub>2</sub> NAs electrodes is extended from the UV to visible region after being co-sensitized by PbS and Bi<sub>2</sub>S<sub>3</sub> NPs. This means that more sunlight can be used to generate the photocurrent. Secondly, as shown in Fig. 4b, Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> and PbS/Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> heterojunctions have been formed, and then the Fermi levels of TiO<sub>2</sub>, Bi<sub>2</sub>S<sub>3</sub>, and PbS tend to reach balance, producing efficient charge transfer channel. Thirdly, the interfacial electric field in Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> and PbS/Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> heterojunction may prevent the recombination of photogenerated electron-hole pairs. However, higher concentration of Bi<sup>3+</sup> would cause conglomeration of the crystal nucleus, and moreover, excess Bi<sub>2</sub>S<sub>3</sub> would act as potential barrier for charge transfer. As a result, the photoelectric properties of PbS/Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub> NAs would not be further improved. The best photoelectrical performance is obtained from PbS/Bi<sub>2</sub>S<sub>3</sub>(5)/TiO<sub>2</sub> NAs electrode, in which  $J_{sc}$  and  $\eta$  are 4.70 mA/cm<sup>2</sup> and 1.13 %, respectively.

## 4 Conclusions

PbS/Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> NAs has been fabricated by a three-step process containing hydrothermal and sonication-assisted SILAR method. PbS NPs were attached to Bi<sub>2</sub>S<sub>3</sub>( $n$ )/TiO<sub>2</sub>

NAs, in which the size of PbS NPs was maintained lower than 4 nm by controlling the SILAR cycles. PbS and Bi<sub>2</sub>S<sub>3</sub> NPs were co-sensitized on TiO<sub>2</sub> NAs to enhance their photoelectric property. When the concentration of Bi<sup>3+</sup> was 5 mmol/L, the best photoelectric property was obtained.  $J_{sc}$  and  $\eta$  of PbS/Bi<sub>2</sub>S<sub>3</sub>(5)/TiO<sub>2</sub> NAs were respectively 4.70 mA/cm<sup>2</sup> and 1.13 % under an illumination of 100 mW/cm<sup>2</sup>.

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