

Photocatalytic Degradation of Isopropanol Over PbSnO₃ Nanostructures Under Visible Light Irradiation

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Abstract Nanostructured PbSnO₃ photocatalysts with particulate and tubular morphologies have been synthesized from a simple hydrothermal process. As-prepared samples were characterized by X-ray diffraction, Brunauer–Emmet–Teller surface area, transmission electron microscopy, and diffraction spectroscopy. The photoactivities of the PbSnO₃ nanostructures for isopropanol (IPA) degradation under visible light irradiation were investigated systematically, and the results revealed that these nanostructures show much higher photocatalytic properties than bulk PbSnO₃ material. The possible growth mechanism of tubular PbSnO₃ catalyst was also investigated briefly.

Keywords Nanostructures · Photocatalysts

Introduction

Since the Honda–Fujishima effect was reported in 1972, considerable efforts have been paid to develop semiconductor photocatalysts for water splitting and degradation of organic pollutants in order to solve the urgent energy and environmental issues [1–9]. However, to date, most of the photocatalysts reported only respond to UV light irradiation (<420 nm). For visible light accounts for about 43% of the

solar spectrum, the utilization of visible light is more significant than UV light and thus developing visible light-driven photocatalyst is one of the most important and meaningful subjects in this field. The fundamental steps for photocatalytic reaction of oxide semiconductor mainly include the following processes: (i) the generation of photoexcited charges in the semiconductor materials, (ii) the separation and migration of the generated charges without recombination, and (iii) the redox reaction on the surface of the semiconductor. The first and second steps are associated with the electronic structures of the oxide semiconductor, while the third step is strongly relevant to the surface properties of the catalyst [10–12].

Generally, the improvement of surface area always contributes to more reaction sites, which is beneficial to the photocatalytic reaction. With particular microstructures, nanomaterials have recently gained much attention to be used as high-performance photocatalysts with enhanced photocatalytic activities. For example, in our previous work, we reported the synthesis of perovskite SrSnO₃ nanostructures [13] from a facile hydrothermal method. Compared with the catalyst from the traditional solid state route, nanostructured SrSnO₃ catalysts with larger surface areas showed higher photocatalytic activities for water splitting under UV light irradiation. Undoubtedly, the enhanced photocatalytic activities are mainly attributed to the increased surface areas, which are believed to be one of the efficient approaches to enhance the activity of catalysts. From a similar hydrothermal process, we reported here the preparation of a new visible light-responded photocatalyst, PbSnO₃ nanostructures including particulate and tubular shapes. Experimental results confirmed that these nanostructures show distinguished photocatalytic oxidation activity upon mineralizing isopropanol (IPA) into CO₂ in the visible light region.

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Experimental Section

Synthesis of PbSnO₃ Nanostructures

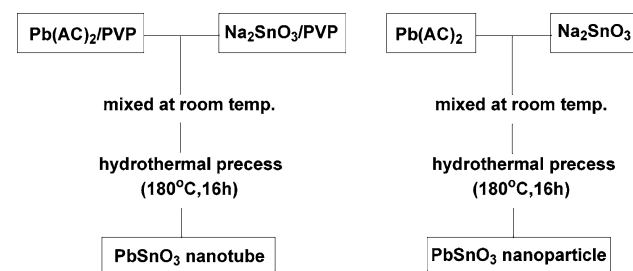
For the synthesis of tubular PbSnO₃ nanostructures, two same surfactant–water solutions were first prepared by dissolving 0.2 g poly(vinyl pyrrolidone) (PVP) surfactant in 25 mL distilled water, respectively. Then, equivalent amounts of Pb(AC)₂ and Na₂SnO₃ (2 mmol) were dissolved in the above surfactant–water solution at room temperature, separately. After stirred for 30 min, the solutions were mixed together and kept stirring for another 30 min, which were then transferred into a Teflon-lined stainless steel autoclave and subsequently heated at 180 °C for 16 h in an oven. After cooling to room temperature, the yellow precipitate was filtered and washed for several times with distilled water and ethanol, respectively, then dried in air at 70 °C. PbSnO₃ nanoparticles were also synthesized in this work using a similar process without the use of surfactant PVP. Brief flowcharts illustrating the formation of PbSnO₃ nanostructures are shown in Scheme 1.

Synthesis of Bulk PbSnO₃ from SSR

To compare the photocatalytic properties, bulk PbSnO₃ was also synthesized by selecting optimal experimental parameters including calcinations temperature and time. For the synthesis of PbSnO₃ bulk material, we first dissolved equivalent amounts of Pb(AC)₂ and Na₂SnO₃ into distilled water under stirring, and then mixed them to obtain the white precursor. Heating the white precursor at 500 °C for 5 h in a quartz tube under Ar flow resulted in yellow powders. In this process, temperature is very important for the formation of yellow powders due to the instability of PbSnO₃ at high temperature.

Characterization

The crystal structure of the as-prepared sample was confirmed by the X-ray diffraction pattern (JEOL JDX-3500



Scheme 1 Flowchart for preparing PbSnO₃ nanostructures by the hydrothermal process

Tokyo, Japan). The morphology and size of the sample were characterized by transmission electron microscope (HRTEM, JEM-3000F) equipped with an X-ray dispersive spectrometer (EDS). UV–Vis diffuse reflectance spectra were recorded on a UV/Vis spectrometer (UV-2500, Shimadzu) and were converted from reflection to absorbance by the standard Kubelka–Munk method. The surface area of the sample was measured by the BET method (Shimadzu Gemini Micromeritics).

Evolution of Photocatalytic Property

The photoactivities of the obtained PbSnO₃ nanostructures were evaluated by decomposition of gaseous IPA under visible light irradiation. Typically, 0.1 g PbSnO₃ catalyst was spread uniformly in a quartz-made vessel with an irradiation area of 7.8 cm². Prior to light irradiation, the vessel was kept in dark for 2 h until an adsorption–desorption equilibrium was finally established. The visible light with light intensity of about 1.8 mW/cm² was obtained by using a 300 W Xe lamp with a set of combined filters (L42 + B390 + HA30) and a water filter. The products in the gas phase were analyzed with a gas chromatograph system (GC-14B, Shimadzu, Japan), using a flame ionization detector (FID) for organic compounds determination.

Results and Discussion

Crystal Structure and Morphology

The crystal structure of both as-synthesized PbSnO₃ nanostructures from the hydrothermal process and bulk material from the solid-state route were characterized by XRD and the results are shown in Fig. 1. In these patterns, all peaks can be indexed as cubic phase PbSnO₃ with pyrochlore-type structure (space group: *Fd3m*). The calculated lattice constant $a = 10.67 \text{ \AA}$ is in agreement with previously reported value (JCPDS 17-060). From the XRD patterns, it can be clearly seen that the PbSnO₃ nanostructures are of better crystallinity than the bulk material, which might be one of the reasons why nanostructured PbSnO₃ show higher photocatalytic activities (detailed contents in the part of discussion). Inset in Fig. 1 is a typical SEM image of the product from the SSR. Scheme 2 shows the crystal structure of pyrochlore-type PbSnO₃, an anion-deficient three-dimensional framework consisting of corner-sharing SnO₆ octahedra.

Figure 2a shows a TEM image of as-prepared PbSnO₃ nanoparticles from the hydrothermal process. Obviously, the products are consisted of many small nanoparticles with dimensions in the range of 10–15 nm. The corresponding

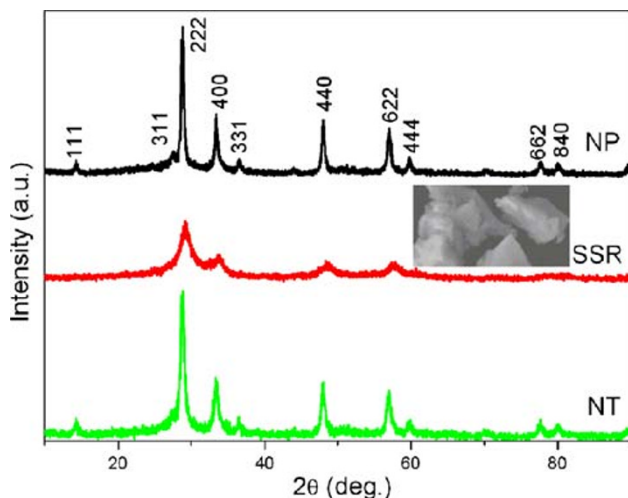
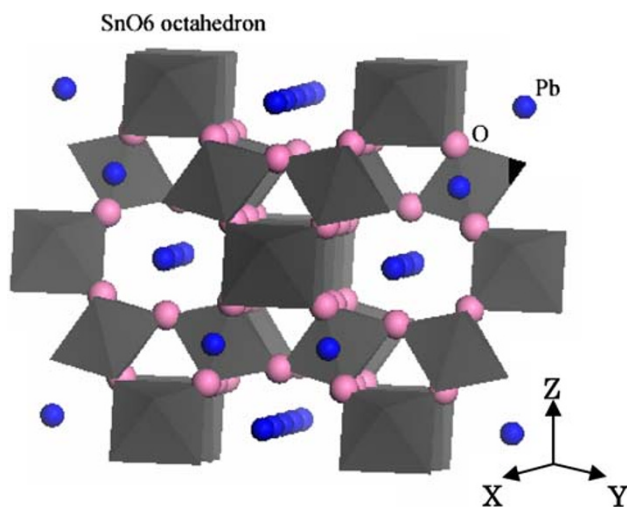


Fig. 1 XRD patterns of the as-prepared PbSnO_3 nanostructures from the hydrothermal route and bulk samples from the solid-state route, respectively. Inset shows SEM image of bulk material from SSR



Scheme 2 Crystal structure of pyrochlore PbSnO_3

selected-area electron diffraction (SAED) pattern (Fig. 2b) can be readily indexed as cubic phase PbSnO_3 , which is in agreement with the XRD result. An EDS spectrum in Fig. 2c depicts the presence of Pb, Sn, and O elements, indicating the formation of PbSnO_3 . In this spectrum, the signals corresponding to Cu arise from the TEM grid. The microstructures of the produced PbSnO_3 nanoparticles were investigated using high-resolution TEM. As indicated in Fig. 2d, the nanoparticles are well-crystallized and of good crystallinity. The marked lattice fringes of 0.32 and 0.25 nm correspond well to the (311) and (331) crystalline planes of cubic PbSnO_3 .

In the presence of surfactant PVP, polycrystalline PbSnO_3 nanotubes were obtained instead of nanoparticles. Panels (a) and (b) of Fig. 3 are typical TEM images of as-obtained PbSnO_3 nanotubes, which reveal that the

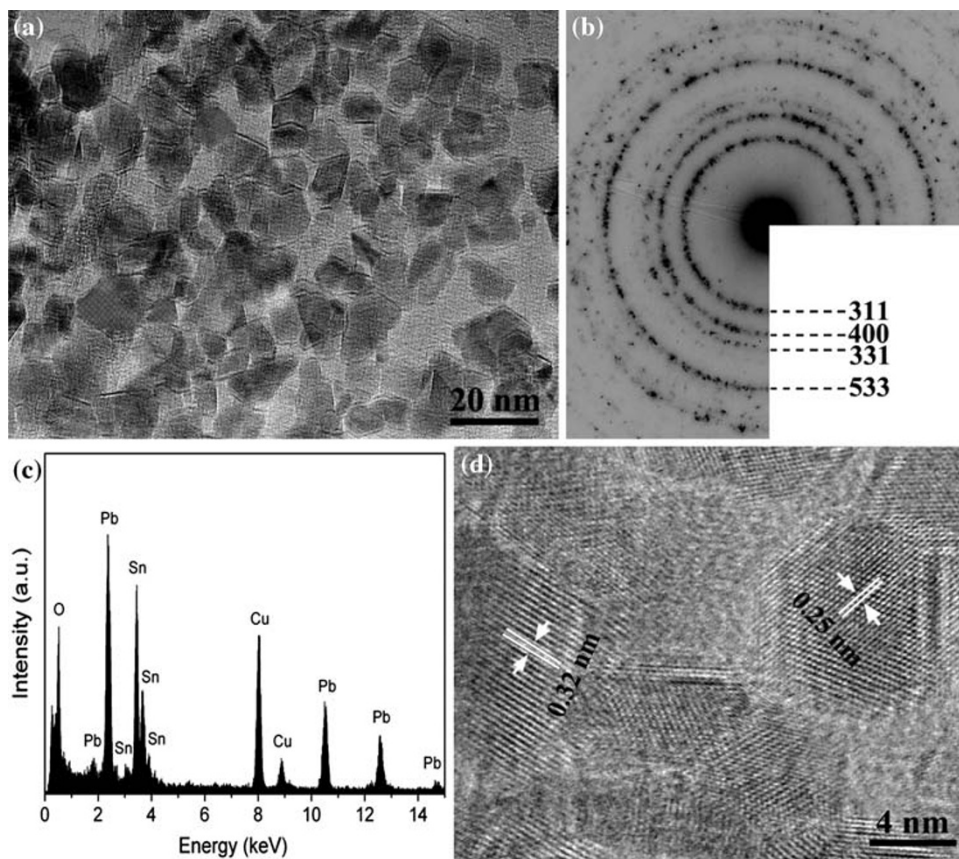
nanotubes are polycrystalline with typical diameters of 300–340 nm and wall thickness of 40–80 nm. Figure 3c is the corresponding SAED pattern taken from a single PbSnO_3 nanotube, confirming the formation of polycrystalline nanotube. The three polycrystalline rings are in accordance with those of (311), (400), and (533) of cubic phase PbSnO_3 . Typical HRTEM images of the nanotubes are shown in Fig. 3d and e. It can be seen that the polycrystalline PbSnO_3 nanotubes are composed of numerous nanoparticles with diameters of several to ten nanometers. The interplanar spacing was calculated to be about 0.32 nm, corresponding to the (311) plane of cubic PbSnO_3 , in accordance with the SAED result.

UV–Vis spectra of all three PbSnO_3 samples were checked and the spectra are displayed in Fig. 4. It is evident that PbSnO_3 nanostructures could absorb much more visible light than bulk sample at the present condition. Corresponding band gaps of PbSnO_3 are determined to be 2.8 eV for bulk material, 2.8 eV for nanotubes, and 2.7 eV for nanoparticles from the absorption edges, respectively (as shown in Table 1).

Growth Mechanism

One-dimensional micro- or nanosized tubular materials with hollow interior structure have attracted extraordinary attention owing to their unique properties and potential applications [14–16]. Many kinds of growth mechanisms have been proposed for the formation of nanotubes. For example, the rolling mechanism and template-assisted mechanism have been reported to explain the formation of tubular structure with layered or pseudo-layered structures such as BN [17], NiCl_2 [18], Nb_2O_5 [19], Se [20], etc. During the growth of PbSnO_3 nanotubes, surfactant PVP was used and was found to be the key issue for nanotube growth. Thus, the surfactant-assisted growth process can be used to explain the formation of these nanotubes. The possible formation process of PbSnO_3 nanotubes may involve three following distinctive stages: (i) the generation of PbSnO_3 particles, (ii) the adsorption of PVP molecules on the surface of particles and subsequently self-assembly into tubular microstructure, and (iii) the formation of uniform PbSnO_3 nanotubes. In the initial stage, cubic PbSnO_3 tiny nuclei could easily crystallize and serve as the seeds for the growth of nanotubes. Meanwhile, PVP molecules in the solution would strongly and rapidly adsorb on the surfaces of these nascent nuclei, which confined the crystal growth and efficiently controlled the dimension and morphology of the final products. Then, these particles with high free energy aggregated and self-assembled into tubular structures with the help of PVP template molecules. As a result, the growth of PbSnO_3 nanotubes would form eventually by a typical oriented

Fig. 2 **a** TEM image; **b** SAED pattern; **c** EDS spectrum; **d** HRTEM image of the as-prepared PbSnO_3 nanoparticles from the hydrothermal process



attachment process under the hydrothermal conditions. Meanwhile, the existence of PVP in this solution can alter the surface energies of various crystallographic surfaces to promote selective anisotropic growth of nanocrystals [21].

Photocatalytic Degradation of IPA

The photocatalytic activities of the PbSnO_3 nanostructures were evaluated by IPA mineralization under visible light irradiation. Under visible light irradiation, gaseous IPA was gradually oxidized through an acetone intermediate to CO_2 , and the concentration changes of IPA, acetone, and CO_2 versus time over PbSnO_3 nanoparticles are shown in Fig. 5. It was clear that the concentration of IPA in the reaction system almost decreased from the initial concentration to zero; the concentration of acetone also decreased continually while the concentration of CO_2 increased with the long-term irradiation. Inset in Fig. 5 shows that almost no additional CO_2 gas was detected under dark test, suggesting that degradation of IPA over the catalyst was driven by light irradiation. Figure 6 further displays the concentration changes of evolved acetone over different PbSnO_3 nanostructures and bulk material with the increasing of irradiation time. Clearly, acetone was detected over all these catalysts when light was turned on. Among them,

particulate PbSnO_3 performs the best activity for degradation of IPA under the present conditions.

In this case, the photocatalytic activities for IPA degradation over these catalysts were in the order of nanoparticle > nanotube > bulk material, which was in consistent with that of BET surface areas. As mentioned earlier, BET surface area of catalyst is closely related to its photoactivity. Usually, larger surface area means much more active sites, at which the photocatalytic reaction occurs. Thus, as shown in Table 1, PbSnO_3 nanostructures with larger surface areas as $68 \text{ m}^2/\text{g}$ for nanoparticles and $50 \text{ m}^2/\text{g}$ for nanotubes, respectively, resulted in enhanced photocatalytic activities than bulk material with $10 \text{ m}^2/\text{g}$ of surface area. Meanwhile, the improved crystallinity of PbSnO_3 nanostructures (shown in XRD patterns) resulted in the increase of photocatalytic activity since it could reduce electron-hole recombination rate.

The wavelength dependence of the rate of acetone evolution from IPA degradation over PbSnO_3 nanoparticles was investigated by using different cutoff filters, as shown in Fig. 7. The intensity variation of the incident light with different cutoff filters is given as an inset figure for reference. It is notable that the rate of acetone evolution decreased with increasing cutoff wavelength, which is in good agreement with the UV–Vis diffuse reflectance

Fig. 3 **a, b** TEM images; **c** SAED pattern; **d, e** HRTEM images of the as-prepared PbSnO_3 nanotubes in the presence of surfactant PVP

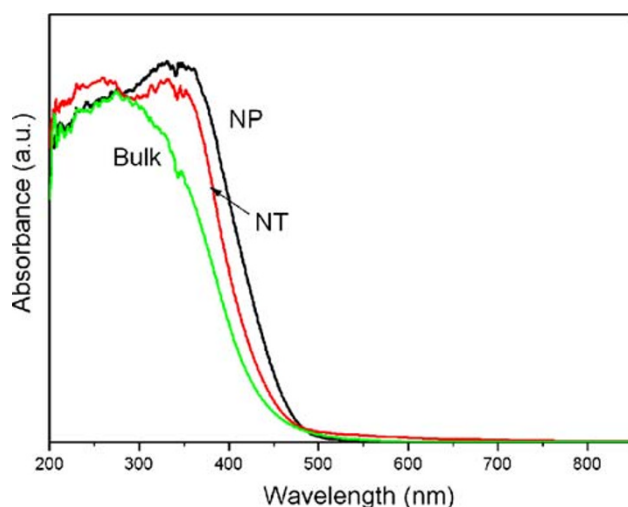
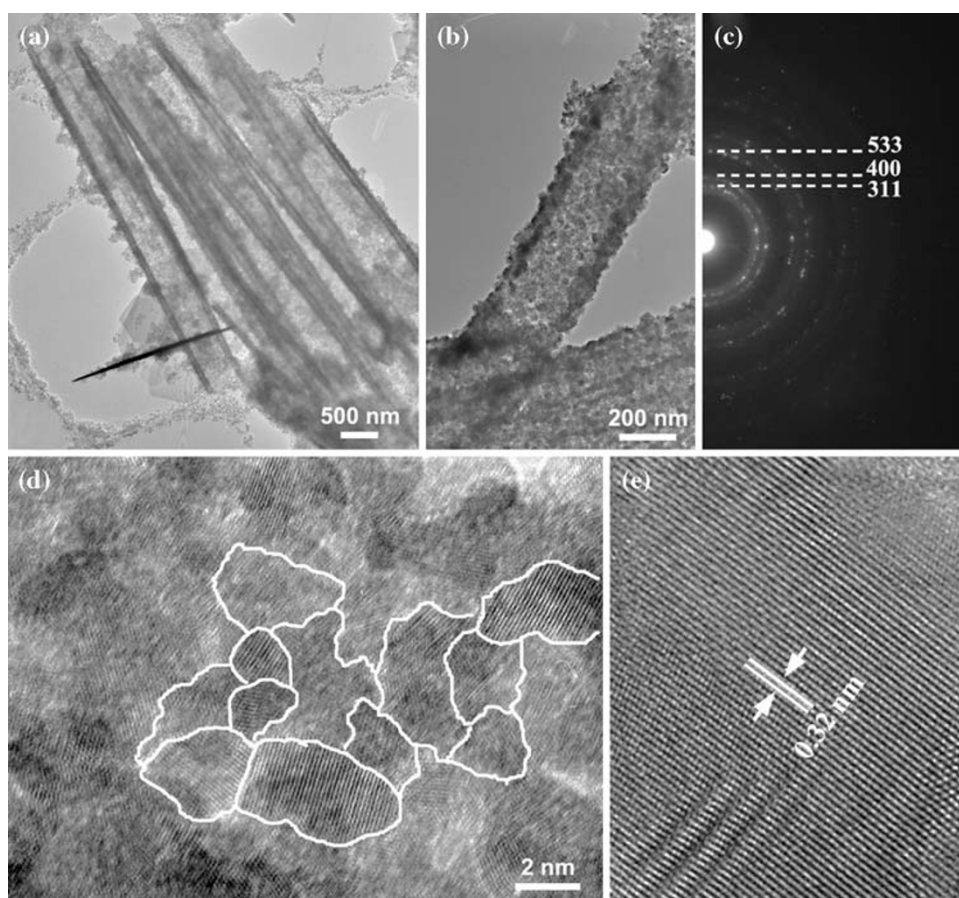


Fig. 4 UV-Vis diffuse reflectance spectra of PbSnO_3 nanostructures from the hydrothermal route and PbSnO_3 particles from the solid-state route, respectively

spectra of PbSnO_3 nanoparticles, indicating the present reaction is driven by a visible light absorption. The used catalysts were again checked by XRD and UV-Vis reflectance spectroscopy to explore the stabilities of

Table 1 Physical and photocatalytic properties of PbSnO_3 samples

Sample	Band gap (eV)	BET (m^2/g)	Rate of acetone (ppm/h)
NP	2.7	68	42.2
NT	2.8	50	18.5
Bulk ^a	2.8	10	5.1

^a Bulk PbSnO_3 are prepared from the solid-state route

samples. There was no detectable change between the spectra of PbSnO_3 before and after the photodegradation of IPA gas, suggesting that the catalyst was fairly stable for the degradation of organic compounds. For many p-block metal oxides photocatalysts with d^{10} configuration, the VB and CB are the 2p orbital of the oxygen atom and the lowest unoccupied molecular orbital (LUMO) of p-block metal center, respectively [22–24]. Meanwhile, for the lead-containing compounds, it was found that an additional hybridization of the occupied Pb 6s and O 2p orbitals seems to push up the position of the valence band and result in a narrower band gap [25]. Based on the above depiction, we assumed that the VB of PbSnO_3 is composed of hybridized Pb 6s and O 2p orbitals, whereas the CB is composed of Sn 5s orbitals, and these bands meet the potential requirements of organic oxidation.

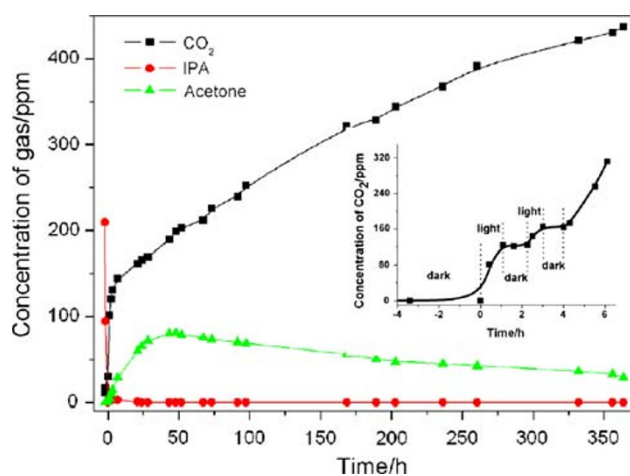


Fig. 5 Changes of IPA, acetone, and CO₂ concentrations as a function of time in the presence of PbSnO₃ nanoparticles from the hydrothermal process under visible light irradiation (catalyst: 0.1 g, 300 W Xe lamp, 420 nm cutoff filter and water filter). Inset shows that no CO₂ gas was evolved turning off the light

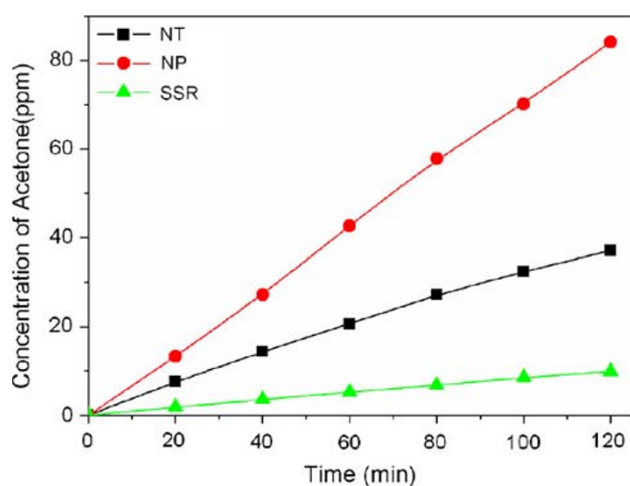


Fig. 6 Acetone evolution from IPA photodegradation over various PbSnO₃ samples (catalyst: 0.1 g, 300 W Xe lamp, L42 + B390 + HA30 and water filter)

Conclusion

In summary, we have successfully synthesized pure phase PbSnO₃ nanoparticles and nanotubes from the facile hydrothermal process at low temperature. The surfactant PVP used as the capping reagent plays a crucial role in the formation of tubular PbSnO₃ structure. PbSnO₃ nanostructures with better crystallinity and larger surface areas show enhanced photocatalytic activity for the decomposition of organic pollutant isopropanol under the visible light irradiation than the catalyst prepared by the solid-state method.

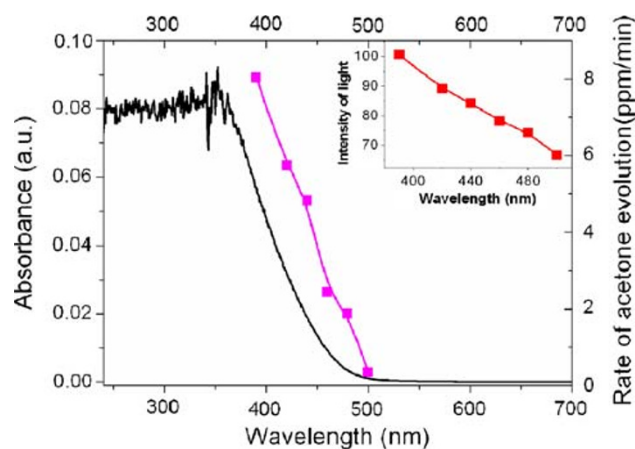


Fig. 7 Wavelength dependence of acetone evolution from isopropanol photodegradation on the cutoff wavelength of incident light, and UV-Vis diffuse reflectance spectrum of PbSnO₃ samples. The inset shows the wavelength dependence of light intensity with different cutoff filters (catalyst: 0.1 g, 300 W Xe lamp, 400 nm <math>\lambda < 500\text{ nm}</math>)

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References

1. A. Fujishima, K. Honda, *Nature* **238**, 37 (1972). doi:10.1038/238037a0
2. T. Kawai, T. Sakata, *Nature* **286**, 474 (1980). doi:10.1038/286474a0
3. M.R. Hoffmann, S.T. Martin, W. Choi, *Chem. Rev.* **95**, 69 (1995). doi:10.1021/cr00033a004
4. C.C. Wong, W. Chu, *Environ. Sci. Technol.* **37**, 2310 (2003). doi:10.1021/es020898n
5. W. Ho, J. Yu, J. Lin, P. Li, *Langmuir* **20**, 5865 (2004). doi:10.1021/la049838g
6. M.A. Fox, M.T. Dulay, *Chem. Rev.* **3**, 341 (1993). doi:10.1021/cr00017a016
7. Z.G. Zou, J.H. Ye, K. Sayama, H. Arakawa, *Nature* **414**, 625 (2001). doi:10.1038/414625a
8. J. Sato, S. Saito, H. Nishiyama, Y. Inoue, *J. Phys. Chem. B* **105**, 6061 (2001). doi:10.1021/jp010794j
9. J. Sato, S. Saito, H. Nishiyama, Y. Inoue, *J. Photochem. Photobiol. Chem.* **148**, 85 (2002). doi:10.1016/S1010-6030(02)00076-X
10. H.B. Fu, C.S. Pan, W.Q. Yao, Y.F. Zhu, *J. Phys. Chem. B* **109**, 22432 (2005). doi:10.1021/jp052995j
11. H.L. Xu, W.Z. Wang, W. Zhu, *J. Phys. Chem. B* **110**, 13829 (2006). doi:10.1021/jp061934y
12. J.Q. Yu, A. Kudo, *Adv. Funct. Mater.* **16**, 2163 (2006). doi:10.1002/adfm.200500799
13. D. Chen, J.H. Ye, *Chem. Mater.* **19**(18), 4585 (2007). doi:10.1021/cm071321d
14. D.R. Bae, S.J. Lee, S.W. Han, J.M. Lim, D.M. Kang, J.H. Jung, *Chem. Mater.* **20**, 3809 (2008). doi:10.1021/cm703674d

15. T.W. Ebbesen, P.M. Ajayan, *Nature* **358**, 220 (1992). doi: [10.1038/358220a0](https://doi.org/10.1038/358220a0)
16. B. Chi, E.S. Victorio, T. Jin, *Chem. Lett.* **35**, 350 (2006). doi: [10.1246/cl.2006.350](https://doi.org/10.1246/cl.2006.350)
17. N.G. Chopra, R.J. Luyken, K. Cherrey, V.H. Crespi, M.L. Cohen, S.G. Louie, A. Zettl, *Science* **269**, 966 (1995). doi: [10.1126/science.269.5226.966](https://doi.org/10.1126/science.269.5226.966)
18. Y.R. Hacoheh, E. Grunbaum, R. Tenne, J. Sloan, J.L. Hutchison, *Nature* **295**, 336 (1998). doi: [10.1038/26380](https://doi.org/10.1038/26380)
19. Y. Kobayashi, H. Hata, M. Salama, T.E. Mallouk, *Nano Lett.* **7**, 2142 (2007). doi: [10.1021/nl0708260](https://doi.org/10.1021/nl0708260)
20. Y.R. Ma, L.M. Qi, J.M. Ma, H.M. Cheng, *Adv. Mater.* **16**, 1023 (2004). doi: [10.1002/adma.200400071](https://doi.org/10.1002/adma.200400071)
21. S.H. Yu, B. Liu, M.S. Mo, J.H. Huang, X.M. Liu, Y.T. Qian, *Adv. Funct. Mater.* **13**, 639 (2003). doi: [10.1002/adfm.200304373](https://doi.org/10.1002/adfm.200304373)
22. J.M. Herrmann, J. Disdier, P. Pichat, *J. Catal.* **113**, 72 (1988). doi: [10.1016/0021-9517\(88\)90238-2](https://doi.org/10.1016/0021-9517(88)90238-2)
23. D.S. Muggli, L. Ding, M.J. Odland, *Catal. Lett.* **78**, 23 (2002). doi: [10.1023/A:1014954114655](https://doi.org/10.1023/A:1014954114655)
24. X.Z. Li, F.B. Li, *Environ. Sci. Technol.* **35**, 2381 (2001). doi: [10.1021/es001752w](https://doi.org/10.1021/es001752w)
25. H.G. Kim, O.S. Becker, J.S. Jang, S.M. Ji, P.H. Borse, J.S. Lee, *J. Solid State Chem.* **179**, 1214 (2006). doi: [10.1016/j.jssc.2006.01.024](https://doi.org/10.1016/j.jssc.2006.01.024)