ORIGINAL ARTICLE

Morphology control in precision polyolefins

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Abstract Herein, we review the major advances in controlling polyethylene morphology through precise control of branch frequency and identity. This control is made possible by the acyclic diene metathesis reaction.

Keywords ADMET · Olefin metathesis · Precision polyolefins · Polyethylene

Introduction

Polymer chemistry has changed the world. From the polyamides of Wallace Carothers to synthetic rubbers to the plastics that are ubiquitous in modern life, polymer science has provided the world with materials that have made modern life more convenient, affordable, and advanced. Despite being a relatively young field (less than a hundred years ago Hermann Staudinger was chided for even believing that macromolecules existed [1]), polymer science has built a considerable body of knowledge regarding structure-property relations. This knowledge begins with a thorough understanding of the primary structure of the polymer, i.e., the polymer repeat unit. Increasing levels of understanding can be built on this foundation (Fig. 1). How a single polymer chain organizes itself gives rise to the secondary structure. Multiple chains interact to give a tertiary structure. Finally, there is the morphological structure, and it is this structure that ultimately determines the material properties. Understanding, however, begins at the most fundamental level—the repeat unit.

ADMET polymerization

We study olefin metathesis. Our understanding of this reaction provides the foundation on which we have built the precision polyolefin paradigm. Our group has pioneered the field of acyclic diene metathesis (ADMET) polymerization [2], in which a diene is condensed via metathesis to give an unsaturated polymer and a small molecule condensate (usually ethylene) (Fig. 2).

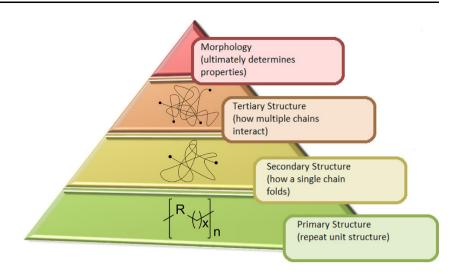
The development of well-defined catalysts was key. Initial work using WCl₆ and a Lewis acid to promote olefin metathesis produced a mixture of products—low molecular weight oligomers, insoluble material, and gaseous ethylene. It was discovered that to form high molecular weight polymer, there could be no side reactions (particularly vinyl addition) competing with the olefin metathesis. Serendipitously, this realization coincided with the report of the first Schrock catalysts, enabling the rapid development of the burgeoning field of metathesis polymerization.

ADMET polymerization is a reversible reaction. Consequently, to achieve high molecular weights the reaction equilibrium must be driven toward polymer formation by the removal of the small molecule condensate. Thus, high vacuum conditions are almost always required. ADMET polymerization is concentration dependent, with bulk conditions often being the optimum. Like other condensation polymerizations, ADMET polymerization results in relatively broad molecular weight distributions (PDI \approx 2.0). Unlike ring-opening metathesis polymerization (ROMP), the metathesis chain polymerization, ADMET produces

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Fig. 1 Structural hierarchy



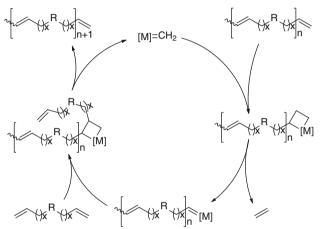
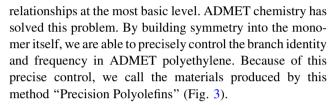


Fig. 2 ADMET polymerization mechanism

polymers that are most often semicrystalline, with molecular weights typical of other polycondensations. As ring-strain in the monomer is not required, a wide variety of diene monomers is possible. ADMET has been used to produce polymers containing simple hydrocarbons [3, 4], aromatic amines [5], branched acids [6, 7], ethers [8, 9], silanes [10], acetals [11], esters [12–15], aromatic moieties [16–20], thioethers [21], and ketones [14, 22], to name a few.

What are "precision polyolefins"?

Polyolefins are usually made by chain polymerization. In the case of polyethylene, the polymerization conditions of ethylene can be varied to produce materials with a wide variety of properties (HDPE, LDPE, LLDPE, just to name a few). These properties are almost always related to branching—branch frequency and branch identity. While the amount of branching can be grossly controlled and characterized, the primary structure of the polymer is random. This makes is it impossible to systematically study the structure–property



Effect of branch identity

Among the first polymers to be synthesized and studied using ADMET was a precision analog of an ethylene/ propylene copolymer, which may be described as polyethylene with a methyl branch on every 21st carbon. This material was then compared to a polymer where the methyl group was on average on every 21st carbon. The difference between the thermal properties of these two materials is striking. The precise polymer showed a sharp melt transition, whereas the random analog had a very broad transition (Fig. 4). It is clear that the precise placement of branches affects material properties. But are these methyl branches included in the crystal lattice of the polymer?

ADMET polyethylene is orthorhombic. When methyl groups are incorporated on every 21st carbon, they are indeed included in the crystal lattice; however, there is a change to a triclinic lattice with a hexagonal subcell of methylene groups. Using transmission electron microscopy (TEM), we determined that the lamella thickness in this material is between 10 and 20 nm [24].

We subsequently broadened our study to include a wide variety of alkyl groups, to determine the effects of branch identity (Table 1). This effort was aided by the development of a two-step monomer synthesis by Giovanni Rojas [25]. We synthesized polymers with alkyl branches of various sizes ranging from a methyl group to an adamantyl group. Despite the diversity of branch identities, these polymers could be simply divided into two groups—those



in which the branch is included in the unit cell, and those in which the branch is excluded from the unit cell. When the branch is included in the unit cell, the crystal lattice is triclinic. When the branch is excluded, the crystal lattice is orthorhombic. For the series of polymers with a branch on every 21st carbon, three branch types were included in the unit cell: methyl, gem-dimethyl, and ethyl [26]. The

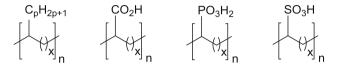


Fig. 3 Selected examples of precision polymers

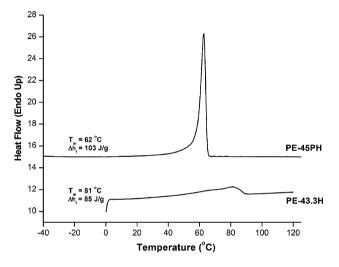


Fig. 4 ADMET methyl-branched polyethylene: precision (*top*) versus random (*bottom*) [23]

Table 1 Thermal and crystallographic data of alkylbranched precision polyethylene [26, 28, 29]

polymers with branches included in the unit cell display decreasing melting temperatures with increasing branch size. The trend continues as the series is extended from the ethyl to the propyl branch; however, even as the branch size continues to increase, the polymers with branches excluded from the unit cell all display very similar melting temperatures.

The crystalline structures of ADMET polyethylene with ethyl or *n*-hexyl branches were studied using TEM [27]. These measurements showed that the ethyl-branched sample displayed straight lamellar crystals with an average thickness of 55 Å. This length corresponds to twice the ethylene sequence length between branches, indicating that one lamellar stem includes three ethyl branches, one of which is within the lamella. Similar measurements were performed on ADMET polyethylene with *n*-hexyl branches on every 21st carbon. This material displayed a narrow lamella thickness distribution, with lamellae averaging 25–26 Å in thickness. This length indicates that one lamella stem comprises a single ethylene sequence between two branches, further supporting the conclusion that the larger hexyl branch is entirely excluded from the crystalline phase.

Effect of branch spacing

We know that methyl branches are included in the polymer crystal lattice, but what effect does the branch frequency have on polymer morphology? To answer this question, a series of precise methyl-branched polymers was synthesized (Table from slide 30), ranging from having a methyl branch on every fifth carbon, to having a branch on every

Alkyl branch on every 21st car	Alkyl branch	Alkyl branch on every 39th carbon				
Branch inclusion in unit cell	T _m (°C)	$\Delta h_{\rm m} ({\rm J/g})$		T _m (°C)	$\Delta h_{\rm m} ({\rm J/g})$	Branch inclusion in unit cell
Included triclinic system	134	204	No branch	134	204	Included
	63	104	Methyl	92	137	
	45	61	gem- Dimethyl	-	-	Excluded
	24	65	Ethyl	76	93	
Excluded orthorhombic system	12	60	Propyl	78	71	
	12	57	Butyl	76	128	
	14	58	Pentyl	74	74	
	12	49	Hexyl	73	73	
	_	_	Decyl	71	76	
	_	_	Pentadecyl	70	83	
	11	37	iso-Propyl	77	74	
	13	50	tert-Butyl	_	_	
	9	43	sec-Butyl	-	_	



21st carbon, and ultimately having no branches (linear ADMET polyethylene). These precise ethylene-*co*-propylene copolymers displayed a melting point range of almost 200 °C. When a methyl group was incorporated on every fifth carbon, the polymer was completely amorphous. As the distance between the branches increased, the melting temperature also increased. This indicates that as the frequency of branch defects decreases, the crystallinity of the polymer is increased.

A series of alkyl-branched polymers with precise branches on every 39th carbon were also synthesized (Table 1) [28, 29]. Unlike the previous alkyl-branched series, with branches on every 21st carbon, the ethyl branch was excluded from the unit cell of these polymers. This demonstrates the importance of branch frequency. As the distance between branches increases, the crystalline region becomes more likely to exclude branch defects. Again, each polymer with branches excluded from the unit cell displayed similar melting temperatures (around 74 °C). The distance between precise branch points was even further extended to 74 methylene units between butyl branches [30]. From this work, we can conclude that melting point is determined by branch-to-branch distance—up to 75 carbons (Fig. 5) [27]. We believe chains fold to form lamellae devoid of large defects. Thus, the larger alkyl groups are excluded from the crystal lattice. However, the distance between branch points must be large enough to allow crystallization to occur. When there are no branches, orthorhombic crystals form; when there are small, precisely placed branches, they are included in the crystal, resulting in a triclinic unit cell; when there are large, precisely placed branches, they are excluded from the crystal, and the polymer reverts to an orthorhombic crystal.

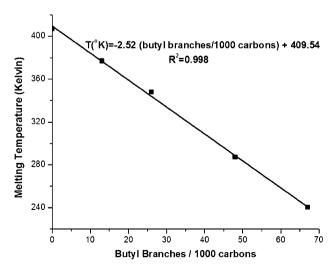


Fig. 5 Linear relationship between melting temperature and branch frequency in precision butyl-branched polyethylene [30]



We know that large precision branches are excluded from the crystalline region of the polymer, but is there some special "ordering" of these branches in the amorphous region? Such interactions in the amorphous region were first observed at Sumitomo Chemical Company [31]. When ADMET polyethylene with either an ethyl branch or an *n*-hexyl was isothermally crystallized at the crystallization temperature, the formation of a transient-ordered mesophase and subsequent crystallization through packing optimization was observed. This raises the question, if alkyl branches interact in the amorphous region of the polymer, how would strongly interacting groups affect the polymer morphology?

Precision Acid and ionomer functional group branches

To investigate the effect of strongly interacting groups precisely placed on a polyethylene backbone, a series of polymers were designed and synthesized with pendant carboxylic acid groups. The synthesis of these polymers was straight forward, and involved the polymerization of a protected carboxylic acid (Fig. 6). In this case, hydrogenation not only reduced the double bonds in the polymer backbone, but also removed the protecting group, yielding the final polymer. Carboxylic acid branches were included on ever 9th, 15th, and 21st carbon. In the case of the polymer with carboxylic acid groups on every 21st carbon, the acid groups were excluded from the polyethylene-like orthorhombic unit cell. When the acid branches were more frequent, the polymers were amorphous due to increased steric congestion along the polymer backbone (Fig. 7) [6]. Analogous polymers were created using ROMP, which had the same concentration of acid groups along the polymer backbone, but without precise spacing.

From FTIR evidence, we know the carboxylic acid groups on the polymer dimerize. Interestingly, these acid-acid correlations persist above the melt [22]. The correlations between acid dimers in the amorphous matrix result in an X-ray scattering peak, which is not observed in the random ROMP analogs, due to the fact that the dimer positions are uncorrelated. Analysis of the X-ray scattering pattern of a sample made from the ADMET ethylene/acrylic acid copolymer indicated a layered structure, where each lamellae is approximately 2.6–4.2 nm thick and contains 2–3 acid-

Fig. 6 Precision carboxylic acid-branched polyethylene [6]

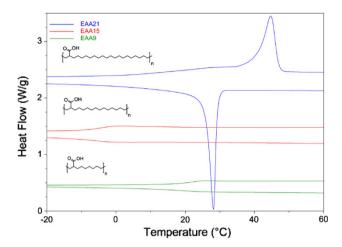


Fig. 7 Differential scanning calorimetry analysis of carboxylic acidfunctionalized precision polyethylene [6]

rich layers. This layered morphology was absent from the ROMP analogs, in which the random placement of the acid groups accommodates the H-bonding acid dimerization without producing periodic acid-rich layers.

These precise carboxylic acid polymers were subsequently transformed to ionomers. An ionomer is a polymeric material that contains ionic units. Typically, such materials exhibit ionic clustering due to polar functional groups trapped within a nonpolar matrix. From the precise and random series of ethylene/acrylic acid copolymers, a new series of zinc-neutralized ionomers were created and studied. The ionic aggregates in an amorphous, precise ionomer with 22 mol% acid and 66 % neutralization adopt a cubic lattice. This was a remarkable observation, given that it is the first report of ionic aggregate self-assembly onto a lattice in an ionomer with an all-carbon backbone [32].

Because the primary structure of these precision ionomers is known exactly, they are particularly useful for comparison between theoretical simulation and experimental results. At Sandia National Laboratories, a set of coarse-grained molecular dynamics simulations of ionomer melts with varying polymer architectures was performed. These results were then compared to experiments to understand ionic aggregation at the molecular level [33]. Our materials are ideal for clear comparisons with simulations because they have a much stronger ionomer scattering peak than randomly spaced analogs, and because of their highly ordered morphology. In this work, the effect of varying the spacing between charges and the effect of randomness in the spacing was studied. Simulations of both precise and random Na⁺neutralized poly(ethylene-co-acrylic acid) ionomers were compared with X-ray scattering data. In both the simulation and experimental data it was found that increasing the spacing between ions along the chain moves the peak to a lower wave vector for precise materials. Additionally, moving from precise to pseudorandom materials broadens the ionomer peak and moves it to a lower wave vector. This is also seen in simulation when moving from periodic to random block models.

The characterization of our precise carboxylic acid AD-MET polymers established that strongly interacting pendant groups have a profound effect on polymer morphology. However, a carboxylic acid has only one acidic proton. What if each pendant group had even greater H-bonding capabilities? To answer this question, we synthesized a series of precision polymers with pendant phosphonic acid groups [7, 34]. In this series, both frequency of acid appearance along the polymer backbone and the architectures associated with each position (single, germinal, and benzyl) were varied. The variation in the primary structure had a direct effect on thermal behavior, particularly crystallization behavior. A logical next step would be the synthesis of a sulfonic acidbranched polymer. This polymer, with triprotic-interacting branches, would potentially have even stronger interactions between the acid groups. Additionally, data generated through studying such a polymer would have implications for fuel cell membranes and proton conduction membranes, which utilize sulfonic acid groups. Synthesis of this material is ongoing in our lab.

Conclusion

Using ADMET polymerization, we have expanded our understanding of fundamental structure-property relations. Our ability to set the primary structure has allowed us to systematically study morphology with a greater degree of control. The key is precision. The precise placement of specific branches at precise increments along the polymer backbone leads to significant and quantifiable effects in the polymer morphology. Our understanding of the connection between variations in the primary structure and variations in morphology would not be possible, however, without first knowing the repeat unit of the polymer. Thus, understanding is built on the fundamental chemistry involved: Understanding the metathesis reaction allows the design of the primary structure, which allows control over the higher levels of organization, leading ultimately to control over morphology.

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